University of Stavanger FACULTY OF SCIENCE AND TECHNOLOGY			
MASTER	RTHESIS		
Study program/specialization: Master of Science in Engineering Structures and Materials/Mechanical Systems	Spring semester, 2019 Confidential		
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Title of Master Thesis: Investigation of Material Property changes in HSLA Steel due to Weld Proximity.			
Credits: 30 p			
Keywords: Weld, Mechanical testing, S420, HSLA, Fatigue, Residual Stress, Ultrasound, Microstructure, Optical Microscope, WPQ, WPQR, WPS, NS-EN ISO 15614	Number of pages: 170 + appendices/other: 124 Stavanger, 15.08.2019 date/year		

Acknowledgements

Several key personnel from both business and academia has been helping us out in this project.

The thesis project was proposed by Professor Chandima Ratnayake Ratnayake Mudiyanselage at the University of Stavanger. He has always made sure to have time to help and guide us throughout the project.

Welding Engineer Arild Finnesand at KIWA (formerly Teknologisk Institutt AS) in Stavanger has been very helpful throughout the project. He has supported us with the welding of the steel plates, theoretical and technical knowledge. He has also been very adamant about teaching us the proper welding procedures and designing a complete WPS.

Arild also introduced us to metallurgist and CEO Petter Lunde at Quality Lab in Forsand. Qlab is a test facility that performs standardized testing for the industry. Petter offered to help us out with the machining and testing of all specimens.

Dr. Xavier Ficquet, principal development engineer at Veqter in Bristol, England, conducted several residual stress measurements; including ultrasonic testing, incremental deep-hole drilling and x-ray diffraction.

Halvar Frøvold and Jan Heggum from NST (Norsk Sveiseteknikk AS) supplied us with free filler material for the welding of all the plates.

Smed T. Kristiansen AS and Watech offered their water cutting services.

IKM Inspection performed NDT X-ray analysis of all welded plates.

Trio Oiltech Services performed the precision grinding of the fatigue test specimens.

All these companies helped us and was eager to offer us all assistance that we needed in order to complete the project.

Professor Morten Andre Langøy was supportive during the project and supplied us with his metallurgical expertise.

We would like to thank Engineer Jørgen Grønsund, Emil Surnevik Kristiansen and Johan Andreas Håland Thorkaas from the Department of Mechanical and Structural Engineering and Materials Science at the University of Stavanger.

We want to give a sincere thanks to our parents who have always supported and given us motivation.

Magnus also wants to give his wife Roselyn a warm hug. Without your patience and support, it would not have been possible for me to carry out this project. I am incredibly fond of you. Also a big hug for my two young children, Summer and William, for the energy and joy they give me.

Abstract

Standards and specifications apply a rule of thumb when determining the distance of two adjacent weld joints, but they rarely give any clear technical justification for the predetermined values. It is vital to understand the implications it has on mechanical and material properties of the weld joint when exceeding these set requirements. The objective of this thesis was aimed at obtaining quantifiable data regarding the implications of having two adjacent weld joints in close proximity.

An experimental analysis was designed where the resulting material behavior was analyzed and documented. The test consisted of six welded 500x300x15 mm S420G2+M steel plates, each having two adjacent parallel butt welds along its full length. The distances between two adjacent welds were 50, 15 and 5 mm in order to analyze the effect a variation in distance have on the properties of the initial weld pass.

All plates were welded at KIWA in Stavanger, and subsequently mechanically tested at Quality Lab in Forsand in accordance with NS-EN ISO 15614-1:2017. A WPQ was established in order to maintain conformance in all subsequent welding operations and ensure replicability of the experiment. Fatigue test and microstructural analysis (optical microscope) was conducted at the Department of Mechanical and Structural Engineering and Materials Science at the University of Stavanger. Residual stress analysis was conducted at Veqter in Bristol using ultrasound (US), Incremental Centre-Hole Drilling (ICHD) and X-ray Diffraction (XRD) technique.

The mechanical testing results from Qlab showed no reduction in the material properties from the tested specimens based on tensile tests, Charpy V Impact toughness test and Vickers hardness test. This was despite that the closest adjacent weld toes were 1.3 mm. There was no clear difference between the different results from the tests.

An optical microscope was used during the microstructural examination of three specimens with different weld to weld distances; 44 mm, 12 mm and 1.3 mm. Based on the observed microstructure in the WM and HAZ the welds at 44 mm and 12 mm between the weld toes had no visible microstructural effect. The weld that had 1.3 mm between the weld toes had a clear HAZ overlap. It did not seem to be any harmful microstructural changes in the HAZ and the heat from the secondary weld appeared to have had a normalizing effect on the initial HAZ. The results from the mechanical testing supported this observation. Because of the limitations in magnification of the optical microscope, further testing should be conducted in order to assure that this is the case. This also requires the addition of SEM and TEM results.

The objective of the fatigue test was to assess if any reduction in fatigue strength had occurred due to the weld proximity situation. The fatigue test specimens were prepared with the intention to isolate and study the microstructure. The preparation of the sample were conducted in accordance with ASTM E466-15 for a homogenous material subjected to high-cycle fatigue. Weld discontinuities was avoided in order to isolate and focus on the HAZ microstructure, and any geometrical and angular weld stress concentrations was removed. The result from the test was that the specimen at 5 mm distance with overlapping welds seemed to have failed in the

secondary weld. The reason for the crack initiation was not possible to determine. Important to note though, was that the failure did not occur in the first weld, which was the focus of the study, but the secondary overlapping weld. In the absence of stress concentrations, the remaining fatigue specimens failed in the parent metal. The sample size was limited, but the results indicated that at 12 mm between two weld toes, the weld has the higher fatigue strength.

The majority of ultrasonic (US) residual stress tests that were performed showed longitudinal tensile and a compressive transverse residual stress in both the weld cap and the root of tested specimens. It was surprising to find compressive transverse residual stress in the weld, but all the subsequent residual stress test methods performed supported this result. This included XRD and ICHD. A Contour measurement planned in the future will provide the longitudinal residual stress and will be used as an additional source for calibration.

The findings in this report can assist in the assessment of a weld proximity or weld-on-weld scenarios. The welding operation did not seem to introduced any degradation of the material properties in the WM and HAZ. The mechanical properties of similar weld joints in the field could be assumed to be unaffected due to the welding of an adjacent weld.

Important to note is that this is for this type of method and materials. More testing has to be performed in other to ascertain the findings. The results can be used as a baseline for further research.

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Abbreviations

AF	Acicular Ferrite
AWS	American Welding Society
BM	Base Material
CE, CET	Carbon equivalent [%]
CGHAZ	Coarse Grain Heat-Affected Zone
FGHAZ	Flux-Cored Arc Welding
GF	Grain boundary Ferrite
GMAW	Gas Metal Arc Welding
HAZ	heat-affected zone
HV	Vickers Hardness
ICHAZ	Inter-Critical Heat-Affected Zone
ICHD	Incremental Hole-Drilling
ISO	The International Organization for Standardization
IWE	International Welding Engineer
J	Joule
MAG	Metal Active Gas
MIG	Metal Inert Gas
MMA	Manual Metal Arc
MT, MPI	Magnetic-Particle Testing
PT	Penetrant Testing
pWPS	preliminary Welding Procedure Specification
SCHAZ	Sub Critical Heat-Affected Zone
SEM	Scanning Electron Microscope
T _{8/5}	Cooling time (from 800 °C to 500 °C)
TEM	transmission electron microscope or Transmission electron microscopy
UTS	Ultimate Tensile Strength
WF	Widmanstätten Ferrite
WPS	Welding Procedure Specification
XRD	X-Ray Diffraction

1 Introduction

1.1 Motivation and background

Restrictions have been set on weld proximity and overlapping welds during the fabrication of steel structures in most industrial codes, standards and specifications. Two weld joints are required to have a minimum required distance between each other to comply with the design requirements. However, no technical explanation has been given in the literature for the predetermined distances. The reasoning behind the set distance is to avoid the adverse effects weld proximity or weld overlapping has on a weld joint, also known as weld-on-weld. However, during the fabrication stage this issue may be unavoidable and can also be found on existing structures. It is vital to maintain code compliance in both these scenarios in order to make an informed decision for the appropriate course of action.

Weld-on-welds are a frequent occurrence in the industry. For instance, a multi-pass weld is just several weld passes on top of each other. There is a clear difference between weld-on-welds and multi-pass welds. A multi-pass weld has been tested and approved, while a weld-on-weld scenario is when two, separately approved welds, conflict with each other by physically overlapping. An initial weld could experience unsatisfactory changes in the HAZ due to the secondary weld. Especially if the weld has been performed under controlled conditions. There are several accepted procedures for weld-on-welds that have been tested and approved. In pipelines, the overlapping of a longitudinal seam weld and a circumferential girth weld have been thoroughly documented and welding procedures have been developed for the specific cases. The nozzles in pressure containing vessels, must sometimes be in an area with several butt welds. Weld overlapping is also common in split tee connections.

We need to differentiate between weld-on-welds and weld proximity. Weld proximity issues arise when two initially approved welds do not physically overlap, but conflicts with the required minimum design distance. This is covered in several international standards. BS 2633 "Class I arc welding of ferritic steel pipework for carrying fluids", states that the toes of adjacent butt welds shall, whenever possible, be no closer than four times the nominal thickness of the pipe [1]. BS 4515 'Specification for welding of steel pipelines on land and offshore' (Section 11), states that the proximity of weld toe-to-toe distance shall not be less than four times the pipe thickness [2]. BS 2971 'Class II arc welding of carbon steel pipework for carrying of fluids' (Section 10), states that if design factors are such that the meeting of more than two welded seams cannot be avoided, then appropriate precautions shall be taken which shall be agreed between the contracting parties [3]. PD5500 'Specification for Unfired fusion welded pressure vessels' (Section 4.1.3), states that where any part of a vessel is made in two or more courses, the longitudinal seams shall be completed before commencing the adjoining circumferential seam(s) and, where practicable, the longitudinal seams of the adjacent courses shall be staggered by four times the nominal thickness or 100 mm, whichever is the greater, measured from the toe of the welds [4].

It is vital to have a good understanding of the implications on the material properties in a weldon-weld and weld proximity scenario in a weld joint. In this area there is a lack of clear guidance in structural standards, codes and specifications, which has led to uncertainties on how to address the specific scenarios. Being able to identify the specific affected material properties is essential when determining what precautions to take in the assessment of a structure.

1.2 Problem statement

Predetermined distances for weld proximity have been set in most standards, codes and specifications in the industry today. There is still a lack of information regarding the implications when divergences from the requirements are necessary or have been detected in the service-life of a structure. More information is needed regarding the consequences on the material properties.

1.3 Scope of the thesis

Design an experimental procedure to test and analyze the effect two parallel adjacent welds have on each other at varying distances.

Approve a WPS for the weld configuration.

Mechanical testing in accordance with ISO 15614-1:2017:

- Visual testing
- Radiographic or ultrasonic testing
- Surface crack detection
- Vickers Hardness Test
- Charpy V Impact Test
- Transverse Tensile test
- Bending test
- Macroscopic examination

In addition, the following tests will be performed:

- Fatigue test
- Microscopic analysis
- Residual stress measurement

1.4 Organization of this thesis

The organisation of the chapters are presented in following figure. Every box corresponds to a certain chapter. A dashed line sorrounding several chapters serves the purpose of grouping the chapters into the categories "Part I: Theory" and "Part II: Experimental Investigation". Each chapter in category "Part II: Experimental Investigation" consists of a methodology, result and conclusion. In the end, a main conclusion summarizes the conclusions from the chapters. The results section of each chapter remains in each sub-chapter.

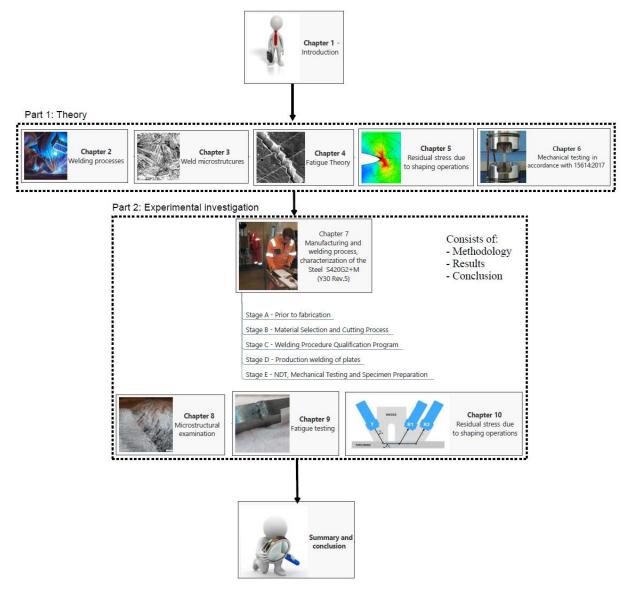


Figure 1-1 - Organisation diagram

1.5 Main challenges

- Develop a viable model for physical testing.
- Involve external partners in the project.
- Develop a welding procedure specification. In order to achieve this a preliminary welding procedure specification must be developed and tested to make sure all parameters are correct.
- Maintain control over all variables by having a systematic and careful working process. This is especially important since the margin of error is very small and there are a lot of uncertainties.
- Conduct extensive testing on finished weldment.

Part I: Theory

2 Welding processes

A large variety of different welding methods exists today but the most important is arc welding. Arc welding is a process that joins two materials by heating them above the melting point and thereby fusing them together [5].

2.1 Manual Metal Arc welding (MMA)

Manual Metal Arc, method nr 111 in EN-ISO is often refered to as stick welding. The MMA method establishes an arc between a sticklike fluxcovered electrode and the base material as seen in Figure 2-1.

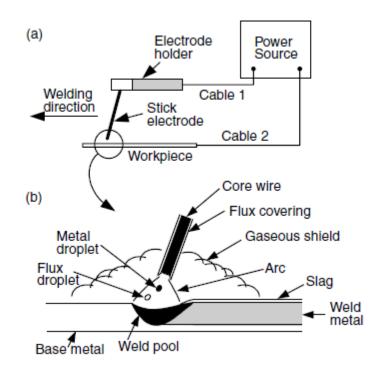


Figure 2-1 – Schematic diagram of Manual Metal Arc welding [5]

The benefits of this method is that it is relatively easy to set up, meets high quality requirements, is suited for outdoor use due to less weather sensitivity and has a high selection of additives.

The negative aspects of the method is that it is slow compared to MIG/MAG welding, and requires regular breaks for electrode replacement. It also requires a relatively skilled welder. It tends to generate more smoke than other methods, which places higher demands on the ventilation system. Basic electrodes are moisture sensitive, which requires extra good storage and handling.

2.2 Gas-metal arc welding (GMAW)

Gas-metal arc welding is the method of joining two metals by heating them with an arc established between a continuously fed filler wire electrode and a weld pool. The shielding mechanism is obtained by using an externally supply of inter gas, such as argon and helium,

hence why it's called metal-inert gas (MIG) welding. Since CO_2 , which isn't inert, is also sometimes used the more suitable name is gas-metal arc welding (GMAW). The method can be used to fuse almost all metals of all thicknesses. Since GMAW doesn't use any flux, it can be used to weld aluminum [5].

Gas-metal arc welding was introduced in the 1920s, but didn't become commercially available until 1948. It was initially implemented as a high-current-density, small-diameter, bare-metalelectrode process using an inert gas for arc shielding. MIG was initially used to weld aluminum, but further progression in the industry offered an alternative with reactive gas welding [6].

2.3 Flux-Core Arc Welding (FCAW)

Flux-Core Arc Welding is similar to Gas-Metal Arc Welding in the sense that it uses a continuous filler material in the welding process as seen in Figure 2-2 [5]. The key difference is that the filler electrode has a flux core similar to manual metal arc welding, which combined with gas in the welding process [5].

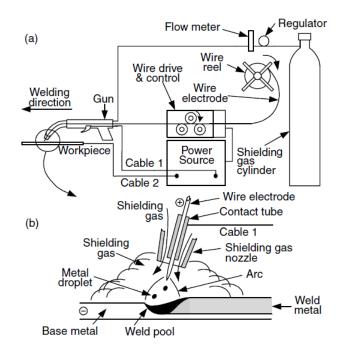


Figure 2-2 – Flux-Core Arc welding: (a) overall process; (b) welding area enlarged

[5].

3 Weld microstructures

3.1 Introduction

The weld joint is divided into two major regions: the fusion zone and the heat-affected zone (HAZ). The fusion zone includes the melt of both deposited weld and adjacent base material. The heat-affected zone is the area around the fusion zone that has not melted, but whose microstructure and mechanical properties has transformed. The area outside the HAZ is called the base material (BM). In this region the temperature does not reach the levels that result in tempering of the base material [7].

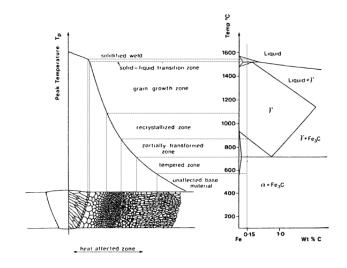


Figure 3-1 – Schematic diagram of the various sub-zones of the heat-affected zone approximately corresponding to 0.15 weight % C [8].

3.2 Fusion zone

The microstructure of the weld metal in the fusion zone is called the as-deposited or primary microstructure. Figure 3-2 illustrates the major microstructural constituents in the primary microstructure. These are grain boundary (or allotriomorphic) ferrite (GF), Widmanstätten ferrite (WF), and acicular ferrite (AF). There are also traces of martensite, retained austenite and degenerate pearlite in the deposited weld microstructure, but due to its limited occurrence, these are referred to as microphases [7].

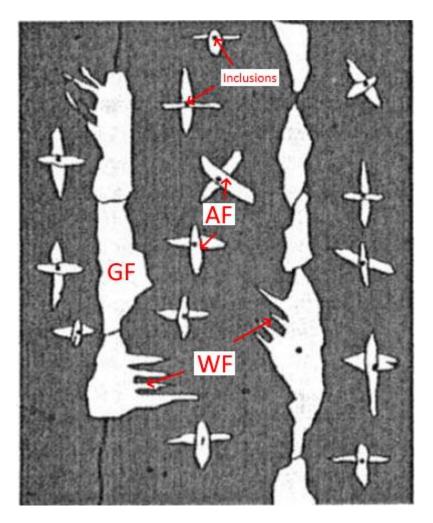


Figure 3-2 – Schematic diagram of the primary microstructure in the columnar austenite grains of a steel weld deposit. The terms *GF*, *WF* and *AF* refers to allotriomorphic ferrite, Widmanstätten ferrite and acicular ferrite, respectively. Modified from Bhadeshia [7].

Hardenability is the ability in a steel to form martensite. The two most significant variables which effect hardenability are austenite grain size, carbon composition and alloying elements. The hardenability of a microstructure increases with growing grain size due to the ratio of grain boundary per unit volume decreases. Locations for ferrite and pearlite nucleation is slowed down which increases the hardenability. The importance of alloying elements is often expressed by its carbon equivalence (CE). An increase in CE promotes the forming of a martensitic grain structure since it retards the austenite to ferrite transformation [7]. A steel is considered weldable if CE < 0.4 [8].

Ito and Besseyo formulated an expression for carbon equivalence at carbon weight < 0.18 % C

$$CE = C + \frac{Si}{30} + \frac{Mn + Cu + Cr}{20} + \frac{Ni}{60} + \frac{Mo}{15} + \frac{V}{10} + 5B$$

This expression is more suitable for steels with a lower carbon content since the carbon sensitivity increases at lower carbon content levels.

3.3 The heat-affected zone (HAZ)

The HAZ is the region adjacent to the weld that has not melted, but instead has undergone a solid-state transformation or been tempered. The microstructure and mechanical properties in these regions have been altered due to various degrees of temperature and has therefore been divided into the various subzones illustrated in Figure 3-1 [7].

3.3.1 Heat flow

The effect of the heat input decreases as a function of distance from the fusion boundary, but the cooling rate on the other hand is not as affected. The formula Δt_{8-5} is the time it takes to cool from 800°*C* to 500°*C*, which is the temperature where many weldable steels experience a solid-state phase transformation from austenite to ferrite [7].

The thermal cycle within any location within the HAZ can be described with the two parameters; peak temperature T_p and the time period Δt_{8-5} . Both are dependent on the heat input q.

$$T_p \propto \frac{q}{r}$$
$$\Delta t_{8-5} \propto q^n$$

Where r is the distance from the fusion boundary and n depends on the nature of the heat dispersion into the material. The dispersion is dictated by the thickness of the weld bead compared to the base material and whether the heat flow is two- or three-dimensional [7].

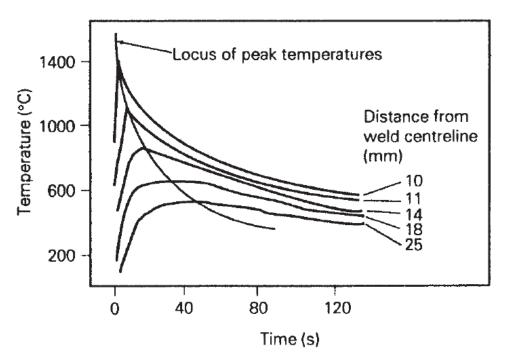


Figure 3-3 – Temperature changes over time depending on the distance from the weld zone [7].

3.3.2 Phase transformations in the HAZ

The phase transformations that occur during the welding process are as Bhadeshia and Honeycomb [7] describes. The subsequent microstructural regions after the fusion boundary has a distinct characteristic dependent on the nature of the peak temperature and duration as seen in Figure 8-5.

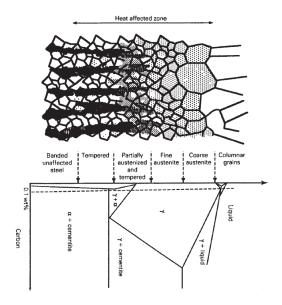


Figure 3-4 – Schematic diagram of the heat-affected zone [7].

- 1. The region adjacent to the fusion boundary is heated to temperatures that has allowed for complete austenitization of the grain structure (CG HAZ). The temperature for complete austenitization during rapid heating is different from a quasistatic heating rate. In order to achieve full austenitization the temperatures need go above the $A_{c3} = 950 \,^{\circ}C$, and begins to form at $A_{c1} = 800 \,^{\circ}C$. The peak temperatures in the fusion boundary is well above this temperature that results in a very coarse-grained austenitic microstructure. The forming temperature which is usually defined above 1100 °C.
- 2. The grain-size decreases rapidly when moving away from the fusion boundary resulting in a fine-grained microstructure (FG HAZ). The mechanical properties in this zone tends to be higher than in the coarse-grained zone. The temperature here is defined as being between 1100 °C and Ac3.
- 3. Further away from the fusion boundary the temperatures drop to the region in between the A_{c1} and A_{c3} curve called the intercritical heat-affected zone (IC HAZ). The microstructure in this region experiences a partially austenitizing process with a rather high carbon content. The grains that do not transform into austenite experiences a tempering effect instead.
- 4. At temperature below A_{c1} the microstructure only experiences tempering. This zone is called the subcritical heat-affected zone (SC HAZ).

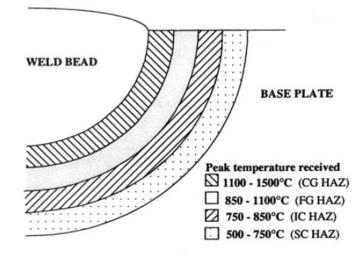


Figure 3-5 - Schematic diagram of a single pass weld [9].

3.3.3 Multipass welds

Each weld pass heat-treats the underlying microstructure in a multirun weld. This results in a very complicated microstructure where the layers can be fully austenitised and transformed into a different microstructure. Areas further away might only experience a tempering process. These microstructures are called secondary or reheated microstructures [9].

Extra zones are identified regarding the CG-HAZ as seen in Figure 3-6.

- Unaltered CG HAZ (U CG HAZ) is the region where the CG HAZ has been reheated to a region above 1100 °C.
- Supercritical reheated CG HAZ (SC CG HAZ) that has been reheated to a temperature between Ac3 and 1100 °C.
- Intercritical reheated CG-HAZ (IC CG HAZ) is the zone that has been reheated to a temperature between Ac1 and Ac3.
- Subcritical reheated CG-HAZ (S CG HAZ) defined as the zone heated to a temperature below Ac1.

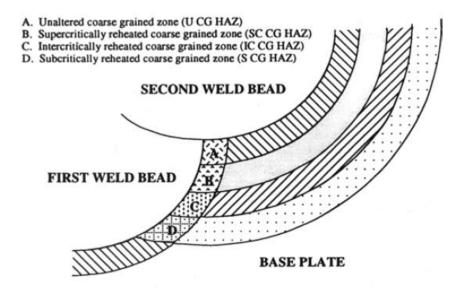


Figure 3-6 - Schematic diagram of a multipass weld [9].

3.3.4 Intercritical reheated coarse grained heat-affected zone (IC CG HAZ)

A reverted austenitization occurs in the CG HAZ when it is reheated to the temperatures between Ac1 and Ac3. The austenite grows from the former austenite grain boundaries and the composition of this phase has a high carbon content due to the high solubility of carbon in austenite. Under these conditions the reverted austenite can transform back to either retained austenite or martensite. This localized transformation requires a sufficiently high carbon content and high cooling rate in order to occur. These localized areas are hard and brittle due to the inherent properties of martensite [9].

4 Fatigue Theory

4.1 General fatigue theory

Fatigue is the progressive, localized and permanent structural change that occurs in a material after a certain number of loading cycles have been reached. Unless cyclic stress, tensile stress or plastic stress are acting simultaneously in the material neither crack initiation nor crack propagation will take place.

The three stages of fatigue are [10]:

- Initial fatigue damage leading to crack initiation.
- Progressive cyclical growth of a crack (crack propagation) until the remaining uncracked cross section of a part becomes too weak to sustain the loads imposed.
- Final, sudden fracture of the remaining cross section.

The total number of cycles required for a material to fail is called its fatigue life. The term failure here is different from fracture and damage. A part can be damaged but still usable. Failure on the other hand is when a part is damaged to such a degree that it isn't viable for service. Fracture is when the failed part physically splits in two. The fatigue life depends on several variables such as stress level, stress state, cyclical wave form, fatigue environment and the metallurgical condition of the material [10].

Laboratory fatigue tests can be divided into crack initiation and crack propagation. Crack initiation testing is the study of the time it takes for a crack to initiate and result in a failure. In crack propagation testing an initial crack is introduced in the surface of the material in order to study the growth rate [10].

The crack initiation phase is defined by the creation of a crack or by the sharpening of a rounded imperfection into a crack. This process can account for over half of the life of a fatigue specimen with a smooth surface, but is generally not relevant for welded structures because welds contain flaws that are sufficiently sharp to eliminate the entire crack initiation phase [11].

There are two common stress ratios in a stress cycle

$$R = \frac{Minimum \ stress}{Maximum \ stress} = \frac{S_{min}}{S_{max}}$$
(4-1)

and

$$A = \frac{Alternating \ stress \ amplitude}{Mean \ stress} = \frac{S_a}{S_m}$$
(4-2)

The variable S_{min} and S_{max} is the minimum and maximum stress respectively. In the second stress ratio formula the variable S_m is the mean stress and S_a is the alternating stress amplitude. The mean stress is the average of maximum and minimum stress

$$S_m = \frac{S_{max} + S_{min}}{2} \tag{4-3}$$

And the formula for the alternating stress amplitude

$$S_a = \frac{S_r}{2} = \frac{S_{max} - S_{min}}{2}$$
(4-4)

 S_r is the stress range and S_a is the stress range amplitude. When running a constant amplitude fatigue test a normal cyclical range is $S_m \pm S_a$.

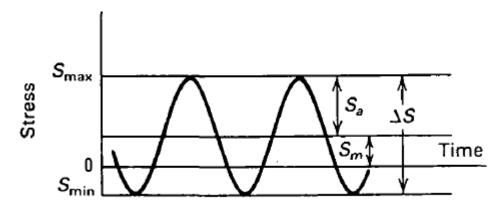


Figure 4-1 – Constant amplitude stress cycle [10].

Table 4-1 - Stress ratios

$\frac{\text{Stress ratios}}{R = S_{min}/S_{max}}$	
R = 0	The stress range is between maximum stress and 0 stress.
0 < R < 1	Cycled between two tensile or two compressive stresses.
R = -1	The stresses are fully reversed. Mean $= 0$.
-1 < R < 0	Partially reversed.
R = 1	Fully tensile.

4.2 Fatigue test regimens

Two common methods that are used during fatigue testing are high- and low-cycle. The low-cycle fatigue testing regime is characterized by high overstress in the plastic stress range, while the high-cycle fatigue test is usually characterized by being in the elastic stress range. The dividing line is approximated to be around about $10^4 - 10^5$ cycles [10].

4.3 Fatigue mechanisms in solids

Fatigue cracks initiate at locations with maximum local stress and minimum local strength and has been observed to nucleate due to a variety of crystallographic features. During a certain number of loading cycles, dislocations as seen in Figure 4-2, pile up and form persistent slip bands (PSB) [10].

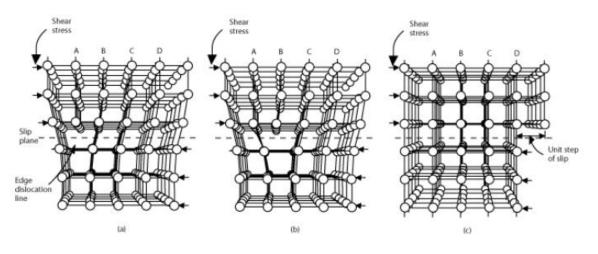


Figure 4-2 – Dislocations [12].

These PSBs, as seen in Figure 4-3, serves as the nucleation point for fatigue cracks. Crack initiations always nucleate in zones with high plastic deformation concentration where the main contributing factor is the surface roughness.

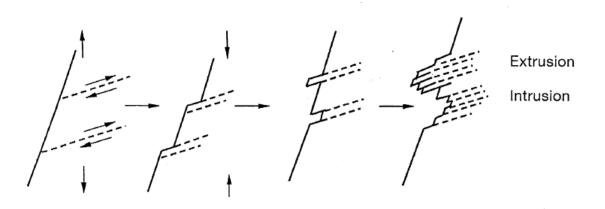


Figure 4-3 – Persistent Slip Bands (PSB) [13].

Cracks initiate at the surface except for instances where internal defects, or particular surface hardening treatments are applied [10].

4.4 Fatigue failure in welded joints

During design of a component or structure the avoidance of fatigue failure is a factor which limits the design stresses. This is especially true in weldments due to the reduced fatigue strength compared to unwelded components. Figure 4-4 shows a comparison of the fatigue strengths between unwelded and welded components. While the endurance life of a component may be around 0.5xUTS, a welded component may be around 30 MPa in some cases. In addition, fatigue failure can occur during compressive stresses [11].

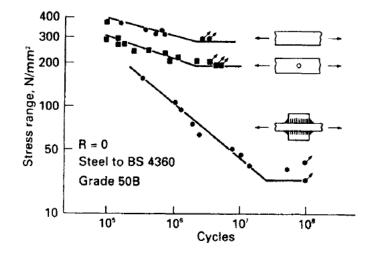


Figure 4-4 – Comparison of fatigue strengthsof plain steel plate, notched plate and with fillet welded attachments [11].

Stress concentrations are introduced in a weldment due to a sudden change in geometry in the weld toe. The severity is dependent on the shape of the weld bead and the type of joint. If the weld joint is smooth, then the stress concentration is lower. The opposite is mostly true for normal weld joints though and should be accounted for in the design. The weld itself does not have to be load carrying for a stress concentration to occur, and just the change in the geometry introduces a stress concentration [11].

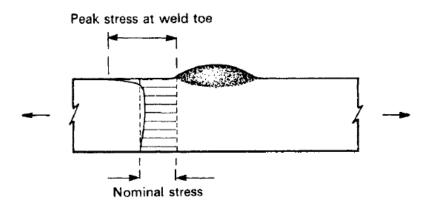
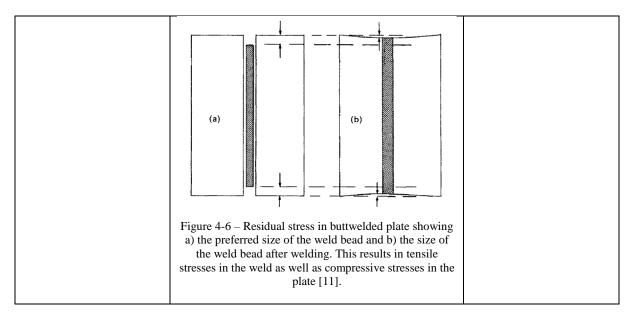


Figure 4-5 - Stress Concentration due to weld geometry [11].

Inherent flaws called weld intrusions increase the stress concentrations drastically. These intrusions are very small crack-like discontinuities and exist in the weld toe with varying depth. In addition, undercutting of the weld toe and residual stresses are also an issue with welding. All these issues are considered when determining the representative SN-curve for the weld joint. It is important though that care is taken to follow the welding procedure in order to avoid other issues in the weld [11].

4.5 Residual stresses

Residual stresses can be induced in a material from various processes such as casting, forging and rolling. These "locked-in" stresses are independent of external loading, balanced out by each other and the system of compressive and tensile forces is thus in equilibrium. In a welded structure there are two types of stresses – reaction stresses and residual stresses. The reaction stresses are the overall distribution of compressive and tensile forces throughout the welding structure assembly and are sometimes called "long range residual stresses". Residual stresses on the other hand is a form of localized stress in the sense that it occurs in each weld joint. The residual stresses arise due to the restraining effect the adjacent material has on the weld as seen in the buttweld of two plates in Figure 4-6. During the weld heating and cooling cycles, the weld metal wishes to contract and expand, but since the surrounding material inhibits this process it results in "locked-in" residual stresses [11].



The longitudinal stresses reaches yield point level are distributed as shown in Figure 4-7 with tensile residual stresses counteracted by compressive stresses further out in the plate where the area A is equal to $B_1 + B_2$. The transverse stresses may vary from the figure but does include high stress areas coincident with the weld zone.

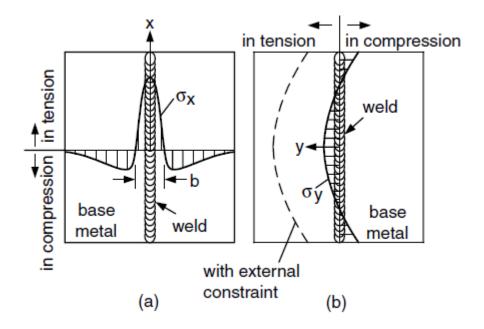


Figure 4-7 - Residual stress distribution in a welded plate [5].

When a plate is loaded above the yield point level due to a high load and a high mean residual stress, the location of highest stress starts to yield. The load is then redistributed to the adjacent material, which in turn starts to yield as well. When the loading is removed, the former residual stress level is reduced. The residual stress in this area has been relaxed. No plastic straining occurs when a compressive loading is applied, but the stress range is equal. This emphasizes the importance of residual stresses of yield point magnitude. Fracture is possible during both compressive and tensile loading conditions, and that the fatigue strength of a welded joint is mainly governed by the stress range and not the nominal stress level. The formation of a crack might redistribute the residual stresses such that it offers stress relief. This could stop the crack grow th in a fully or partially compressive load scenario due to the requirements for a crack to grow is tensile stresses. In the same sense that tensile residual stresses facilitate crack initiation, compressive residual stresses could do the opposite [11].

4.6 Stress relief

According to Maddox [11], the need for stress relief is only necessary for welded joints under partly or fully compressive loading. This is due to the significance of crack propagation and that crack growth is marginally influenced by mean stress level. The stress range must still increase when the mean stress is lower.

4.7 Material properties

One of the most distinct differences between a welded joint and a solid component is that the fatigue strength of a solid component normally increases with increasing tensile strength, whereas a welded joint doesn't. This is due to the significance of the crack initiation stage in the fatigue life of an unwelded component. In a welded component it is assumed that a

preexisting crack already exist, so the fatigue life primarily consists of crack propagation. The crack growth doesn't vary significantly with material tensile strength [11].

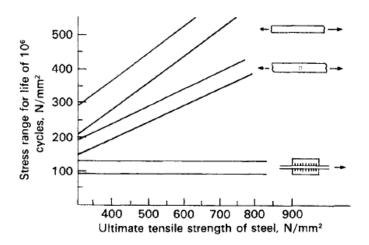


Figure 4-8 - Effect of increasing tensile strength between welded and unwelded components [14].

4.8 Weld quality

Weld flaws in a weld joint, such as porosity, slag inclusions, lack of fusion and incomplete weld root penetration, provide additional stress concentrations. If these are more severe than the weld toe, they will act as alternative sources for crack initiation. In dressed flush butt welds these are the critical crack initiation locations [11].

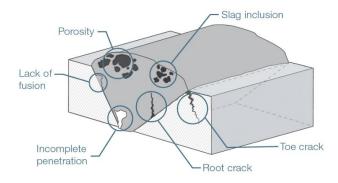


Figure 4-9 - Weld flaws [15].

In addition, weld misalignment increases the local secondary stress concentration by bending, either by axial eccentric misalignment or angular misalignment. The distinction between misalignment and weld flaws is that the former does not introduce an alternative crack initiation site, but rather enhances existing stress concentrations. Due to the bending being higher in the top surface, this is especially true in the weld toe [14].

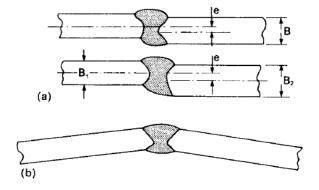


Figure 4-10 - Weld axial and angular misalignment [11].

5 Residual stresses due to shaping operations

Operations such as grinding, cutting, milling, turning, drilling and processes such as shot peening and blasting, involves local plastic deformation. Residual stresses from shaping can be compressive or tensile and can also have additional large shear components [16].

Grinding is an abrasive method used to remove small portions of material on a surface. Machines that are used are grindstones, angle grinder, grinding machines, etc. There are similarities between the methods mentioned in the previous section and grinding. The energy from the cutting operation results in temperature increase and plastic deformation. Compressive stresses in the surface tend to be the results of plastic deformation due to smearing. Tensile stresses tend to appear when the heat from the operation expands the material, which in turn is constrained by the adjacent material. The subsequent cooling of this area results in tensile residual stresses. A third factor is the forming of martensitic microstructure due to rapid cooling, which causes compressive residual stresses. The resulting residual stresses depends on the balance between these three factors as seen in Figure 5-1.

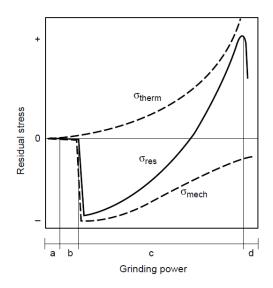
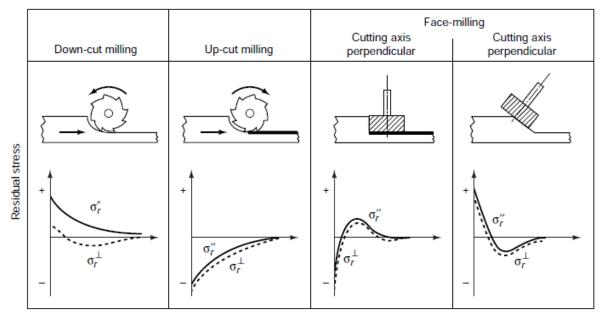


Figure 5-1 - Residual stresses during shaping operation. The higher the grinding power the higher the tensile residual stresses [16].

Important parameters during grinding is the cutting speed, depth of cut and feed rate, the mechanical properties of the workpiece and the heat conductivity of the grinding wheel and workpiece. Normally, the affected depth is relatively small; less than 100 μ m. The level of the residual stresses can reach values below the yield point [16].

Milling removes material by using rotary cutters. The factors that control the resulting residual stresses are the same as for grinding; the smearing effect, temperature and martensitic formation process. The mechanism is divided into chip formation and smearing of the workpiece surface. The chip formation process tends to induce tensile residual stresses, while the smearing process results in compressive stresses. Most of the heat generated during the cutting process is stored in the removed chip. The affected depth is usually thicker than grinding [16].



Distance from surface

Figure 5-2 - Residual stress distribution depending on the cutting process. The longitudinal stress is the solid line while the dashed line is the transverse stress [16].

6 Mechanical testing in accordance with 15614-1:2017

All new welding procedures were to be carried out in accordance with 15614-1:2017 Specification and qualification of welding procedures for metallic materials - Welding procedure test - Part 1: Arc and gas welding of steels and arc welding of nickel and nickel alloys [17]. The main objective of welding qualification is to validate that the joining process intended for construction is capable of producing joints having the necessary mechanical properties for proposed application. The standard consists of two levels; level 1 and 2. Level 1 is based on requirements of ASME IX and level 2 is based on the previous issues of standard 15614-1:2017. The extent of testing is greater in level 2 and the ranges of qualification are more restrictive than in level 1. Procedural tests for level 2 automatically qualify level 1 requirements.

There are two main categories of testing – non-destructive testing and destructive testing. As the names imply, the different methods either destroy or has no lasting effect on the test sample.

Activity Description	Specification/ Procedure	Acceptance Criteria		
Visual testing	ISO 17637:2016	EN ISO 5817:2014		
Radiographic	ISO 17636-2:2013	ISO 10675-1:2016		
Surface Crack Detection Test (Magnetic particle inspection)	NS-EN ISO 17638:2016, NS-EN ISO 3452-1:2013	ISO 23278:2015 Level B		
Macro examination	NS-EN 17639:2013	EN ISO 5817:2014 (quality level B)		
Transverse Tensile Test	ISO 4136:2012, ISO 6892-1:2016 Method A1	NS-EN 15614-1: 2017, NS-EN 10225:2009		
Charpy V Impact Test	NS-EN ISO 148-1:2016, ISO 9016:2012	NS-EN 15614-1, NS-EN 10225:2009		
Side bend test	NS-EN ISO 5173:2010	NS-EN 15614-1:2017		
Vickers hardness test	NS-EN ISO 9015:2011	NS-EN 15614-1:2017		

Table 6-1 - Extent of mechanical testing procedures and correspondence	onding acceptance criterias in order to approve WPS.
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6.1 Visual testing

The procedural steps for visual testing is documented in NS-EN ISO 17637:2016 [18]. The inspection and testing is performed after the initial joint preparation before welding, during the welding operation and on the finished weld. The examiner shall have access to the necessary inspection and product documentation required. Several weld discontinuities can be detected by the examiner in this stage.

The finished weld shall be examined to determine whether it meets the requirements of the application or product standards or other agreed acceptance criterias. The acceptance criterias in this project was NS-EN ISO 5817:2014 [19] and NS-EN ISO 15614-1:2017.

6.2 Radiographic testing

A non-destructive test performed in accordance with Welding - Fusion-welded joints in steel, nickel, titanium and their alloys (beam welding excluded) - Quality levels for imperfections (NS-EN ISO 5817:2014) [19] and acceptance criterias in Non-destructive testing of welds - Acceptance levels for radiographic testing - Part 1: Steel, nickel, titanium and their alloys (NS-EN ISO 10675-1) [20].

An X-ray tube sends out radioactive waves that penetrates the tested specimen and hits a photographic film. Cracks and pores in the weld let more X-rays through, which results in more exposed areas on the film.

6.3 Magnetic particle inspection testing (MT)

A non-destructive test performed in accordance with Non-destructive testing of welds - Magnetic particle testing NS-EN ISO 17638:2016 [21] and acceptance criteria Non-destructive testing of welds - Magnetic particle testing - Acceptance levels (NS-EN ISO 23278:2015) [22].

This method is used to detect surface flaws in magnetic materials. The test specimen is magnetized, and an iron powder is distributed on the surface. Cracks in the metal surface and 2 mm into the surface disturbs the magnetic field and attract the iron powder. The cracks become visible as dark stripes on the surface.

6.4 Penetrant testing (PT)

A non-destructive test performed in accordance with Non-destructive testing - Penetrant testing - Part 1: General principles (NS-EN ISO 3452-1:2013) [23] in accordance with acceptance criterias in Non-destructive testing of welds - Magnetic particle testing - Acceptance levels (NS-EN ISO 23278:2015) [22].

The testing method is used to exclusively detect surface defects, such as pores and cracks, not large enough to be visible in the visual inspection. The method uses a liquid penetrant that is sprayed on the surface in combination with an absorbing component that reveals weld discontinuities.

6.5 Macroscopic and microscopic examination

Macroscopic examination is the analysis of a test specimen by the naked eye, or under low magnification (generally less than x 50), with or without etching. Microscopic examination is the analysis of a test specimen with a magnification of x50 to x500 with or without etching [24].

A destructive test performed in accordance with Destructive tests on welds in metallic materials - Macroscopic and microscopic examination of welds (NS-EN ISO 17639:2003) [24] and acceptance criteria's in Welding - Fusion-welded joints in steel, nickel, titanium and their alloys (beam welding excluded) - Quality levels for imperfections (NS-EN ISO 5817:2014) [19].

An optical microscope can be used to assess the weld and base material microstructure (including grain structure, morphology and orientation, precipitates and inclusions), with several thousand times magnification. The test piece is initially prepared by being cut into small pieces, and then grinded and treated with acid in order to reveal grain structure, precipitations and microscopic flaws. The difference between macroscopic and microscopic examination is the level of magnification. A macroscopic examination studies the fusion zone, the heat-affected zone and the weld passes in a low magnification. In a microscopic examination the magnification is increased such that it's possible to study the grain structure itself in both the weld and HAZ [25].

6.6 Tensile test

A destructive test performed in accordance with Destructive tests on welds in metallic materials - Transverse tensile test (NS-EN ISO 4136:2012) [26] and acceptance criterias in Metallic materials - Tensile testing - Part 1: Method of test at room temperature (NS-EN ISO 6892-1:2016) [27].

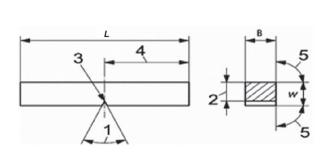
The tensile test demonstrates the static strength of a specimen. The usual characteristic data that is obtained from the test is the yield point, ultimate tensile strength and total elongation. The test result is presented in a stress-strain diagram.

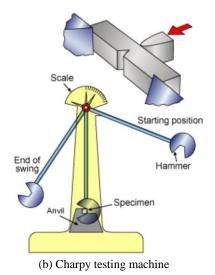
6.7 Charpy V Impact Toughness test

A destructive test performed in accordance with Metallic materials - Charpy pendulum impact test - Part 1: Test method (NS-EN ISO 148-1:2016) [28] and Destructive tests on welds in metallic materials - Impact tests - Test specimen location, notch orientation and examination (NS-EN ISO 9016:2012) [29]. Acceptance criteria's are presented in Specification and qualification of welding procedures for metallic materials Welding procedure test Part 1: Arc and gas welding of steels and arc welding of nickel and nickel alloys (NS-EN ISO 15614-1) [17] and Weldable structural steels for fixed offshore structures - Technical delivery conditions - Part 1: Plates (NS-EN 10225-1) [30].

A Charpy impact test is a standardized high strain-rate test to measure a materials toughness, which is the ability to absorb energy from an impact before fracturing. The test specimen is designed according to ISO 148-1:2016 and ISO 9016:2012. There are different designs, but the most conservative one is where a V-notch is introduced in the surface. A standard V-notch geometry shall have an included angle of 45°, a depth of 2 mm and a root radius of 0.25 mm, see Figure 6-1(c) [28].

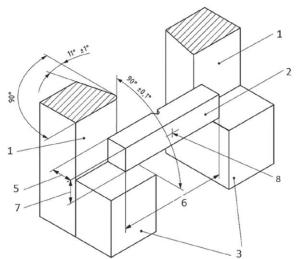
The Charpy test is about hitting a notched test piece with a swinging pendulum. The notched test piece is placed in a test bit support, Figure 6-1 (c). The notch should be placed in the middle between two anvils, see Figure 6-1 (b), so that the center of strike is positioned on the opposite side of the notch. From the test you can find out how much energy was absorbed in the impact and the appearance of the shear fracture.





(a) Charpy pendulum impact test piece, V-notch geometry

Thickness, B = 10mmWidth, W = 10mmLength, L = 55mmAngle of notch, $1 = 45^{\circ}$ Ligament, 2 = 8mmNotch radius, 3 = 0.25mm



(c) Test piece terminology showing configuration of test piece support and anvils of a pendulum impact testing machine 1. Anvil

- 2. Standardized test piece
- 3. Test piece support
- 4. Shroud
- 5. Width of test piece, W
- 6. Length of test piece, L
- 7. Thickness of test piece, B
- 8. Centre of strike
- 9. Direction of pendulum swing

Figure 6-1 - Charpy test

As seen in Figure 6-2 the energy absorption is very temperature dependent. A decrease in temperature negatively affects the materials ability to absorb an impact and decreases the amount of energy needed to fracture the specimen. The ductile to brittle transition phase is in a relatively small window called the Impact Transition Temperature. The energy absorbed around the Impact Transition Temperature for steel is normally around 27 J (40 J for high strength

steel). The thickness of the specimen also affects the toughness of the material whereas a thinner specimen has a higher toughness [25].

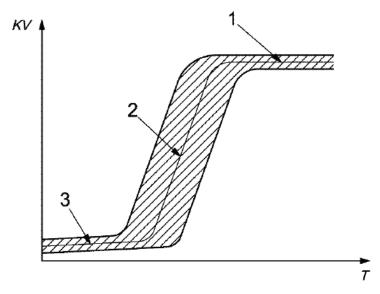


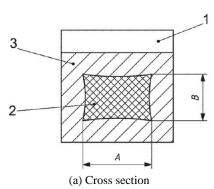
Figure 6-2 - Absorbed energy/temperature curve. The Impact Transition Temperature is a relatively narrow window [28].

Three Charpy specimens are tested at a certain design temperature and at the same location; WM, FL,FL+2 mm and FL+5 mm. The average of these values need to meet a set minimum value that is dependent on the strength value of the material and is typically 27 J or 40 J of absorbed energy.

Fracture appearance

The Charpy tests fracture area is mostly evaluated by the shear fractures that occur. The greater the amount of shear fracture, the greater the toughness. The fracture surface of a Charpy specimen has a mixture of shear and planar fracture areas. The shear areas are concidered to be fully ductile, but the planar fracture area tends to be more ductile, brittle in a combination of fracture modes. Since the classification is very subjective, it is advised not to be used it in specifications.

According to NS-EN ISO 148-1: 2016 [28], there are several methods for assessing the proportion of shear fractures. One of the methods compare the fracture with an appearance chart. Another method measures the length and width of the cleavage section, see Figure 6-3.



		Α																	
В		mm																	
mm	1,0	1,5	2,0	2,5	3,0	3,5	4,0	4,5	5,0	5,5	6,0	6,5	7,0	7,5	8,0	8,5	9,0	9,5	10
								ł	Per c	ent s	hear	•							
1,0	99	98	98	97	96	96	95	94	94	93	92	92	91	91	90	89	89	88	88
1,5	98	97	96	95	94	93	92	92	91	90	89	88	87	86	85	84	83	82	81
2,0	98	96	95	94	92	91	90	89	88	86	85	84	82	81	80	79	77	76	75
2,5	97	95	94	92	91	89	88	86	84	83	81	80	78	77	75	73	72	70	69
3,0	96	94	92	91	89	87	85	83	81	79	77	76	74	72	70	68	66	64	62
3,5	96	93	91	89	87	85	82	80	78	76	74	72	69	67	65	63	61	58	56
4,0	95	92	90	88	85	82	80	77	75	72	70	67	65	62	60	57	55	52	50
4,5	94	92	89	86	83	80	77	75	72	69	66	63	61	58	55	52	49	46	44
5,0	94	91	88	85	81	78	75	72	69	66	62	59	56	53	50	47	44	41	37
5,5	93	90	86	83	79	76	72	69	66	62	59	55	52	48	45	42	38	35	31
6,0	92	89	85	81	77	74	70	66	62	59	55	51	47	44	40	36	33	29	25
6,5	92	88	84	80	76	72	67	63	59	55	51	47	43	39	35	31	27	23	19
7,0	91	87	82	78	74	69	65	61	56	52	47	43	39	34	30	26	21	17	12
7,5	91	86	81	77	72	67	62	58	53	48	44	39	34	30	25	20	16	11	6
8,0	90	85	80	75	70	65	60	55	50	45	40	35	30	25	20	15	10	5	0
NOTE 1	TE 100 % shear shall be reported when A and B are zero.																		

1. notch

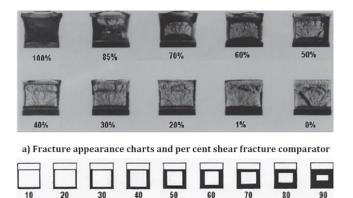
2. cleavage area (brittle)

3. shear area (dull) A. dimension measured to estimate the

cleavage area

B. dimension measured to estimate the cleavage area

(a) Percent shear for measurement in mm



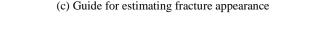


Figure 6-3 – Methods for determining percentage of shear fracture

6.8 Bending test

The bending test is a destructive test performed in accordance with Destructive tests on welds in metallic materials - Bend tests (NS-EN ISO 5173:2010) [31] and acceptance criterias in Specification and qualification of welding procedures for metallic materials Welding procedure test Part 1: Arc and gas welding of steels and arc welding of nickel and nickel alloys (NS-EN ISO 15614-1:2017) [17].

Bending tests are conducted in order to assess the ductility and the presence of imperfections in the weld. ISO 5173:2010 specifies various methods of testing the welded buttjoint by performing transverse root, face and side bend test. A standard test is transverse across the weld, but in cases where there is a significant difference in the filler and base material's physical and mechanical properties a longitudinal test may be performed.

6.9 Vickers hardness test

Hardness test performed in accordance with the procedure in Destructive tests on welds in metallic materials Hardness testing Part 1: Hardness test on arc welded joints (NS-EN ISO 9015-1:2011) [32] and acceptance criterias in Specification and qualification of welding procedures for metallic materials Welding procedure test Part 1: Arc and gas welding of steels and arc welding of nickel and nickel alloys (NS-EN ISO 15614-1:2017) [17].

A materials ability to resist deformation from a harder object is defined as its hardness. The size of the resulting imprint is larger the softer the material is. There are three main hardness testing methods, Brinell, Rockwell and Vickers. Vickers is the method used in the majority of cases.

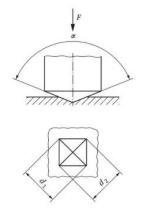


Figure 6-4 - The pyramide diamond indenter of a Vickers hardness tester [33].

The Vickers hardness test is performed by applying a diamond cone tip into the surface of the material. After the load has been applied the diagonals of the imprint is measured and the hardness is calculated with the formulas:

$$Vickers hardness = \frac{Test force (kgf)}{Surface area of indentation (mm^2)}$$
$$= \frac{1}{g_n} * \frac{Test force (N)}{Surface area of indentation (mm^2)}$$
$$= \frac{1}{g_n} * \frac{F * 2 * \sin(\alpha/2)}{d^2}$$
$$= \frac{1}{g_n} * \frac{2 * F * \sin(\alpha/2)}{d^2}$$

Where α is the angle in 136°, d is the arithmetic mean of the two diagonal lengths and g_n is the gravitational constant in 9.80665 m/s² which results in

Vickers hardness $\approx 0.1891 * F/d^2$

Part II: Experimental Investigation

7 Manufacturing and welding process, characterization of the Steel S420G2+M (Y30 Rev.5)

The following experimental chapter describes the process of developing a WPQ for S420G2+M steel plates and subsequent mechanical testing. It aims to provide information on the manufacturing process and the mechanical testing that has been carried out.

The test data obtained from the mechanical testing in this chapter is necessary in order to form a conclusion in the following chapters:

- Experimental Investigation: Microstructure
- Experimental Investigation: Fatigue
- Experimental Investigation: Residual Stress Measurements

This chapter is divided into five sub-chapters that describe the different steps that have been completed during the project. All sub-chapters have a corresponding appendix. This chapter and the following chapters in experimental work should be read together with the document, Inspection & Test Plan (I&TP). The I&TP is a quality management document that describes all activities, procedures, acceptance criteria in the project.

To see a summary of all documents that have been produced during the project, see the document list in the beginning in the appendices.

7.1 Introduction: Strategy and objectives

Initially, an experimental investigation was performed in order to test and analyze parallel welds at various distances. The choice of test setup and material was two parallel buttwelds in a S420G2+M (Y30 Rev.5) offshore steel plate.

A common minimum distance between two adjacent welds are 50 mm or 4x the material thickness. This minimum distance is set for them to be regarded as independent and to ensure that the two welds do not adversely affect each other. The interests of the project were therefore to see if the material properties were affected when the welds were closer than this minimum distance. Three sets of two steel plates with parallel buttwelds with different distances were manufactured and assessed -50 mm, 15 mm and 5 mm.

In order to maintain conformity of all welded plates a welding procedure qualification record (WPQR) was initially established. This was achieved by welding an initial qualifying plate that was tested and qualified in accordance with the requirements in NS-EN ISO 15614-1:2017 [17]. Following welding procedures were then subsequently based of the approved welding procedure qualification (WPQ).

The manufactured plates were subjected to both destructive and non-destructive testing; visual testing, radiographic, hardness testing, Charpy V Impact testing, tensile testing, fatigue testing, ultrasonic measurement, macro and microscopic examination.

7.2 Experimental program

The experimental program was conducted in five stages. Every stage is presented in a dedicated sub-chapter as seen below:

Stage A - Prior to fabrication
Stage B - Material Selection and Cutting Process
Stage C - Welding Procedure Qualification Program
Stage D - Production welding of plates
Stage E – NDT, Mechanical Testing and Specimen Preparation

7.3 Stage A - Prior to fabrication

A test plate was welded prior to the manufacturing of the production plates. It served the purpose to act as a concept plate, before the actual test began. By doing this it was possible to gather some experience before the actual welding was started.

7.3.1 Objective

- Weld and test an initial "prior to fabrication plate" before the main "production plates" were welded.
- Develop an "Inspection and Test Plan" based the results and conclusions from this stage.

7.3.2 Fabrication and welding process of Test Plate

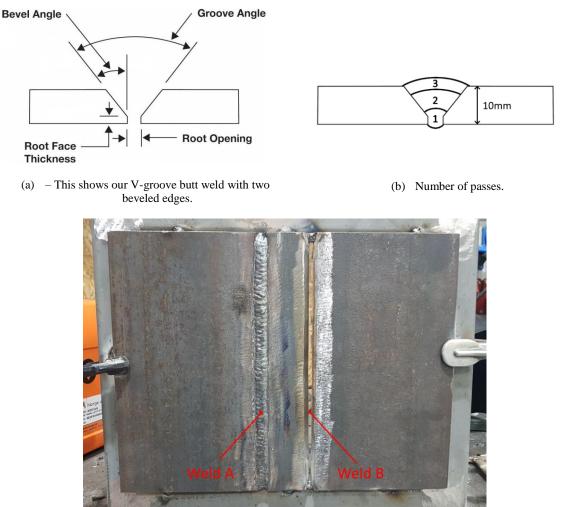
The main objective with the initial welding that was performed at Vest Norge Doors AS in Sandnes, was to determine if the weld joint chosen for the experiment was functional. This included factors such as welding method, if machining was possible on a deformed plate due to weld distortion and if it was possible to conduct mechanical testing on the finished plate. Weld logging was performed due to the necessary equipment to measure the weld parameters was not available.

7.3.2.1 Welding process

First, three plates were cut from a Flat bar P-150X10. An angle grinder was used to bevel angles of 30 degree. These plates were joined together and welded to a frame, see Table 7-1. The joints had a groove angle of 60° with a root face thickness of 0.5 mm and between the plates was a root opening of 2.5 mm. Welding was performed in PF position (rising) and without a welding procedure specification. The full penetration weld was made by using the shield metal arc welding (SMAW) technique and a consumable electrode covered in flux, type SPEZIAL. All welding was done indoors, and the welding machine used was a KEMPPI - Minarc Evo 180.

Welding process	111 - SMAW (Shielded Metal Arc Welding)
Base material	S355J2+M
Filler metal	Consumable electrode covered in flux. Type SPEZIAL -
	2.50X0350XVPMD.
Welding Current (A)	70-80
Interpass Temperature (° C)	≤ 250
Number of Passes	3

Table 7-1 -	Welding	procedure	characteristics.
-------------	---------	-----------	------------------



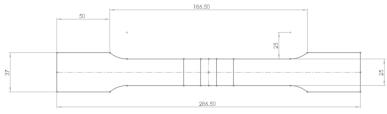
(c) - The distance between weld toes is 50 mm. The left weld is called A and is performed first. Weld number two is seen as a repair weld and is called B.

Figure 7-1 – V- groove butt weld for Prior To Fabrication Test Plate).

After welding, the plate was transported to the University in Stavanger (UIS) for preparation. The plates were cut into 6 pieces and the sides of the specimen was machined, see Figure 7-2. When the specimens were finished, they got an ID.



(a) The test plate was cut into 6 parts and each specimen received a ID. The plate thickness was 10mm. What is missing in the picture is the letter F (fatigue) on specimen F1-F5 and the letter T (Tensile) on specimen T6.



(b) Test specimen drawing.



(c) The sides of the specimen are milled out.

Figure 7-2 – Test Specimens for the "Prior To Fabrication Test Plate".

7.3.3 Material properties of the S355J2+N and filler metal

The base material used for fabrication of "Prior To Fabrication Test Plate" was steel grade S355J2+M. This is a commonly used high strength structural steel and in critical components or major structural members. S355J2+M is an unalloyed fully killed structural steel with good weldability and machinability [34]. The yield point of the steel in room temperature is 355 MPa, with a minimum impact energy value of 27 J at -20 °C (J2) and thermomechanically rolled (M) delivery condition. The final rolling process in thermomechanically rolled steel is performed in a certain temperature range. This results in material properties that can't be achieved by normal cooling conditions [34].

The filler metal that was chosen for welding was an electrode from OERLIKON by the name SPEZIAL. SPEZIAL is a basic, double coated multi-purpose MMA electrode meant for welding of S235 to S355.

7.3.3.1 Identification of base material and filler metal

Identification of base material and filler metal are given in Table 7-2 and Table 7-3.

Identification of base material S355J2+M							
Heat No. and Plate No.	Туре	Name/Grade	Standard	Group	Delivery condition		
43831-9133182	Flat bar P- 150X10	S355J2+M	NS-EN 10025-2:2004	1.2	М		

Table 7-3 - Identification of filler metal, SPEZIAL.

Identification of filler metal	
Brand name	Specification/Classification
SPEZIAL Electrodes 2.50X0350XVPMD	EN ISO 2560-A: E 38 3 B 1 2 H10
	AWS A5.1: E 7016-H8

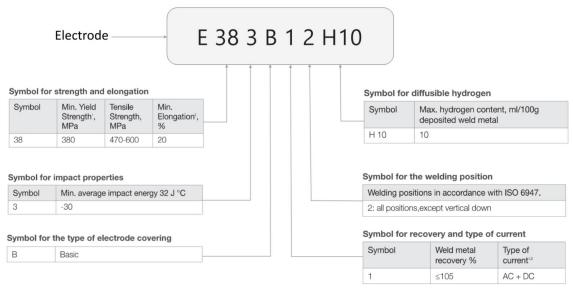


Figure 7-3 – Specification NS-EN ISO 2560:2009 - Welding consumables - Covered electrodes for manual metal arc welding of non-alloy and fine grain steel.

7.3.3.2 Chemical composition

The chemical composition of the base material is presented in Table 7-4. It shows the maximum allowed composition in accordance with NS-EN 10025-2:2004 [34], and the chemical composition from the ladle analysis.

The chemical composition for the filler metal, SPEZIAL, is given in Table 7-5.

The MDS for base material and filler metal is included in Appendix A.

Elements	Requirements	Ladle analysis
	EN 10025:2004:2005 [%]	[%]
Carbon C	≤ 0.20	0.15
Silicon Si	≤0.55	0.190
Manganese Mn	≤1.60	1.29
Phosphors P	0.025	0.026
Sulfur S	≤0.025	0.012
Nitrogen N	No spec.	0.005
Aluminum Al	No spec.	0.003
Copper Cu	≤0.55	0.04
Chromium Cr	No spec.	0.07
Nickel Ni	No spec.	0.02
Vanadium V	No spec.	0.05
Titanium Ti	No spec.	< 0.001
Niobium Nb	No spec.	< 0.001
Boron B	No spec.	0.0004
Molybdenum Mo	No spec.	0.01
Arsenic As	No spec.	0.002
Tin Sn	No spec.	0.002
Lead Pb	No spec.	-
Calcium Ca	No spec.	0.0010

Table 7-4 – Chemical content of base material, Steel S355J2+M.

Antimony Sb	No spec.	-		
Bismuth Bi	No spec.	-		
EV2 (CEV)	0.45	0.39		
EV2: CEV=C+Mn/6+Mo/5+Ni/15+Cr/5+V/5+Cu/15				

Table 7-5 - Filler metal composition, SPEZIAL.

Elements	Requirements MDS – OERLIKON [%]
Carbon C	0.06
Silicon Si	0.7
Manganese Mn	0.9
Phosphors P	≤0.020
Sulfur S	≤0.015

The composition of steel provides important information in order to determine the transformation behavior. Carbon, manganese and nickel below 1 %, silicon below 1.1/2 % and copper move the transformation curve to the right, but do not change the slope.

Chromium, molybdenum, vanadium and other strong carbide forming elements move the curve to the right as well, but they do not change the shape of the curve.

According to NS-EN 10025:2005, the carbon equivalence value, CEV, in the ladle analysis for S355J2 in % with a plate thickness \leq 30 mm should be less than 0.45.

One of the easiest ways to improve the strength of a steel, is to increase the carbon content. The weldability of the steel limits the amount of carbon to 0.25 % [25]. Based on the composition we can see that S355J2+M is a low carbon steel since the carbon content is lower than 0.30 %. The risk of forming brittle martensite increases with higher carbon content. The higher the carbon content, the more brittle and harder the formed martensite will become during rapid cooling or quenching.

Steel that contains between 0.9-1.7 % manganese is called carbon-manganese steels (C-MN Steel). It can be desirable to increase the manganese content due to increased depth of hardening, improved strength and toughness. Manganese also assists during the production process, when it bonds with impurities such as oxygen and sulfur [25].

7.3.3.3 Mechanical properties

The mechanical properties and the requirements for S355J2+M and filler metal that's presented below, is from NS-EN 10025:2-2005, EN ISO 2560:2009 and MDS S355J2+M [34, 35].

In order to make sure that the choice of filler metal was correct and the weld was quality sufficient, a tensile test had to be performed on specimen T6, Figure 7-4. Fracture occurred in

the base material. The result is shown Table 7-6. The five specimens left, F1-F5, was later used for fatigue testing.

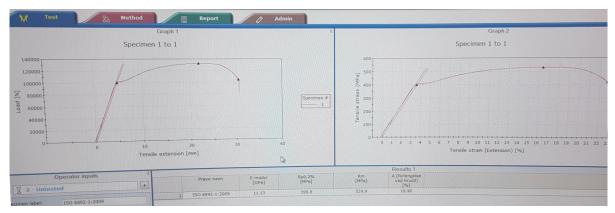


Figure 7-4 - Tensile test of specimen T6. The test results are presented below in the "Tensile test" table.

Tensile test Base material - S355J2+M							
Code / certificate /	Specimen	Heat	Yield stress,	Tensile strength,	R _{eH} /	Elongation,	
test	No.	No.	R_{eH}	$R_m [N/mm^2]$	R _m	Ā	
			$[N/mm^2]$			[%]	
Requirements in NS-			>355	470-630		>22	
EN 10025:2:2004 1)							
MDS - S355J2+M 1)		86336Y	414	541	0.77	30.8	
Test results 1) 2)	T6	86336Y	ca. 410	529.9	0.77	-	
 Test results 1) 2) To 80330 Y ca. 410 529.9 0.77 - 1) Tensile test acc. to EN ISO 6892-1:2009. 2) The test was for the welded "Prior to fabrication test plate". Fracture was in the base material. 							

Table 7-6 - Tensile test data for base material, S355J2+M.

Table 7-7 - Tensile test data for MMA electrodes, SPEZIAL

Tensile test						
Filler metal – OERLIKON SPEZIAL						
Code / Certificate	Yield stress, R _{eH}	Tensile strength, R _m	Elongation			
	$[N/mm^2]$	$[N/mm^2]$	[%]			
Requirements in EN ISO 2560-	Min. 380	470-600	≥ 20			
A: E 38 3 B 1 2 H10						
Requirements acc. to MDS-	Min 380	470-600	≥25			
OERLIKON SPEZIAL (as						
welded)						

Table 7-8 - Charpy test data for base material (S355J2+M)

Charpy Impact test				
Base material - S355J2+M				
Code / Certificate	Heat No.	Test temp	Single values [J]	

		(C °)	1	2	3	Avarage [J]
Requirements in NS-EN 10025:2:2004 1)		-20	Min. 27	Min. 27	Min. 27	
Inspection certificate 1)	86336Y	-20	118	132	104	118
1) Tensile test acc. to EN ISO 148-2010						

 Table 7-9 - Charpy test data for MMA electrodes SPEZIAL

Charpy Impact test Filler metal – OERLIKON SPEZIAL						
Code / Certificate	Test temp	Sir	ngle values [J]	Avarage	
	(C °)	1	2	3	[J]	
Requirements acc. to EN ISO	-30	32	32	32		
2560-A: E 38 3 B 1 2 H10						
Requirements acc. to MDS-	+20	≥150	≥150	≥150		
OERLIKON SPEZIAL (as	-30	≥60	≥60	≥60		
welded)						
1) Tensile test acc. to EN ISO 148-2010						

7.3.4 Inspection and test plan

After the "Prior To Fabrication Test Plate" was completed and tested, an I&TP was developed. The I&TP was implemented in order to keep track of all the steps in the project. All activities in the project was carried out according to the "I&TP". An I&TP is a tool used for planning a project and maintaining control of responsibilities and relevant documents.

The I&TP is divided into the various steps of the project. The document should be read together with "Experimental work" as it describes in detail all the activities that have been carried out.

The "I&TP" can be found in Appendix A.

7.3.5 Appendix A

In step A-Prior to fabrication the following documents have been created. See Appendix A.

- I&TP
- MDS Base material S355 J2+M
- MDS Filler metal OERLIKON SPEZIAL

7.3.6 Summary

• The initial test of the prior to fabrication showed that the test setup was viable for further experimentation with some modifications:

- This test plate was welded prior to the manufacturing of the productions plates. It served the purpose to act as a concept plate and by doing this it was possible to gather some experience before the start of the production welding.
- The cutting operation was performed using a metal cutting band saw and an angle grinder. The operation was done in two stages, the plate was first cut then subsequently grinded with a flap disk. After this experience we choose to find another method, due to that the operation was very time consuming, unprecise and unrepeatable.

Oxy-fuel cutting was a viable method, but the microstructural implications from the high temperature that is induced in the material might also apply unwanted effects. A more favorable alternative was waterjet cutting. This method doesn't introduce significant heat into the material since it uses an abrasive high-pressure cutting process. It is also very efficient when cutting several plates in a series.

- We discussed the choice of steel for the base material and decided to choose S420G2+M. This is a steel that is often used for critical constructions and therefore very interesting for the offshore industry. The steel is more complex than S355J2+M as it is higher alloyed, which makes it more challenging to weld as the risk for cracks increases. We also wanted a steel that is more likely to be sensitive to weld-on-weld or weld proximity interference.
- We decided to use the FCAW welding method due to it being more productive and more common in offshore welding operations as it is cost effective. This method also makes it possible to use a thicker material.
- As we changed the base material and welding method, we also needed to change to a different filler metal. The choice was NSSW SF-3AM as it is commonly used offshore, has preferable properties and works well with the chosen base material.
- Important to note is that post weld heat treatment can lower the strength values of a thermomechanically rolled steel [36].
- The weld joint in this chapter was visually tested and subjected to tensile testing. More extensive testing will be performed on future plates in order to develop a WPQ.

7.4 Stage B – Material Selection and Cutting Process

The manufacturing and delivery conditions of the steel and filler metal used in the project will be discussed in this sub-chapter. It also includes the process of casting the designated steel and the subsequent waterjet cutting operation.

7.4.1 Objective

• Order materials and prepare for welding.

7.4.2 Introduction (Stage B)

The choice of steel was chosen based on its commonality in the offshore industry and its potential sensitivity to the following welding operations. S420G2+M (NORSOK Y30 Rev. 5) is one of the most commonly sold steels from Norsk Stål AS. It is included in the European Standard NS-EN 10225:2009 [30, 37] and is a high strength plate used for offshore platforms, wind-power installations, pressure vessels and FPSO:s where high strength and corrosion resistance is a requirement. NS-EN 10225:2009 is applicable to steels for offshore structures, designed to operate in the offshore sector, but not to steels supplied for the fabrication of subsea pipelines, risers, process equipment, process piping and other utilities. It is primarily applicable to the North Sea Sector, but may also be applicable in other areas provided that due consideration is given to local condition e.g. temperature [36].

The dimension of the plates was 15x300x500 mm which were waterjet cut from a heavy plate with dimensions of 15x2500x12000 mm at Smed T. Kristiansen in Randaberg. The waterjet cutting method does not introduce any adverse heating that could affect the base material due to it being a cold working process. This is wanted for the future analysis of the microstructure. The intention to analyze the changes the microstructure experiences from the welding operation heat source.

Welding method 13, gas-shielded metal arc welding was chosen as the welding process. The filler metal was one of Norsk Sveiseteknikk AS main products from Nippon Steel & Sumikin Welding, NSSW SM-47A and NSSW SF-3AM. This filler metal is common both offshore and in ship construction.

7.4.3 Material properties of the S420G2+M and filler material

The base material used for welding of qualification and production plates was the steel type S420G2+M (MDS-Y30 Rev. 5). It is classified according to NS-EN 10225:2009 and EN 10020:2000 [38] as an special steel alloy in group 3 and steel number 1.8857+M. The construction steel has a specified minimum yield point in room temperature at 420 MPa, and a minimum average impact energy value of 60 J at -40 °C. The steel is of grade 2 and delivery condition thermomechanically rolled (M) [36].

Method 13 was chosen for welding, gas-shielded metal arc welding and the filler metal was the following:

- For root passes method 138 was used. MAG welding with metal cored electrode (Gas Metal Arc Welding using active gas and metal cored electrode). The filler metal used was NST:s NSSW SM-47A. A metal cored wire for low temperature pipe and steel applications down to -60 °C.
- For the hot, fill and cap passes, welding method 136 was used. This is MAG welding with flux cored electrode (gas metal arc welding using active gas and flux cored electrode). Filler metal used was NST:s SF-3AM, flux cored wire for low-alloyed steel, offshore applications, piping, etc.

7.4.3.1 Identification of base material and filler metal

Identification of base material and filler metal are presented in Table 7-10 and Table 7-11.

Heat No. and Plate No.	Туре	Name/Grade	Standard	Group	Delivery condition
43831-	Heavy Plate	S420G2+M	NS-EN	2.1	М
9133182	15X2500X12000	(MDS-Y30	10025-2: 2004		
		Rev.5)			

Table 7-10 - Identification of base material (S420G2+M)

Table 7-11 - Identification of filler material (NSSW SM-47A and NSSW SF-3A)

Brand name	Specification/Classification
NSSW SM - 47A	EN ISO 17632-A-T 46 6 1Ni M M 1 H5
	AWS A5.36 E80T15-M21A8-Ni1-H4
NSSW SF - 3AM	EN ISO 17632-A-T 46 4 Z P M 2 H5
	EN ISO 17632-A-T 46 6 Z P M 2 H5
	AWS A5.36 E81T9-M21A8-Ni1-H4

7.4.3.2 Chemical composition

The chemical composition of the base material is presented in Table 7-12. The table shows the maximum allowed value according to NS-EN 1025:2005 and the chemical composition from the ladle analysis. The chemical composition of the filler metal is presented in Table 7-13, Table 7-14 and Table 7-15.

Elements	Requirements (% by mass), EN	Ladle analysis (% by mass),
(Base material)	10025:2004:2005	MDS
Carbon C	≤0.14	0.10
Silicon Si	0.15-0.55	0.28
Manganese Mn	≤1.65	1.49
Phosphors P	≤0.020	0.011
Sulfur S	≤0.005	0.002
Nitrogen N	≤0.010	0.005

Table 7-12 – Chemical content of base material (Steel S420G2+M)

Aluminum Al	0.015-0.055	0.041			
Copper Cu	≤0.30	0.02			
Chromium Cr	≤0.25	0.05			
Nickel Ni	≤0.70	0.04			
Vanadium V	≤0.080	0.003			
Titanium Ti	≤0.025	0.002			
Niobium Nb	≤0.040	0.026			
Boron B	0.0005	0.0001			
Molybdenum Mo	≤0.25	0.002			
Arsenic As	≤0.030	0.002			
Tin Sn	0.020	0.005			
Lead Pb	0.010	0.001			
Calcium Ca	0.005	0002			
Antimony Sb	0.010	0.000			
Bismuth Bi	0.010	0.001			
Nb+V	0.09	0.03			
EV1 (P _{cm})	≤0.22	0.19			
EV2 (CEV)	≤0.42	0.36			
EV3	≤0.11	0.03			
EV4	≤0.90	0.11			
The levels of the resi	dual elements arsenic, antimony,	tin, lead, bismuth and calcium shall not			
exceed 0.03% As, 0.0	010% Sb, 0.020% Sn, 0.010% Pb	b, 0.010% Bi and 0.005% Ca. Boron (B)			

exceed 0.03% As, 0.010% Sb, 0.020% Sn, 0.010% Pb, 0.010% Bi and 0.005% Ca. Boron (B) shall not exceed 0.0005%. These elements shall be checked at least once every 5000 tones at each manufacturing location and shall be reported as a cast analysis [36].

EV1:	Pcm=C+Mn/20+Mo/15+Ni/60+Cr/20+V/10+Cu/20+Si/30+5*B
EV2:	CEV=C+Mn/6+Mo/5+Ni/15+Cr/5+V/5+Cu/15
EV3:	V+Nb+Ti
EV4:	Cr+Cu+Mo+Ni

Elements (Filler material NSSW SM- 47A)	Requirements (% b mass), spec.	y Chemical composition (%), Manuf.No.7U341AW996
Carbon C	0.04-0.10	0.07
Silicon Si	0.40-0.80	0.59
Manganese Mn	1.1-1.4	1.26
Phosphors P	≤0.020	0.009
Sulfur S	≤0.020	0.008
Copper Cu	≤0.3	0.25
Nickel Ni	0.80-1.10	1.02
Chromium Cr	≤0.15	0.02
Molybdenum Mo	≤0.15	0.01
Vanadium V	≤0.05	0.01
Niobium Nb	≤0.05	0.01

Table 7-13 - Chemical content of filler material.

Hydrogen Content of Deposit	ed Metal (Acc. To ISO 3690)	
HDM (ml/100g)	Ave.	Spec Max.
1.0, 1.7, 1.0	1.2	5 Max

Table 7-14 - Filler metal composition for NSSW SF-3AM Manuf.No.7S041MP960 (% of weight), only nominal values are available, see Appendix B.

Elements (Filler material)	Requirements (% by mass), spec.	Chemical composition for NSSW SF-3AM (%), Manuf. No. S041MP960
Carbon C	0.03-0.07	0.06
Silicon Si	0.25-0.60	0.31
Manganese Mn	1.0-1.5	1.22
Phosphors P	≤0.020	0.010
Sulfur S	≤0.020	0.005
Copper Cu	≤ 0.40	0.26
Nickel Ni	0.80-1.10	1.02
Chromium Cr	≤0.15	0.02
Molybdenum Mo	≤0.35	0.01
Vanadium V	≤0.05	0.01
Niobium Nb	Ref.	0.01

Hydrogen Content of Deposite	ed Metal (Acc. To ISO 369	0)
HDM (ml/100g)	Ave.	Spec.
1.4, 1.2, 1.3	1.3	5 Max

Table 7-15 - Filler metal composition for NSSW SF-3AM Manuf.No.8X221MP996 (% of weight), only nominal values are available, see Appendix B.

Elements (Filler material)	Requirements (% by mass), spec.	Chemical composition for NSSW SF-3AM (%), Manuf. No. 8X221MP996
Carbon C	0.03-0.07	0.05
Silicon Si	0.20-0.55	0.27
Manganese Mn	1.0-1.5	1.15
Phosphors P	≤0.020	0.010
Sulfur S	≤0.020	0.004
Copper Cu	≤ 0.40	0.30
Nickel Ni	0.80-1.10	1.03
Chromium Cr	≤0.15	0.03
Molybdenum Mo	≤0.35	0.01
Vanadium V	≤0.05	0.01
Niobium Nb	Ref.	0.02
Boron B	Ref.	0.003

Hydrogen Content of Deposited Metal (Acc. To ISO 3690)		
HDM (ml/100g)	Ave.	Spec.
1.8, 1.2, 1.0	1.3	5 Max

The easiest way to increase the strength of a steel is to increase the carbon content. A too high amount of carbon can decrease the weldability and increases the risk for solidification cracking in the weld and cold cracking in the fusion boundary in the HAZ. By comparing the chemical composition of S420G2+M and S355J2+M, used for welding of the previous plate, we can see that the carbon content has been reduced from 0.15 % to 0.10 %, while Mn has increased from 1.29 % to 1.49 %, Al has increased from 0.003 % to 0.041 % and Nb has increased from 0.001 % to 0.026 %.

Higher strength of pure carbon steel is achieved by increasing the manganese content. Both carbon and manganese increases the strength, while also reducing the weldability. When a steel is alloyed with manganese it can go from being a carbon steel to becoming a carbon-manganese steel at 0.9-1.7% [25]. The microstructure will still be ferritic-perlitic up to 1.7 % where it changes to a bainitic microstructure. The increase in aluminum and niobium content raises the yield point. Steel that has been treated with Al and Nb is called microtreated steels. Microtreated steels are a special type of microalloyed steels. Micro alloying means that small amounts of bonding elements is introduced in the steel melt. Micro alloying elements are Vanadium (V) and Titanium (Ti) with a higher density than steel. These elements react with C and N and forms carbides, nitrides and carbonnitrides, which forms small precipitations in the austenite.

When these particles become large enough, they lock the grain boundaries and prevent grain growth in the austenite during hot rolling and during the cooling period after hot rolling a more fine-grained ferrite structure is formed. All this means that the material has a better impact resistance but also a lower transition temperature.

S420G2 + M has less Boron than S355J2 + M. For residual element control, Boron (B) shall not be intentionally added to the steel. Nor are any other elements listed in Table 7-12 allowed.

7.4.3.3 Mechanical properties

The mechanical properties of S420G2 + M and filler materials presented below are from the material certificates and the requirements from NS-EN 10225: 2009.

Tensile test					
For base material $-S^2$	420G2+M				
Code / certificate /	Heat	Yield stress	Tensile	$R_{eH\!/}R_m$	Elongation %
test	&plate No.	$(N/mm^{2}),$	strength		
		R_{eH}	$(N/mm^2) R_m$		
Requirements in NS-	-	420-540	500-660	≤0.93	≥19
EN 10225:2:2004 1)					
Inspection	43831-	489	557	0.88	31
certificate 1)	9133182				

Table 7-16 - Tensile test data for base material – S420G2+M

Table 7-17 - Tensile test data for NST filler material

Tensile test			
Filler material – OERLIKON S	PEZIAL		
Code / Certificate	Yield stress	Tensile strength	Elongation
	$(N/mm^2), R_{eH}$	$(N/mm^2) R_m$	%
Requirements in EN ISO	Min. 380	470-600	20
2560-A: E 38 3 B 1 2 H10			
Electrode SPEZIAL (As	≥380	470-600	20
welded)			

Table 7-18 - Charpy test data for base material S420G2+M

Charpy Impact Test For base material – S420	G2+M					
Code / Certificate	Heat	Test temp	Sing	gle values	[J]	Avarage
	No.	(C °)	1	2	3	[J]
Requirements in NS-EN	-	-40	Min. 42	Min.	Min.	Min. 60
10025:2:2004 1)				42	42	
Inspection certificate 1)	43831-	-40	118	111	114	114
	9133182					

Charpy Impact test, Condition: As welded					
Filler material – NSSW SM-47A and		1			
Code / Certificate	Test	S	ingle values		Average
	temp	1	2	3	[J]
	(C °)				
Requirements according to material	-40	Min.	Min. 32	Min. 32	Min. 47
test report (as welded)		32			
	-60	Min.	Min. 32	Min. 32	Min. 47
		32			
NSSW SM-47A, Manuf. No.	-60	93	104	115	104
U341AW996					
(as welded)					
NSSW SF-3AM, Manuf. No.	-40	112	126	142	127
7S041MP960	-60	82	80	87	83
(as welded)					
NSSW SF-3AM, Manuf. No.	-40	137	151	162	150
8X221MP996	-60	115	95	74	95
(as welded)					

Table 7-19 - Charpy test data for NST filler material

7.4.4 Plate manufacturing process

The steel used has been made by the basic oxygen process. It is a steel making process where the carbon-rich molten pig iron is made into low carbon steel. The plate has been manufactured at Salzgitter Flachstahl GmbH and later mill rolled at Ilsenburger Grobblech. In Figure 7-5 we can see the manufacturing procedure for the plate. The material presented below on the manufacturing process comes from the I&TP Manufacturing Procedure for Plate S420G2 +M. I&TP is manufactured by Salzgitter Flachstahl and Ilsenbruger Grobblech.

General information:

Steel making:	Basic oxygen steelmaking process Continuous casting Vacuum
	treatment by ladle degassing
Rolling:	Quattro mill stand with 36 m length Descaling at several passes with 180 bar water pressure Radiometric thickness control TMCP
Plate Stacking:	Stock cooling for hydrogen effusion (min. 12 hours)
Inspection:	Surface inspection on both sides of plates Marking of defects and grinding if necessary After grinding a thickness test will be performed to assure min. wall thickness

Ultrasonic testing:	Automatic UT-Application will be applied to ascertain soundness of the plates. Acceptance criteria S1/E2 acc. To EN 10160
Plate Marking:	Low stress die stamping
Cutting:	Flame cutting

Manufacturing Procedure for Plate S420G2+M

Salzgitter Flachstahl

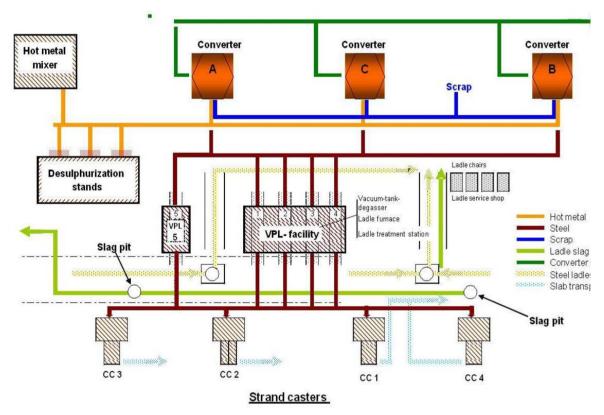
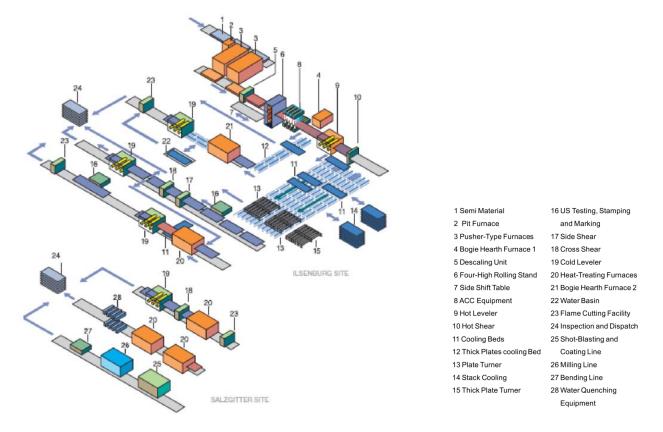


Figure 7-5 - Schematic overview of equipment and material flow in Salzgitter steel plant

- 1) <u>Pig Iron Production (Blast furnace)</u>: Pig iron is first melted into a blast furnace and then transported to the steel works.
- 2) <u>Desulphurisation (Pig Iron)</u>: Pig iron is desulphurised in a ladle by injection of a mixture of CaC2 and Mg powder.
- 3) <u>Steel Production (Converter A, B and C):</u> The converter is loaded with pig iron and scrap. The steel used during the project was fully killed and this is done through the addition of Al and Si. This is the precondition of a fine-grained microstructure. Pure oxygen is blown into the iron bath through a water-cooled lance and inert gas (argon) is blown through bottom. The red hot molten pig iron is further heated until the desired carbon content is reached. The process is called carbon drop. At the end of the process the pig iron has been converted into crude steel. Heat number is allocated. After completion of the production process, crude steel is transferred into a casting ladle.

- 4) <u>Degassing (VPL 1-5 facilities)</u>: Here vaccum treatment is performed by ladle degassing. This is done according to NS-EN 10225: 2009. Vacuum is used to extract the gases from the steel. Gas that is not removed becomes like frozen gas bubbles in the material. Alloying elements are also added to this station. This is done to give the steel its unique character. Before sending, a ladle check is made for alloying correction. A CA-SI treatment for inclusion shape control and desulphuristation is also performed.
- 5) <u>Continuous Casting (CC 1-4)</u>: These have a bow type with a radius of 10.5 m. The steel is transferred from teeming ladle to tundisched in the plant. Continuous casting occurs in sequence and all process parameters are continuously monitored. Before a new casting campaign, a centerline segregation control is implemented. The continuous casting lines create slabs for the liquid steel. The liquid steel is protected against re-oxidation by slag covering, intert gas and mold powder. The casting has a solid shell a few centimeters thick while most of the cross-section is still liquid. A slab dimension control and transverse cutting by automatic flame cutting equipment is then carried out. Generally the first and last slab of one sequence will be rejected by key quality change.
- 6) <u>Flame Scarfing (Scarfing workshop)</u>: The slab is transferred to the Scarfing Shop. Here identification of the incoming slabs and visual inspection of surface and edges is carried out. Slab is cut into mother slabs and automatic paint marking with heat resistant paint with heat and slab number.
- 7) <u>Transport:</u> The S420G2 + M sheet is transported to sister company Ilsenburger Grobblech GMbH where it will be rolled into steel plates.



Ilsenbruger Grobblech

Figure 7-6 - Schmatic overview of equipment and material flow (Ilsenbrug and Salzgitter facilities) - rolling mill

- 8) <u>Slab Reheating (pos. 3)</u>: The slab is transported to a pusher type furnace. In the furnace the plate is heated a temperature between 1100-1200 °C.
- 9) <u>Descaling (pos.5)</u>: Descaling in 2x2 lanes with water pressure of 200 bar.
- 10) Plate rolling (pos. 6&8): The material is exclusively TMCP processed. No accelerated cooling is done on the project plates. Accelerated cooling is only done on plates with a thickness over 25 mm. The sense of TMCP is to roll the plate in a temperature frame, where due to Nb/ Ti or V as micro alloying elements no recrystallisation takes place. Den first rolling är i intervallet 950-800 °C and the second rolling is done in the interval of 780-700 °C and cools in still air. Figure 7-7 and Figure 7-8 illustrates the rolling procedure in a schematic diagram.

Information about the rolling process:

- Quattro mill stand with 36 m length
- Thermo mechanical controlled process
- Finish temperature: 700-800° C

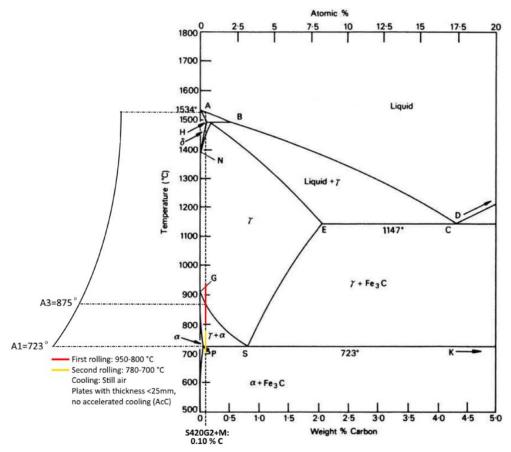


Figure 7-7 - Phase diagram for S420G2 + M 0.10% C showing the temperature interval for rolling. The Iron -Carbon Diagram is only valid for a state of equilibrium condition. During the rolling we have a more dynamic behavior so the temperature is not transferable, but the figure shows the general idea.

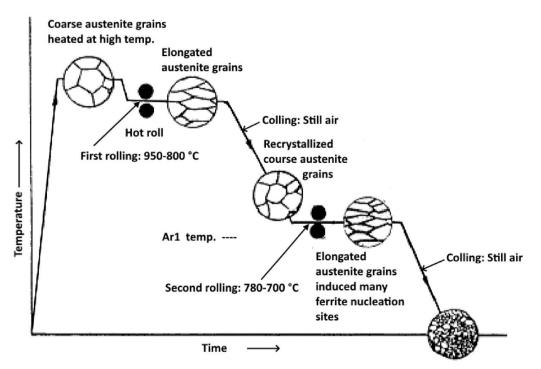


Figure 7-8 - Schematic representation of thermomechanically rolled steel plate S420G2+M

- 11) <u>Thickness Control (pos. 6):</u> A radiometric thickness control is conducted.
- 12) Hot Levelling, Plate Cutting and Hot Marking (pos. 9, 10 &11)
- 13) <u>Plate Stacking (pos. 12):</u> The plates will be stocked for hydrogen effusion.
- 14) <u>Non-Destructive testing (pos. 11, 16)</u>: Visually Inspection and Automatic Ultrasonic Testing. All plates shall be 100% visually controlled and approved according to EN 10163-2. Acceptance criteria for the Ultrasonic Test is EN 10160.
- 15) Marking (pos. 16): Paint marking and low stress die stamping.
- 16) <u>Cutting (pos. 18, 17)</u>: Sampling and preparation of test coupons and cutting to final plate dimensions and final dimension check.
- 17) <u>Material testing and final inspection</u>: Product analysis, transverse tensile test and impact test. Final inspection is according to NS-EN 10225:2009.
- 18) <u>Dispatch and Certification</u>: The preparation of the mill test certificate commences as soon as the order has started production. The certificate is finally printed in parallel with the release for dispatch.

7.4.5 Waterjet cutting

The breakthrough for water cutting came with the fact that you started to cut diapers. At the start of water cutting, you only cut with water. The materials that could be cut were only softer materials such as diapers, paper, plastic and thin plywood. Just over 20 years ago, the water jet began to add an abrasive in the form of sand [39].

A little later, the idea came to mix abrasive agents in the form of sand in the jet stream and then materials such as metal, stone and hard plastic could also be cut [39].

Abrasive water cutting works so that a high pressure pump provides a water pressure of about 4000 bar which is pressed against a jewel orifice. The jewel has a hole and a thin water jet with a diameter of 0.1 mm is created. To get the abrasive into the water, the stream passes through a chamber shaped like a tunnel, where the abrasive is picked up by the water stream and accelerated. The water stream, now carrying abrasive particles, then goes through a focusing tube, or nozzle, which gets all of the abrasive fully involved in the water stream and moving in the same direction. Once it has the focusing tube, you have a supersonic stream of water and abrasive that will quickly be its way through virtually any material [39].

Some advantages of abrasive water cutting versus competitive methods are that it is a coldworking method that does not give rise to internal stresses, straightness of the cut, the lack of a heat affect zone, giving no residual stress, melting edges or curing of the material [39].



Figure 7-9 - Waterjet Garnet

Equipment specifications

Kimtech Water-jet Machine Type: XY-2560 Year: 1012

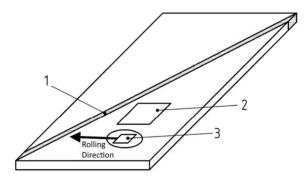


Figure 7-10 - Water-jet cutting på Smedtkristiansen i Dusavik, Stavanger.

Cutting process

All cutting was done with waterjet cutting. The reason why this method was chosen was to not add any extra heat that could affect the material and the end result.

Waterjet cutting was performed at Smed T Kristiansen on a 2.5x3 m sheet metal. The cutting done on this plate was according to "Drawing - Joint preparation 1", see Appendix A. The roller direction has to be perpendicular to the welding direction according to NS-EN ISO 15614: 2017- Specification and qualification of welding procedures for metallic materials. One can find the rolling direction through the heat number, see Figure 7-11 - Waterjet cutting process. In Figure 7-10(c) the cut pieces and the heat numbers are shown.



(a) Marking of plate according to NS-EN 10225:2009. 1. Colour of band according to steel type; 2. Marking; 3. Die stamp (ringed with white paint), text line to be 90 $^\circ$ to rolling direction.

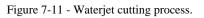


(b) Heat and plate number.



(c) Here we see the finished sheet and heat the number.

(d) Qualification plate, PL1.



7.4.6 Appendix B

I Stage B-Material Selection and Cutting Process the following documents have been produced. See Appendix B.

- Mill test report Base Material S420G2+m HT43831-9133182
- Certified Material Test Report Filler material NSSW SM-47-A Manuf.No. 7U341AW996 MIT 9118
- Certified Material Test Report Filler material NSSW SF-3AM Manuf.No. 7S041MP960 MIT 9188
- Certified Material Test Report Filler material NSSW SF-3AM Manuf.No. 8X221MP996 MIT 9588

7.4.7 Summary

- The objective in this stage was to decide on the suitable steel, filler metal and preparation process before initial welding.
 - The steel type is a commonly used offshore steel. It is thermomechanically rolled, which gives it material properties that cannot be achieved by heat treatment alone.
 - Waterjet cutting was an efficient method that doesn't introduce any adverse heat into the material. This is beneficial due to the interest of the project is to analyze the adverse effect a secondary weld has on the primary weld. The effect oxy-fuel The operation was also able to cut several plates at a single operation. Due to the welding operation is conducted in two stages, the waterjet cutting operation also has to be conducted after the first weld.

7.5 Stage C – Welding Procedure Qualification Program

In this section, the qualification process of the project's WPQ will be presented. The WPQ is backed by a WPQR that is manufactured and tested in accordance with NS-EN 15614:2017. The chapter will begin with a presentation of the material handling and the standard we used to produce welding procedures. The qualification program we produced will then be described and all of the project's WPS's will be presented. Two different types of WPS's were produced. A WPS for the original weld, weld A, and a repair WPS for weld B. The repair WPS was based on the original WPS and was produced in accordance with NORSOK M-101: 2011 (Rev. 5). The results of non-destructive and destructive testing for the qualification plate will be presented at the end of the work together with results from the production plate.

7.5.1 Objective

Develop and approve a WPS.

7.5.2 Material handling

Cold cracking, also called hydrogen cracking, can occur after welding unalloyed steel. Basic theory of how cold crack formation occurs is described in NS-EN 17642-1: 2004 and NS-EN 10025-2: 2005 [34, 40]. Hydrogen cracks are usually formed in the coarse-grained part of the heat-affected zone of HAZ, as a result of a critical combination of microstructure, applied stresses and hydrogen [30].

Important factors affecting the risk of hydrogen cracks are:

- The alloy composition of the steel
- Cooling rate
- Tensile stress concentrations in the welded joint
- The amount of diffused hydrogen in the weld metal [34]

The danger of hydrogen cracks in an unalloyed steel can be expressed with the so-called carbon equivalent CEV, generally used for normal solid C / Mn steels [41].

The cooling rate determines how hard the HAZ will be. The faster the cooling, the harder the structure, and thus the greater danger of hydrogen cracks. The cooling rate is dependent on welding parameters, plate thickness and preheating [41]. The welding parameters can be set to the so-called Heat Input. Heat input is often described with two different concepts, in ISO / TR 18491: 2015 [42] and these terms are defined as arc energy and heat input. Heat input uses a unitless thermal efficiency factor which arc energy does not. Arc energy is determined as shown in equation (7-1). We have used the method for Arc Energy in ISO / TR 18491:2015 [42] when preparing WPQ and logging.

$$E = (U * I)/v * 10^{(-3)}$$
(7-1)

E Arc energy [kJ/mm]

U Arc voltage [V]

- I Arc welding current [A]
- v Traveling speed [mm/s]

Increased heat supply and / or increased preheating results in slower cooling at the same plate thickness, which reduces the risk of hydrogen cracks. This means that thicker sheets have a greater risk of hydrogen cracks.

Moisture is always present in small amounts during welding and this is a prerequisite for hydrogen cracks to occur.

The most common cause of moisture (hydrogen) is [41]:

- Water bound to the filler metal.
- Lubricants in the electrodes.
- Moisture in air or in the protective gas.
- Condensate, rust, oil or coating in the weld zone.

By proper storage, handling and drying procedures, the content of moisture can be minimized and thus reduce the risk of hydrogen cracks.

The hydrogen content is measured in ml H2 / 100 g. The content of the weld varies and is commonly used is <10, <5 and <3 ml / 100g. The hydrogen content for the filler metal of the project can be found in MDSs in Appendix B. The toughest requirement of <3 ml H2 / 100 g comes from the offshore industry. With low hydrogen values one can greatly reduce the heating requirements when welding in thicker materials.

In welding, it is important to understand the relationship between heat input, material thickness, CE / CET, maximum hydrogen content in filler metal and minimum preheating to avoid hydrogen cracks, described in NS-EN 1011-2: 2001 [41]. We will present below how we have gone about counteracting hydrogen cracks.

7.5.3 Material, welding consumable and specimens handling plan

All materials and specimens used have been handled according to the material handling plan below.

Base material - S420G2+M (MDS Y30 Rev 5)

The material was stored indoors in a dry environment to avoid corrosion. All material was also stored on a pallet to prevent material from disappearing and being damaged.

Filler Wire (NSSW-SM47A and NSSW SF-3AM)

Storage and handling of filler material was done in accordance with supplier NST recommendations. Unopened packages of filler material were stored indoors in a heating cabinet to ensure optimal performance during use. After welding, the filler material was stored in a heating cabinet until it was used again. The manufacturer has presented hydrogen content of Deposited Metal on MDS filler metal see Chemical composition 7.4.3.2.

Electrode – ESAB OK48.08 E7018-G, Low Hydrogen Welding Rod

The electrodes used for welding strongbacks were ESAB OK48.08. These were taken from unopened vacuum-packed packages and before use, dates were checked.

Prior to all welding, the filler material was checked against the stated filler material in the welding procedure. We also checked that the filler material had not begun to corrode or had defects.

Test specimens

- Test specimens for mechanical testing were taken care of by Qlab.
- Test specimens for fatigue testing were taken care of at UIS. Finished prepared fatigue specimens were lubricated in anti-rust oil and stored in a dry environment. Fatigue specimens had undergone thorough surface preparation and were handled with great care. Specimens were rolled into towels in anticipation of testing. Installation of specimens in the fatigue machine was done with nitrile gloves so as not to apply grease or moisture to the surface which could cause surface corrosion.

7.5.4 Production flow of Welding Procedure Test

The figure below shows the different steps the qualification plate (PL1) went through. When we picked up the plate after waterjet cutting, we were on step 2 Figure 7-12. The plate was then transported to KIWA TI. where welding and logging would be carried out.

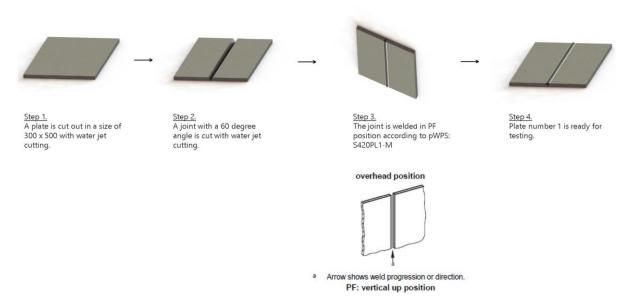
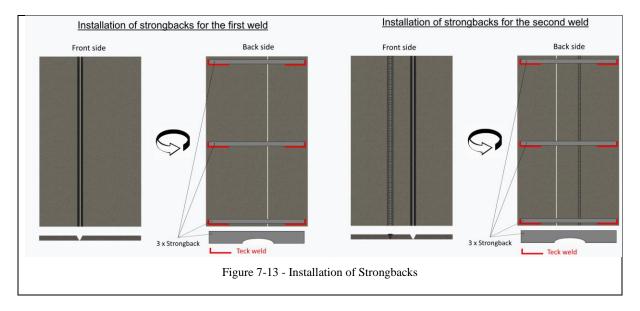


Figure 7-12 - Production flow of Welding Procedure Test

Before welding, strong backs were installed on the back for fixing the plates. The strong backs were formed so that after installation, during and after welding, it was possible to examine the weld joint and the weld root side. Three strongbacks were welded. According to standard for manufacturing welding procedure, NS-EN 15614: 2017, the discard shall be 25 mm at the beginning and end of the plate. Two strongbacks were welded in areas that would later be discarded and the last strongbacks installed in the middle of the plate. Tack weld was carried out in accordance with guidelines in NS-EN 1011-1: 2009, where the recommended minimum

length of a tack weld should be 50 mm [41]. For easier disassembly the tack welds were only welded on one side Figure 7-13.

In the following steps 3 and 4 in Figure 7-12, welding in ascending position (PF) as well as non-destructive and destructive testing was performed to look at the mechanical properties. After welding and dismantling of stongbacks, a visual check was made where deformation and defects were noted.



7.5.5 Welding Procedure Qualification Program

Welding procedures are the necessary basis for planning the welding operation and the quality control during welding. Qualification of welding procedures must be carried out prior to actual welding production and can be done according to several methods mentioned in NS-EN ISO15607.2007 [43].

The method used was "Qualification based on a welding procedure test", see Figure 7-14.

There was no pWPS or WPS when the project started, so first a pWPS is produced, this one is shown in Figure 7-15. Plate 1 (PL1-SW) was the qualifying plate and welded according to the pWPS-S420PL1-M.

All new welding procedure tests shall be carried out in accordance with NS-EN ISO 15614: 2017-Specification and qualification of welding procedures for metallic materials Welding procedure test Part 1: Arc and gas welding of steels and arc welding of nickel and nickel alloys. Two levels of welding procedures tests are given in order to permit application to a wide range of welded fabrication and they are desginated by level 1 and 2. Level 1 is based on the requirements of ASME Section IX and level 2 based on the previous edition of NS- EN ISO 15614.2017. Testing was carried out in accordance with the level 2 regulations, which also qualify for lower level 1. During testing it was demonstrated that the used joining process is capable of producing joints that meet standard requirements.

Testing results were used as the basis for the establishment of the Welding Procedure Qualification Record, WPQR. The qualification program was used for the production of WPQR. The tests from the qualification plate had to be approved before the project's WPS's became valid.

The project's WPS is based on WPQR: S420PL1-M. We can see the whole manufacturing process of WPS in Figure 7-14.

Repair WPS is based on original WPS according to NORSOK M-101: 2011 Rev.5. According to NORSOK M-101: 2011 Rev.5, the preheat temperatures used during repair welding was minimum 50 °C higher than the preheat used for the original weld. Preheating was achieved by propane.

The activities in the qualification program was carried out according to the I&TP. The I&TP shows the activities, acceptance criteria, verifying documents that was followed and who carried out the approval.

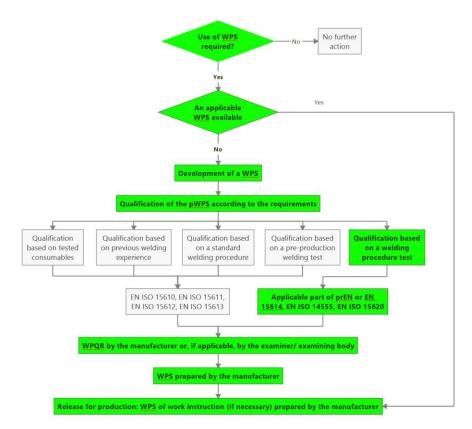


Figure 7-14 - Flow diagram for the development and qualification of a WPS

Welding Procedure Qualification Overview

Table 7-20 - Welding Procedure Qualification Overview shows how the different welding procedures are linked together. Here you can see that PL1 was a qualification plate welded with

pWPS-S420PL1-M. Welding, logging and testing done on this sheet resulted in WPQR-S420PL1-M, as previously mentioned, this became the basis for future WPS's.

In Table 7-20 we also see Qlabs Cert. No., these are Qlabs materials reports.

Plate	pWPS No	WPQR No	Weld A Original WPS No	Weld B Repair WPS No	Notes	Q-labs Cert.No.
1	S420PL1-M	S420PL1-M	PL1-SW		Welding procedure qualification plate.	8612-1
2					Not used	
3		S420PL1-M	PL1-SW	PL3/4-DW50-Rep	The WPS is based on WPQR and the rep- WPS is based on original WPS.	8612-2
4		S420PL1-M	PL1-SW	PL3/4-DW50-Rep	The WPS is based on WPQR and the rep- WPS is based on original WPS.	
5		S420PL1-M	PL1-SW	PL5/6-DW15-Rep	The WPS is based on WPQR and the rep- WPS is based on original WPS.	8612-3
6		S420PL1-M	PL1-SW	PL5/6-DW15-Rep	The WPS is based on WPQR and the rep- WPS is based on original WPS.	
7		S420PL1-M	PL1-SW	PL7/8-DW5-Rep	The WPS is based on WPQR and the rep- WPS is based on original WPS.	8612-4
8		S420PL1-M	PL1-SW	PL7/8-DW5-Rep	The WPS is based on WPQR and the rep- WPS is based on original WPS.	
9					Used for fatigue testing.	
10					Not used.	
	VPS=Preliminary V Dual Weld, Rep=I		e Specification, WPS=W	eding Procedure Specifi	cation, WPQR=Welding Procedure Qualification Reco	ord, PL=Plate, SW=Sing

Table 7-20 -	Welding	Procedure	Qualification	Overview
14010 / 20	i erang	11000000000	2 autoriou	0.001.000

7.5.6 preliminary Welding Procedure Specification

Together with the filler material supplier NST and the KIWA TI, pWPS-S420PL1-M was manufactured, see Figure 7-15.

ι	S		ive Sta		ty nger	Preliminary WELD								VPS No. ef.: ate:		us Larsson	Rev.4
Prod. bj Project: Locatio		KIWA MSc T Gener	hesis	- We	ld Proxim	ity IN		t: spec.: WPQR:		versity of S RSOK M-1				Ref. sta Exam. b		-EN ISO 156′ knologisk Inst	
Welding	g proce	ess			138					136							
Shieldir	ng gas	type		1	Argon	/18%	CO2 (M21)		2	Argon /	18%C	D2 (M21)		3			
Weavin	ig (yes	/no)			yes		max: 7	mm		yes	m	ax: 13	mm			max:	mm
Purging	g gas ty	/pe		N//	4				I/min								
Welding	g positi	ons		PF	, vertical	up							WPQ	R No.:	S420PL1	-M	
Joint ty	pe			в٧	V												
Joint pr	reparat	ion		Wa	aterjet cu	tting,	Grinding						\backslash	60	0	/	
Cleanin	ng metl	nod		Wi	re brush											1	
Backing	-			N//	-								-+	~	up /		
Single/[-	ngle								F				
Back go				N//						<	\geq	_ I		│ Fi	u /	Π	
Flux de	-	on		N//								1		2		11	
Flux ha	-			N//										- ju	3-4mm	Lo mm	
Tungste Torch a		trode		N//	• -90 ⁽	,			mm								
Stand o	-	ance		10	-30				mm								
Nozzle				-	-20				mm								
Tack we		. ,		Ge	neral			Rev: 0									
Identif			oarer	nt m	etal	I: C	E max: 0,42			14 PCM	max:	0.22	: CE ma	IX: 0,42	C max: (0,14 PCM	/ max: 0,2
Part		Nam	e/Gra			Standard Grou		p D	Delivery cond.		Т	Thickness range [mm]			eter range		
1	\$420G	62 +M ,∣	MDS-1	(30 I	Rev.5 El	N 102	10225:2009 2.1		TM	1			15,	00		-	
11	S420G	62+M,∣	MDS-	(30 F	Rev.5 El	N 102	25:2009	2.1 TM				15,00 -					
ldentif	ficatio	n of	iller	met	al												
Index		B	rand	nam	e			Classific	ation			Gro	Group Filler handling				
1	NSS	W SM	-47A			EN	ISO 17632-A	T46 6 1		1 H5		FM1	11 Sup		lers reco	ommendati	on
2		W SF-				EN	ISO 17632-A- ISO 17632-A-	-T 46 4 Z	2 P M 2	2 H5		FM1	Suppliers recommendation				
Weldir	-						-				quipm						
Pass no.	Index	Dia. [mm]	Weld meth		Wire feed speed	I	Current	Volta [V]		Current / Polarity		ing speed	Run Ou	-	1 Gas [l/min]	Н	eat input [kJ/mm]
BW:	_	4.00	40-	_	-	_	-	-	45	P 2	-	-		A			-
I(Root) Fill	1	1,20 1,20	138	-	2,2 - 2.4	_	100 - 120 210 - 230	14.5 - 22 -		DC+ DC+		60 - 80 10 - 200	N	/A /A	20		1.3 - 1.6 1.4 - 2.0
CAP	2	1,20	136	-	6.5 - 7.5	_	200 - 210	22 -		DC+ DC+		60 - 200 60 - 220		/A /A	20		1.4 - 2.0 1.2 - 1.7
				_													
Heat tr	reatm	ent									1		Meth	od.	_		
				0.10	tornana ta			V-	attract	monteres						h-1	
Preheat		20			terpass te			∘c He		ment proc.:		min		emp. co		ital contact th	
-	KS: WPS is		ated		th esti		Soaking: ed and fix	ed par	min/m			Addit Date	/Signatur	enclos e:	ed (Yes/ N e	Cooling rate:):	°C
Positi	ion fi	. x ing	: 3	stro	ongback	s to	be used.					Appr	.04.05 / 1 oved: .04.05 / 1	-	Larsson Larsson		
								P	age	1	of	1					

Figure 7-15 - pWPS

7.5.7 Welding process

All welding and logging was done together with the KIWA TI in Stavanger. KIWA supported the project with welders, welding engineers, welding booths, equipment, accessories, material handling and storage.

Before welding, the welding machine was calibrated, parameters such as Volts, Amperes and welding gas flow were set and a test plate was welded. Before welding PL1, it was checked that the bevel and groove angle were in accordance with pWPS. The welding machine used was a Fronius-Trans Steel 2700 and the logging equipment used was FLUKE 376 TRMS clamp meter, folding ruler, stopwatch, lamp, cam type gauge and a welding log diagram.

The welding process started with the installation of strongbacks on the back side according to Figure 7-13. The plate was then mounted and welded in the welding position PF (vertical up position) according to ISO 6947:2011. Prior to welding, fit-up, joint preparation and welding position was approved by examiner at KIWA TI.

Welding was done according to pWPS: S420PL1-M, Figure 7-15 and all welding was logged (all logs can be seen in Appendix C&D). The root string was welded with method 138, MAG welding with metal cored electrode; gas metal arc welding using active gas and metal cored electrode. Fill and cap were welded using method 136, MAG welding with flux cored electrode; Gas metal arc welding using active gas and flux cored electrode. When welding was completed, visual inspection was performed by IWI/IWE Arild Finnesand, KIWA TI. The visual inspection was performed according to ISO 17637:2016 and the acceptance criteria was EN ISO 5817:2014 B/C.

The strongbacks were later removed when the plate had reached room temperature. The plate was transported to non-destructive testing (NDT) 24 hours after the completion of welding.



(a) Calibration of welding machine



(b) Gas 20 l/min

(c) Test welding



(d) Control of bevel angle



(e) Welding machine

(f) Digital contact thermometer.



(g) Installation of strongbacks

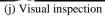


(h) Welding and Logging of qualification plate



(i) Visual inspection







(k) Plate 1

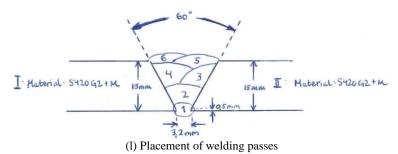


Figure 7-16 - Welding of qualification plate

7.5.8 NDT and mechanical testing

After welding, the plate was transported to IKM Inspection for radiographic testing.

After the approved radiographic test, the plate was transported to Qlab in Forsand where the remaining NDT and mechanical testing were performed. After testing, Qlab produced material report: 8612-1 which is part of the document WPQR: S420PL1-M.

All testing in the project was conducted according to NS-EN ISO 15614: 2017 Level 2. In Figure 7-17 we see the tests that were performed and the location of test specimens for the butt joint plate.

2

3

4

5

4

3

2

EL

1

SW

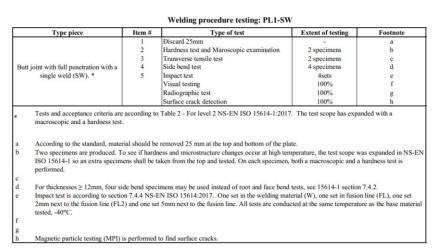


Table 7-21 shows the procedure and acceptance criteria used in qualifying.

Figure 7-17 - Welding procedure testing

Activity Description	Specification/	Acceptance Criteria	Verifying	Result
	Procedure		Document	
Visual testing	ISO 17637:2016	EN ISO 5817:2014 B/C	Material	OK
			report: 8612-1	
Radiographic	ISO 17636-2:2013	ISO 10675-1:2016	Material	OK
			report: 8612-1	
Surface Crack Detection	NS-EN ISO 17638:2016,	ISO 23278:2015 Level B	Material	OK
Test (Magnetic particle	NS-EN ISO 3452-1:2013		report: 8612-1	
inspection)				
Macro examination	NS-EN 17639:2013	EN ISO 5817:2014	Material	OK
		(quality level B)	report: 8612-1	
Transverse Tensile Test	ISO 4136:2012, ISO	NS-EN 15614-1: 2017,	Material	OK
	6892-1:2016 Method A1	NS-EN 10225:2009	report: 8612-1	
Charpy V Impact Test	NS-EN ISO 148-1:2016,	NS-EN 15614-1, NS-EN	Material	OK
	ISO 9016:2012	10225:2009	report: 8612-1	
Side bend test	NS-EN ISO 5173:2010	NS-EN 15614-1:2017	Material	OK
			report: 8612-1	
Vickers hardness test	NS-EN ISO 9015:2011	NS-EN 15614-1:2017	Material	OK
			report: 8612-1	

Table 7-21 - Examination and testing of qualification plate

7.5.9 Welding Procedure Qualification Record

NS-EN ISO 15607 defines a WPQR as "Record comprising all necessary data needed for qualification of a preliminary welding procedure specification". The document WPQR S420PL1-M contains the following:

- WPQ
- Summary
- pWPS
- Results from NDT and mechanical testing
- Material certificate base material
- Material certificate filler material

7.5.10 Welding Procedure Specification

NS-EN ISO 15607 defines a WPS as "A document that has been qualified by one of the methods described in clause 6 and provides the required variables of the welding procedure to ensure repeatability during production welding".

Two different variants of WPSs were produced. The first WPS was for weld A and the second was a repair WPS for approximity welds (weld B). The Repair WPS was based on the same WPQ as the original weld but was converted into a repair WPS according to Norsok M-101: 2011 Rev.5 Section 6.11-Preheat and interpass termperature and Section 10.4-Repair welding procedure.

A total of four WPSs were produced:

- PL1-SW
- PL3-DW50-Rep & PL4-DW50-Rep
- PL5-DW15-Rep & PL6-DW15-Rep
- PL7-DW5-Rep & PL8-DW5-Rep

The WPS documents specified the procedure for performing welding processes for production stage which includes welding method, base material, welding consumable, preparation, preheating, welding method, control of welding, post weld heat treatment and equipment to be used.

7.5.11 Appendix C

In Stage-C Welding Procedure Qualification Program the following documents has been created.

- WPQR: S420PL1-M and including documents:
 - WPQ S420PL1-M
 - Summary
 - WPS: PL7/8-DW5-Rep

- o pWPS S420PL1-M
- Weld log pWPS:S420PL1-M
- Report7967-19-DRT-1
- Mill test report Materialcertifikat S420G2+M HT43831-9133182
- Mill test report Filler material: NSSW SM-47-A Manuf.No. 7U341AW996 MIT 9118
- Mill test report Filler material: NSSW SF-3AM Manuf.No. 7S041MP960 MIT 9188
- Mill test report Filler material: NSSW SF-3AM Manuf.No. 8X221MP996 MIT 9588
- All WPS's:
 - WPS PL1-SW
 - WPS: PL3/4-DW50-Rep
 - WPS: PL5/6-DW15-Rep
 - WPS: PL7/8-DW5-Rep

7.5.12 Summary

The main objective of this step was to present the procedural steps that was performed in other to develop a WPQ. In order to approve WPQ is has to be approved in accordance with NS-EN 15614:2017, which requires extensive mechanical testing. The following production plates would be welded using the WPS based on this WPQ. The WPQ ensured that all the following welding operations carried out met the requirements.

7.6 Stage D – Production welding of plates

In this sub-chapter, the manufacturing and welding process of the project's production plates will be presented. The first weld joint was cut in Stage B and is ready for welding. Production welding shall be done according to the welding procedures manufactured in Stage C.

7.6.1 Objective

Manufacturing of the production plates intended for Non-Destructive Testing (NDT) and Destructive mechanical Testing (DT).

7.6.2 Introduction

In this sub-chapter, the welding and cutting process for the production plates will be described.

In the previous stage C, only one plate was welded and tested. This was carried out to confirm that the welding process was correct and that the mechanical properties of the welds were in accordance with the requirements of NS-EN ISO 15614:2017.

We started Stage D by welding six production plates according to our Welding Procedure Specification (WPS).

7.6.3 Production flow of test plates for production test

Figure 7-18 shows the different fabrication steps that the production plates went through.

In Step 1 and 2 the plates were cut with a waterjet cutter. In step 3, strongbacks was mounted on the backside of the plate and joint A was welded.

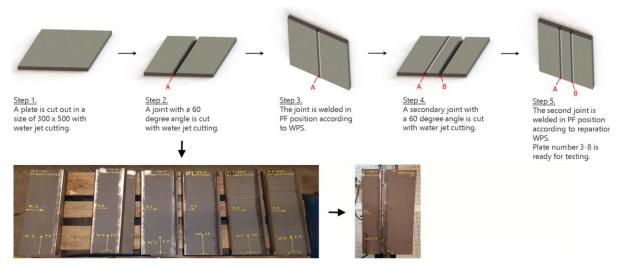


Figure 7-18 - Production flow of test plates for production test.

In step 4 the plates was sent for waterjet cutting of weld joint B. After cutting, the plates were sent back to KIWA TI for welding.

Step 5 is installation of strongbacks and welding of joint B.

Six production plates were prepared with three different spacings between joint edges. The distance between joint edges (D1) was 50, 15 and 5 mm shown in Figure 7-19.

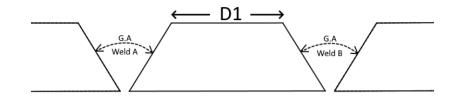
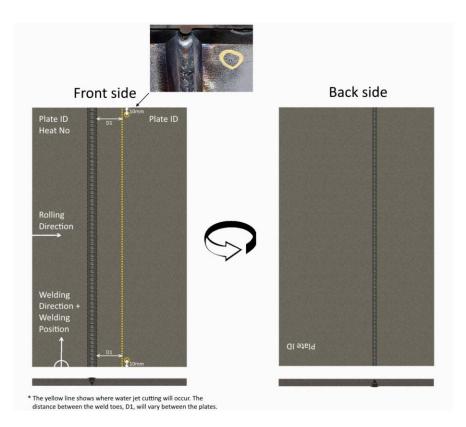


Figure 7-19 – Distance between the weld joints.

To keep track of all parts that were manufactured, a identification and marking system was created, this was important for traceability.

Here is a description how the identification, marking and cutting of plates were carried out. The first qualification plate had only one weld while the production plates had two parallel welds, see Figure 7-20 (c). In the manufacture of weld joint number two (weld B), a parallel cutting was performed alongside the first original weld (weld A), see yellow dotted line in Figure 7-20 (a).



(a) Identification system and yellow cutting line.



(b) Centre punch tools

Figure 7-20 - Identification and marking system

Figure 7-20 - Identification and marking system, shows how the plates were named. The plate ID was written on both of the front side's upper corners, this was important to avoid mixing up the plates after cutting.

Prior to welding, the distance D1 was measured from the joint edge A to the next joint edge B. The three different distances 50, 15 and 5 mm were marked with a center punch. This mark is circled with a yellow circle in Figure 7-20(a) and the tools used are shown in figure b.

After welding, a yellow line was drawn between the two markings, this was the cutting line. The cutting was carried out with a 30 degree bevel angle, see drawing-joint preparation 2. When the cutting was completed, the plate was flipped so that the back side became the front side which gave a groove angle of 60 degrees. Welding of weld B was done in welding position PF (rising welding position).

The rolling direction of the steel affects the microstructure and mechanical properties of the steel, thus affecting it's anisotropy, which means that the strength can differ between directions. Therefore, the test plates have been produced consistently so that all welding passes have been done across the steels rolling direction.

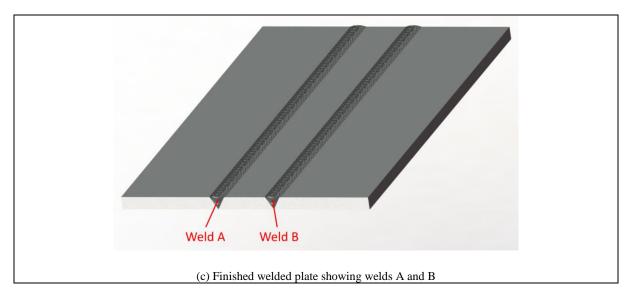
Hard stamping was done on the short sides of the specimen as it is prohibited according to NS 1011-2:2001 to stamp in areas that are highly stressed or where dynamic loads are anticipated.

7.6.4 Welding and the manufacturing process of the production plates

Below is a more detailed presentation of the welding process and illustrated with pictures.

7.6.4.1 Weldability

According to the welding and testing procedure defined in NS-EN ISO 15614:2017, the welds were made on plates of 300 mm width, 500 mm height, 15 mm thickness. The root pass was made with MCAW (energy per unit length / heat input around 1.29-1.61) and before welding, fill and cap of the root was grinded and then welded with FCAW (energy per unit length / heat input around 1.21-1.99) in PF position.



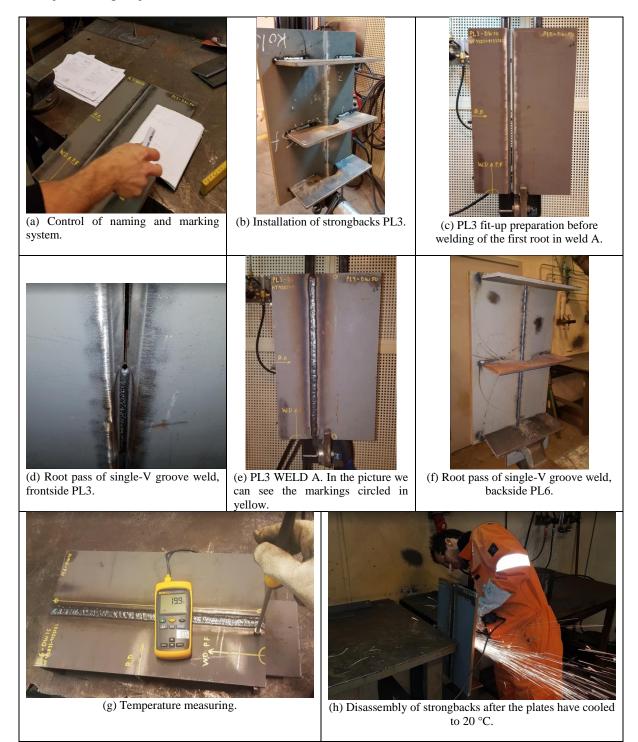
7.6.4.2 Welding of weld joint A

Before welding the first weld A, the identification and marking system was checked, see Figure 7-20. Thereafter preparations of the joints, fit-up and alignment of production test plates and installation of strongbacks. On the first plate, PL3, according to procedure, no strongbacks were installed which led to the tack weld for the plate breaking. Upcoming strongbacks were welded according to the "Installation of strongbacks" procedure, see Figure 7-13. In Figure 7-21 (g) we can see the installation of strongbacks with tack welds.

A summary of deformation, crack in welds, etc. is presented as a summary at the end of the chapter.

All primary welds conducted on PL3-PL8 was done according to original WPS: PL1-SW and full logging was performed. To see WPSs and logs, see Appendix D.

When the plates were cooled to about 20 °C, all strongbacks were removed. The deformation that occurred was measured and a visual inspection was carried out according to the acceptance criteria of EN ISO 5817: 2014. After inspection, the plates were transported to Watech AS for waterjet cutting of joint B, weld B.

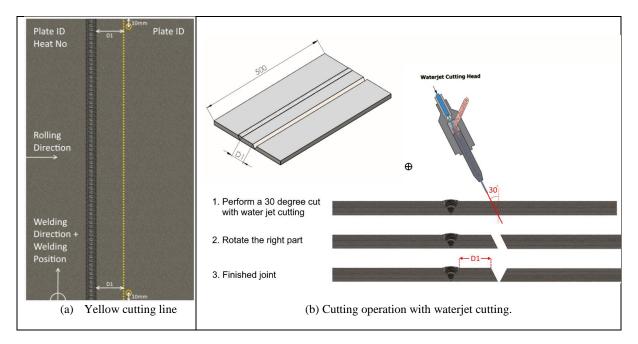


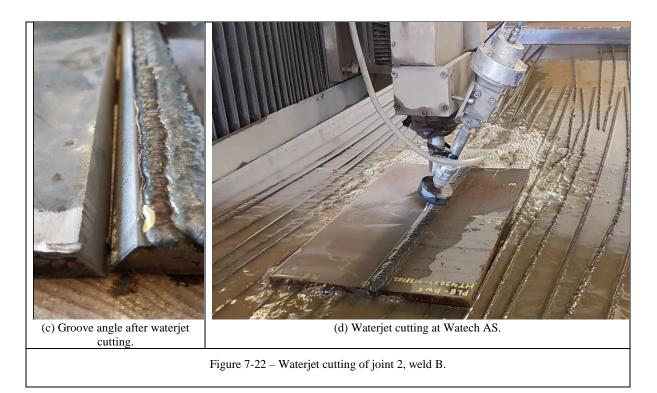


7.6.4.3 Waterjet cutting of weld joint B

Waterjet cutting was carried out at Watech and according to drawing - Joint preparation 2 (Appendix D). Which we can see in Figure 7-21 there were six plates with a yellow cutting line. There were three different distances between the weld joints 50, 15 and 5 mm +/- 2 mm and the distance is called D1. With the waterjet cutting method, a cut with a 30 degree angle was made and after cutting the right plate was flipped, see Figure 7-22. The cut made at Watech varied between 28-30 degrees as the machine had to be adjusted by loosening a screw, see Figure 7-22 (d). The final groove angle ranged between 57-60 degrees.

After cutting, the plates were transported back to KIWA TI. for welding of joint B, weld B.





7.6.4.4 Welding of weld joint B

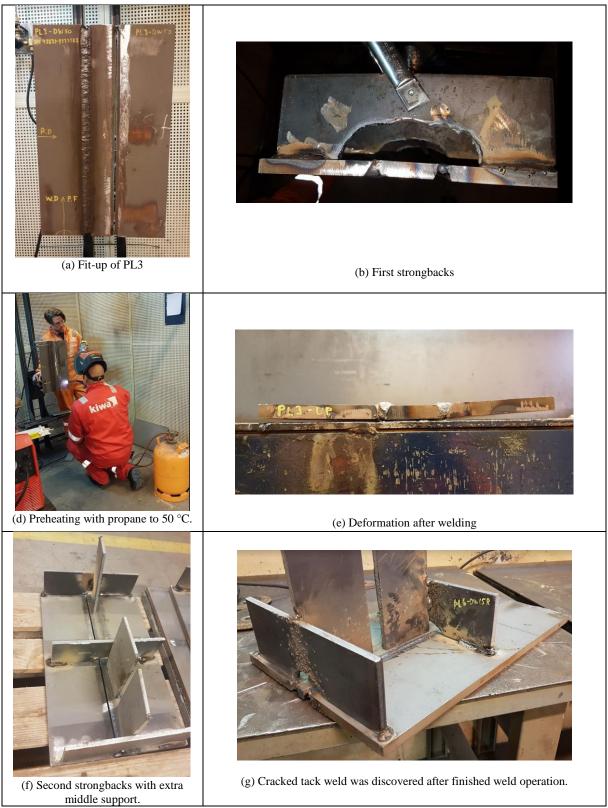
Before welding the second weld B, the markings on the plates were checked. Thereafter, joint preparation, fit-up and alignment of production test plates and installation of strongbacks. Before installation, the strongbacks had to be reshaped as their support was not sufficient. On the first plate a large notch was made in the strongback, but after welding we noticed that the strongbacks could not withstand the deformation forces that arose during welding. On the other plates, therefore, a smaller notch was made in the strongback and an extra support was welded on the strongbacks Figure 7-23(f). This type of strongback installation caused less deformation of the test plates, but instead some of the strongback holding welds ruptured due to the deformation forces.

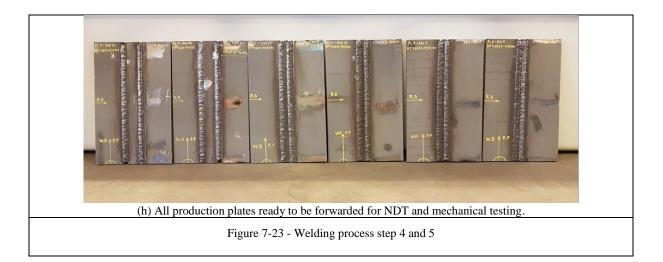
Weld B was welded with a repair WPS and all welding was logged. The repair WPS was based on the same WPQ as the original WPS but had an elevated minimum preheat temperature of 50 °C according to Norsok M-101: 2011 Rev.5. Preheating was done with propane.

To see repair WPS's and logs from welding, see Appendix D.

After the plates had cooled in air down to a temperature about 20 °C, the strongbacks were removed. The deformation that occurred was measured and a final visual inspection was carried out according to acceptance criteria in EN ISO 5817: 2014. After 24 hours after the completion of welding, the plates were sent to IKM Inspection for non-destructive testing (NDT).

A summary of deformations, crack in welds and other results are presented in chapter Results and Discussions.





7.6.5 Appendix D

In step D - Production welding of plates, the following documents have been produced. See Appendix D.

- All Weld log's for Weld A:
 - Weld log: PL3-DW50
 - Weld log: PL4-DW50
 - Weld log: PL5-DW15
 - Weld log: PL6-DW15
 - Weld log: PL7-DW5
 - Weld log: PL8-DW5
- All Weld log's for repair Weld B:
 - Weld log: PL3-DW50-Rep
 - Weld log: PL4-DW50-Rep
 - Weld log: PL5-DW15-Rep
 - Weld log: PL6-DW15-Rep
 - Weld log: PL7-DW5-Rep
 - Weld log: PL8-DW5-Rep

7.6.6 Summary

The objective of this stage was to manufacture the production plates that were intended for further mechanical testing at Qlab. The mechanical testing will be described in detail in the next chapter. We had a heavy focus during this operation to log the different procedural steps in the form of drawings, pictures, weld logs and taking notes. To have this information was essential for the future assessment and evaluation of the project.

7.7 Stage E – NDT, Mechanical Testing and Specimen Preparation

In this chapter, the non-destructive and mechanical testing of the production plates will be presented. NDT testing was done at KIWA TI and IKM Inspection AS while the mechanical testing was done at Qlab in Forsand. Testing was carried out according to the same standard as the qualification plate NS-EN ISO 15614: 2017.

7.7.1 Introduction

This section will present and describe how testing of the production plates has been carried out. Results from all plates will be used for a final discussion and conclusion. The aim of the project has been to look at nearby welds and how they affect each other's properties. There was a total of six production test plates, but it was only three of them that were used for mechanical testing. Remaining plates were used for testing residual stresses.

7.7.2 Objective

The objective of this stage was to perform non-destructive and mechanical testing on the production plates that were produced in previous stage.

7.7.3 Non-Destructive Testing (NDT) and Mechanical Testing

The production plates were tested using the same standard as the qualification plate, but the test scope in NS-EN ISO 156214: 2017 Level 2 had expanded with one macroscopic test, one hardness test and two fatigue tests. The fatigue test has replaced previous bending test. We can see the test program for the production plates in Figure 7-24.

Before, during and after welding, a visual inspection was carried out by examiners at the KIWA Institute of Technology, where all welds passed the examination. Non-destructive and destructive tests were conducted at least 24 hours after welding.

After the visual testing at KIWA, the plates were transported to IKM Inspection for radiographic testing.

After radiographic testing, the plates were transported to Qlab where all mechanical testing accept the fatigue test was carried out. Plates PL3, PL5 and PL7 were tested at Qlab while PL4, PL6 and PL8 were to be used for residual stress tests.

Qlab performed the Surface Crack Detection Test (MPI), Macro examination, Transverse Tensile Test, Charpy V Impact Test and Vickers hardness test. After completion of the tests at Qlab the test specimens and production plates were brought to the University of Stavanger for further investigation and preparation. The specimens that were cut for fatigue testing were prepared and tested. The plates that were intended for residual stress testing were cut, prepared and sent to Veqter LTD in England for residual stress testing.

The test results are presented in chapters; F - Microstructural, G - Fatigue Testing and H - Residual Stress Measurements.

Table Table 7-22 shows the activities, procedures, acceptance criteria and verifying documents used during testing of the production plates.

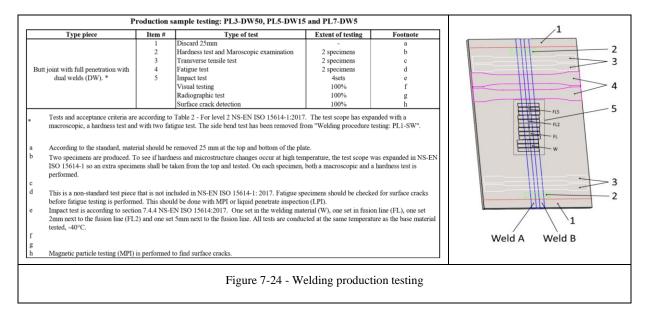


Table 7-22 -	Examination	and testing	of prod	uction p	lates

Activity Description	Specification/ Procedure	Acceptance Criteria	Verifying Document	Result
Visual testing	ISO 17637:2016	EN ISO 5817:2014	Material report:	OK
C		B/C	8612-2, 8612-3	
			& 8612-4	
Radiographic	ISO 17636-2:2013	ISO 10675-1:2016	Report serie:	OK
			8076	
Surface Crack	NS-EN ISO 17638:2016,	ISO 23278:2015 Level	Material report:	OK
Detection Test (MPI)	NS-EN ISO 3452-1:2013	В	8612-2, 8612-3	
			& 8612-4	
Macro examination	NS-EN 17639:2013	EN ISO 5817:2014	Material report:	OK
		(quality level B)	8612-2, 8612-3	
			& 8612-4	
Transverse Tensile Test	ISO 4136:2012, ISO 6892-	NS-EN 15614-1: 2017,	Material report:	OK
	1:2016 Method A1	NS-EN 10225:2009	8612-2, 8612-3	
			& 8612-4	
Charpy V Impact Test	NS-EN ISO 148-1:2016,	NS-EN 15614-1:2017,	Material report:	OK
	ISO 9016:2012	NS-EN 10225:2009	8612-2, 8612-3	
			& 8612-4	
Vickers hardness test	NS-EN ISO 9015:2011	NS-EN 15614-1:2017	Material report:	OK
			8612-2, 8612-3	
			& 8612-4	

7.7.4 Appendix E

In Stage E - NDT, Mechanical Testing and Specimen Preparation. Se Appendix E

- NDT report Radiograpic Examination
 - o Report8076-19-DRT-1
 - o Report8076-19-DRT-2-REV1
 - o Report8076-19-DRT-3-REV1
 - Report8076-19-DRT-5
 - o Report8076-19-DRT-6
 - o Report8076-19-DRT-7
- Qlabs Material Report 8612-2 (PL 3)
- Qlabs Material Report 8612-3 (PL 5)
- Qlabs Material Report 8612-4 (PL7)

7.7.5 Summary

The objective of this stage was to perform non-destructive and mechanical testing on the production plates that were produced in previous stage. The planning and execution were intensive and required good coordination with our supporting partners. The testing went well and was executed according to plan. The test results are presented in chapter Results and Discussions.

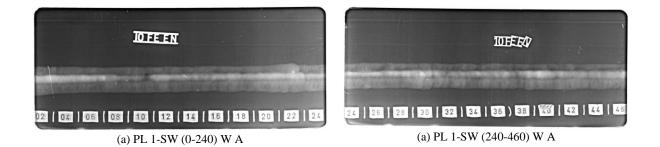
7.8 Results and discussion

The results presented here are from Stage-C, Stage-D and Stage-E.

7.8.1 Results from Stage C – Welding Procedure Qualification Program

This chapter concerns the testing and approval process of the weld qualifying plate, with one weld joint used for the WPQ. Table 7-21 summarizes the procedures, acceptance criterias and verifying documents used during the stage. All tests met the requirements, and has been verified by KIWA TI, IKM Testing and Qlab.

Plate	Weld No	Film location 20-240	Film location 240-460	Defect location	Remarks
S420PL1-M	Weld A	Accepted	Accepted	106-110	515



Magnetic particle inspection (MPI)

No findings, accepted.

Macro examination

No weld imperfections was visible on the cross section.

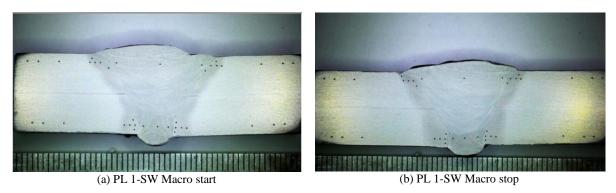


Figure 7-25 - Macro Examination of PL1

Transverse Tensil	e test		
Specimen No.	Acceptance criteria [MPa]	R _m [MPa]	Fracture
Cross weld sample 1	500-660	522	Base material
Cross weld sample 2	500-660	528	Base material

Table 7-25 - Charpy test data for PL1.

Charpy V Impact Toughness test, KV ₈								
Test ident.	Dimension	Notch	Test temp	Sin	gle values	[J]	Avarage	
	[mm]	Orientation	(C °)	1	2	3	[J]	
Acceptance criteria			-40	Min. 42	Min. 42	Min. 42	Min. 60	
Weld	10x10x55	Т	-40	80	131	106	106	
FL	10x10x55	Т	-40	113	186	193	164	
FL+2	10x10x55	Т	-40	253	130	200	194	
FL+5	10x10x55	Т	-40	234	236	173	214	

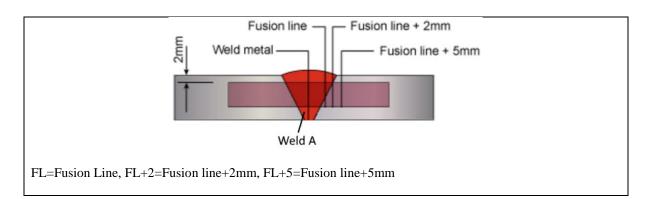
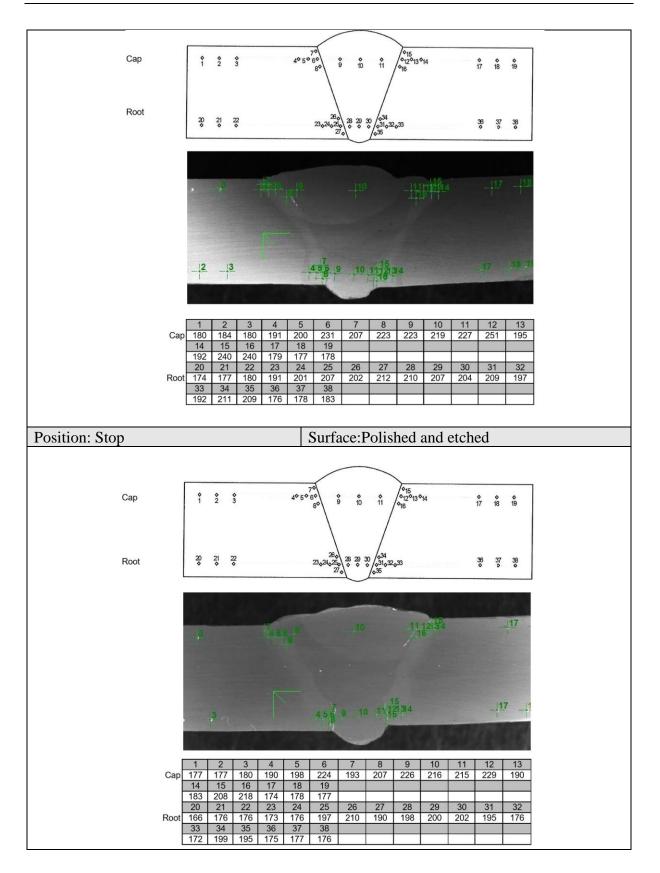


Table 7-26 - Side bend test on PL1.

Side ben	d test		
Former	Dimension [mm]	Bend angle [°]	Comment
4T	10	180	Accepted

Table 7-27 - Vickers hardness test data, HV10, for PL1 .

Vickers Hardness Test, HV10	
Position: Start	Surface:Polished and etched



7.8.2 Results from Stage D – Production welding of plates

Below are the remarks from the welding operation of the production plates. During welding the distortional forces from the elastic strain cracked the tack weld of the strongback supports on several welds. It is relevant information when determining the residual stresses, and the source for the stress distribution profiles.

Strongbacks was installed as seen in Figure 7-26. These figures should be read in conjuction with the deformation values in Table 7-20. Every tack weld has a unique ID and belongs to either w1 (weld A) or w2 (weld B).

Ex. ID w1:SB1.1: Strongback (SB) was installed for welding of weld A (w1) and is placed in the far left corner.

All groove angles for weld A was 60 degrees while the groove angle for weld B varied between 57-60 degrees, see table below.

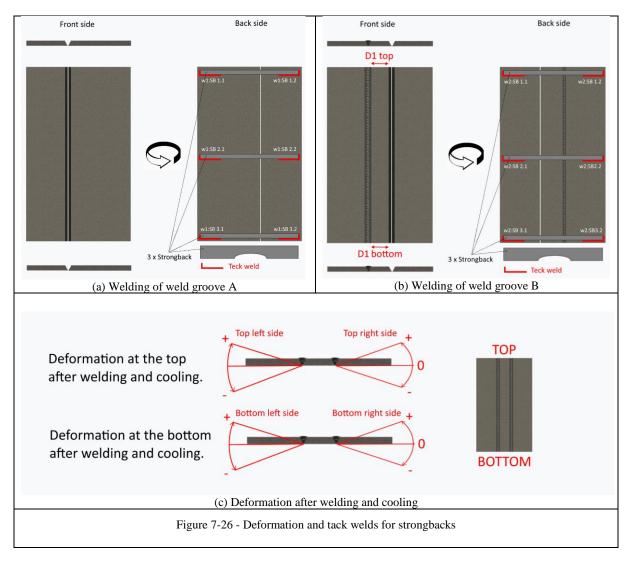


Table 7-28 - Strongback weld rapture PL3-PL8.

History of teck welds for weld 1 and weld 2

	SB 1.1	SB 1.2	SB 2.1	SB 2.2	SB 3.1	SB 3.2
PL3 Weld 1(1)(2)	ok	ok	ok	ok	-	ok
PL3 Weld 2 ₍₃₎	ok	ok	ok	ok	ok	ok
PL4 Weld 1	ok	ok	ok	ok	ok	ok
PL4 Weld 2	ok	ok	ok	ok	ok	- (4)
PL5 Weld 1	ok	ok	ok	ok	ok	ok
PL5 Weld 2	ok	- (5)	ok	- (6)	- (7)	ok
PL6 Weld 1	ok	ok	ok	ok	ok	ok
PL6 Weld 2	ok	ok	-	ok	ok	ok
PL7 Weld 1	ok	ok	ok	ok	ok	ok
PL7 Weld 2	ok	ok	ok	ok	ok	ok
PL8 Weld 1	ok	ok	ok	ok	ok	ok
PL8 Weld 2	ok	ok	ok	ok	ok	ok

Notes:

Strongback removed when plate had reached room temperatire, if not see extra notes.

OK – Strongback held

- Strongback weld rapture

Extra note:

(1) Strongbacks for weld A was removed before the plate temperature was 20 $^\circ \text{C}.$

(2) One strongback fractured during cooling, no remark was made which one it was.

(3) No strongback fractured but because of the design some bending deformation occured.

(4) Fractured after approx. 1h after welding.

(5) Tack weld fractured during welding of fill. Tack weld was repaired and held.

(6) Tack weld fractured during welding of cap. Tack weld was not repaired.

(7) No tack weld was welded, misstake.

Table 7-29 - Distance between weld joints

Distance between weld	joints (D1)				
	D1 Should be	Real distance	Real distance		
	[mm]	$D_{1 \text{ Top}} [mm]$	D _{1 Bottom} [mm]		
PL3	50	49	48		
PL4	50	50	50		
PL5	15	15	- (1)		
PL6	15	- (1)	- (1)		
PL7	5	7	- (1)		
PL8	5	3	- (1)		
			• • • •		

(1) Measurement was done after welding. It wasn't possible to measure D1.

Table 7-30 - Deformation	after welding a	nd cooling
--------------------------	-----------------	------------

Deform	ation after weldi	ng and cooling		GA Wed A ↓ D1 →	GA Weld B
	Top,	Top,	Bottom,	Bottom,	Groove angle,
	left side [°]	right side [°]	left side [°]	right side [°]	weld B [°]

PL3	+3	+2	+3	+1	58 (1)			
PL4	+3	0	+3 +0.5		57 (1)			
PL5	+2	+0.5	+3	+3	57 (1)			
PL6	+3	+1	+3	+2	57 (1)			
PL7	-1	-3 -1.5 -2 57						
PL8	L8 +1 0 +2 0 57 (1)							
(1) Before welding the weld groove B was tiltet 3 degrees to compensate for deformation.								
Note: M	easurements were	done with an degree	ee angle ruler.					

7.8.3 Results from Stage E – NDT, Mechanical Testing and Specimen Preparation

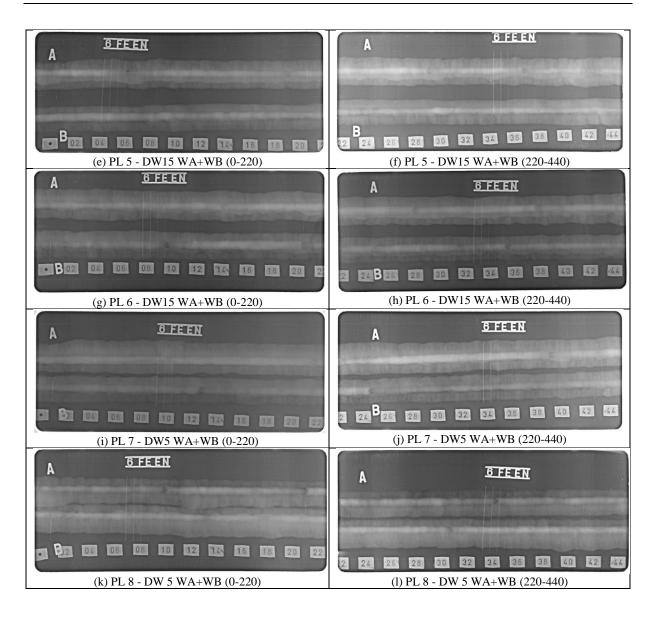
This chapter concerns the testing and approval process of the production plates with adjacent welds. In Table 7-22 we see what procedures, accept criterias and verifying documents that were used. All tests met the requirements, and has been verified by KIWA TI, IKM Testing and Qlab..

7.8.3.1 Radiographic testing

All welds were tested at radiographically tested at IKM Testing according with ISO 17636-2:2013. The examination was passed for all welds.

Plate	Weld No	Film location 0-220	Film location 220-440	Defect location	Remarks		
PL 3 - DW 50	Weld A	Accepted	Accepted				
	Weld B	Accepted	Accepted				
PL 4 - DW50	Weld A	Accepted	Accepted	270	517		
	Weld B	Accepted	Accepted				
PL 5 - DW15	Weld A	Accepted	Accepted	60	2011		
	Weld B	Accepted	Accepted				
PL 6 - DW15	Weld A	Accepted	Accepted				
	Weld B	Accepted	Accepted	100, 360	515,515		
PL 7 - DW5	Weld A	Accepted	Accepted	395	515		
	Weld B	Accepted	Accepted	130, 250	517, 517		
PL 8 - DW 5	Weld A	Accepted	Accepted	100, 340	517/2011, 517		
	Weld B	Accepted	Accepted				
A 22 23 23 (a) 1	<u>6 FFEN</u> 22 24 25 28 30 52 34 35 58 40 52 44						
(a) PL 3-DW 50 (0-220) W A + W B (b) PL 3 - DW 50 WA+WB (220-440) (c) PL 4 - DW50 WA+WB (0-220) (d) PL 4 - DW50 WA+WB (220-440)							

Table 7-31 - Radiographic test of PL3-PL8.



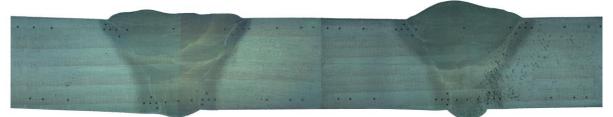
7.8.3.2 MPI examination

MPI was performed on all welds according to NS-EN ISO 17638:2016 and NS-EN ISO 3452-1:2013. The examination was passed for all welds. No findings of defects or impurities, specimens accepted.

7.8.3.3 Macro examination

Macro examination was performed on all plates according with ISO NS-EN 17639:2013. No weld imperfections was visible on the cross sections. Futhermore, an almost straight fusion line can be observed in all cases.

NOTE: The white lines on the macro specimens are purposely introduced on the surface with a a nail in order to being able to determine the fusion line at high magnifications. It was made by Qlab during examination.



(a) Macro start – PL3 Weld A left, Weld B right.



(b) Macro start – PL5 Weld A left, Weld B right.



(c) Macro start - PL7 Weld A left, Weld B right.

Figure 7-27 – Macro examination of PL3-PL8. Weld cross section 1.21-1.99 kJ/mm heat input as welded. The macro was polished and etched with Nital and were examinated and verified at Qlab, but the pictures were scanned at UIS.

7.8.3.4 Transverse Tensile Test

- Testing according to NS EN ISO 4136-1 / ISO 6892-1:2016 Method A.1
- Name one the tensile specimens are S1, F2, F3 och S4, see Figure 7-28.
- In Figure 7-30 we see the results from the tensile tests.
- Only specimen F was elongation controlled.

All specimens passed the test. Fracture was always in the base material.

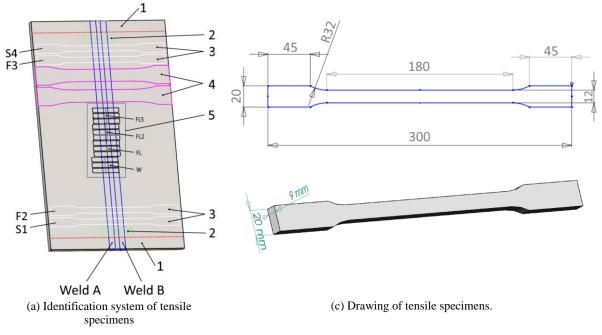


Figure 7-28 – Tensile identification system.

Transverse Tens	sile test					
Specimen No.	Acceptance criteria [MPa]	R _{eH} [MPa]	R _m [MPa]	R _{eH} / R _m	A5 [%]	Fracture
8612-2 S1	500-660	440.54	528	0.83	-	BM
8612-2 F2	500-660	458.94	535	0.86	23.03	BM
8612-2 F3	500-660	-	534		24.45	BM
8612-2 S4	500-660	444.78	525	0.85	-	BM
8612-3 S1	500-660	448.19	526	0.85	-	BM
8612-3 F2	500-660	-	537		28.06	BM
8612-3 F3	500-660	-	530		28.50	BM
8612-3 S4	500-660	442.26	517	0.86	-	BM
8612-4 S1	500-660	461.07	531	0.87	-	BM
8612-4 F2	500-660	-	534		28.13	BM
8612-4 F3	500-660	-	533		29.11	BM
8612-4 S4	500-660	455.23	525	0.87	-	BM

R_{eH} – Yield strength (420-540MPa) R_m – Tensile strength (500-660MPa)

BM - Base Material

A5 – Elongation, $L_0=5.65\sqrt{S_0}$ (Min. base material 19%.

This requirement is for the base material and not welds, see standard NS-EN 10225:2009

 R_{eH}/R_m – Yield ratio (Max: 0.93).

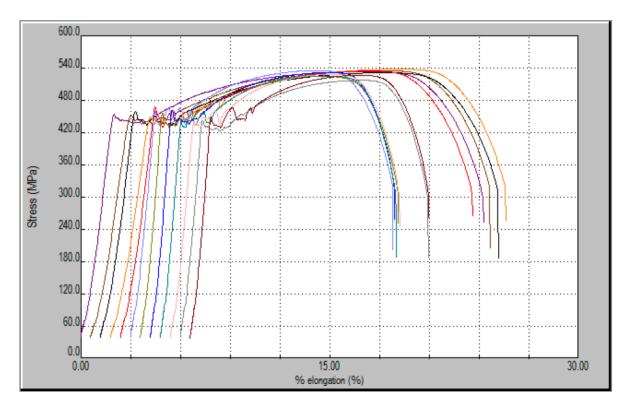


Table 7-32 - Results of tensile testing and acceptance criteria

Figure 7-29 - Tensile test graph



(e) PL7 – DW5

(f) Tensile specimens S1 and S4. Fracture in base material.

Figure 7-30 - Tensile test. I Figure 2 34 we see the dimensions of the tensile specimens.

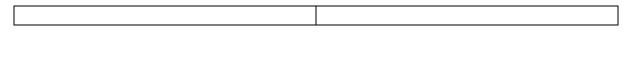
7.8.3.5 Charpy V Impact Toughness Test, KV₈

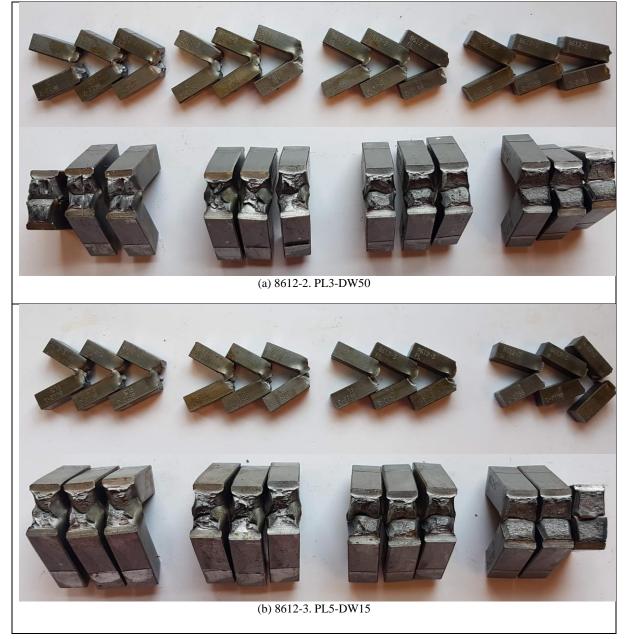
The samples from the Charpy tests are extracted from the initial weld, Weld A, in accordance with NS-EN ISO 148-1:2016, ISO 9016:2012. The locations where the specimen should be extracted is in the Weld, Fusion Line, Fusion Line +2 mm and Fusion Line +5 mm.

- Before testing, all V-notch specimens was controlled with a NO-GO gauge.
- The marking of the test pieces was performed on surfaces that weren't in contact with support, anvils or strikers. Plastic deformation and surface discontinuities caused by marking did not affect the absorbed energy.

Charpy V Impact Toughness test, KV ₈								
Plate	Test ident.	Dimension	Notch	Test	Single values [J] Avera			
		[mm]	Orientation	temp (C°)	1	2	3	[J]
Requirements in					Min.	Min.	Min.	Min.
MDS-S420G2+M					42	42	42	60
S420PL1-M (1)	Weld	10x10x55	Т	-40	80	131	106	106
PL3-DW50 (2)	Weld	10x10x55	Т	-40	121.3	120.1	112.2	117.9
PL5-DW15 (2)	Weld	10x10x55	Т	-40	100.4	84*	91.4	91.9
PL7-DW5 (2)	Weld	10x10x55	Т	-40	107.3	83.7*	81.3*	90.8
S420PL1-M (1)	FL	10x10x55	Т	-40	113	186	193	164
PL3-DW50 (2)	FL	10x10x55	Т	-40	186.3	123.5	163	157.6
PL5-DW15 (2)	FL	10x10x55	Т	-40	168.7	206.2	143.4	172.8
PL7-DW5 (2)	FL	10x10x55	Т	-40	202.9	280.8	141*	208.2

S420PL1-M (1)	FL+2 10x10x55		Т	-40	253	130	200	194	
PL3-DW50 ₍₂₎	FL+2	10x10x55	Т	-40	394.6	340.5	404.3	379.8	
PL5-DW15 (2)	FL+2	10x10x55	Т	-40	365.2	365.5	372.4	367.7	
PL7-DW5 (2)	FL+2	10x10x55	Т	-40	244.7*	354.3*	395.9	331.6	
S420PL1-M (1)	FL+5	10x10x55	Т	-40	234	236	173	214	
PL3-DW50 (2)	FL+5	10x10x55	Т	-40	196.2*	214.5	212	207.6	
PL5-DW15 (2)	FL+5	10x10x55	Т	-40	360.9	319.1	326.9	335.6	
PL7-DW5 (2)	FL+5	10x10x55	Т	-40	342.6*	305.9*	301.3*	316.6	
Weld A Weld B									
FL=Fusion Line, FL+2=Fusion line+2mm, FL+5=Fusion line+5mm, KV ₈ =, T=notch through the thickness (1) Qualification plate (2) Production plates *Broken (complete fracture)									





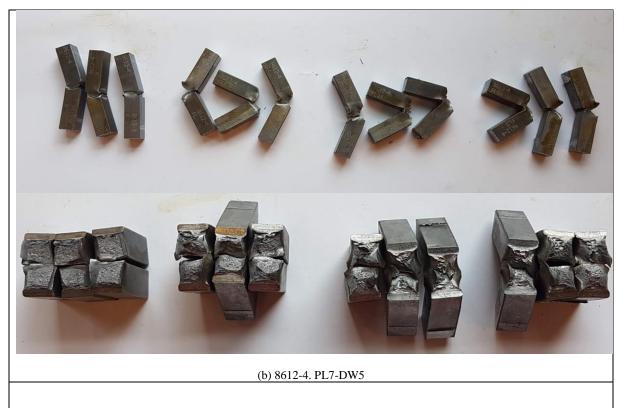
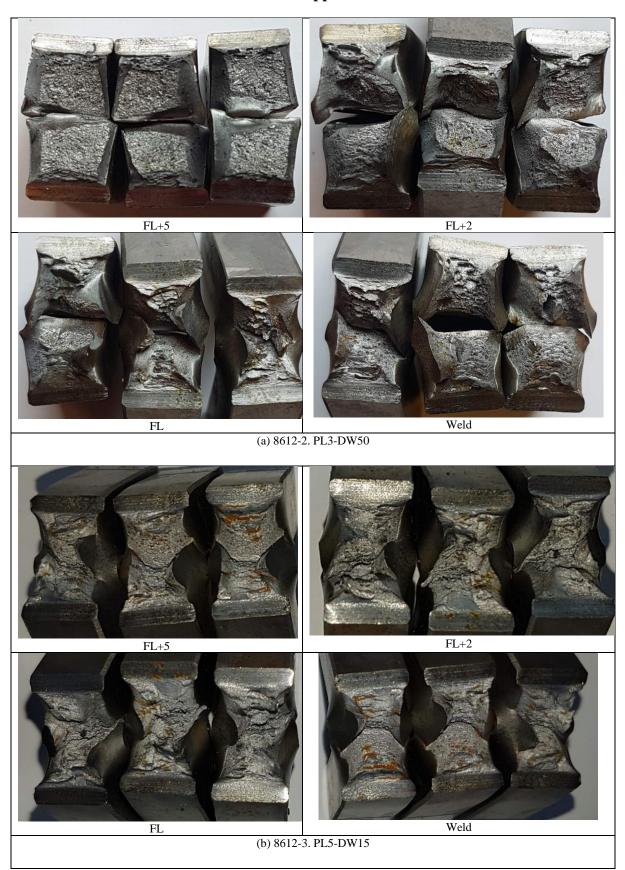
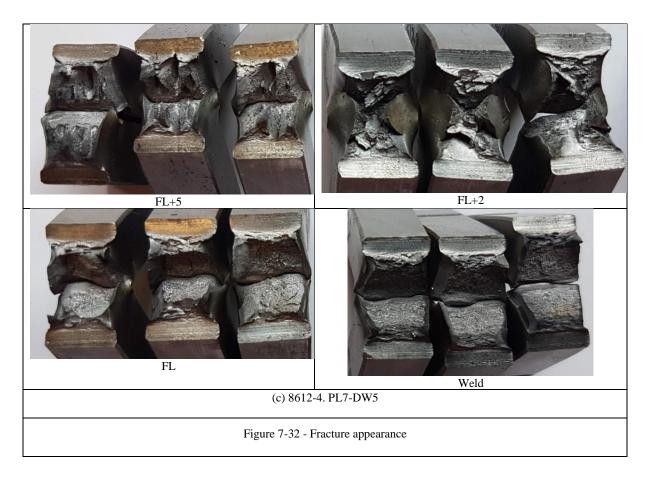


Figure 7-31 - Charpy V Impact test

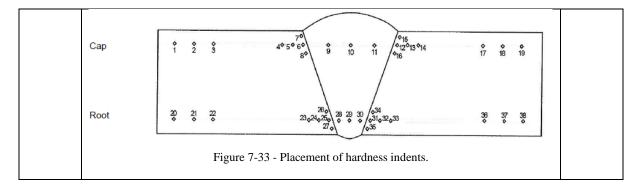


Fracture appearance



7.8.3.6 Vickers hardness Test

No indication of hardening in the microstructure.



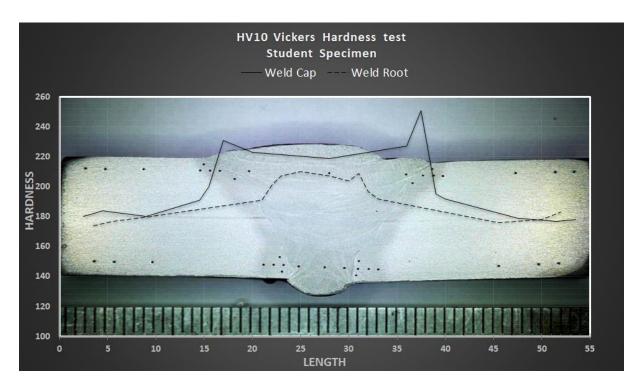


Figure 7-34 - HV Vickers Hardness Test Student Specimen.

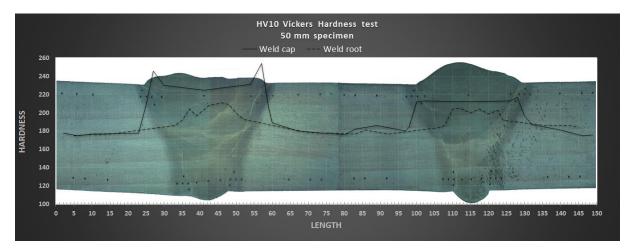


Figure 7-35 HV Vickers Hardness Test 44 mm between the welds.

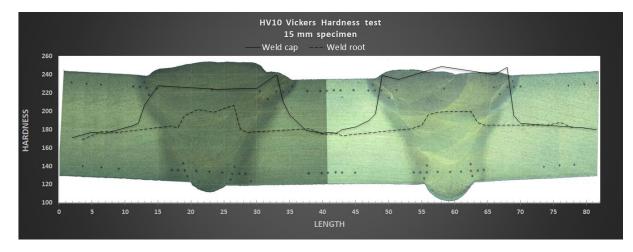


Figure 7-36 - HV Vickers Hardness Test 12 mm between the welds.

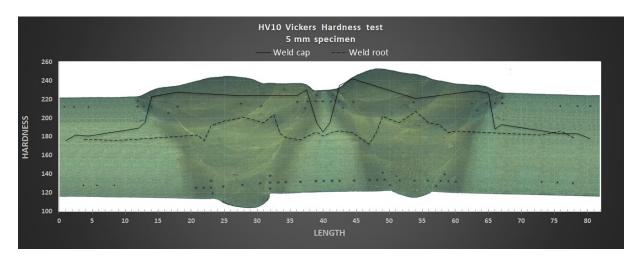


Figure 7-37 - HV Vickers Hardness Test 1.3 mm between the welds.

The results from the hardness tests are shown in Table 7-34 and Table 7-35. The locations for the indents are as shown in Figure 7-33.

Table 7-34 - Vickers hardness test results taken from the upper row in line with the weld cap. The welds are divided into weld A and Weld B. Weld A is the initial weld and Weld B is the secondary parallel weld. Weld DW = Dual Weld, SW = Single Weld, DW50 = Dual Weld 50 mm, DW15 = Dual Weld 15 mm, DW5 = Dual Weld 5 mm. NOTE DW5 Weld Cap A is missing indent 17-19 and Weld Cap B DW5 is missing 1-3 due to lack of base material in the area between the welds.

		CAP WELD A		CAP WE	LD B	TEST PLATE		
#	Description	DW50	DW15	DW5	DW50	DW15	DW5	SW
1	Base material	178	171	176	176	176	-	180
2	Base material	175	177	182	182	180	-	184
3	Base material	177	176	181	186	183	-	180
4	Fusion line + 2	227	184	189	207	190	-	191
5	Fusion line + 1	207	187	195	201	197	195	200
6	Fusion line	178	207	223	180	239	235	231
7	Fusion line Upper	209	209	205	184	219	242	207
8	Fusion line Lower	246	206	244	212	253	216	223
9	Center of Weld	230	228	228	213	235	224	223
10	Center of Weld	225	224	224	212	249	221	219
11	Center of Weld	231	225	224	213	240	229	227
12	Fusion line	246	240	230	213	248	227	251
13	Fusion line + 1	221	211	195	204	195	188	195
14	Fusion line + 2	254	196	185	217	187	193	192
15	Fusion line Upper	205	201	198	197	235	194	240
16	Fusion line Lower	189	229	217	187	251	219	240
17	Base material	180	179	-	181	183	183	179

18	Base material	178	176	-	175	182	183	177
19	Base material	178	177	-	176	180	178	178

		ROOT WE	ROOT WELD A		ROOT WE	LD B	SINGLE WELD	
#	Description	DW50	DW15	DW5	DW50	DW15	DW5	TEST PLATE
20	Base material	175	177	169	176	177	185	174
21	Base material	176	181	178	173	177	186	177
22	Base material	177	177	177	175	176	184	180
23	Fusion line + 2	204	205	184	181	182	172	191
24	Fusion line + 1	201	205	182	185	176	184	201
25	Fusion line	186	184	195	196	192	202	207
26	Fusion line Upper	192	187	206	197	200	197	202
27	Fusion line Lower	204	204	214	195	193	198	212
28	Center of Weld	197	205	202	199	197	195	210
29	Center of Weld	208	200	199	200	201	207	207
30	Center of Weld	211	204	203	200	195	195	204
31	Fusion line	207	200	207	200	204	193	209
32	Fusion line + 1	208	197	181	188	183	184	197
33	Fusion line + 2	209	199	177	185	179	186	192
34	Fusion line Upper	200	203	209	202	208	205	211
35	Fusion line Lower	193	192	208	196	203	207	209
36	Base material	182	186	181	185	177	182	176
37	Base material	179	186	175	188	185	186	178
38	Base material	178	184	175	182	181	179	183

Table 7-35 – Vickers hardness test from weld root

7.8.4 Summary and conclusion

The mechanical testing results from the NS-EN ISO/IEC 17025 accredited Qlab, indicated no degradation in material properties due to weld proximity. This statement was based on results from tensile tests, Charpy V Impact toughness test and Vickers hardness tests. The distance between the weld toes of the butt welded joints was 44, 12 and 1.3 mm.

All **Tensile Test** specimens failed in the BM outside the HAZ. This was as assumed since the UTS of the WM is higher than the BM. The result seemed to indicate that the influence of the adjacent secondary weld did not have a negative impact on the tensile strength properties of the initial weld. The fracture also initiated a considerable distance away from the weld showing no sign of reduction of tensile strength in the HAZ. The WM were machined flush with the BM.

The results from the **Charpy V Impact Toughness test** showed no reduction in toughness properties from the adjacent secondary weld. Due to inherent spread of the test results it was not possible to determine any distinct variation between the results at the different distances. All tests showed that the weld joint had toughness properties well above the requirements. The fracture appearance and lateral expansion was well within the limits. The test results are shown in Table 7-33.

The results from the **Vickers Hardness test** did not indicate any reduction in material hardness properties due to the adjacent secondary weld. As seen in Figure 7-37 the hardness values between a weld joint with 1.3 mm between the weld toes did not have any significant increase in hardness. Due to the spread in results in the WM and HAZ it was not possible to see if the secondary weld had any softening effect on the initial weld. The welds at 12 mm and 44 mm from weld-toe to weld-toe did not show any adverse effect due to the weld proximity, and showed a similar toughness pattern as the welds at 1.3 mm from eachother. More tests are needed in order determine if there is any difference between the various distances. Important to note also is that only one specimen at 1.3 mm was hardness tested. This is sufficient to approve a WPQ, but due to the nature of this test more tests are needed in order to ascertain the initial test results.

These results indicate that weld proximity as close as 1.3 mm does not have any degrading impact on the weld joint. This is based on level 2 mechanical testing in accordance with 15614-1:2017, used to approve WPQs.

It is important to note that these results are for this set of parameters and materials, but the findings can act as a baseline for further testing at various HI, welding positions, distances, etc.

8 Experimental Investigation: Microstructural examination

8.1 Objective

- Observe the microstructural properties of the weld and corresponding heat-affected zone.
- Determine any adverse microstructure in the area of overlapping.

8.2 Experimental procedure

Standardized transverse macro samples were extracted from the welded plates in order to study the microstructure in an optical microscope. The samples were taken from both the weld start and stop location and were large enough to contain both parallel welds in accordance with NS-EN 15614-1:2017 Specification and qualification of welding procedures for metallic materials [17]. After extraction, the samples were molded in epoxy and subsequently grinded and polished with a 1 μ m finish according to the procedure in Table 8-1. When the desired surface finish was reached, the samples were prepared for etching with 2 % nital.

Surface	Suspension	Lubricant	Time
SiC-Pap #80	-	Water	5 min
SiC-Pap #500	-	Water	3 min
SiC-Pap #4000	-	Water	5 min
Mol	DiaPro AllegroLargo 9 µm	-	20 min
Dap	DiaPro Dac 3 µm	-	30 min
Mol	DiaPro Nap-B 1 µm	-	20 min



Figure 8-1 - Struers grinding and polishing machine including TegraForce-5, TegraPol-35 and TegraDoser-5.



Figure 8-2 - Olympus GX53 Optical microscope.

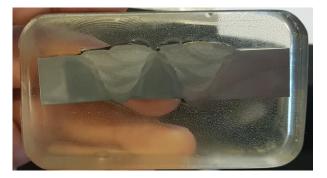


Figure 8-3 – Polished with 1 μm polishing cloth and subsequently etched with 2 % nital.

8.3 Results and discussion

8.3.1 Microscopic specimen 8612-1-SW-Start

This section presents the results obtained from the macro specimen 8612-1-SW-Start 25 mm from the weld starting location. The macrograph of the weld section is shown in Figure 8-4 and displays the weld passes and corresponding heat-affected zones. The multipass weld consists of six weld passes in accordance with the schematic diagram in the weld log Figure 8-4. Each multipass weld has used the same weld layer setup and using the flux-cored arc welding method, with exception from the root pass where metal-core arc welding was used instead.

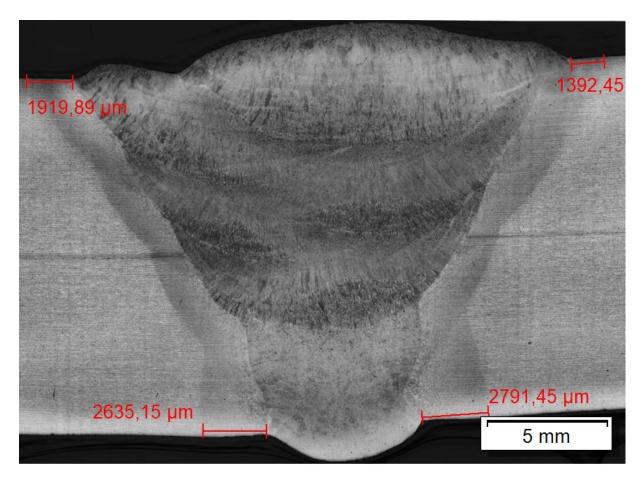


Figure 8-4 – Macrograph of a multipass weld with corresponding heat-affected zones. The specimen extracted from the plate the welding procedure qualification was based on with a single weld. Etching reagent: 2% nital.

The resulting approximate lengths of the heat-affected zones can be seen in Table 8-2 and corresponding weld log in Table 8-3. Each test plate has been welded in accordance with the same WPQR and been logged and documented.

Weld string, #	Heat input [kJ/mm]	Approximate length of HAZ [mm]
1 - Root	1.57	2.6-2.8
2 – Hot-Pass	1.21	1.6
3 - Fill	1.44	1.5
4 - Fill	1.97	1.7
5 – Weld Cap	1.68	1.9
6 – Weld Cap	1.61	1.7

Table 8-2 - Heat input and corresponding lengths of heat-affected zones in weld qualification plate.

Table 8-3 - Weld log from test plate.

Welding procedure qualification record procedure



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Ò

1.99

1.97

1.44

1.46

1,21

1.32

1.57

1.35

WPQR No .: 5420 PL1 - M

3 1

4 1

6 1

Fich

Cab

1.2

1,2

1.2

1.2

1.2

1.2

1.2

7.1

7.1

7.1

7.1

7.1

7.1

7.1

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Det

Det

Det

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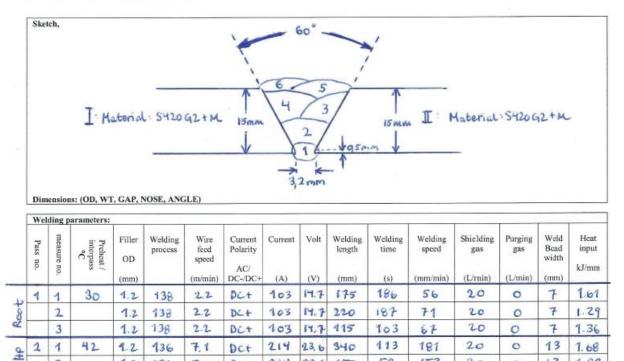
DC+

23.6

23.6

23.6

23.6



The Figure 8-5 below shows an extruded view of the weld toe outside the final weld cap. The
various sub-regions in the HAZ are visible and has a length of approximately 1.7 mm.

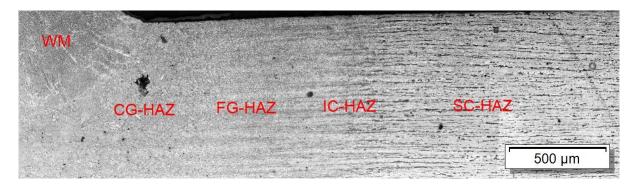


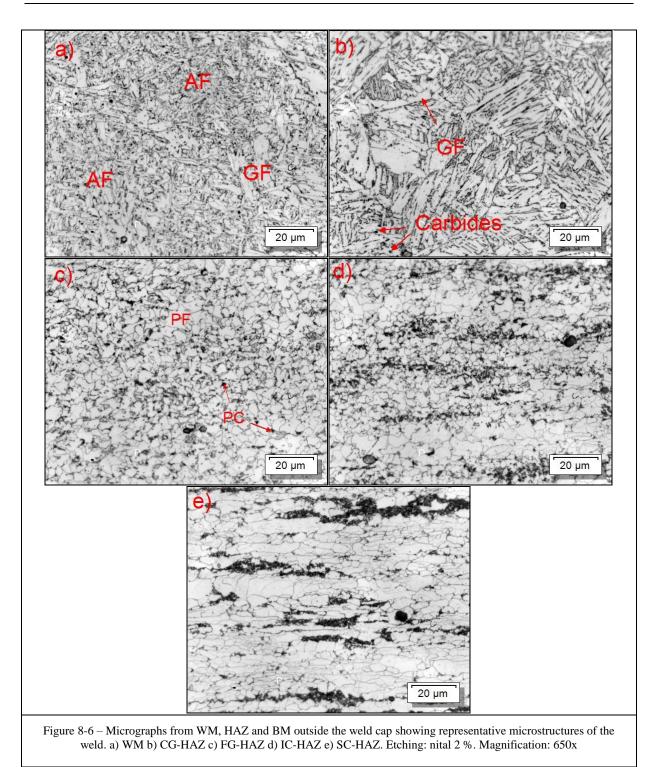
Figure 8-5 - Heat-affected zone outside the weld toe. The weld metal, coarse-grained HAZ, fine-grained HAZ, tempered and partially austenitised HAZ and unaffected base material is visible.

8.3.1.1 Microstructures in the weld metal, HAZ and base material of test plate

The Figure 8-6 shows the microstructure of the weld metal (WM), coarsed-grained heat-affected zone (CG-HAZ), fine-grained heat-affected zone (FG-HAZ), intercritical heat-affected zone (IC-HAZ) and sub-critical heat-affected zone (SC-HAZ).

Classification of microstructures:

- Grain boundary (or allotriomorphic) ferrite (GF)
- Polygonal (or equiaxed) ferrite (PF)
- Widmanstätten ferrite (WF)
- Acicular ferrite (AF)
- Pearlite (PC)



8.3.1.2 Weld metal

The microconstituents in the weld metal constituted mainly of grain boundary (or allotriomorphic) ferrite (GF), acicular ferrite (AF), polygonal (or equiaxed) ferrite (PF), Widmanstätten ferrite (WF), ferrite plus carbides and as well as pearlite (PC).

Figure 8-7 is showing the distribution between the microconstituents in the weld metal based on carbon content.

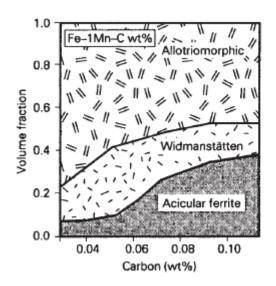


Figure 8-7 - Graph showing the prevalence of the different phases in a primary microstructure as a function of carbon content [7].

Side plate Widmanstätten ferrite and allotriomorphic ferrite reveals the prior austenite grain structure in Figure 8-8. These grains have a columnar grain structure in the direction of the heat gradient.

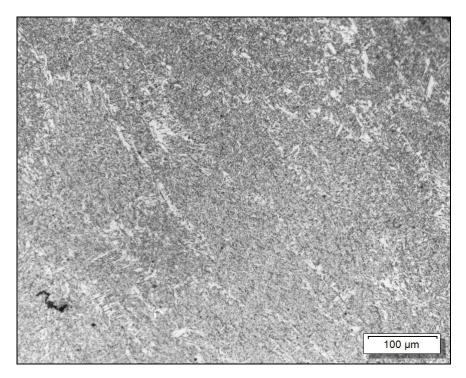


Figure 8-8 - The prior austenite grain profile in the weld metal.

8.3.1.3 Coarse-grained HAZ

As with the case of lower CE-steels the CG-HAZ is predominately proeutectoid ferrite networks (white phases) in the grain boundaries as allotriomorphic ferrite (GF) and polygonal ferrite (PF).

The grain size in this area is due to the high temperatures over 1200 °C, allowing for a complete austenitization of the microstructure despite being subjected to these temperatures only for a short moment [8].

8.3.1.4 Fine-grained HAZ

The FG-HAZ consists of refined ferrite and pearlite in the form of polygonal ferrite (PF) and fine pearlite FC (P). This is the result of a recrystallization during the heat cycle. The region has completely austenitised, but at a lower peak temperature and cooling rate. This inhibited the growth of large grains, but instead resulted in smaller grains that are more refined. The following phase transformation from austenite to proeutectoid ferrite and posterior pearlite is followed by a process of intensive nucleation rate with slow rate of growth, resulting in a microstructure with small grain size, lower than the original base material [44].

8.3.1.5 Intercritical HAZ

The IC-HAZ is formed in the range between the A_{c1} to A_{c3} curves. In comparison to the CGand FG-HAZ this region only gets partially austenitised. This process results in the pearlite and only some of the ferrite will undergo a transformation to austenite as seen in Figure 8-6. Depending on the cooling rate, this region can form martensite when transforming from austenite. This is due to the austenite that is formed has a higher carbon concentration [44]. During microscopic examination however no martensite was found in this region and the only microconstituents remaining was equiaxed ferrite and pearlite.

8.3.1.6 Sub-critical HAZ

The SC-HAZ is the region affected by temperature below A_{c1} . The area experiences tempering during the heat cycle which results in the forming of spheroidization, also known as degenerated pearlite. No phase transformation occurred in this region. The ferritic and perlitic banded microstructure comes from the hot rolling process of a hypoeutectoid steel.

8.3.1.7 Weld root WM and HAZ

The filler metal used in the weld root was metal-cored instead of flux-cored as seen in Figure 8-9. The distance of the HAZ was further than at the weld cap. This could be due to the localized heating during the heating cycle due to the joint bevel configuration. The grain structure did not have any columnar growth properties.

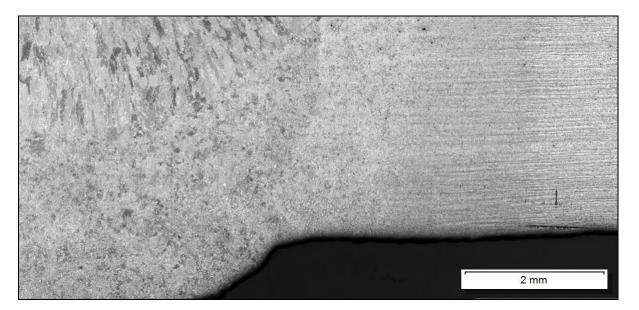


Figure 8-9 - Weld root from test plate specimen.

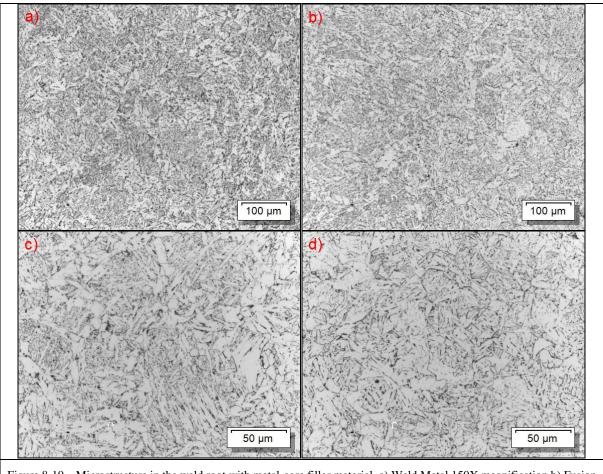


Figure 8-10 – Microstructure in the weld root with metal-core filler material. a) Weld Metal 150X magnification b) Fusion boundary 150X c) Weld Metal 300X and d) CG-HAZ at 300X.

8.3.2 Results and discussion - Microscopic specimen 8612-4 PL7

The macrograph of the common HAZ between weld A and weld B is shown in Figure 8-11. The region had experienced a complete overlap of heat-affected zones. The distance between the weld toes were 1.3 mm with a depth from top surface about 1.7 mm. No detrimental microconstituents were observed in the region that would affect the microstructural properties. Microstructural grain structure did not seem to contain any brittle phases. The region appeared to have experienced a tempering and partially austenitizing effect from the welding procedure.

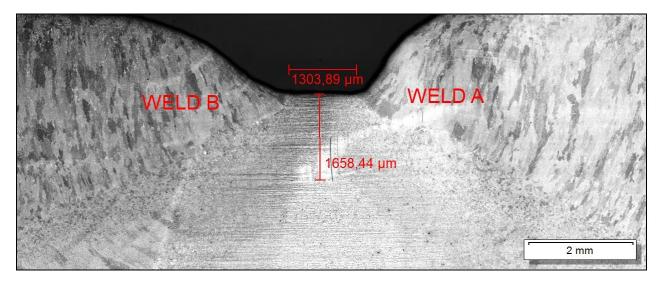


Figure 8-11 - Overlapping HAZ in specimen 8612-4 PL7. Weld A is the initial weld and Weld B is the secondary overlapping weld. Distance between the weld toes are 1.3 mm and HAZ are overlapping 1.7 mm from the top surface.

Figure 8-12 below is showing the fusion boundary of weld A. It did not seem like the coarsegrained region had experienced any brittle transformation due to reheating to the region between A_{c1} and A_{c3} .

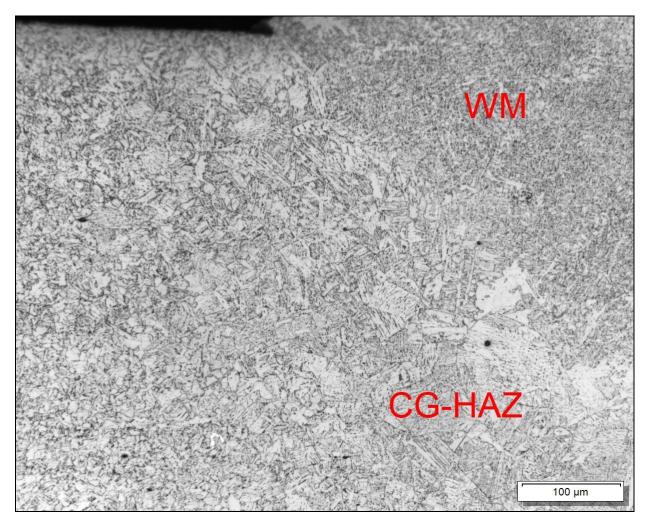


Figure 8-12 - Fusion boundary of weld A.

Figure 8-13 is showing the fine-grained and partially austenitized HAZ. This region still had the banded grain structure from the initial hot rolling after being subjected to two heating cycles.

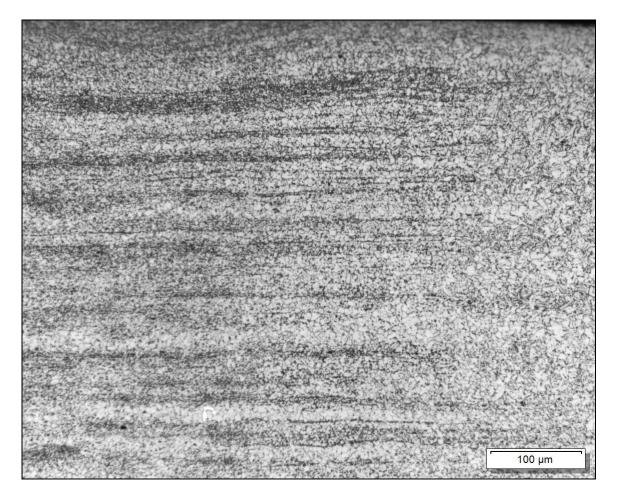


Figure 8-13 - Overlapping heat-affected zones between weld toes. The initial HAZ has experienced a normalizing effect from the overlapping HAZ.

Figure 8-14 shows the root of the weld joint. Due to the distance between the regions, the heat-affected zones had no effect on each other.



Figure 8-14 - Distance between weld toes was approximately 16 mm.

8.3.3 Results and discussion - Microscopic specimens 8612-3 PL5 and 8612-2 PL3

Figure 8-15 shows the heat-affected zones from the 8612-3 PL5 sample. There was no sign of interaction between the two heat-affected zones at this distance between the two welds. The

microstructure was unchanged in both instances. The same can be stated about 8612-2 PL3 that had approximately 50 mm between the weld toes. At these distances, the heat generated from the heat source will not be sufficient to have any impact on the microstructure.



Figure 8-15 - Common HAZ of 8612-3 PL5. There was no interaction between the two heat-affected zones from weld A and weld B when the distance between the weld toes was 12 mm.

8.4 Chapter conclusion

The microconstituents in the weld metal and HAZ are typical, and a common combination in the industry. In only one specimen did the weld HAZ overlap. This was the weld with 1.3 mm between the weld toes. No visible harmful phases were detected with the optical microscope in either the WM or the HAZ. The impact of the overlap only seemed to have a normalizing effect of the previous HAZ of weld A.

Between subsequent weld passes and unaffected region of the coarse-grain HAZ microstructure can be distinguished. The reheated pockets of the CG HAZ regions are small and discontinuous, which make their microstructure difficult to identify and investigate. The reheated IC CG HAZ is very susceptible to failure, due to the fact that the phase transformation into austenite began on the grain boundaries; these small areas are rapidly cooled and have a strong possibility to become hard and brittle. The subcritical regions are mainly tempered bainite and martensite with precipitated carbides and therefore represent no danger to the structural integrity of the weld.

The welds with a distance of 12 mm and 44 mm between the weld toes had no overlap of HAZ. There never was any overlap of visible HAZ of the weld roots, despite the larger area of effect.

Important to note though is that the magnification of the optical microscope is limited and further analysis with a scanning electron microscope (SEM) and a transmission electron microscope (TEM) is necessary in order to assure that this is the case. Investigation in these microscopes are needed in order to confirm the exact nature of all phase transformations occurring in the overlapping HAZ.

9 Experimental Investigation: Fatigue Testing

This chapter will present the experimental procedure and results from the fatigue test conducted at the University of Stavanger. Three different fatigue test series were performed as part of the experiment. In the first test the main objective was to test the viability of the experiment and to gain a basic understanding of the mechanisms involved in fatigue testing. The purpose of the second test was to determine a rough SN-curve for the base material. In the final test the welded production plates were tested and analyzed.

9.1 Specimen appendix

Specimen Serie	Fatigue test log	Roughness test	Pre-test visual examination	Appendix
Prior to fabrication:				
F1	Yes	No	No	Appendix G
F2	Yes	No	No	Appendix G
F3	Yes	No	No	Appendix G
F4	Yes	No	No	Appendix G
F5	Yes	No	No	Appendix G
Unwelded base material:	Fatigue test log	Roughness test	Pre-test visual examination	
BM1	Yes	Yes	Yes	Appendix G
BM2	Yes	Yes	Yes	Appendix G
BM3	Yes	Yes	Yes	Appendix G
BM4	Yes	Yes	Yes	Appendix G
BM5	Yes	Yes	Yes	Appendix G
Production:	Fatigue test log	Roughness test	Pre-test visual examination	
8612-2 PL3 2A(B)	Yes	Yes	Yes	Appendix G
8612-2 PL3 2A(BB)	Yes	No	Yes	Appendix G
8612-3 PL5 A3	Yes	Yes	Yes	Appendix G
8612-3 PL5 3A	Yes	Yes	Yes	Appendix G
8612-4	Yes	Yes	Yes	Appendix G

Table 9-1 - Related information in appendix.

9.2 Equipment specifications

The fatigue testing machine was installed in the workshop of the Department of Mechanical and Structural Engineering and Materials Science at the University of Stavanger. The machine specifications are shown in table 9-2.

Table 9-2.

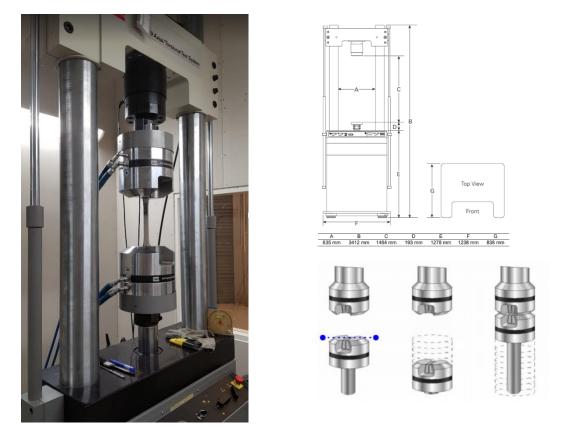


Figure 9-1 – MTS 809 Axial/Torsional Test System and dimensions. The lower grip is the actuator and the upper grip is static while testing, but can be manually moved by adjusting the overhead crossbeam.

	MTS 809 Axial-Torsional Test System						
Model 319.25 Axial-Torsional Load Frame							
Dynamic force rating	±250 kN						
Dynamic torque rating	±220 Nm						
Axial displacement	±75 mm						
Rotation	$\pm 45^{\circ}$ (Theoretical – In practice, the hydraulic hoses are too short to allow a						
	dynamic rotation larger than 15°.)						
Controller hardware	FlexTest 40						
Wedge Grip Model	647.25 Axial-Torsional						
Grips for plate specimen	Thickness of plate 1-11.9 mm and Grip depth: 89 mm						
Grips for Compact Tension	Part number D071727-45, model number 640.20B-21. Designed for specimens						
specimen	with $B = 12.7$ mm, $W = 50.8$ mm and hole diameter $D = 12.7$ mm according to						
	ASTM E1820.						

Table 9-2 - Specifications fatigue testing machine MTS 809 Axial-Torsional Test System

9.3 Fatigue test – Prior to fabrication test specimens

9.3.1 Objective

- Test the viability of the experiment
- Gain an understanding of the different factors influencing the fatigue life.

9.3.2 Experimental procedure

In the initial stages of the project, a 10 mm S355 test plate was prepared in order to determine the viability of the test. This operation was performed at Vest Norge Doors AS in Sandnes. The plate was initially cut into three parts using a metal cutting band saw and subsequently grinded until the weld groove was 60° with 2.5 mm root spacing with an angle grinder. The width of the middle section was estimated in order to obtain a weld toe distance of approximately 50 mm. The plates were subsequently welded by shielded metal arc welding. No weld logging was performed at this stage due to a lack of necessary equipment.



Figure 9-2 - Initial test plate welded at Vest Norge Doors AS, Sandnes.

The finalized plates where then transported to the University of Stavanger for machining and subsequent tensile and fatigue testing.



Figure 9-3 – Finished prior to fabrication test specimens.

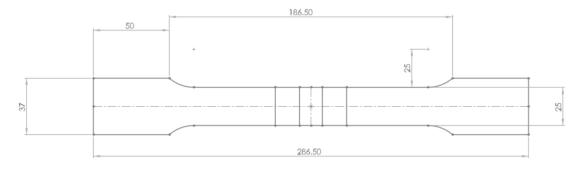


Figure 9-4 – Dimensions of prior to fabrication specimens.

9.3.3 Results and discussion

The results obtained from the fatigue tests are presented in

Table 9-3.

Fatigue Test R=0.1									
Prior to fabrication test plate fatigue specimens									
Specimen ID	F1	F2	F2	F2	F3	F4	F5		
Test Number	1	3,6	11	11	5	4	2		
Run sequence	1 st run	1 st run	2 nd run	3 rd run	1 st run	1 st run	1 st run		
Stress range [MPa]	481	200	300	350	300	300	481		
Area [<i>mm</i> ^2]	205.0	201.3	193.44	193.44	220	212.5	205.0		
Max load [MPa]	534.4	222.2	333.3	388.9	333.3	333.3	534.4		
Min Load [MPa]	53.4	22.2	33.3	38.9	33.3	33.3	53.4		
Mean load [MPa]	293.9	122.2	183.3	213.9	183.3	183.3	293.9		
Machine Load [MPa]	293.9±240.5	122.2±100	183.3±150	213.9±175	183.3±150	183.3±150	293.9±240.5		
Cycles	12 195	9 659 157	2 300 000	747 488	924 104	312 278	38 570		
Displacement range	-	0.3	0.40	0.47	0.48	0.46	-		
Run out/Fracture	Fracture	Run out	Run out	Fracture	Fracture	Fracture	Fracture		
Specimen ID	Description		•	•	•	Reference	es		
F1	Fracture at 48					Figure 9-5	5		
			scalculation of						
	resulted in an extreme overload with plastic fracture as a result. The								
	specimen faile								
F2			59 157 cycles.			Figure 9-5	,		
			00 000 cycles.						
			7 488 cycles (r						
			nanual grinding						
	the weld root.	geometric stre	ess concentration	on leatures. Cra	ick initiated in				
F3	Fractured at 3	00 MPa after 9	24 104 cycles			Figure 9-5	5		
15			aterial. Likely	cause for the c	ack to initiate	I igure y=c	,		
			ılar misalignme						
			tress. In additio						
					sulted in some				
	undercut.								
F4	Fracture at 30	0 MPa after 31	2 278 cycles.			Figure 9-5	5		
			obable cause of	f crack initiatio	n is due to	U			
	stress concent	ration in the w	eld toe.						
F5	Fracture at 48	1 MPa after 38	570 cycles.			Figure 9-5	5		
			scalculation of	the specimen a	area. Fracture				
	occurred in the	e root of the w	eld joint.	-					
F6	Fracture due t	o tensile test.				Figure 9-5	5		
		tensile strengt	h in the steel w	as 410 MPa an	d 540 MPa				
	respectively.								

Table 9-3 – Prior to fabrication test plate fatigue test specimens

In Figure 9-5 the fractured specimens with corresponding stress range and number of cycles at fracture. Five fatigue tests and one tensile test were conducted. In the tensile test the specimen failed in the base material, which is supported by Maddox for one buttweld in [11]. A transverse buttweld subjected to transverse loading during static loading conditions will under normal conditions fail in the weld metal and will experience no reduction in strength due to the weld. However, if the weld is subjected to fatigue stress the result will be different. Fatigue failure consists of crack initiation at one or more weld toes, with following crack propagation normal to the direction of loading due to stress concentrations, including the effect of weld toe intrusions. Complete removal of the weld cap and weld toe intrusions can restore the weldment

to a state where fatigue failure instead occurs in the base material itself. The presence of weld flaws in this region may be significant.



Figure 9-5 - Fractured specimens from the prior to fabrication test plate and tensile tested specimen. Specimen F1-F5 fractured in the weld metal or HAZ during fatigue testing. Specimen T6 fractured in the base material after being subjected to a tensile test.

9.3.4 Appendix G

- Doc-G701Fatigue test Prior to fabrication specimens
- Doc-G702Fatigue test Unwelded base material specimens
- Doc-G703Fatigue test Production test plate specimens
- Doc-H800The chapter "Residual Stress Measurements" has no attachments.

9.3.5 Summary

The implications of stress concentrations and misalignments in the specimens was the dominating features that resulted in fracture of the specimens. All but one fatigue specimen failed in the weld in either the weld cap or the weld root. The fatigue life improved substantially by improving the surface finish and removing all major stress concentrations.

9.4 Fatigue test – Unwelded base material plate specimens

9.4.1 Objective

- To gain an understanding of the fatigue properties of a pure base material sample from S420G2+M steel.
- To approximate a SN-curve based on the results.

9.4.2 Experimental procedure

In order to determine the appropriate stress range for the main welded specimens, some fatigue tests were conducted on pure base material. The preparation was performed accordance to the requirements in ASTM E466-15 [45]. The standard covers the procedural steps in order to design a specimen in the fatigue stress range where the strains are predominately elastic. It is limited to axially loaded specimens subjected to constant amplitude loading at ambient temperature. The specimen can be either notched or unnotched. It is designed for measuring the effects of variations of material, geometry, surface condition and residual stress of metallic specimens at a rather large amount of cycles.

Careful considerations of the procedural steps are vital in order to verify that the results are viable and reproducible. To achieve this, a tight control of variables is necessary; such as hardness, cleanliness, grain size, composition, directionality, surface residual stress, surface finish, etc. All obtained data should be logged and stored for future evaluation [45].

The design of the specimen dimensions should be such that the eventual failure occurs in the reduced area in the test section. It is therefore vital when reducing the area of the test section, that the radius introduced from the machining doesn't cause any detrimental stress concentrations. In addition, using a square or rectangular cross section might reduce the fatigue life of the specimen, due to reduced resistance to plane slippage at the edges. In a circular cross section, the surrounding grains confines the material and inhibits the process and might increase fatigue life [45].

The specimen dimensions were designed as a rectangular cross section with tangentially blended fillets between uniform test section at the ends. The radius of the specimen is supposed to be eight times the specimen test section width to minimize the stress concentration. The specimen test section width should be 2-6 times the thickness, whereas the resulting area should lie in between 19.4-645 mm². The test section length should be 2-3 times the test section width. The width of the grip should be 1.5 times the test section width. The length of the whole specimen was 300 mm, resulting in a grip length of 40 mm on each side.

This test dimension is designed for pure base material specimens, but due to the following tests had to fit two welds, some adjustments had to be made of the original design. In order to fit both welds in the test specimen section the length had to be increased. The finished design of the specimen was as shown in Figure 9-6.

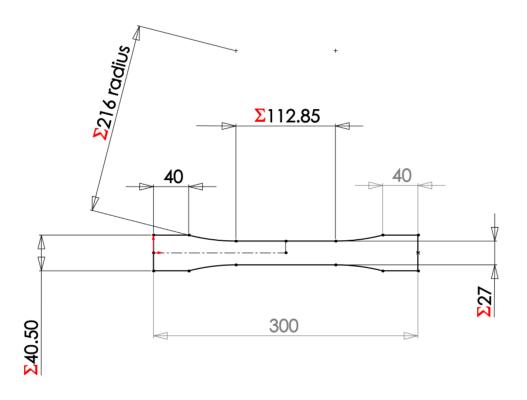


Figure 9-6 - Specimen dimensions according to ASTM E466.

The following procedural steps was taken to machine and prepare the said specimen. according to ASTM E466-15 Appendix X1. Example of the machining procedure:

- 1. X1.2.1 Machining was done gradually where the second-last step was 0.4 mm and last 0.2 mm.
- 2. X1.2.2 Removed the next 0.1-0.2 mm on the front and backside of the specimens with cylindrical grinding (plane Surfaces Grinding Machine). It was not possible to use the plane surface grinding machine on the sides of the samples, so this was done later with abrasive paper.
- 3. X1.2.2 The final grinding was done manually with abrasive paper. Sanding was done with Hermes WS Flex Waterproof P180, Struers FEPA P # 500, Hermes WS Flex Waterproof P1000, Struers FEPA P # 1200 and Silicon Carbide 1200/4000
- 4. X1.2.4 Requirements after grinding were that all slip marks would be along the test direction of the test specimens. Finally, a visual check was made with a magnifying glass, where no transverse grinding marks were accepted. A visual log was made where all errors were written down.
- 5. A roughness check was carried out. Requirements for Maximum surface roughness were 0.2um in the longitudinal direction
- 6. After surface treatment and control, specimens were lubricated into grease while waiting for testing. Specimens were stored in towels.



Figure 9-7 - Finished test specimen sample.

9.4.3 Results and discussion

The results from the tests on the pure base material fatigue specimens are as shown in Table 9-4.

Fatigue test R=0.1 Unwelded base material plate specimens									
Specimen ID BM1 BM2 BM3 BM4 BM5									
Test Number	8	7	16	10) 9				
Run sequence	1 st run	1 st run	1 st run	1 st rur	n 1 st run				
Stress range [MPa]	430	350	400	375	5 400				
Area [<i>mm</i> ^2]	237.5	234.8	240.14	233.9	228.2				
Max load [MPa]	477.8	388.9	444.4	416.7	7 444.4				
Min Load [MPa]	47.8	38.9	44.4	41.7	7 44.4				
Mean load [MPa]	262.8	213.9	244.4	229.2	2 244.4				
Machine Load [MPa]	262.8 ± 215	213.9 ± 175	244.4 ± 200	229.2 ± 187.5	5 244.4 ± 200				
Cycles	15 643	2 760 746	394 701	6 770 000	426 615				
Displacement range	0.93	0.63	0.82	0.70	0.77				
Run out/Fracture	Fracture	Run out	Fracture	Run ou	t Fracture				
Specimen ID	Description			R	References				
BM1	Fracture at 430	after 15 643 cycles							
	Plastic fracture.	Low-cycle fatigue ra	inge.						
BM2	Run out at 350 I	MPa after 2 760 746	cycles.						
BM3	Fracture at 400 MPa after 394 701 cycles.								
	Crack initiated in the side face.								
BM4	Run out at 375 I	Run out at 375 MPa after 6 770 000 cycles.							
BM5		MPa after 426 615 cy	/cles.						
	Crack initiation	at surface face.							

Table 9-4 - Results from fatigue test of unwelded base material plate specimens.

For a crack to initiate, the stress range had to be above yield point level. At a stress range of 375 MPa, which equals to 417 MPa in maximum load due to stress ratio R=0.1, the test ran out at 6 770 000 cycles. The yield point of the metal was approximately 430 MPa which means that the specimen was subjected to dynamic loads equaling 97 % without fracturing.

Three tests were loaded until fracture; one at 430 MPa and two at 400 MPa. With stress ratio R=0.1 the resulting maximum load was 478 MPa and 444 MPa. The specimen at stress range 430 MPa failed prematurely in the low-cycle fatigue stress range.

The final crack initiation in the high-cycle fatigue specimens may have nucleated in intrusions due to plane dislocation slippage. The fatigue life of the high-cycle fatigue specimens consisted mainly in the crack initiation stage. When a crack finally was initiated the subsequent crack

propagation growth rate was extremely fast. It wasn't possible to detect any visible crack before fracture, despite regular check-up during testing.



Figure 9-8 - Unwelded base material fatigue specimens after testing.

9.4.4 Summary

The main fatigue life of the unwelded base material test specimens consisted of the crack initiation stage. This was increased even further by following the procedural steps in ASTM E466-15. The surface finish greatly improved the fatigue life.

These tests stands in direct contrast to the previous fatigue tests on the welded "prior to fabrication specimens" having reduced fatigue strength due to discontinuities and defects in the weld.

To create an approximated SN-curve based on the testing results was not possible due to the limited sample size. It was also not advisable to conduct further fatigue tests on the used specimens at an increased stress range due to uncertainties from the influence of strain hardening.

9.5 Fatigue test - Production plate specimens

9.5.1 Objective

- Determine if the fatigue strength of the microstructure is reduced by overlapping the heat-affected zones of two parallel welds.
- In order to focus on the microstructure, it is necessary to design the fatigue test specimen such that the test excludes the influence of geometrical stress concentrations, weld discontinuities as well as linear and angular misalignments.
- Determine the location of crack initiaion.

9.5.2 Experimental procedure

The fatigue specimens were extracted from the welded production plates in chapter 7. The objective of the test was to see if overlapping heat-affected zones influences the fatigue behavior in the absence of stress concentrations due to geometry, weld discontinuities, angular and linear misalignments. In order to achieve this, it was decided to remove any influence from stress concentrations. It was decided to machine the top and bottom surface of the specimens before fatigue testing, mainly from the bottom face. The fatigue machine grip limit was 11 mm and the thickness of the plate was 15 mm so 4 mm had to be removed from the bottom face. This also removed the weld cap and introduced compressive stresses in the weld surface.

Some angular misalignment due to distortion was observed after welding as documented in Table 9-5 and in figures below.

Welding Distortion in [°]									
Plate ID	8612-2 PL3	PL4	8612-3 PL5	PL6	8612-4 PL7	PL8			
Bottom Weld A	3	3	3	3	-1.5	2			
Bottom Weld B	1	0.5	3	2	-2	0			
Top Weld A	3	3	2	3	-1	1			
Top Weld B	2	0	0.5	1	-3	0			

Table 9-5 - Welding distortion in production plates



Figure 9-9 – Weld distortion after second weld PL3.



Figure 9-10 - Weld distortion after second weld PL5.



Figure 9-11 - Weld distortion after second weld PL7.

The specimen was machined in sequence as shown in Figure 9-12. Initially the top face was machined down until parallel and subsequently the bottom face was machined until the surface was parallel to the top. In order to minimize the effect on the microstructure the machining was performed at lower speeds and with an increase amount of coolant.



Figure 9-12 – Machining of fatigue specimens from welded S420G2+M plates in order to remove stress concentrations and misalignments the top face was initially machined until parallel followed by the machining of the bottom face.

The specimen dimensions were the same as for the unwelded pure base material specimens as seen in Figure 9-6 and method of preparation was also the same.

9.5.3 Results and discussion

The sample size from the fatigue testing of the welded production plates are presented in Table 9-6. Due to the distortion of the plate 8612-4 PL7 DW5 the machining of the top face resulted in the removal of the critical area of the conjoining HAZ. Due to time constraints it was not possible to machine more specimens that had this critical distance between the welds.

Fatigue test samples							
Specimen ID Sample size		Distance between welds					
8612-2 PL3 DW50	2	44 mm					
8612-3 PL5 DW15	2	12 mm					
8612-4 PL7 DW5	1	1.2 mm					

The results from the fatigue tests are shown in Table 9-7 and

Table 9-8.

		Fatigue te Production pla					
Specimen ID	8612-2 2A (B)	8612-2 2A (B)	8612-2 2A (BB)	8612-3 A	3	8612-3 A3	
Weld distance	44 mm	44 mm	44 mm	12 mm		12 mm	
Test number	12	13	17	14		14	
Run sequence	1 st run	2nd run	1 st run	1 st run		2 nd run	
Stress range [MPa]	375	400	400		375	400	
Area [<i>mm</i> ^2]	232.0	232.0	243.1		235.25	235.25	
Max load [MPa]	416.7	444.4	444.4	416.7		444.4	
Min Load [MPa]	41.7	44.4	44.4	41.7		44.4	
Mean load [MPa]	229.1	244.42	244.42	229.1		244.42	
Machine Load [MPa]	229.1 ± 187.5	244.42 ± 200.0	244.42 ± 200.0	229.1 ± 187.5		244.42 ± 200.0	
Cycles	5 000 000	439 404	322 187	4 953 349		483 574	
Displacement range	0.69	-	0.80	0.70		0.76	
Run out/Fracture	Run out	Fracture	Fracture		Run out	Fracture	
Specimen ID	Description				Reference	ces	
8612-2 2A (B)	Run out at 375 MPa after 5 000 000 cycles. Continued second run at 400 MPa and fractured after 439 404 cycles. Fracture occurred in the base material at a previously detected surface defect.						
8612-2 2A (BB)	Fractured at 400 MPa after 322 187 cycles. Fracture occurred in the base material on the side face.						
8612-3 A3	Run out at stress range 375 MPa after 4 953 349 cycles. Continued second run at 400 MPa and fractured after 483 574 cycles. Fracture at surface defect in base material. An additional crack initiated in the base material at another surface defect.				Figure Figure 9	9-13, Figure 9-14, -15	

Table 9-7 - Results from fatigue test of the production plate specimens.

		gue test R=0.1 on plate specimens	
Specimen ID	8612-3 3A	8612-4	8612-4
Weld distance	12 mm	Initial distance was 1.2 mr	n. Initial distance was 1.2 mm.
Test number	18	1	15 15
Run sequence	1 st run	1 st ru	in 2 nd run
Stress range [MPa]	400	37	400
Area [<i>mm</i> ^2]	244.07	205.8	35 205.85
Max load [MPa]	444.4	416	.7 444.4
Min Load [MPa]	44.4	41	.7 44.4
Mean load [MPa]	244.42	229	.1 244.42
Machine Load [MPa]	244.42 ± 200.0	229.1 ± 187	.5 244.42 ± 200.0
Cycles	306 418	4 938 99	2 084 237
Displacement range	0.80	0.6	66 0.73
Run out/Fracture	Fracture	Run o	ut Fracture
Specimen ID	Description		References
8612-3 3A	Fractured at 400 MPa after 306 41 Fracture initiated in the base mater		
8612-4	Initial run out at 375 MPa after 4 9 Fracture at 2 084 237 cycles at 400 The fracture initiated between the what seemed to be a weld defect in This sample was also subjected to miscalculation of the cross-section) MPa. welds inside the specimen in n weld B. a lower load level due to	

Table 9-8 - Results from fatigue test of the production plates.

The results from the fatigue testing indicates that the welded joints had a higher fatigue strength than the base material. This agrees with Maddox, that writes [11] that if a weld joint would be machined and grinded flush, the most normal crack initiation location would be the base material. This statement is based on one single weld. The results from the fatigue test showed that this was the case for 12 mm and 44 mm between two parallel butt-welds.

The preparation procedure might have adversely affected the results due to introducing compressive residual stresses that inhibited crack initiation.

The fatigue test of the sample with 1.2+ mm between the welds failed in what seemed was a weld defect on the side face of the specimen, so the result from this was inconclusive.

The effect of residual stresses did not result in a fracture in the weld joint.

The number of cycles needed to fracture the specimen was very similar to that of pure base material. This may be due to the careful preparation process that all specimens underwent and since the base material was the weakest part of the specimen.

A visual inspection of each specimen was performed before every test, and some (Figure 9-13, Figure 9-14 and Figure 9-15) visible surface defects was detected and recorded. After fracturing, these specimens where examined and in several cases these locations were the source of crack initiation.





Figure 9-13 – Final fracture and location of crack initiation of 8612-3 A3 specimen.



Figure 9-14 – Location of a small defect in the base material surface where fracture occurred.

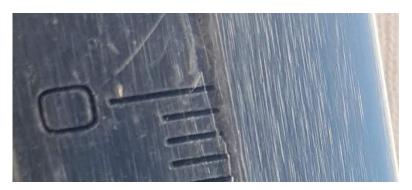


Figure 9-15 - Location where the crack initiated was at the location of the defects in the base material.

9.5.4 Summary

The results from the fatigue test indicates that having two parallel buttwelds, in the absence of stress concentrations and welds discontinuities, does not negatively impact the fatigue performance of the weld joint at the distances of 12 mm and 44 mm.

The specimen with an initial distance between the welds of 1.2 mm failed due to a weld defect, and therefore was inconclusive.

The preparation method might have adversely affected the results due to introducing compressive residual stresses that inhibited crack initiation.

The main fatigue life of the "production plate specimens" composed of the crack initiation stage.

9.6 Stress range vs total cycles of S420G2+M base material

The crack initiation occurred in the base material during the fatigue testing of the unwelded base material and production plate fatigue specimens. The total cycles required for fracture to occur was also very similar. In addition, all preparation and prior documentation of the state of the metal was also the same and in accordance with ASTM E466-15 [45]. This made it plausible to combine the two results in a single graph, to estimate the fatigue life of the S420G2+M metal.

Figure 9-16 shows the combined results in a Stress Range vs total Cycles graph from the main welded and base material fatigue specimens. Both the results from the fractured and run out specimens are shown in the diagram.

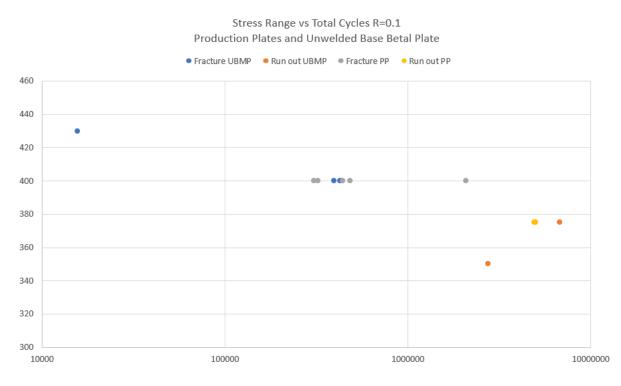
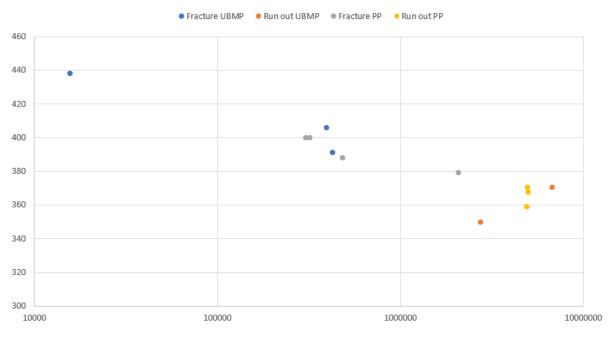


Figure 9-16 – Stress range vs total cycles graph of production plate specimens and unwelded base material specimens. R=0.1.

When adjusting the stress ranges based on the averaged registered displacements it was possible to derive a trend line using linear regression. By adjusting the stress ranges based on this, we get the approximated stress range vs total cycles curve as seen in Figure 9-17. The lowest value at fracture was 380 MPa and the highest value at run out was 370 MPa. Endurance limit was determined to be at 5 million cycles.



Adjusted Stress Range vs Total Cycles R=0.1 Production Plates and Unwelded Base Betal Plate

Figure 9-17 - Adjusted stress range vs total cycles. R=0.1.

10 Experimental Investigation: Residual Stress Measurements

Six specimens as seen in Table 10-1 were sent to Veqter in Bristol, England for residual stress testing. The ASTM E466-15 specimens had an identical preparation as the fatigue specimens in previous chapter. The as-welded samples just had the weld cap grinded of in order to be able to perform ultrasonic residual stress testing.

Weld Proximities	ASTM E466-15	As-welded
50 mm	1	1
15 mm	1	1
5 mm	1	1

10.1 Performed Tests

- CNC-machined, machine precision grinded and manually grinded test specimen (ASTM E466-15)
 - Ultrasonic testing (US)
 - Average Longitudinal residual stress from 0 to 2 mm on the top and bottom face
 - Average Transverse residual stress from 0 to 2 mm on the top and bottom face
 - X-ray diffraction testing (XRD)
 - 50 µm from top face
- As-welded specimen weld cap removed by flap disk grinding
 - Incremental Centre-Hole Drilling testing (ICHD)
 - Bi-axial residual stress profile 1 mm from the top face on the weld centerline side B of the 5mm apart seams weld specimen
 - X-ray diffraction testing (XRD)
 - Bi-axial residual stress at 50 µm from top and bottom face
 - Bi-axial residual stress at 150 µm from top and bottom face

10.2 Ultrasound measurement technique

10.2.1 Ultrasound procedure

The Ultrasound (US) residual stress measurement technique exploits the acousto-elastic property of common materials, in which the speed of sound through a material changes with stress. It was determined [46, 47] that the speed of a longitudinal ultrasound wave through a material was the most sensitive to stress changes in the direction of propagation of the wave. In essence, the speed of the longitudinal wave would decrease through tensile stress regions and increase through compressive stress regions [48]. By using Snell's law it is possible to generate a critically refracted longitudinal (L_{cr}) wave travelling parallel to the surface of a specimen reaching depths below the surface roughly equal to its wavelength [49, 50].

The US probe head used during this project is shown schematically in Figure 10-1 which comprises of 2 MHz ultrasound probes attached to a single acrylic wedge at specific angles according to Snell's law. The transmitting probe (T) emits the longitudinal wave, which is then critically refracted at the specimen surface, travelling parallel to and just beneath the surface of the specimen before being detected by receivers 1 and 2 (R1 and R2). The time needed for the L_{cr} wave to travel between the transmitter and receivers is the Time-of-Flight (ToF) and is used to calculate the change in average stress experienced in the specimen material between the ultrasound probes when referenced against a known stress field, i.e. ideally a stress-free region.

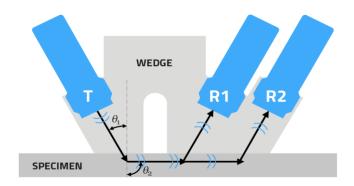


Figure 10-1 - A diagram showing the US residual stress measurement probe head.

When measuring the ToF at different locations within a specimen using the same gauge volume, it is possible to determine the relative change in stress experienced using the following formula:

$$\Delta \sigma = \frac{E(T - T_0)}{L_{11} \cdot T_0} \tag{10-1}$$

Where $\Delta \sigma$ is the change in stress between the two stress states, *T* is the Time of Flight in the unknown stress state, T_0 is the Time of Flight in the known stress state (usually the stress free state, but can be another known stress state) and L_{11} is the acousto-elastic coefficient. L_{11} is a constant property specific to the specimen material and is calibrated experimentally or using a different residual stress measurement technique [51].

For this project, the equipment was arranged to provide results from a gauge area of 5 mm long, 4 mm wide and 2.8 mm deep below the material surface. Only the Longitudinal ToF was measured using the US system due to the gauge size. Ultrasonic couplant was applied over the measurement location to facilitate the transmission of the ultrasound. The transmitted and received signals were recorded by a fast sampling oscilloscope and processed off-line using inhouse software. To convert the Time of Flight to residual stress to known stress value need to be obtain. This can be done using another residual stress measurement technique. The results presented in this report where calibrated using an Incremental Centre Hole-Drilling (ICHD) measurement carried out on as welded specimen. As measurement was taken only 15 mm from an edge only the Transverse stress was taken into consideration as the Longitudinal stress would have been reduced to the proximity of a free surface.

A Contour measurement planned in the future will provide the longitudinal residual stress and will be used as an additional source for calibration.

10.2.2 Technique Accuracy

The uncertainty of the UT residual stress measurement technique is determined by several parameters, with only some being accounted for in this analysis. The uncertainty is dependent upon the thickness of the ultrasound couplant, the gradients of residual stresses within the gauge volume, the material texture, the temperature and the resolution/accuracy of the oscilloscope recording equipment. It was assumed that the microstructure was the same throughout the material and therefore the L_{11} coefficient and T0 were assumed to be constant. The UT equipment was designed to apply a constant contact force each time to the acrylic wedge so controlling the thickness of couplant and was monitored and recorded during the tests. The oscilloscope recording equipment provides a state-of-the-art resolution of measurement of 0.4 ns, which is highly accurate.

10.3 Results: Residual Stress Measurements

10.3.1 CNC-machined, precision grinded and manually grinded specimens

The time of flight (ToF) measured on the three CNC-Machined specimens on the top and bottom surface across the welds are shown in Figure 10-2. The measured ToF was then shifted to show about 0 ns in the base material. The increases in ToF relates to increases in stresses and decreases in ToF relates to decreases in stresses. As it can been seen, clear changes occur at the weld location. In the longitudinal direction the ToF increases by about 10 to 15 nanoseconds in a M shape form center at the weld centerline. Some drop in ToF was generally found on either side of the weld. Each on the max peaks where found to be center to the weld centerline at the bottom and top side of the specimen. In the transverse direction the ToF looks like a near mirror to the longitudinal direction at the exception that, each negative drop where balanced with a sharp increase in ToF.

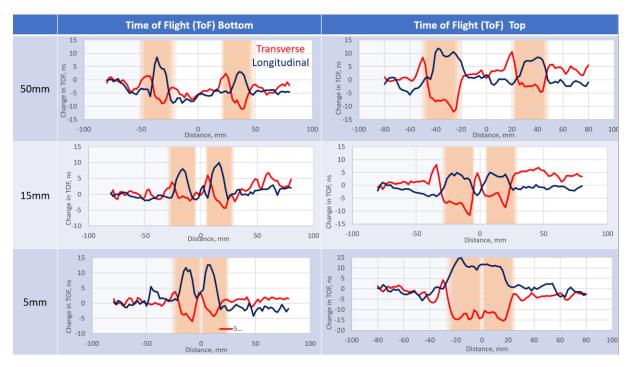


Figure 10-2 - Time of Flight results from US testing. Weld A is the leftmost weld.

The ToF shown about where then converted to residual stress using two values. Zero in the base material and -100 MPa for the max compressive peak found over the three specimens. The Ultrasonic measurement can be severely affected by changes in texture which occurs naturally in welds. Often several additional calibration values need to be used. A Contour measurement will be performed in the near future to provide further confidence in the results. One of the clear results provided by the ICHD measurement was that the longitudinal residual stress was in tension while the transverse direction was in compression, which is also shown in the Ultrasonic measurement. It should be noted that the near surface of the specimen within 150 μ m was in high compression near yield. This compressive level was introduced by the grinding process. The XRD carried out on the surface of the CNC-machined specimen measuring up to 15 μ m

show that all surfaces were between -320 MPa and -450 MPa. As the Ultrasonic technique is a relative technique, which measure the changes in residual stress, the addition of a constant layer of compressive stress will not influence the results.

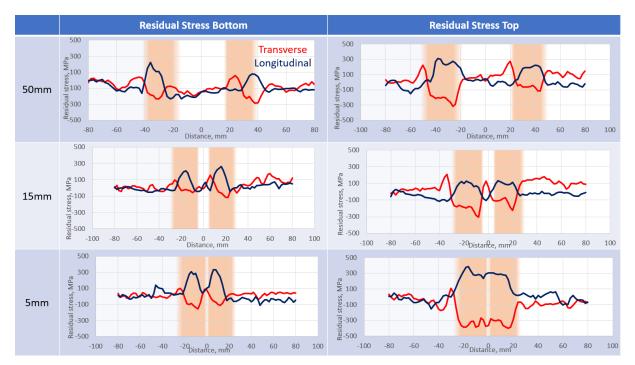


Figure 10-3 - Residual stress test results from US testing. Weld A is the leftmost weld.

	XRD	XRD
	5 mm Top transverse	5 mm Top longitudinal
Weld Center Line	-336 MPa	-440 MPa
-5 mm from WCL	-400 MPa	-392 MPa
-60 mm from WCL	-300 MPa	-346 MPa

Table 10-2 - Results X-ray diffraction machined and grinded specimen.

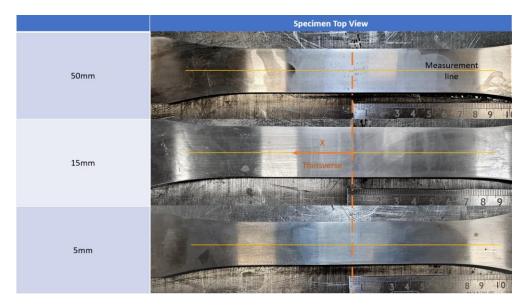


Figure 10-4 - ASTM E466-15 Test Sample measuring direction.



Figure 10-5 - Ultrasonic Residual Stress Measurement.

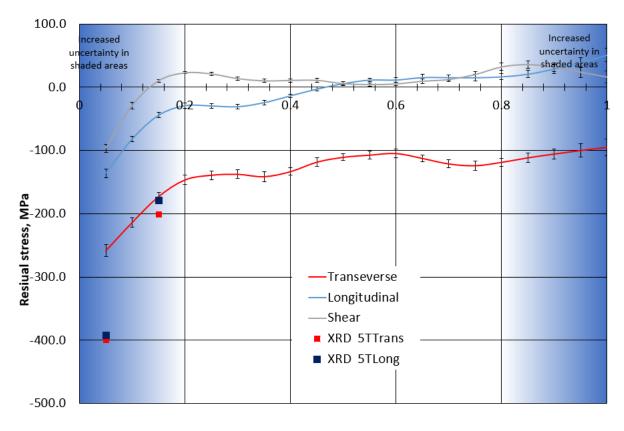
10.3.2 As-welded specimen

XRD and ICHD measurement were carried out on the 5 mm weld apart specimen at 15 mm from the edge at the weld centerline. The XRD measurements were carried out at 15 μ m and 150 μ m and the ICHD measurement was performed to provide 1 mm of residual stress measurement as shown in Figure 10-6.

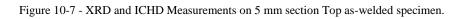
Figure 10-7 shows a comparison of both the XRD and ICHD results. As it can be seen both transverse and longitudinal residual stress are in high compression at the surface which was caused by the grinding of the weld cap. The effect of the grinding was found up to about 150 μ m the residual stress from the weld was measured thereafter. The longitudinal stress was found negligible which was expected due to the proximity of the weld. The transverse stress was found to be about -100 MPa.



Figure 10-6 - Specimen test locations in the as-welded specimen.



Depth through specimen from back surface, mm



10.4 Summary: Residual Stress Measurement

10.4.1 CNC-Machined, precision grinded and manually grinded specimen ASTM E466-15

US testing was performed on all the specimens on both root and weld cap. XRD was performed on the 5 mm specimen in weld cap B.

The preparation of the specimens was in accordance with E466-15 Standard Practice for Conducting Force Controlled Constant Amplitude Axial Fatigue Tests of Metallic Materials. Similar to the fatigue test specimens. It was CNC-milled, precision grinded and subsequently manually grinded until a surface roughness of 0.2 μ m was achieved. The influence of the preparation operation on the residual stresses was assumed to be 150 μ m into the specimen surface.

In the majority of US tests performed the longitudinal residual stress was in tensile and the transverse residual stress was in compression. The ICHD measurement seemed to confirm this finding.

The effect weld proximity had on the residual stress profile was not noticeable in the 50 mm and 15 mm specimens. But, in the specimen with 1.2 mm between the weld toes there was a clear increase in both tensile and compressive residual stress distribution.

The ultrasonic test showed a clear transition from parent to unwelded metal.

The US test only provide relative changes which needs to be calibrated with a minimum of 2 know stress values. The US will also be affected by changes in texture in material which occurs in the weld.

The XRD shows that we have yield level compressive and tensile residual stresses $50 \,\mu m$ from the top face. This is the result from the grinding and milling operations.

10.4.2 As-welded specimen

XRD and ICHD was performed on the top surface of the 5 mm as-welded specimen.

The weld cap of the specimen was removed with a flap disk.

XRD and ICHD was performed on weld A and weld B, respectively at the locations shown in Figure 10-1. ICHD was performed down to 1000 μ m and XRD was performed at 50 μ m and 150 μ m.

The results differed slightly, but both showed that there were high compressive residual stresses in the top surface from the grinding operation at an approximated depth of 0-150 μ m.

The ICHD showed the residual stress profile 1000 μ m into the specimen surface in the weld B center line. From 500 μ m the longitudinal and shear residual stresses was in tension, while the compressive still was in compression. This was not as expected. The location the test was performed might have reduced the longitudinal residual stress result.

10.5 Further work

- Perform a contour test to determine the residual stress distribution.
- Conduct further US testing on as-welded plates.

11 Summary and main Conclusion

The experimental chapters consists of a methodology, result and conclusion of the findings from each experiment. This chapter serves the purpose of summarising the key findings.

Predetermined distances for weld proximity have been set in most standards, codes and specifications in the industry today. There is still a lack of information regarding the implications when divergences from the requirements are necessary or have been detected in the service-life of a structure. It is vital to understand the implications it has on mechanical and material properties of the weld joint when exceeding these set requirements.

The objective of this thesis was aimed at obtaining quantifiable data regarding the implications of having two adjacent weld joints in close proximity.

11.1 Test Setup

An experimental analysis was designed where the resulting material behavior was analyzed and documented. The test consisted of six welded 500x300x15 mm S420G2+M steel plates, each having two adjacent parallel butt welds along its full length. The distances between two adjacent welds were 50, 15 and 5 mm in order to analyze the effect a variation in distance have on the properties of the initial weld pass.

11.2 Welding Procedure Qualification

In order to maintain conformity of all welded plates a welding procedure qualification record (WPQR) were initially established. This was achieved by welding an initial qualifying plate that was tested and qualified in accordance with the requirements in NS-EN ISO 15614-1:2017 [17]. Following welding procedures were then subsequently based of the approved welding procedure qualification (WPQ).

11.3 Range of Testing

All plates were welded at KIWA in Stavanger, and subsequently mechanically tested at Quality Lab in Forsand in accordance with NS-EN ISO 15614-1:2017.:

- Visual testing
- Surface crack detection
- Vickers Hardness Test
- Charpy V Impact Test
- Transverse Tensile test
- Bending test
- Macroscopic examination
- Radiographic or ultrasonic testing was performed at IKM Testing

Fatigue test and microstructural analysis (optical microscope) was conducted at the Department of Mechanical and Structural Engineering and Materials Science at the University of Stavanger. Residual stress analysis was conducted at Veqter in Bristol using ultrasound (US), Incremental Centre-Hole Drilling (ICHD) and X-ray Diffraction (XRD) technique.

11.4 Mechanical Testing

The mechanical testing results from the NS-EN ISO/IEC 17025 accredited Qlab, indicated no degradation in material properties due to weld proximity. This statement was based on results from tensile tests, Charpy V Impact toughness test and Vickers hardness tests. The distance between the weld toes of the butt welded joints was 44, 12 and 1.3 mm.

All Tensile Test specimens failed in the BM outside the HAZ. This was as assumed since the UTS of the WM is higher than the BM. The result seemed to indicate that the influence of the adjacent secondary weld did not have a negative impact on the tensile strength properties of the initial weld. The fracture also initiated a considerable distance away from the weld showing no sign of reduction of tensile strength in the HAZ. The WM were machined flush with the BM.

The results from the Charpy V Impact Toughness test showed no reduction in toughness properties from the adjacent secondary weld. Due to inherent spread of the test results it was not possible to determine any distinct variation between the results at the different distances. All tests showed that the weld joint had toughness properties well above the requirements. The fracture appearance and lateral expansion was well within the limits. The test results are shown in Table 7-33.

The results from the Vickers Hardness test did not indicate any reduction in material hardness properties due to the adjacent secondary weld. As seen in Figure 7-37 the hardness values between a weld joint with 1.3 mm between the weld toes did not have any significant increase in hardness. Due to the spread in results in the WM and HAZ it was not possible to see if the secondary weld had any softening effect on the initial weld. The welds at 12 mm and 44 mm from weld-toe to weld-toe did not show any adverse effect due to the weld proximity, and showed a similar toughness pattern as the welds at 1.3 mm from eachother. More tests are needed in order determine if there is any difference between the various distances. Important to note also is that only one specimen at 1.3 mm was hardness tested. This is sufficient to approve a WPQ, but due to the nature of this test more tests are needed in order to ascertain the initial test results.

These results indicate that weld proximity as close as 1.3 mm does not have any degrading impact on the weld joint. This is based on level 2 mechanical testing in accordance with 15614-1:2017, used to approve WPQs.

It is important to note that these results are for this set of parameters and materials, but the findings can act as a baseline for further testing at various HI, welding positions, distances, etc.

11.5 Weld Microstructure

The microconstituents in the weld metal and HAZ are typical, and a common combination in the industry. In only one specimen did the weld HAZ overlap. This was the weld with 1.3 mm between the weld toes. No visible harmful phases were detected with the optical microscope in either the WM or the HAZ. The impact of the overlap only seemed to have a normalizing effect of the previous HAZ of weld A.

Between subsequent weld passes and unaffected region of the coarse-grain HAZ microstructure can be distinguished. The reheated pockets of the CG HAZ regions are small and discontinuous, which make their microstructure difficult to identify and investigate. The reheated IC CG HAZ is very susceptible to failure, due to the fact that the phase transformation into austenite began on the grain boundaries; these small areas is rapidly cooled and have a strong possibility to become hard and brittle. The subcritical regions are mainly tempered bainite and martensite with precipitated carbides and therefore represent no danger to the structural integrity of the weld.

The welds with a distance of 12 mm and 44 mm between the weld toes had no overlap of HAZ. The was never any overlap of visible HAZ of the weld roots, despite the larger area of effect.

Important to note though is that the magnification of the optical microscope is limited and further analysis with a scanning electron microscope (SEM) and a transmission electron microscope (TEM) is necessary in order to assure that this is the case. Investigation in these microscopes are needed in order to confirm the exact nature of all phase transformations occurring in the overlapping HAZ.

11.6 Fatigue test

The objective of the fatigue test was to assess if any reduction in fatigue strength had occurred due to the weld proximity situation. The fatigue test specimens were prepared with the intention to isolate and study the microstructure. The preparation of the sample were conducted in accordance with ASTM E466-15 for a homogenous material subjected to high-cycle fatigue. Weld discontinuities was avoided in order to isolate and focus on the HAZ microstructure, and any geometrical and angular weld stress concentrations was removed. The result from the test was that the specimen at 5 mm distance with overlapping welds seemed to have failed in the secondary weld. The reason for the crack initiation was not possible to determine. Important to note though, was that the failure did not occur in the first weld, which was the focus of the study, but the secondary overlapping weld. In the absence of stress concentrations, the remaining fatigue specimens failed in the parent metal. The sample size was limited, but the results indicated that at 12 mm between two weld toes, the weld has the higher fatigue strength.

11.7 Residual Stress Measurements

Six specimens as seen in Table 10-1 were sent to Veqter in Bristol, England for residual stress testing. The ASTM E466-15 specimens had an identical preparation as the fatigue specimens in previous chapter. The as-welded samples just had the weld cap grinded of in order to be able to perform ultrasonic residual stress testing.

11.7.1 CNC-Machined, precision grinded and manually grinded specimen

US testing was performed on all the specimens on both root and weld cap. XRD was performed on the 5 mm specimen in weld cap B.

The preparation of the specimens was in accordance with E466-15 Standard Practice for Conducting Force Controlled Constant Amplitude Axial Fatigue Tests of Metallic Materials. Similar to the fatigue test specimens. It was CNC-milled, precision grinded and subsequently manually grinded until a surface roughness of 0.2 μ m was achieved. The influence of the preparation operation on the residual stresses was assumed to be 150 μ m into the specimen surface.

In the majority of US tests performed the longitudinal residual stress was in tensile and the transverse residual stress was in compression. The ICHD measurement seemed to confirm this finding.

The effect weld proximity had on the residual stress profile was not noticeable in the 50 mm and 15 mm specimens. But, in the specimen with 1.2 mm between the weld toes there was a clear increase in both tensile and compressive residual stress distribution.

The ultrasonic test showed a clear transition from parent to unwelded metal.

The US test only provide relative changes which needs to be calibrated with a minimum of 2 know stress values. The US will also be affected by changes in texture in material which occurs in the weld.

The XRD shows that we have yield level compressive and tensile residual stresses 50 μ m from the top face. This is the result from the grinding and milling operations.

11.7.2 As-welded specimen

XRD and ICHD was performed on the top surface of the 5 mm as-welded specimen.

The weld cap of the specimen was removed with a flap disk.

XRD and ICHD was performed on weld A and weld B, respectively at the locations shown in Figure 10-1. ICHD was performed down to 1000 μ m and XRD was performed at 50 μ m and 150 μ m.

The results differed slightly, but both showed that there were high compressive residual stresses in the top surface from the grinding operation at an approximated depth of 0-150 μ m.

The ICHD showed the residual stress profile $1000 \,\mu\text{m}$ into the specimen surface in the weld B center line. From $500 \,\mu\text{m}$ the longitudinal and shear residual stresses was in tension, while the compressive still was in compression. This was not as expected. The location the test was performed might have reduced the longitudinal residual stress result.

12 Future research

This project served the purpose of being a baseline in the matter of weld proximity research. The findings can assist in the assessment of a weld proximity or weld-on-weld scenario.

The welding operation did not seem to introduced any degradation of the material properties in the WM and HAZ. The mechanical properties of similar weld joints in the field could be assumed to be unaffected due to the welding of an adjacent weld.

Important to note is that this is for this type of method and materials. More testing has to be performed in other to ascertain the findings. The results can be used as a baseline for further research.

Future research regarding this topic is listed below:

Mechanical testing

Further mechanical testing is needed in order to confirm results obtained in the project. Especially at a distance where we have HAZ overlap. The sample size of hardness tests conducted in this area with overlapping HAZ consisted solely of one macro specimen. This is sufficient when approving a WPQ, but lacking when performing an experiment of this nature. Hardness tests would have to be performed at distances that increases the risk of forming brittle reheated grain structures.

Further study and experimentation can be performed on welded joints of plates with various HI, welding positions, environmental conditions, pre-heat, welding method, filler material, plate thickness, use of a pipe instead of plate, fixations, etc.

Instead of using a 15 mm thick plate, it would have been interesting to test a 10 mm plate. This would have reduced the deformation due to the welding process. However, the side effect this could have been that the residual stresses would be reduced, which was a part of the study. Thicker goods may be more representative of thick-walled pipe sections. The welding time had been reduced because not as many passes had been needed. This would have been more economical and you might have time to produce more plates. In addition, the plates would not have been machined to accommodate the fatigue machine due to a maximum grip distance of 11 mm.

These results could be used as a guideline for other normalized steels. Further research could be conducted on QT steel which has a higher hardenability.

Microstructural examination

Due to the limitations of the optical microscope more thorough analysis is needed in the SEM and TEM to confirm the results. The results would have to be verified and compared with the results from the mechanical testing.

Fatigue test

The welded joint subjected to fatigue testing was prepared in accordance with ASTM E466-15. This introduced high compressive residual stresses based on the results from the XRD and ICHD. For future research the analysis of a welded specimen in its original state would be more representative for a structure. Fatigue test on plates with remaining weld cap.

In this test the stress range R=0.1. For future research other stress ratios could be tested.

Residual stress measurement

Use the findings from the residual stress test to model and calibrate a FEA. The welding operation, including the location of all fixations, was documented on all test results. This also offers a possibility to determine the reason why we had tensile transverse compressive stresses in the weld joints. The residual stresses obtained could be difficult model due to the additional introduction of grinding which introduced compressive residual stresses in the surface of the specimen. A residual stress test would have to be performed on an un-grinded specimen in order to map the original residual stress distribution.

More residual stress measurements was still being performed after the delivery of this thesis. This included contour plotting, which is a destructive test method that measures the residual stress on the surface of a specimen after it has been cut. This method provides information of the residual stresses on the whole surface.

In addition, further ultrasonic testing was being performed on the as-welded specimens. These results could be used to calibrate the method.

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Appendices

- A. Manufacturing and welding process, characterization of the Steel S420G2+M Prior to fabrication
- B. Manufacturing and welding process, characterization of the Steel S420G2+M Material Selection and Cutting Process
- C. Manufacturing and welding process, characterization of the Steel S420G2+M - Welding Procedure Qualification Program
- D. Manufacturing and welding process, characterization of the Steel S420G2+M - Production Welding of Plates
- E. Manufacturing and welding process, characterization of the Steel S420G2+M - NDT, Mechanical Testing and Specimen Preparation
- F. Microstructural Examination
- G. Fatigue Testing
- H. Residual Stress Measurement

Instruction

The project Weld proximity INVES. has undergone different stages, which in this document are shown in one hundred series. The different steps are represented by letters and can be seen in the Inspection and Test Plan. The documents are grouped under the stage in which they were produced.

A 10	0-199: E	Experimental i	nvestigation:	Manufacturing and	l welding processes -	 Prior to 	fabrication
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- Stage (Appx.)
 Series

 A
 100-199

 B
 200-299

 C
 300-399

 D
 400-499
 Experimental investigation: Manufacturing and welding processes - Prior to fabrication Experimental investigation: Manufacturing and welding processes - Material Selection and Cutting Process Experimental investigation: Manufacturing and welding processes - Welding Procedure Qualification Program Experimental investigation: Manufacturing and welding processes - Production Welding of Plates Experimental investigation: Manufacturing and welding processes - NDT, Mechanical Testing and Specimen Preparation Experimental investigation: Microstructural Examination Experimental investigation: Fatigue Testing Experimental investigation: Residual Stress Measurements 200-299: 300-399: 400-499:

 - 500-599: 600-699: E F
 - 700-799:
 - G H 800-899:

Document No.	Title
Doc-A101	Inspection and Test Plan for Master Thesis-Weld Proximity Investigation
Doc-A102	MDS Base material - S355J2+M
Doc-A103	MDS Filler metal - OERLIKON SPEZIAL
Doc-B201	Mill test report - Base Material S420G2+m - HT43831-9133182
Doc-B202	Certified Material Test Report - Filler material NSSW SM-47-A Manuf.No. 7U341AW996 MIT 9118
Doc-B203	Certified Material Test Report - Filler material NSSW SF-3AM Manuf.No. 7S041MP960 MIT 9188
Doc-B204	Certified Material Test Report - Filler material NSSW SF-3AM Manuf.No. 8X221MP996 MIT 9588
Doc-C301	WPQR: S420PL1-M and including documents:
	- WPQ \$420PL1-M
	- Summary
	- pWPS \$420PL1-M
	- Weld log - pWPS:S420PL1-M - Report7967-19-DRT-1
	- Mill test report - Materialcertifikat S420G2+M - HT43831-9133182
	- Mill test report - Filler material: NSSW SM-47-A Manuf.No. 7U341AW996 MIT 9118
	- Mill test report - Filler material: NSSW SF-3AM Manuf.No. 7S041MP960 MIT 9188
	- Mill test report - Filler material: NSSW SF-3AM Manuf.No. 8X221MP996 MIT 9588
Doc-C302	All WPS's:
	- WPS PL1-SW
	- WPS: PL3/4-DW50-Rep
	- WPS: PL5/6-DW15-Rep
	- WPS: PL7/8-DW5-Rep
Doc-D401	All Weld log's for Weld A:
	- Weld log: PL3-DW50
	- Weld log: PL4-DW50
	- Weld log: PL5-DW15
	- Weld log: PL6-DW15
	- Weld log: PL7-DW5
	- Weld log: PL3-DW5
Doc-D402	All Weld log's for repair Weld B:
	- Weld log: PL3-DW50-Rep
	- Weld log: PL4-DW50-Rep
	- Weld log: PL5-DW15-Rep
	- Weld log: PL6-DW15-Rep
	- Weld log: PL7-DW5-Rep
	- Weld log: PL3-DW5-Rep
Doc-E501	NDT report - Radiograpic Examination
	- Report8076-19-DRT-1
	- Report8076-19-DRT-2-REV1
	- Report8076-19-DRT-3-REV1
	- Report8076-19-DRT-5
	- Report8076-19-DRT-6 - Report8076-19-DRT-7
Doc-E502	- Réport80/6-19-DR1-7 Qlabs Material Report 8612-2 (PL 3)
Doc-E502	Qlabs Material Report 8612-2 (PL 5) Qlabs Material Report 8612-3 (PL 5)
Doc-E504	Qlabs Material Report 8612-9 (PL7)
200 12004	
Doc-F600	The chapter "Microstructural Examination" has no attachments.
	•
Doc-G701	Fatigue test - Prior to fabrication specimens
Doc-G702	Fatigue test - Unwelded base metal specimens
Doc-G703	Fatigue test - Production test plate specimens
Doc-H800	The chapter "Residual Stress Measurements" has no attachments.



Manufacturing and welding process, characterization of the Steel S420G2+M

- Prior to fabrication

- Doc A101 Inspection and Test Plan for Master Thesis Weld Proximity Investigation
- Doc A102 MDS Base material S355J2+M
- Doc A103 MDS Filler metal OERLIKON SPEZIAL

• Doc A101 - Inspection and Test Plan for Master Thesis - Weld Proximity Investigation

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Inspection and Test Plan for Master Thesis - Weld Proximity Investigatior

A101 ITP-W&W-002 01.02.2019 Rev 4 09.05.2019 Doc No: ITP No: Issue Date: Revision no: Rev date:

Dwg=Drawing, p\ analyse, UIS=Uni

			is, SJA=Sikker jobb analyse, UIS=Univeristy of Stavanger, attias Larsson (Master Student UIS), Magnus Larsson (Ma				y leader Q-lab, metalurg)	Emil Surnevik (Senior Engineer), Jørgen Grøn	nsund (Se	nior Engine	eer), Dr Xa	ier Ficquet (CEng MIMe	echE)		
AGE	Ref.	Activity Description	Specification / Procedure / Drawing	Acceptance Criteria	Verifying Document	Responsible	M&M Sign.	Comments / Notes				Inv				
STAGE	Rei.	Activity Description	Specification / Procedure / Drawing	Acceptance Criteria	verrying bocument	Responsible	With Sign.	comments / Notes	Pro	of. C.R Sign	A	1&M Sign	K A	WA Sign	Q A	lab Sign
	1.1	Pre Inspection meeting for weld on weld project	Meeting agenda	Schedule	Minutes of meeting	Magnus		Carried out before the project begins. The scope of the project is defined.	н		н					
		Perform a test welding on proximity welds.						The goal of this activity is to gain some understanding of the task and its								
	1.2	Welding of two adjacent welds will be performed at Vest Norway Doors AS.			Pictures			limitations before making the I&TP and contact partners. After welding carried out the plate is transported to UIS.	D		н					
	1.3	Transport plate to UIS				Magnus		Different methods for obtaining a good			н					
	1.4	Try to machine specimens to see if it is possible to implement this on UIS.			Pictures	Magnus and Emil		surface finish should be tested. After machining and surface preparation, some	D		н					
		possible to implement this on old.						specimens are prepared which can be tested later in the project.								
								Collaboration partners and expertise: KIWA - Welding, interpretation of standards and production of WPQ. Q-lab								
	1.5	Find partners for the project.				M&M		Mechanical testing, interpretation of WPQ.	D		н		н		н	
fabrication								VEQTER-Interpretation of residual stresses and fatigue testing.								
2	1.6	Make an Inspection and Test Plan.		M&M:s approval	Test and Inspection Plan for project Weld Proximity Investigation	Magnus		The document should describe the project's progress and act as a quality-	D		н		D		D	
A. Prior	1.7	Verify and approve documents to be used in		M&M:s approval	Document list	Mattias		assuring document. Document list gives an overview of all the documents produced relating to the			н		n		D	
		the project.						project The SJA document should be used for								
	1.8	Produce SJA och meldeskjema			SJA / Meldeskjema.	Magnus		tasks that are considered critical. The meldeskjema should be used for			н					
	1.9	Produce a method for carrying out welding			Figures: Production flow of test plate for welding procedure test / Production flow	M&M, KIWA and Q-lab.		unwanted events in the UIS workshop. These two figures show roughly the various steps for producing the WPQ and							D	
	1.5	procedure test and production test.			of test plates for production test Naming and marking system / Installation	Math, KIWA and Qrab.		the production plates.							5	
	1.10	Prepare drawings and illustrations for how the steel plates should be cut, fixed, prepared and named before welding.			of strongbacks / Drawings - Joint Preparation 1 / Drawings - Joint	M&M, KIWA and Q-lab		Design documentation must have AFC status before the project begins.			н		D		D	
		named before weiding.			preparation 2			Three versions of test plans will be								
	1.11	Produce test plans for WPQ qualification plate	NS-EN ISO 15614-1:2017 Level 2 / ASTM E466-15 / ASTM E468-11	NS-EN ISO 15614-1:2017 Level 2 / ASTM E466-15	Welding procedure and production	M&M and KIWA		produced. The first must comply with NS- EN ISO 15614: 2017 to qualify as a WPQ. The other two will be based on this			н		н		D	
		and production plates.	ASIM E408-11	ASIM E400-15	sample testing			standard but adapted for fatigue and ultrasonic testing.								
*	2.1	Approval and ordering of materials.	Supplier procedure	NS-EN 10225:2009	Receipt / BOM	Magnus		Steel is delivered to Smedtkristansen and welding filler metal to KIWA.	D		н		D			
and Cutting Process	2.2	Plan for how materials should be handled and stored.		Mill test reports	Material handling	Magnus and IWE Arild		Steel must be stored indoors by KIWA and filler material must be stored according to			н		н			
Cutting								the material handling document. Both the parent and filler material must be checked. Parent metal: S420G2+M: Filler								
	2.3	Check incoming material. Material identification includes visual inspection, type	Supplier procedure	NS-EN 10225:2009	BOM / Mill test reports / Drawing /	Manus & IWE Arild		metal: NSSW SM-47A and NSSW SF-3AM. Verify that materials comply to PO / BOM			н		н			
Select		of material & traceability dimensional check.			Pictures			upon receipt. Filler material must be handled according to doc-material								
B. Material Selection		Cut out plates and prepare joints with Water-						handling. Conducted at SmedtKristiansen. Method								
8.8	2.4	jet cutting. Transport plates to KIWA.	Drawing - Joint preparation 1	Drawing - Joint preparation 1	Pictures	Magnus Magnus		to check the result is through visual check and control measurement. Plates should be stored indoors.			w		-			
	3.1	Create pWPS	pWPS NST		pWPS: S420PL1-M	M&M		pWPS was produced together with KIWA and NST.	D		н		н			
	3.2	Create and prepare the weld log sheet.			Weld log - pWPS:S420PL1-M	IWE Arild & Magnus		The same log template to be used for all welding.			н		т			
	3.3	Check and calibrate equipment to be used for welding and logging.	pWPS: S420PL1-M	pWPS: S420PL1-M	Pictures	IWE Arild Finnesand		Equipment used for logging and welding is arranged by KIWA. The welding machine			w		н			
	3.4	Perform joint preparation, fit-up & alignment of qualification test plate.	Drawing - Joint preparation 1 / Installation of strongbacks / pWPS: S420PL1-M	Drawing - Joint preparation 1 / pWPS: S420PL1-M	Pictures	Welder and Magnus		is a Fronius-TransSteel 2700. Fit-up of plates must be approved by Mattias before welding can start.			w		н			
	3.5		pWPS: S420PL1-M	pWPS: S420PL1-M				All deviations must be reported. Deviations can be cracks in the strongback								
	3.5	Welding & Logging of WPS qualifying plate	pWPS: \$420PL1-M	pWPS: S420PL1-M	Weld log - pWPS:S420PL1-M / Pictures	Welder and Mattias		welds, unexpected deformation, all weld parameters that deviate from pWPS, etc.	D		н		н			
	3.6	Name and mark the plate according to instructions.	Naming and marking system		Pictures	Magnus		If marking disappears during welding, marking should be done again.			н		w			
gram	3.7	Perform visual testing	ISO 17637:2016	EN ISO 5817:2014 B/C	WPQR: S420PL1-M	IWE Arild Finnesand		Visual check is made by IWE Arild Finnesand at KIWA. If the plate is approved visually, it is transported to IKM	D		w		н			
ion Pro	3.8	Plates are transported to IKM inspection for X-				Magnus		Inspection for X-Ray testing.								
Qulaifica	3.0	ray inspection.				magnas		After an approved test, IKM Inspection								
dure Q	3.9	Perform X-ray testing	ISO 17636-2:2013	ISO 10675-1:2016	WPQR: S420PL1-M	IKM Inspections AS		will deliver an inspection report that will be a basis for the WPQR. If the test is approved, the qualification plate is			D					
g Proce	3.10	The plates are transported to Q-lab for						transported to Q-Lab.								
Weldin	3.10	mechanical testing. Perform the Surface Crack Detection Test	NS-EN ISO 17638:2016, NS-EN ISO 3452-1:2013	ISO 23278:2015 Level B	WPOR- \$420PI 1-M	Magnus Q-Lab					w				н	
ن ن	3.12	(Magnetic particle inspection) Specimens preperation		10 EST 0.1015 COCID	W1 Q1. 3420 C2 W	Q-lab		Complete the test samples for all			w				н	
	3.14	Perform Vickers Hardness test Perform Transverse Tensile Test	NS-EN ISO 9015:2011 ISO 4136:2012, ISO 6892-1:2016 Method A1	15614-1:2017 15614-1, Mill test report	WPQR: S420PL1-M WPQR: S420PL1-M	Q-lab Q-lab Q-lab		mechanical tests .			w w	-			H H	
	3.15	Macroscopic testing Perform Charpy V Impact test, KV8	ISO 17639:2013 NS-EN ISO 148-1:2016, ISO 9016:2012	EN ISO 5817:2014 15614-1, Mill test report	WPQR: S420PL1-M WPQR: S420PL1-M	Q-lab Q-lab					W				H	
	3.17	If all tests are approved, Qlab writes a material report that verifies that all mechanical tests have been approved.	NS-EN ISO 15614-1.2017 Level 2	NS-EN ISO 15614-1.2017 Level 2	Material report: 8612-1			This report will be the basis of the WPQR-	D	_	D		D		н	
		have been approved. Compile all documentation that has been produced in the Welding Procedure						S420PL1-M.			1	1				
	3.18	Qualification Program and material data and manufacture a WPQ and WPQR.	NS-EN ISO 15614-1.2017 Level 2	NS-EN ISO 15614-1.2017 Level 2	WPQR: S420PL1-M	Magnus och Arild			D	L	н		н		D	L
	3.19	Produce welding procedures for production welding stage	NS-EN ISO 15614-1.2017 Level 2	NS-EN ISO 15614-1.2017 Level 2	WPS / WPS-Rep			Produces 12 WPS for plate 3-8. Six WPSs for the first weld, weld A, and six Repair- WPSs for the second weld, weld B.	н		н		н		н	
	4.1	Check and calibrate equipment to be used for	WPS	WPS	Pictures	IWE Arild Finnesand		WPSs for the second weld, weld B. Equipment used for logging and welding is arranged by KIWA. The welding machine			w	-				-
	4.1	welding and logging.		WF3	FREUTES	one concerning and		is a Fronius-TransSteel 2700. The plates were pre-cut in Stage B and		<u> </u>	-	-				
	4.2	Perform joint preparation, fit-up & alignment of production test plates for the first weld, weld A.	Drawing - Joint preparation 1 / Installation of strongbacks / WPS /	Drawing - Joint preparation 1 / WPS	Pictures	Welder and Magnus		transported to KIWA. Fit-up of plates must be approved by master student Mattias	t		w		н			
								before welding can start. All deviations must be reported.		-	+	+				-
	4.3	Perform production welding and logging on the first weld, weld A.	WPS	WPS	Weld logg / Pictures	Welder and Mattias		Deviations can be if welds for strongback cracks, unexpected deformation, all welds that deviate from wps, etc.			н		н			
	4.4	Name and mark the plate according to instructions.	Naming and marking system		Pictures	Magnus		If marking disappears during welding, marking should be done again.	L		н					L
olates	4.5	Perform visual inspection.	ISO 17637:2016	EN ISO 5817:2014 B/C	Test report	IWE Arild Finnesand		If visual inspection is approved, the plates are forwarded for cutting.			w		н			
welding of plate:	4.6	Transport plates to Watech for water jet cutting of joint 2, weld B.				Mattias		Cut a new parallel joint next to the first			н	-				
	4.7	Water jet cut weld joint B at Watech.	Drawing - Joint preparation 2	WPS-Rep	Pictures	Watech		weld, weld A. Method to check the result is through visual check and control			w					
D. Production	4.8	Transport back plates to KIWA for welding				Mattias		measurement.		<u> </u>	н	-			<u> </u>	
D. P.		welding joint 2, weld B. Check and calibrate equipment to be used for	1000 D	1105 D	Printer and			Equipment used for logging and welding is	-			-				
	4.9	welding and logging. Perform joint preparation, fit-up & alignment	WPS-Rep	WPS-Rep	Pictures	IWE Arild Finnesand		arranged by KIWA. The welding machine is a Fronius-TransSteel 2700.			w	-	rt			
	4.10	of production test plates for the second weld, weld B.	Drawing - Joint preparation 2 / Installation of strongbacks / WPS-Rep	Drawing - Joint preparation 2 / WPS- Rep	Pictures	Welder and Magnus		Fit-up of plates must be approved by Mattias before welding can start.	L		w		н			
	4.11	Perform production welding and logging on	WPS-Rep	WPS-Rep	Weld logg	Welder and Mattias		All deviations must be reported. Deviations can be cracks in the strongback	c.		н		н			
		the second weld, weld B.						welds, unexpected deformation, all weld parameters that deviate from pWPS, etc.		<u> </u>	<u> </u>	<u> </u>			<u> </u>	
	4.12	Final visual inspection of welds, marking, compilation of data and preparation of plates	ISO 17637:2016	WPS / WPS-Rep / EN ISO 5817:2014 B/C	Test report			Conducted on KIWA . Interest is, among other things, to see how the plates have deformed because of the heat, polish the	D		н		н			
		before being sent off for testing. Plates are transported to "IKM Inspection" for				KM prostine of		welds, mark the plates. IKM Inspection AS picks up the plates at				+				<u> </u>
	5.1	Non Destructive Testing.				IKM Inspections AS		KIWA.	I		w	1		I	I	L

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	5.	5.2	Perform X-ray testing	ISO 17636-2:2013	ISO 10675-1:2016	NDT report, pictures		Conducted on IKM Testing	D	D	D		D
	5.	5.3	Plates are transported to Q-lab for mechanical testing.				Magnus and Mattias			н			
ration	5.	5.4	Perform the Surface Crack Detection Test (Magnetic particle inspection)	NS-EN ISO 17638:2016 / NS-EN ISO 3452-1:2013	ISO 23278:2015 Level B	WPQR: S420PL1-M	Q-lab	Conducted on Q-lab		w	D		н
Prepa	5.	5.5	Specimens preperation for mechanical testing.	NS-EN ISO 15614-1.2017 Level 2	NS-EN ISO 15614-1.2017 Level 2		Q-lab	Completes the test samples for all mechanical tests.					
Len	5.	5.6	Perform Vickers Hardness test	NS-EN ISO 9015:2011	NS-EN ISO 15614-1.2017 Level 2	Qlab Material report	Q-lab	mechanical tests.		w	D		н
pedir	5.	5.7	Perform Transverse Tensile Test	Specimens preparation - Tensile test / ISO 4136:2012 /	15614-1 / Mill test report	Qlab Material report				w	D		н
g and S ₁	5.	5.8	Macroscopic testing	ISO 17639:2013	EN ISO 5817:2014	Qlab Material report	Q-lab	The hardness samples from Qlab are analyzed.		w	D		н
Testin	5.	5.9	Perform Charpy V Impact test, KV8	NS-EN ISO 148-1:2016 / ISO 9016:2012	NS-EN ISO 15614-1:2017 / Mill test report	Qlab Material report	Q-lab			w	D		н
nical	5.:	.10	Qlab produces material report for sheet PL3, PL5 and PL7.	Welding procedure and production sample testing	NS-EN ISO 15614-1.2017 Level 2	Material report: 8612-2, 8612-3 and 8612 4.	Q-lab		D	н	D		н
lec hi	5.3	.11	Specimens preperation for fatigue testing.	Specimens preparation - Fatigue testing / ASTM E466-15	ASTM E466-15	Pictures	Q-lab			н			н
E. NDT, N	5.:	.12	Transport the proximity weld fatigue specimens from Q-lab to UIS. All tested and untested material shall also be sent to UIS.				Magnus	A total of 5 specimens. 2pcs of 50mm between the welds, 2pcs of 15mm between the welds and 1st of 5mm					
	5.:	.13	Transport all material stored at KIWA to UIS.				Magnus	between the welds. All material stored at KIWA is transporteted to UIS, this includes unwelded plate PL2, PL3 and PL10 and welded plates PL4, PL6 and PL8.		н			
scopic	6.	6.1	Preparation and interpretation of microstructure. The macros we received from Q-lab are embedded in Epoxy in a silicone				M&M	It is one macros from the start and one from the stop of each plate. Q-lab had 4					
Micre	UNESTI		form.			Report - Macroscopic and microscopic		plates which gives 8 pieces of macros. The equipment used for this study is an	<u> </u>	н			
	6.	6.2	Analysis of microstructure Machine the sides of the weld poximity	ISO 17639:2003		examinations of welds	Mattias	Inverted Metallurgical Microscope GX53	<u> </u>	н			D
	7.	7.1	fatigue specimens. Machine five fatigue specimens from plates without welding.	Doc - Fatigue test 2 / Doc - Fatigue test 3 / ASTM E466- 15 / ASTM E468-11	ASTM E466-15	Pictures	Emil (UIS) and Magnus	The maching should be done according to ASTM E466-15 Appendix X1, step X1.2.1.		н			
	7.	7.2	Transports all fatigue specimens to CASTOLIN TRIO AS for precision grinding.				Magnus	Five weld proximity fatigue specimens and five base material fatigue specimens.					
								The maching should be done according to ASTM E466-15 Appendix X1. This step		n			
5	7.	7.3	CASTOLIN TRIO AS performs precision grinding with a Surface grinding machine.	Doc - Fatigue test 2 / Doc - Fatigue test 3 / ASTM E466- 15 / ASTM E468-11	ASME E466-15	Pictures	CASTOLIN TRIO	goes under X1.2.2. Ten specimens receive precision grinding. Five welded specimens and five from base material.		w			
stigati	7.	7.4	Transports fatigue specimens from CASTOLIN TRIO AS to UIS for final surface treatment.				Magnus			н			
F. Fatigue Test Inve	7.	7.5	Last surface treatment of fatigue specimens is carried out at the lab at UIS.	Doc - Fatigue test 2 / Doc - Fatigue test 3 / ASTM E466- 15 / ASTM E468-11	ASME E466-15	Visual inspection log / Picture / Roughness check	Magnus	Last surface treatment is carried out at the lab at UIS. This includes step X1.2.2. X1.2.4 in ASTM 466-15, roughness check were maximum surface roughness is .2um in the longitudinal direction and a final visual check with magnifying glass. Test is lubricated in rust protection while waiting for test.		г			
	7.	7.6	Perform fatigue testing	Doc -Fatigue test 1 / Doc-Fatigue test 2 / Doc- Fatigue test 3	DNVGL-RP-C203 / ASME E466-15 / ASTM E468-11	Fatigue test log	Magnus och Jörgen	Conducted at the University of Stavanger. There are three different tests to be carried out. These are: Doc-Fatigue test 1 (the specimens we produced in Stage 1- Prior to fabrication), Doc-Fatigue test 2 (base material) and Doc-Fatigue test 3 (weld proximity investigation test).		н			
-	8.		From plate PL4, PL6 and PL8 two pieces were cut out per plate. These should be prepared and sent to the VEQTER - Residual Stress Specialist.	Doc - Ultrasonic measurement test		Picture	Magnus and Xavier (VEQTER)	Two different types of samples are produced for each plate. The first is cut- out part where only the weld is sanded down with flap disc, the other is a fatigue specimen. The goal of the test is to see how much residual stress we have between the welds and how much residual stress is bot when we perform machining of fatigue specimens.		Т			
stress test investigatio	8.		Preparation of Specimen 1 - Fatigue specimen for Ultrasonic measurement test.	Doc-Ultrasonic Measurement Test / ASTM E466-15 / ASTM E468-11	ASME E466-15	Picture	Magnus	This is prepared according to ASTM E466- 15. Unlike previous fatigue specimens, precision grinding will be performed on the UIS. Step 2-X1.2.2 will be performed with cylindrical grinding (Surfaces Grinding Machine).		н			
ual str	8.	8.3	Preparation of Specimen 2 - Residual stress specimens for Ultrasonic measurement test.	Doc-Ultrasonic Measurement Test		Picture	Magnus	Weld end machined flush with the plate edges (flap disc).		н			
F. Residu	8.	8.4	Six specimens are sent with DHL to VEQTER in UK for analysis of residual stresses between welds.			Pictues	Magnus	The plates from which the samples are taken are PL4, PL6 and PL8. There are two samples per plate.		н			
1	8.	8.5	VEQTER conducts Residual Stress Analysis.			Residual Stress Report	VEQTER	The method used to measure residual stress is the Ultrasonic Measurement Test.		D			
1	8.	8.6	Specimens are sent back to UIS (Stavanger) from the UK with DHL.				VEQTER	stress is the Ultrasonic Measurement Test. Specimens should be handled with care when surface preparation can be					
	8	8.7	Fatigue tests are performed on the 3 specimens that have been tested for residual stresses	Doc-Ultrasonic Measurement Test	DNVGL-RP-C203 / ASME E466-15 / ASTM E468-11		Magnus	Before testing is carried out, the specimens must be inspected visually and in case of defects, surface preperation is					
			an carca.					done before testing.		н		1	

• Doc A102 - MDS Base material - S355J2+M

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37 Heat No. Test type C70 C [56] MN [56] S [56] N [56] CU [56] NI [56] CR [56] MO [56] V [56] AL [56] B [56] TI [56] NB [56] B08 86338Y H 0 0.16 1.29 0.026 0.026 0.025 0.01 0.06 0.001 0.000 Pieces Bur 86338Y H 0 0.16 1.29 0.100 0.026 0.012 0.005 0.04 0.02 0.07 0.01 0.06 0.003 0.0004 0.001 64 0 107 Heat No. Test type C70 AS [96] SN [96] CR [96] 0.02 0.021 0.020 0.001 64 0 0 0 0 0 0 0 0.001 0.001 0.001 0.001 0.001 0.001 0.001 0.001 0.001 0.001 0.001 0.001 0.002 0.0010 0.39 0 0 0 0 0 </th <th>AN/</th> <th>AL Y 515</th> <th></th> <th></th> <th></th> <th></th> <th></th> <th></th> <th>010011</th> <th></th> <th>0-08</th> <th></th> <th></th> <th>JORDI</th> <th></th> <th>EN 100</th> <th>25-2/20</th> <th>JU4 WI</th> <th></th> <th>GATION</th> <th>1141</th>	AN/	AL Y 515							010011		0-08			JORDI		EN 100	25-2/20	JU4 WI		GATION	1141
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Image: Section of the sector of the	07 He	at No.	12 17 180	C70	C [%]	MN [%]	SI [%]	P [%]	S [%]	N [%]	CU [%]	NI [%]	CR [%]	MO [%]	V [%]	AL [%]	B [%]	TI [%]	NB [%]	B	80
86338Y H 0 0.15 1.29 0.192 0.021 0.005 0.04 0.02 0.07 0.01 0.05 0.003 0.004 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001 < 0.001			- 37		>0 <0.2	>0 <1.6														Pieces	Bund
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5 Test results 2 Tensile test acc.to EN ISO 6892-1:2009 4 Charpy impact test acc.to EN ISO 148-1:2010 Heat No. C00 Specimen No. C02 Vield or limit C12 Tensile Elongati proof strength C13 on A5 C13 Elongati on A5 C13 Test temper sture (C) C40 KV2 C41 7.50 Main A5 A5 A5 A5 A6 C04 C04 J (J/om2) min Max C04 Regulation >365 >470 <030																					
ISO 148-1:2010 Heat No. C00 Specimen No. C11 Yield or limit C12 Tensile Elongati strength C13 on A5 C13 Tensile Elongati strength C13 on A5 C40 kV2 C41 7.50 Image: No. No. V A6 Image: No. °C C04 min C04 max J (J/om2) Image: No. S365 >470 <630 22.0 Image: No. Provide Signal Provide Signal Provide Signal Image: No. J (J/om2) Image: No. S365 >470 <630 22.0 Image: No. Provide Signal Image: No. Image: No. J (J/om2) Image: No. S6336Y 20423404 0 414 541 30.8 Image: No. Provide Signal Image: No. Provide Signal Image: No.	863	_					D.0010	0.39													
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C04 Regulation >365 >470 <630 22.0 22.0 21.00 86336Y 20423404 0 414 541 30.8		1 Cor	est resu CD0 Specim	Its	2 C11 Yield of proof	C12 Tensile	C13 Elongat	2 Tensil	e test a	acc.to E		6892-1	2009			CD3 Test temper sture	1 <u>sc</u>	148-1	:2010		
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1 Continuation see Attachment	Hes 863	5 Te t No. 36Y 36Y	cov st resu st resu Specim No. Cov Regular 204234 204234	Its en CO ion 404 0 407 0	2 C11 Yield or proof limit >365 414	t Tensile strengti >470 <€30 541	C13 Elongat on A5 A5 22.0	2 Tensil	e test a	acc.to E		6892-1	2009			CD3 Test temper sture (*C) *C -20.00	C04 min 21.00 C43	0 148-1 40 KV2 C04 max	2010 C41 J (J4 C42	7.50 (om2)	
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Bend test according to EN ISO 7438:2005	Hea 863 863 863	5 Te 5 Te t No. 36Y 1 Con	tinuation est resu Specin No. C04 Regular 204234 204234 204234 204234	Its Its CO tion 104 0 407 0 see At	2 C11 Yield or proof limit >365 414 tachmen	C12 Tensile strengti >470 <030 541	2 Elongat on A5 22.0 30.8	2 Tensil	e test a	acc.to E		6892-1	2009			CD3 Test temper sture (*C) *C -20.00	C04 min 21.00 C43	0 148-1 40 KV2 C04 max	2010 C41 J (J4 C42	7.50 (om2)	
52 Bend Test	Hea 863 863 Ben	5 Te 5 Te t No. 36Y 1 Con d test a Bend 1	coo specim No. CO4234 204234 204234 tinuation accordin Test	Its Its CO tion 104 0 407 0 see At	2 C11 Yield or proof limit >365 414 tachmen	C12 Tensile strengti >470 <030 541	2 Elongat on A5 22.0 30.8	2 Tensil	e test a	acc.to E		6892-1	2009			CD3 Test temper sture (*C) *C -20.00	C04 min 21.00 C43	0 148-1 40 KV2 C04 max	2010 C41 J (J4 C42	7.50 (om2)	
52 Bend Test 53 Rebend test	Hea 863 863 863 863	5 Te 5 Te t No. 36Y 36Y 1 Cor d test a Bend T	coo specim No. CO4 Regular 204232 204232 204233 tinuation accordin Test nd test	Its hen CO hen CO hon C	2 C11 Yield or proof limit >365 414 tachmen	C12 Tensile strengtr >470 <830 541 1	C13 Elongat on A5 22.0 30.8	2 Tensil		acc.to E		6892-1	2009			CD3 Test temper sture (*C) *C -20.00	C04 min 21.00 C43	0 148-1 40 KV2 C04 max	2010 C41 J (J4 C42	7.50 (om2)	
52 Bend Test	Hea 863 863 863 863	5 Te 5 Te t No. 36Y 36Y 1 Cor d test a Bend T	coo specim No. CO4 Regular 204232 204232 204233 tinuation accordin Test nd test	Its hen CO hen CO hon C	2 C11 Yield or proof limit >365 414 tachmen	C12 Tensile strengtr >470 <830 541 1	C13 Elongat on A5 22.0 30.8	2 Tensil		acc.to E		6892-1	2009	Cavil and Intervied Weblishi Ovarante	i machina e to be used liry: e i through	CD3 Test temper sture (°C) °C -20.00 -20.00	plication bolted and calent(Cr	0 148-1 40 KV2 C04 max 132	(J) C41 J (J) C42 104	7.50 (om2)	

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ArcelorMittal Ostrava a.s. Vratimovska 659 707 02 Ostrava-Kunčice Issuerci 18 99250 n document / 017

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WORKS INSPEKTOR IDENTIFICATION No. 14 Zdeněk Podešva PHONE: +420 595682303

replaces seal and signature Issued by: Sylvie Tkáčová

D01The inspection and the test were caried out on the delivered product or on a product test unit.

Z02, Z03, A05

1/3

e 1 o

• Doc A103 - MDS Filler metal - OERLIKON SPEZIAL



MMA Electrodes C-Mn and low-alloy steels

SPEZIAL is a basic, double-coated multi-purpose MMA electrode. The composition of the double coating confers exceptionally good welding characteristics and a highly stable and directional arc. Very good gap bridging and ideally suited for root passes and positional welding. The glassy slag is easily removed from the finely-rippled weld seams, the excellent welding characteristics and ISO-V toughness to -30°C.

Structural steelwork, production and assembly jobs in industry and for pipeline construction for decades. Very good gap bridging and ideally suited for root passes. Material to be welded S(P)235; S(P)355; GP 240; GP 280; L 245; L260.

ISO-V toughness at -30°C. Deposit free from porosity and good of X-ray quality. Optimum AC weldability requires an OCV > 65V. Very good gap bridging and ideally suited for root passes and positional welding. The glassy slag is easily removed from the finely-rippled.

Classif	ication						
EN ISO	2560-A: E 38 3 B 12 H10						
AWS A5.1: E 7016-H8							
Approv	als	Grade					
ABS		3YH10					
BV		3YH10					
DB		•					
DNV		3Y40H10					

Approvals	Grade
GL	3YH10
LRS	3YmH15
RMRS	ЗҮНН
ΤÜV	•
CE	

Chemical analysis (Typical values in %)

С	Mn	Si	Р	S
0.06	0.9	0.7	≤ 0.020	≤ 0.015

All-weld metal Mechanical Properties

Heat Treatment	Yield Strength	Tensile Strength	Elongation	Impact Energy ISO - V (J)	
	(MPa)	(MPa)	A5 (%)	+20 °C	-30 °C
As Welded	≥ 380	470-600	≥ 25	≥ 150	≥ 60

Materials

S(P)235-S(P)355; GP240-GP280; L245-L360

Storage

Keep dry and avoid condensation.

 $HD \le 10$: Re-dry at 300-350°C for 2 hours, 5 times max.

Current	t condit	tion and	l weldir	ig posit	tion
AC; DC+					
PA	PB	PC	PD	PE	PF



MMA Electrodes C-Mn and low-alloy steels

Packaging data

Diam.	Length	Current	Approx. weightn(kg/1000)	CBOX		VPMD	
(mm)	(mm)	(A)		PC	Code	PC	Code
2.0	350	55-65	12.73	330	W000287401	160	W000287407
2.5	350	55-95	19.50	200	W000287402	100	W000287408
3.2	350	80-150	32.73	125	W000287403	55	W000287409
3.2	450	95-150	41.82	125	W000287404	55	W000287410
4.0	450	120-190	65.00	80	W000287405	40	W000287411
5.0	450	190-250	100.45	50	W000287406		



Manufacturing and welding process, characterization of the Steel S420G2+M

- Material Selection and Cutting Process

- Doc B201 Mill test report Base Material S420G2+m HT43831-9133182
- Doc B202 Certified Material Test Report Filler material NSSW SM-47-A
- Doc B203 Certified Material Test Report Filler material NSSW SF-3AM
- Doc B204 Certified Material Test Report Filler material NSSW SF-3AM

• Doc B201 - Mill test report - Base Material S420G2+m -HT43831-9133182

CO #	Item #	Del #	Heat	Lot	Your art #	Qty	Description	NORSK STÅL
201904011/SWK	100	RP5367 20	43831- 9133182			1	HR PL S420G2+M/Q-Y30 15X2500X12000 MM	

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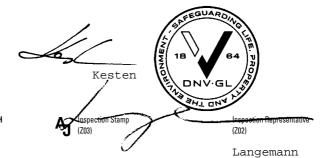
Inspection certificate 3.2 DIN EN 10204 1145686 No. (A03) 1/5 13.11.2018 Page (A02) Date P12001207 28.08.2018 No.(A07) No. (A07) Norsk Stal AS 4683 SOEGNE Purchaser Norsk Stal AS (A06) 4683 SOEGNE Customer (A06) NORWEGEN NORWEGEN Heavy plate Works order No. 0000096751 Product (B01) (A08) 0087902191 Dispatch note No. Steel grade and terms of delivery $\ S420G2{+}M$ 12.11.2018 (B02-B03) EN 10225 08/01 Inspection DNVGL MDS-Y30 Rev.5 DIN EN 10029 A 02/11 DIN EN 10163-2 Kl. A UG3 03/05 (A05) N141JYW8 312604 Application onboard DNV - classed objects is subject to special consideration. Marking of the product (B06)

Trademark/Steelgrade/Heat-No/Product-No/ inspector's stamp

Material data (B01-B99)

ltem	Quantity (B08)	Product No. (B07)	Heat No. (B07)	Cond. of delivery (B04)	Thickness x Width x Length mm x mm x mm (B09–B11)
02 02 02 02	1 1 1 1	913317 1 913317 2 913318 1 913318 2	43831 43831 43831 43831 43831	TM TM TM TM	15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000
Σ	4	Weight 14.1 (B12)	3.2 kgs	TM: thermome	chanically rolled

100



We hereby certify that the delivered material complies with the terms of the order. (Z01)

QM-System: Certification as per ISO 9001 since 28 February 1990

(A04)

Ilsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)

ILSENBURGER GROBBLECH

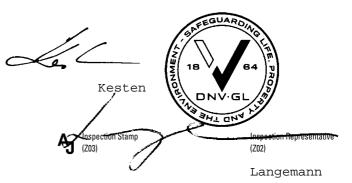
	ection cer EN 10204	tificate 3.2				No. (A03) Page Date	1145686 2/5 13.11.2018
No. (A07) Purchaser (A06)	P1200120 Norsk St 4683 SOE NORWEGEN	al AS GNE	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN	1	
Product (B01) Steel grade (B02–B03)	Heavy pl		Kl. A UG3 (sed objects is	Works order No. (A08) Dispatch note No. Inspection (A05) subject	0000096751 0087902191 12.11.2018 DNVGL N141JYW8 to special

Heat No. (B07)	C % ≤0,14	Si % 0,15-0,55	Mn % ≤1,65	₽ % ≤0,020	S % ≤0,005	N % ≤0,010	Al % 0,015-0,055	Cu % ≤0,30	Cr % ≤0,25	Ni % ≤0,70
43831	0,10	0,28	1,49	0,011	0,002	0,005	0,041	0,02	0,05	0,04
Heat No. (B07)	V % ≤0,080	Ti % ≤0,025	Nb % ≤0,040	B % ≤0,0005	MO % ≤0,25	As % ≤0,030	Sn % ≤0,020	Pb % ≤0,010	Ca % ≤0,005	EV1 1) % ≤0,22
43831	0,003	0,002	0,026	0,0001	0,002	0,002	0,005	0,001	0,002	0,19
Heat No. (B07)	Sb % ≤0,010	EV2 2) % ≤0,42	Bi % ≤0,010	EV3 3) % ≤0,11	EV4 4) % ≤0,90	Nb+V % ≤0,09				
43831	0,000	0,36	0,001	0,03	0,11	0,03				
	/n/20+Mo/15+Ni/60 n/6+Mo/5+Ni/15+C		0+Si/30+5xB	1	-, -	V+Nb+Ti Cr+Cu+Mo+Ni	1	1	1	1
Steel mał (C70)	king: Ba	sic oxy	gen pro	cess						

We hereby certify that the delivered material complies with the terms of the order. (Z01) QM–System: Certification as per ISO 9001 since 28 February 1990

(A04)

llsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)



ILSENBURGER GROBBLECH

-	ction cer N 10204	ctificate 3.2				No . (A03) Page Date	1145686 3/5 13.11.2018
No. (A07) Purchaser (A06)	P1200120 Norsk St 4683 SOE NORWEGEN	al AS GNE	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN		
Product (B01) Steel grade (B02–B03)	Heavy pl		Kl. A UG3 (,	ed objects is	Works order No. (A08) Dispatch note No. Inspection (A05) subject	0000096751 0087902191 12.11.2018 DNVGL N141JYW8 to special

Specimen No. (COO)	Heat No. (B07)	C % ≤0,14	Si % 0,15-0,55	Mn % ≤1,65	₽ % ≤0,020	S % ≤0,005	N % ≤0,010	Al % 0,015-0,055	Cu % ≤0,30
913146 *) 913146 *)	43831 43831	0,11 0,10	0,28 0,27	1,46 1,49	0,007 0,014	0,003 0,000	0,004 0,006	0,042 0,050	0,02 0,02
Specimen No. (C00)	Heat No. (B07)	Cr % ≤0,25	Ni % ≤0,70	V % ≤0,080	Ti % ≤0,025	Nb % ≤0,040	MO % ≤0,25	B % ≤0,0005	EV1 1) % ≤0,22
913146 *) 913146 *)	43831 43831	0,05 0,05	0,05 0,03	0,002 0,005	0,001 0,002	0,029 0,025	0,002 0,00	0,0000 0,0005	0,20 0,19
Specimen No. (C00)	Heat No. (B07)	EV2 2) % ≤0,42	EV3 3) % ≤0,90	EV4 4) % ≤0,11	Nb+V % ≤0,09				
913146 *) 913146 *)	43831 43831	0,37 0,36	0,12 0,10	0,03 0,03	0,03 0,03				

*) The sample product is not part of the delivery

We hereby certify that the delivered material complies with the terms of the order. (Z01) QM–System: Certification as per ISO 9001 since 28 February 1990



llsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)



Langemann

ILSENBURGER

	ection cer EN 10204	ctificate 3.2				No . (A03) Page Date	1145686 4/5 13.11.2018
No. (A07) Purchaser (A06)	P1200120 Norsk St 4683 SOE NORWEGEN	al AS GNE	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN	1	
Product (B01) Steel grade (B02-B03)	Heavy pl	S420G2+M EN 10225 08/01 MDS-Y30 Rev.5 DIN EN 10029 A	Kl. A UG3 (sed objects is	Works order No. (A08) Dispatch note No. Inspection (A05) subject	0000096751 0087902191 12.11.2018 DNVGL N141JYW8 to special

Tensile test (C10-C29)

							-		
Specimen No. (COO)	Heat No. (B07)	Location (C01) 1) 2) 3)	Direct. (CO2) 4)	Cond. (B05) 5)	Type (C10) 6)	Yield point (C11) ReH N/mm2 420 - 540	Tensile strength (C12) Rm N / mm2 500 - 660	ReH/Rm ≤ 0,93	Elongation (C13) A.5 7) % ≥ 19
913318	43831	K4G	Q	TM	P	489	557	0,88	31
1) K: Top 2) 4: 1/4 Width 3) G: Thickness of product					•	5) TM : thermo 6) P : prisma 7) A5 : Lo=5,65			

4) Q: transversal

Specimen No. (C00)	Heat No. (B07)	Location (C01) 1) 2) 3)	Direct. (C02) 4)	Cond. (B05) 5)	Type of specimen (C40–C41)	Temperature (C03) ° C	Impact energy (C42–C43) ⊥ J ≥ 42	2	3	MW 6) J ≥60
913318	43831	к40	Q	TM	KV450	-040	118	111	114	114

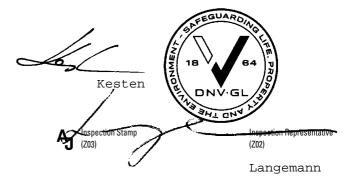
All plates have been ultrasonically tested according to EN 10160 07/99 in area and edges Class of areatesting: S1 Class of edgestesting: E2 Results: No faults were found.

We hereby certify that the delivered material complies with the terms of the order. $\ensuremath{\text{(Z01)}}$

QM–System: Certification as per ISO 9001 since 28 February 1990

(A04)

Ilsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)



ILSENBURGER GROBBLECH

	ction certificate 3.2 N 10204				No . (A03) Page Date	1145686 5/5 13.11.2018
No. (A07) Purchaser (A06)	P12001207 Norsk Stal AS 4683 SOEGNE NORWEGEN	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN	3	
Product (B01)	Heavy plate				Works order No. (A08)	0000096751
Steel grade (B02-B03)	and terms of delivery S420G2+M EN 10225 08/01 MDS-Y30 Rev.5 DIN EN 10029 A DIN EN 10163-2 312604 Application on consideration.	02/11 Kl. A UG3 board DNV -	,	sed objects is	Dispatch note No. Inspection (A05) subject	12.11.2018 DNVGL N141JYW8
do no which given	products are free of rad ot exceed the clearing l guarantees the complia in the Radiation Prote he unrestricted clearan	imit value ince with line ction Ordina	of 100 mit va ance(S) Bq/kg, alues StrlSchV)		

(StrlSchV Annex III, Section 5) for ferrous nuclides.

de l 18 Kesten DN ANT ON llsenburger Grobblech GmbH (Z03) inspection Representative Inspec Stamp Veckenstedter Weg 10 (Z02) D-38871 Ilsenburg Langemann

We hereby certify that the delivered material complies with the terms of the order. (Z01)

QM–System: Certification as per ISO 9001 since 28 February 1990

Trademark (A04)

(A01)

• Doc B202 - Certified Material Test Report - Filler material NSSW SM-47-A

Certified Material	Test Report	Chiba Plant
According to EN 1	0204, type 3.1	Nippon Steel & Sumikin Welding Co., Ltd.
Test Report No. :	N06547	Purchaser's Spec. No. : NSTN-13 Rev.0
Date of issue :	June 29, 2017	Material Classification
Purchaser :	Norsk Sveiseteknikk AS	EN ISO 17632-A-T 46 6 1Ni M M 1 H5
Trade Designation :	NSSW SM-47A	AWS A5.36 E80T15-M21A8-Ni1-H4

1. 1.1.40011415				
Trade Designation	Size	Manuf. No.	Flux lot No.	Manuf. Date
NSSW SM-47A	1.2 mm	7U341AW996	7V311	June 12, 2017
Test type		Test	Unit	
EN ISO	14344 Lot c	elass : T1 (each flux lot), Tes	ting Schedule : 4	(exclude soundness test)

2. Welding Condition (Test type : ISO)

Base Met	al KL3 nm) 20tx250w	-		ding ition	Flat	5	Kind of Gas		80% Ar 20% CO ₂	CTWI) (mm)	20
	Current DC(+) (A)	Volt (V			Speed min)	Flo	w Rate of ((l/min)	Jas	Preheat 7 (°C)	ſemp.	Inter-p	oass Temp. (℃)
	270	30	0	3	0		25		140		135	~ 163
Spec.	270 ± 10	30 =	± 2	30	± 5		$20\sim 25$		$135 \sim$	163	135	$5 \sim 163$

3. Test results of All-Weld-Metal (Test type : ISO)

Tension Test	Conditions	Yield Strength (MPa)	Tensile Strength (MPa)	Elongation (%)
	AW	527	617	28
Spec.		460 Min.	$530 \sim 680$	20 Min. A5
Impact Test	Conditions	Temp. (°C)	Ind. Value (J)	Ave. (J)
I	AW	-60	93, 104, 115	104
Spec.		,	32 Min.	47 Min.
Spec.				

	Chemical composition (%)											
	С	Si	Mn	Р	S	Cu	Ni	Cr	Mo	V	Nb	
	0.07	0.59	1.26	0.009	0.008	0.25	1.02	0.02	0.01	0.01	0.01	
Spee	0.04/	0.40/	1.1/	0.020	0.020	0.3	0.80/	0.15	0.15	0.05	0.05	
Spec.	0.10	0.80	1.4/	Max.	Max.	Max.	1.10	Max.	Max.	Max.	Max.	

S, P and N of the strip : Below specified maximum

4. Hydrogen Content of Deposited Metal (Acc. to ISO 3690)

HDM (ml/100g)	Ave.	Spec.
1.0, 1.7, 1.0	1.2	5 Max.

Statement:

We hereby certify that the contents of this report are correct and accurate, and all operations performed by us or our subcontractors are in compliance with the requirements of the DATA SHEET No. : NSTN-13 Rev.0

No.	170639HFH	(MIT	9118)	Net Weight	
				711941AW006	17 00

Certified by

ISAO KANAUCHI Group Manager, Quality Control Dept. • Doc B203 - Certified Material Test Report - Filler material NSSW SF-3AM

Certified Mate	rial '	Test Report	Chiba Plant					
According to E	N 10	0204, type 3.1	Nippon Steel & Sumikin Welding Co., Ltd.					
Test Report No.	:	N06726	Purchaser's Spec. No. : NSTN-03 Rev.3					
Date of issue	:	September 6, 2017	Material Classification					
Purchaser	:	Norsk Sveiseteknikk AS	EN ISO 17632-A-T 46 4 Z P M 2 H5					
Trade Designati	on :	NSSW SF-3AM	EN ISO 17632-A-T 46 6 Z P M 2 H5 AWS A5.36 E81T9-M21A8-Ni1-H4					

Trade Designation	Size	Manuf. No.	Flux lot No.	Manuf. Date						
NSSW SF-3AM	1.2 mm	7S041MP960	7T051	August 02, 2017						
NSSW SF-3AM	1.2 mm	7S071MP960	7T051	August 03, 2017						
NSSW SF-3AM	1.2 mm	7S101MP960	7T051	August 04, 2017						
Test type	Test Unit									
EN ISO	14344 Lot c	14344 Lot class : T1 (each flux lot), Testing Schedule : 4 (exclude soundness test)								

2. Welding Condition (Test type : ISO)

Base Met	al KL3 nm) 20tx250w	-		ding ition	Flat	;	Kind of Gas		80% Ar 20% CO ₂	CTWI) (mm)	20
	Current DC(+) (A)	Volt	age 7)		Speed min)	Flo	w Rate of ((l/min)	Jas	Preheat 7 (°C)	ſemp.	Inter-p	oass Temp. (°C)
	270	2	7	2	5		25		140		135	~ 163
Spec.	270 ± 10	27	± 2	25	± 5		$20\sim 25$		$135 \sim$	163	135	~ 163

3. Test results of All-Weld-Metal (Test type : ISO)

Tension Test	Conditions	Yield Strength (MPa)	Tensile Strength (MPa)	Elongation (%)
	AW	556	614	26
Spec.		460 Min.	$530 \sim 680$	20 Min. A5
Impact Test	Conditions	Temp. (°C)	Ind. Value (J)	Ave. (J)
	AW	-40	112, 126, 142	127
Spec.			32 Min.	47 Min.
	AW	-60	82, 80, 87	83
Spec.			32 Min.	47 Min.

	Chemical composition (%)											
	С	Si	Mn	Р	S	Cu	Ni	Cr	Mo	V	Nb	
	0.06	0.31	1.22	0.010	0.005	0.26	1.02	0.02	0.01	0.01	0.01	
Spee	0.03/	0.25/	1.00/	0.020	0.020	0.40	0.80/	0.15	0.35	0.05	Ref.	
Spec.	0.07	0.60	1.50	Max.	Max.	Max.	1.10	Max.	Max.	Max.		

S, P and N of the strip \vdots Below specified maximum

4. Hydrogen Content of Deposited Metal (Acc. to ISO 3690)

HDM (ml/100g)	Ave.	Spec.
1.4, 1.2, 1.3	1.3	5 Max.

Statement :

We hereby certify that the contents of this report are correct and accurate, and all operations performed by us or our subcontractors are in compliance with the requirements of the DATA SHEET No. : NSTN-03 Rev.3

No. 170830HFH	(MIT 9188)	Net Weight
		7S041MP9
		7S071MP0

Net Weight	_		
7S041MP960	:	2,575.0	kg
7S071MP960	:	4,275.0	kg
7S101MP960	:	1,375.0	kg

• Doc B204 - Certified Material Test Report - Filler material NSSW SF-3AM

Certified Materia	l Test Report	Chiba Plant
According to EN	10204, type 3.1	Nippon Steel & Sumikin Welding Co., Ltd.
Test Report No. 3	N07653	Purchaser's Spec. No. : NSTN-03 Rev.4
Date of issue :	April 23, 2018	Material Classification
Purchaser :	Norsk Sveiseteknikk AS	EN ISO 17632-A-T 46 4 Z P M 2 H5
Trade Designation	NSSW SF-3AM	EN ISO 17632-A-T 46 6 Z P M 2 H5 AWS A5.36 E81T9-M21A8-Ni1-H4

Trade Designation	Size	Manuf. No.	Flux lot No.	Manuf. Date				
NSSW SF-3AM	1.2 mm	8X221MP996	8Y261	March 08, 2018				
Test type		Test	Unit					
EN ISO	14344 Lot class : T1 (each flux lot), Testing Schedule : 4 (exclude soundness test)							

2. Welding Condition (Test type : ISO)

Base Met		KL33 20tx250wx300L				<u> </u>		5	Kind of Gas		80% Ar 20% CO ₂	CTWI) (mm)	20
	Current DC(+) (A)				Speed Flo min)		w Rate of Gas (l/min)		Preheat 7 (°C)	lemp.	Inter-p	oass Temp. (°C)		
	270		27 2		5		25		140		135	~ 163		
Spec.	Spec. 270 ± 10		±2	25	± 5		$20\sim 25$		$135 \sim$	163	135	$5 \sim 163$		

3. Test results of All-Weld-Metal (Test type : ISO)

Tension Test	Conditions	Yield Strength (MPa)	Tensile Strength (MPa)	Elongation (%)
	AW	535	606	27
Spec.		460 Min.	$530 \sim 680$	20 Min. A5
	0 1:1:		T 1 T7 1 (r)	
Impact Test	Conditions	Temp. (°C)	Ind. Value (J)	Ave. (J)
	AW	-40	137, 151, 162	150
Spec.			32 Min.	47 Min.
	AW	-60	115, 95, 74	95
Spec.			32 Min.	47 Min.

	Chemical composition (%)												
	С	Si	Mn	Р	S	Cu	Ni	Cr	Mo	V	Nb	B	
	0.05	0.27	1.15	0.010	0.004	0.30	1.03	0.03	0.01	0.01	0.02	0.003	
Shoo	0.03/	0.20/	1.00/	0.020	0.020	0.40	0.80/	0.15	0.35	0.05	Ref.	Ref.	
Spec.	0.07	0.55	1.50	Max.	Max.	Max.	1.10	Max.	Max.	Max.			

S, P and N of the strip : Below specified maximum

4. Hydrogen Content of Deposited Metal (Acc. to ISO 3690)

HDM (ml/100g)	Ave.	Spec.
1.8, 1.2, 1.0	1.3	5 Max.

Statement :

We hereby certify that the contents of this report are correct and accurate, and all operations performed by us or our subcontractors are in compliance with the requirements of the DATA SHEET No. : NSTN-03 Rev.4

No.	180424HFH	(MIT	9588)	_	Net Wei

Net Weight			
8X221MP996	:	16,000.0	kg

Certified by <u>SHUSHIRO NAGASHIMA</u> Group Manager, Quality Control Dept.

This is electronically made and is not signed.



Manufacturing and welding process, characterization of the Steel S420G2+M

- Welding Procedure Qualification Program

- Doc C301 WPQR: S420PL1-M
- Doc C302 All WPS's
 - WPS PL1-SW
 - WPS: PL3/4-DW50-Rep
 - WPS: PL5/6-DW15-Rep
 - WPS: PL7/8-DW5-Rep

• Doc C301 - WPQR: S420PL1-M

ι	S		iver Stav	sity anger				ICA	PROCE FION R PQR)			Re Da	te:	Magni 2019.0	us Larsson	Rev.4
Prod. b Project Locatic				Weld Proxin	nity IN		nt: spec.: pWPS:	NO	versity of S RSOK M-1 20PL1-M				Ref. stan Exam. bo		-EN ISO 156 knologisk Ins	
Weldin	a proce	ess		138					136							
Shieldi					n /189	%CO2 (M21)		2		18%C0	D2 (M21)		3			
Weavir				ves		max: 7	mm	-	yes		ax: 13	mm			max:	mm
_	g gas ty			N/A				l/min	,		a. 19				max.	
Weldin	g positi	ions		PF, vertica	l up											
Joint ty	ре			BW												
loint pi	reparat	ion		Waterjet c	utting	g, Grinding					1		60		1	
Cleanir	ng meth	nod		Wire brus	1						F				1	
Backin				N/A							1	1	5			
Single/				Single	-							1		5		
Back g				N/A	_						T		4		TT	2
-lux de Flux ha	signati	on		N/A N/A					F		I	/	-		11	-
	en elec	trode		N/A				mm					2	nose: 0.5 - 1.0	mm	
Torch a				70-90	0								3.	4mm		
Stand of	off dista	ince		10-25			_	mm								
Nozzle	diamet	ter(s)		10-20				mm								
Tack w	elding	proc.		General			Rev: 0									
denti	ficatio	n of	parent	metal	I: C	CE max: 0,42	C ma	ax: 0,	14 PCM	max: (),22 II:	CE ma	x: 0,42	C max: 0	0,14 PC	M max:0,22
Part		Nam	e/Grad	e		Standard		Grou	p D	elivery	cond.	т	nickness [mm]			eter range [mm]
_	S420G	2+M, I	MDS-Y3	80 Rev.5	EN 10	225:2009		2.1	TM				15,0	0		•
1					EN 10	225:2009		2.1	TM				15,0	0		243
	ricatio	n of	filler m	netal												
ndex		B	rand na	ame			Classifi	cation			Grou	ip		Fill	er handling	
C.	NSS	W SM	47A		EN	ISO 17632-A	T 46 6 1	NIM	A 1 H5		-				ommendati	
2	NSS	W SF-	3AM			I ISO 17632-A I ISO 17632-A					7		Supplie	ers reco	ommendati	.on
Noldi			lore							1	4.					
Veldi Pass			weldir	a Wir	2	Current	Mell			quipm	ent: ng speed f		Longth	Gas	P. 10	eat input
no.	Index	[mm]	metho		ł	[A]	Volta [V]	0	Polarity		im/min]			[l/min]		[kJ/mm]
BW:		4.05				-				-	5					-
I(Root) 2 (HP)		1.20	138 136	2,2		103 214	14. 23.		DC+ DC+		6 - 71 3 - 181	N/.		20		.29 - 1.61 .68 - 1.99
(FILL)	2	1.20	136	7,1		214	23.		DC+		4 - 154	N/.		20		.97 - 1.97
(FILL)		1.20	136	7,1		214	23.		DC+		8 - 211	N/		20		.44 - 1.46
5 (CAP) 5 (CAP)	2	1.20	136 136	7,1		204	22.		DC+ DC+		5 - 223 1 - 199	N/.		20 20		.21 - 1.32 .35 - 1.57
(5747)	-		.50	,,,				-				14//		20		
leat t	reatm	ent		-)(t						Metho	d:	-	76	
rehea	t min:	20	°C	Interpass t	emp. r	max: 117	°c He	attreat	ment proc.:			Те	mp. cont	rol: Digital	contact t	hermometer
WHT		-	°C m		°C			min/m			^{min} Hea	ting rate			Cooling rate	
Remark												0		d (Yes/No	-	
				ents, se 3831-913							Det	lanct	d	41-	-	
	able	manu		: 70341		6, 7S041MP	960, 7	S0711	4P960,			Signature	agnus d	arsso		
S101M	,										Approv	led 1		TA	ALADE	Palas

Kiwa Teknologisk Institutt

Summary

pWPS-S420PL1-M_Rev4

Weldlogg_Rev4

DigitalRT - Report 7967-19-DRT-1

Material Report 8612-1_Rev B

Material certificate S420G2+m - 43831-9133182

Consumable certificate - NSSW SM-47A - 7U341AW996 MIT 9118

Consumable certificate - NSSW SF-3AM - 7S041MP960 MIT 9188

Consumable certificate - NSSW SF-3AM - 8X221MP996 MIT 9588

J	S		ivers Stav	sity ange	r		-			S PROCEDURE Ref.: N N (pWPS) Date: 2					L1-M s Larsso 4.05	on Rev.4
Prod. b	y:	KIWA	TI.			Clien	t:	Univ	versity of S	tavang	er	Ref	. stai	nd: NS-	EN ISO	15614-1:2017
Project	:	MSc T	'hesis - V	Veld Pro	kimity	INVES. Ref. s	spec.:	NO	RSOK M-1	01:201	1 Rev.5	Exa	ım. b	ody: Tek	nologisk	Institutt (KIWA)
Locatio	n:	Gener	al			Ref. \	NPQR:	N/A								
Weldin	g proce	ess		138					136							
Shieldi	ng gas	type		1 Arg	on /18	3%CO2 (M21)		2	Argon /1	8%CO	2 (M21)		3			
Weavir	ıg (yes	/no)		yes		max: 7	mm		yes	ma	ax: 13	mm			max:	mm
Purging	g gas t	ype	1	N/A			l	/min								
Weldin	g posit	ions	I	PF, verti	cal up)					W	/PQR N	No.:	S420PL1-	М	
Joint ty	ре			BW												
Joint p	eparat	ion	1	Waterjet	cuttir	ng, Grinding					\setminus		60	0	/	
Cleanir	ng met	hod		Wire bru	sh						4				1	
Backing	-			N/A								\leftarrow	<⊂a			
Single/				Single								F	~			
Back g				N/A					\vee	\geq	- 1		Fi	п /	П	
Flux de	-	ion		N/A N/A							•		$\sqrt{2}$	nose: 0.5 - 1.0	L L	
Flux ha		otrada		N/A				mm						3-4mm		
Torch a		lioue		70-90	0											
Stand of	<u> </u>	ance		10-25			r	mm								
Nozzle			•	10-20			r	mm								
Tack w	elding	proc.		General			Rev: 0									
Identi	ficatio	on of	parent	metal	I:	CE max: 0,42	C max	x: 0,	14 PCM	max: 0	0.22 II: C	E max: (),42	C max: 0	,14	PCM max: 0,22
Part		Nam	ne/Grade)		Standard	,	Grou	p D	elivery	cond.	Thic	knes	s range	Di	ameter range
								0.00	P	0	ooa.		[mr	-	2.	[mm]
I	S4200	32+M,	MDS-Y3	0 Rev.5	EN 1	0225:2009		2.1	тм				15,	00		-
II	S4200	€2+M,	MDS-Y3	0 Rev.5	EN 1	0225:2009		2.1	тм				15,	00		-
Identi	ficatio	on of	filler m	etal												
Index		В	Brand na	me			Classifica	ation			Group			Fille	er handl	ing
1	NSS	W SM	-47A		E	EN ISO 17632-A	T46 6 1N	li M M	I 1 H5	I	7M1	Su	ppli	lers reco	mmenda	ation
2	NSS	W SF-	3AM			EN ISO 17632-A-				I	FM1	Su	ppli	lers reco	mmenda	ation
					E	EN ISO 17632-A-	-1 46 6 Z	P M 2	2 H5							
۱۸/مام!		rome	tora						-							
Weldi Pass	-			a 144	ire	Current	1/04-	70		quipme		n ()	n all	0		Hoot insut
Pass no.	Index		Weldin methoo		ire ed	Current	Volta	ye	Polarity		ng speed Ru		ngth			Heat input
BW:		[mm]			ed	[A] -	[V]			[m	m/min]	[mm]		[l/min]		[kJ/mm] -
вvv: 1(Root)	1	1,20	138	2,2 -		- 100 - 120	- 14.5 - 1	5	DC+	6	- D - 80	N/A		20		- 1.3 - 1.6
Fill	2	1,20	136	7.0	8.0	210 - 230	22 - 2	3	DC+	140	0 - 200	N/A		20		1.4 - 2.0
CAP	2	1,20	136	6.5	7.5	200 - 210	22 - 2	3	DC+	16	0 - 220	N/A		20		1.2 - 1.7
Heat t	reatm	ent										Method:		-		
Prehea	t min:	20	°C	Interpase	temp	. max: 250	°c Hea	ttreat	ment proc.:			Tem	o. co	ntrol: Digi	tal conta	ct thermometer
PWHT	min:		°C m	iax:	°(^C Soaking:		min/m	m	I	^{min} Heatin	g rate:		°C/h C	ooling r	ate: °C/h
Remark			- -		L.2 -						Additiona	al info er	nclos	ed (Yes/ No)):	
The p	VPS is	s cre	ated w	nth es	τıma	ted and fixe	ed para	amete	ers.		Date/Sig	nature:				
Posit	lon f	ixing	r: 3 st	rongba	cks	to be used.					2019.04.		gnus	Larsson		
											Approved	d:				
											2019.04.		tias I	arsson		
											1					

Welding procedure qualification record procedure



Kiwa Teknologisk Institutt

WPQR No .: 5420 PL1 - M

Place, date	Weldeye index: 630
Manufacturer	Stavenger, 08.04.2019
Project	Magnus Larsson and Mattins Larsson
Ref.stand	Master Thesis - Weldon weld
	NS-EN 15614:2017
Ref. Spec. Welders name / ID	NORSOK M-101: 2011 Rev. 5
	Bjørn Ivar Kaarvaag
pWPS Nr.	S420PLI-M
Identification of parent metal I	5420G2+M
Delivery cond./Heat no.	TMCP/43831-9133182
Pipe (T) eller Plate (P):	OD (mm): N/A WT (mm): 15 mm
Identification of parent metal II	\$420 G2+M
Delivery cond./Heat no.	TMCP / 43831 - 97 33 182
Pipe (T) eller Plate (P):	OD (mm): N/A WT (mm): 15mm
Identification of filler metal	NSSW SM-47A NSSW SF-3AM NSSW SF-3AM
Classification, batch/lot/heat	EN 150 17632 AT 466 1NI MM 1 HS/EN 150 17632-A.T 4642PM2HS/ER 150 17632 AT 466 2PM2H5
Filler handling	
Welding equipment, name/brand	Fronius - Trans Steel 2700
Welding process/processes	138, 136
Shield gas type	Argon / 18% CO2 (M21)
Purging gas type	NIA
Weaving, weld bead width	13mm
Welding position	PE
Joint type	BW
Joint preparation	Water Jet cutting, grinding
Cleaning method	Wire brush
Backing	N/A
Single/double	Size
Back gauging	N/A
Flux designation, flux handling	N/A
Tungsten electrode	N/A
Torch angle	70-90
Stand of distance	10-25mm
Nozzle diameter(s)	10-25 mm
Tack welding proc.	111 - Manual metal are welding
Heat treament, preheat, interpass	See Welding procedure quelification record procedure
Heat treament, preneat, interpass Heat treament method, temp.control	N/A
PWHT	N/A N/A
Additional information:	
Auutional information:	3 strongbacks to be used
K-factor:	1,0 (12) 0,8 (111, 131, 135, 114, 136, 137) 0,6 (141, 15
Welding coordinator, signature	dutte
Witnessed by, signature	(head Finneseure)

Welding procedure qualification record procedure



Kiwa Teknologisk Institutt

WPQR No .: S420 PL1 - M

Sketch,	60"
	I Material: S420 G2+M 15mm I Material: S420 G2+M
	3,2mm

Dimensions: (OD, WT, GAP, NOSE, ANGLE)

	Wel	ding p	arameters:														
	Pass no.	measure no.	Preheat / interpass °C	Filler OD	Welding process	Wire feed speed	Current Polarity AC/	Current	Volt	Welding length	Welding time	Welding speed	Shielding gas	Purging gas	Weld Bead width	Heat input kJ/mm	
- (7-4)				(mm)		(m/min)	DC-/DC+		(V)	(mm)	(s)	(mm/min)	(L/min)	(L/min)	(mm)		-
L	1	1	30	1.2	138	2.2	PCt	103	14.7	175	186	56	20	0	7	1.61	
Reet		2		1.2	138	2.2	DC+	103	14.7	220	187	71	20	0	7	1.29	
Q		3		1.2	138	2.2	DCt	103	14.7	115	103	67	20	0	7	1.36	
4b	2	1	42	1.2	136	7.1	Dct	214	23.6	340	113	181	20	0	13	1.68	1.1
		2		1.2	136	7.1	Det	214	23.6	150	59	153	20	0	13	1.99	
	3	1	75	1.2	136	7.1	Det	214	23.6	490	191	154	20	0	12	1.97	
Fich	4	1	100	1.2	136	7,1	Det	214	23.6	225	64	211	20	0	13	1.44	
1C		2		1.2	136	7,1	Det	214	23.6	260	75	208	20	0	13	1.46	
	5	1	103	1.2	136	7.1	DCt	204	22	275	74	223	20	0	13	1,21	
9		2		1.2	136	7.1	DC+	204	22	225	66	205	20	0	13	1.32	
Cap	6	1	117	1.2	136	7.1	DC+	204	22	200	70	171	20	0	13	1.57	
		2		1.2	136	7.1	Det	204	22	295	89	199	20	0	13	1.35	
e profilikanski k																	
			×														
						1	-										
																	14



Digital RT Digital RT

CLIENT / KUNDE KIWA Teknolo AS	ogisk Institutt	CLIENT O. 3517074		KUNDE O.NR			TE OF TEST 19-04-09	ING / KON	NTROLLDAT			RAPPORT NR. 9- DRT-1		GE/S f/av	
DRAWING NO. / TI N/A	EGNING NO.						RK / KONTR on, Bærhe		OPERATO	R / OPERATØ	R	ATTACI O	HMENT	/ VEE	DLEGG
OBJECT / KONTRO WPQR S420PL1-M Plate 15 mm 138/136 C/S	DLL AV														
PROCEDURE / PROSEDYRE BPI-01.Radiogra	aphic Examinatio	on ISO 176	36-2		RE O	V KO	TENT OF TE Ntrollom 0 %		AKSEPT	ANCE STANE STANDARD 5 75-1:2016	DARD /				
MATERIAL TYPE / Carbon steel	MATERIALTYPE		RFAC Wel	CE / OVERFLAT ded	Έ	GRC BW	OVE / FUGI	EGEOMET	RI		LDING I 3/136	PROCESS / SV	EISEPR	OSES	S
EQUIPMENT / UTS X-RAY TUBE / RØI Smart 300 Kv							ADIATION ECHNIQUE			ILDEPOSISJO	ON				
FOCAL SPOT SIZE 3 mm	/ BRENNFLEKK					·			1 2	2 3	4	5	6		7
	TUBE VO		-	B	С	D	I.Q.I. SOU SIDE		Å		7	*			R
EXP. DATA	DISTANO EXP.TIM mAmin. Cimin.	CE mm					FILMS TYPE OF I I.Q.I. TYPI W 10 FE	.Q.I. /		910	01		\mathcal{L}	λ	\checkmark
DETECTOR / DETEKTOR HD-IP Plus	SCREENS AND F SKJERMING OG I N/A			DETECTOR SPATIAL RI 0,063			PIXEL	STØRREL	.SE MIN. 1	SNR	TOTA UNSH 0.1	ARPNESS	MAGNI FORSTØ 1		
WELD NO SVEIS NR	DATE AND TIME DATO KL.	DIA. DIA		FILM LOCATION FILM PLASSERING	G F	ENSITIVITY ØLSOMHET % WIRE NR	WELDER SVEISER	MATERIAL THICKNESS TYKKELSE mm	BEDØMMELS	E FEILENS PLA	CATION SSERING	REMARKS BEMERKNING	GER I	EXP. DATA A-D	TECH. TEGN. 1-7
TEST TEST	09-04-2019	Plat Plat		20 - 240 240 - 460	w 1 w 1	-		15 15	Accepted Accepted	106-110		515 (4mm)			1
	09-04-2019			240 - 400		.5		15	Accepted						<u> </u>
		<u> </u>													<u> </u>
							-				_				
	_		-+							_					
100 = Sprekk 200 = 1	Hulrom porer 300 = 1	Fast inneslutti	ing Sl	agg 400 = Binde	feil og n	nanglende g	viennomsveisi	ng 401 = B	lindefeil						
402 = Rotfeil 500 = 1 COMMENTS / KON ** See weld	Jregelmessig form 50 IMENTARER IOG.)1 = Sårkant 6	500 = 1	Andre uregelmes	sigheter	(spesifiser)	~								
Result: Acce	pted.														
REPAIRS MARKEE			RKET	PÅ											
N3 NAME CERT. N ()	O. / N3 NAVN SERT	Г. NR.		N2 NAME (Sorin Erd			AVN SERT. I	NR.	NR			T. NO. / OPER	ATØR N	AVN	SERT.
APPROVED / GOD	KJENT DATO:			APPROVEI Approved			DATO:2019-	04-09		ERATOR / OP	PERATØ	R DATO:			

		MATE	RIAL REPO	RT	
Certificate no: 8612-1 Client: Master Thesis	Rev B UIS	Order no: 8612 Test date/Report date 07.06.2019	Quality lab	Quality Lab AS Industrivegen 54 4110 Forsand Norge	
		Client contact Mattias Larsson	Equipment identification: Mechanical testing of W	/PQ S420PL1-M	
Pages 4		Surveyor Mattias & Magnus Larsson	Standard		
Additional information: Rev B replaces	Rev A	Quality Lab Confirm . Comment to MPI added	ns that the results are o	connected to the object test	ed.
		Macro examinatio	n according to NS-E	N 17639:2013	
Etchant		Nital	Visual examination		<u>X5</u>
	Con		Macro Stop the acceptance crite	ria of NS-EN ISO 5817:201 vel B.	4

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MATE	RIAL	REP	

Certificate no: 8612-1
B
8612
Client:
Master Thesis UIS
07.06.2019



Quality Lab AS Industrivegen 54 4110 Forsand



quality lab Norge

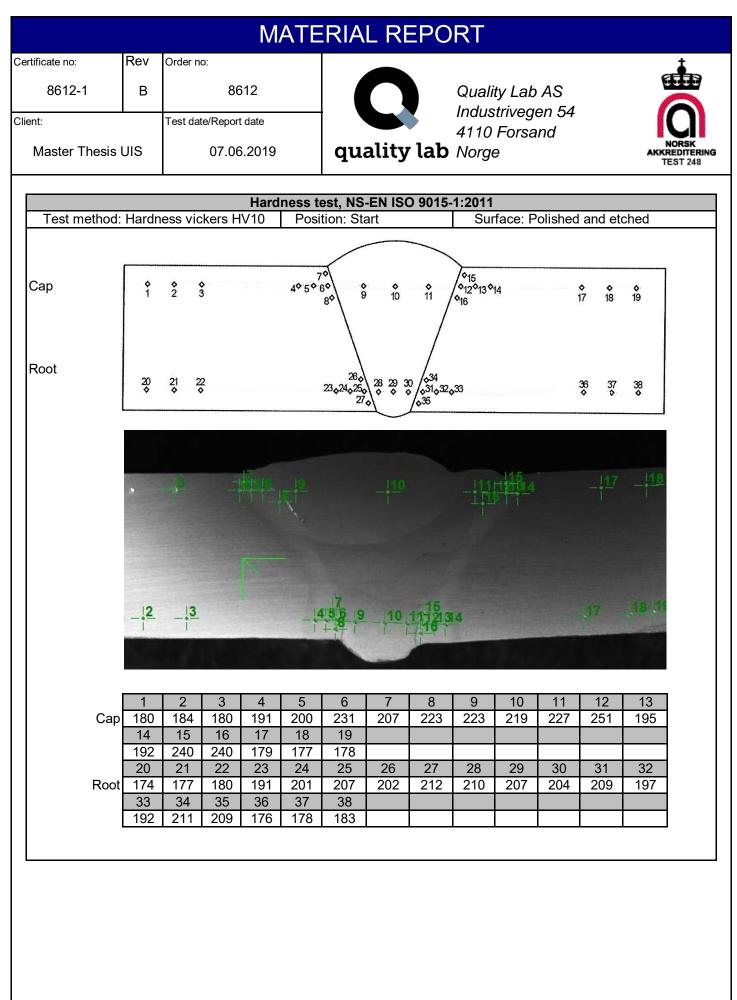
Tensile test	Tensile test, NS EN ISO 4136-1:2012 / ISO 6892-1:2016 Method A.1												
Test ident:	Area [mm ²]	R _m [Mpa]	Fracture										
Cross weld sample 1	260,10	522	Base material										
Cross weld sample 2	255,75	528	Base material										

Charpy V	Charpy V impact test, KV8, NS-EN ISO 148-1:2016; ISO 9016:2012													
Test ident	Dimension	Notch	Test temp	Sing	le value	es [J]	Average							
Test Ident	[mm]	Orientation	[⁰ C]	1	2	3	[J]							
Weld	10x10x55	Т	-40	80	131	106	106							
FL	10x10x55	Т	-40	113	186	193	164							
FL+2	10x10x55	Т	-40	253	130	200	194							
FL+5	10x10x55	Т	-40	234	236	173	214							

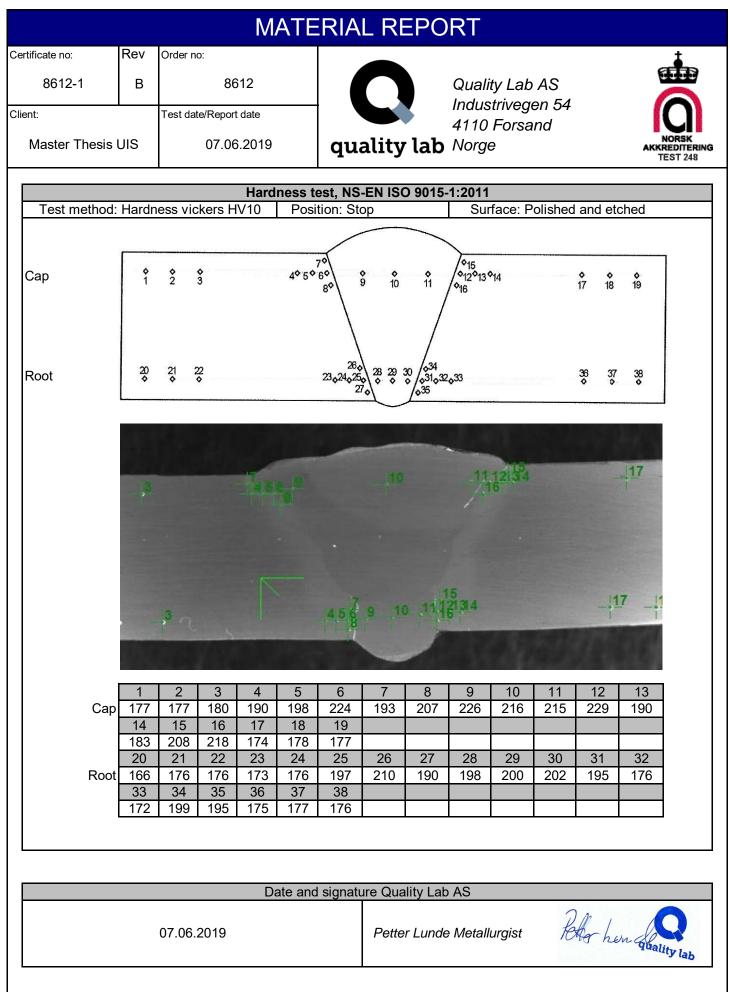
	Be	nd test, NS-EN ISO 5173:2010
Туре:	Side Bend	91.8x 0 +4
Former:	4T Dimension: 10mm	
Bend angle:	180°	
Comments:	Accepted	

MPI examination according to ISO 23278:2015 level B (non accreditated test)

MPI performed, No findings, Accepted



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The test report shall not be reproduced except in full, without written approval by the laboratory Page 4 of 4

CO #	Item #	Del #	Heat	Lot	Your art #	Qty	Description	NORSK STÅL
201904011/SWK	100	RP5367 20	43831- 9133182			1	HR PL S420G2+M/Q-Y30 15X2500X12000 MM	

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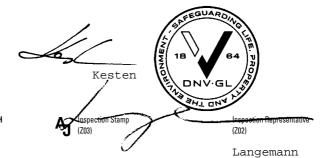
Inspection certificate 3.2 DIN EN 10204 1145686 No. (A03) 1/5 13.11.2018 Page (A02) Date P12001207 28.08.2018 No.(A07) No. (A07) Norsk Stal AS 4683 SOEGNE Purchaser Norsk Stal AS (A06) 4683 SOEGNE Customer (A06) NORWEGEN NORWEGEN Heavy plate Works order No. 0000096751 Product (B01) (A08) 0087902191 Dispatch note No. Steel grade and terms of delivery $\ S420G2{+}M$ 12.11.2018 (B02-B03) EN 10225 08/01 Inspection DNVGL MDS-Y30 Rev.5 DIN EN 10029 A 02/11 DIN EN 10163-2 Kl. A UG3 03/05 (A05) N141JYW8 312604 Application onboard DNV - classed objects is subject to special consideration. Marking of the product (B06)

Trademark/Steelgrade/Heat-No/Product-No/ inspector's stamp

Material data (B01-B99)

ltem	Quantity (B08)	Product No. (B07)	Heat No. (B07)	Cond. of delivery (B04)	Thickness x Width x Length mm x mm x mm (B09–B11)
02 02 02 02	1 1 1 1	913317 1 913317 2 913318 1 913318 2	43831 43831 43831 43831 43831	TM TM TM TM	15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000 15,00 x 2500,0 x 12000
Σ	4	Weight 14.1 (B12)	3.2 kgs	TM: thermome	chanically rolled

100



We hereby certify that the delivered material complies with the terms of the order. (Z01)

QM-System: Certification as per ISO 9001 since 28 February 1990

(A04)

Ilsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)

ILSENBURGER GROBBLECH

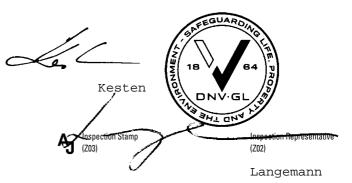
	ection cer EN 10204	tificate 3.2				No. (A03) Page Date	1145686 2/5 13.11.2018
No. (A07) Purchaser (A06)	P1200120 Norsk St 4683 SOE NORWEGEN	al AS GNE	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN	1	
Product (B01) Steel grade (B02–B03)	Heavy pl		Kl. A UG3 (sed objects is	Works order No. (A08) Dispatch note No. Inspection (A05) subject	0000096751 0087902191 12.11.2018 DNVGL N141JYW8 to special

Heat No. (B07)	C % ≤0,14	Si % 0,15-0,55	Mn % ≤1,65	₽ % ≤0,020	S % ≤0,005	N % ≤0,010	Al % 0,015-0,055	Cu % ≤0,30	Cr % ≤0,25	Ni % ≤0,70		
43831	0,10	0,28	1,49	0,011	0,002	0,005	0,041	0,02	0,05	0,04		
Heat No. (B07)	V % ≤0,080	Ti % ≤0,025	Nb % ≤0,040	B % ≤0,0005	MO % ≤0,25	As % ≤0,030	Sn % ≤0,020	Pb % ≤0,010	Ca % ≤0,005	EV1 1) % ≤0,22		
43831	0,003	0,002	0,026	0,0001	0,002	0,002	0,005	0,001	0,002	0,19		
Heat No. (B07)	Sb % ≤0,010	EV2 2) % ≤0,42	Bi % ≤0,010	EV3 3) % ≤0,11	EV4 4) % ≤0,90	Nb+V % ≤0,09						
43831	0,000	0,36	0,001	0,03	0,11	0,03						
	/n/20+Mo/15+Ni/60 n/6+Mo/5+Ni/15+C		0+Si/30+5xB	1	-, -	V+Nb+Ti Cr+Cu+Mo+Ni	1	1	1	1		
2) EV2: CEV=C+Mn/6+Mo/5+Ni/15+Cr/5+V/5+Cu/15 Steel making: Basic oxygen process (C70)												

We hereby certify that the delivered material complies with the terms of the order. (Z01) QM–System: Certification as per ISO 9001 since 28 February 1990

(A04)

llsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)



ILSENBURGER GROBBLECH

-	ction cer N 10204	ctificate 3.2				No . (A03) Page Date	1145686 3/5 13.11.2018
No. (A07) Purchaser (A06)	P1200120 Norsk St 4683 SOE NORWEGEN	al AS GNE	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN		
Product (B01) Steel grade (B02–B03)	Heavy pl		Kl. A UG3 (,	ed objects is	Works order No. (A08) Dispatch note No. Inspection (A05) subject	0000096751 0087902191 12.11.2018 DNVGL N141JYW8 to special

Specimen No. (COO)	Heat No. (B07)	C % ≤0,14	Si % 0,15-0,55	Mn % ≤1,65	₽ % ≤0,020	S % ≤0,005	N % ≤0,010	Al % 0,015-0,055	Cu % ≤0,30
913146 *) 913146 *)	43831 43831	0,11 0,10	0,28 0,27	1,46 1,49	0,007 0,014	0,003 0,000	0,004 0,006	0,042 0,050	0,02 0,02
Specimen No. (C00)	Heat No. (B07)	Cr % ≤0,25	Ni % ≤0,70	V % ≤0,080	Ti % ≤0,025	Nb % ≤0,040	MO % ≤0,25	B % ≤0,0005	EV1 1) % ≤0,22
913146 *) 913146 *)	43831 43831	0,05 0,05	0,05 0,03	0,002 0,005	0,001 0,002	0,029 0,025	0,002 0,00	0,0000 0,0005	0,20 0,19
Specimen No. (C00)	Heat No. (B07)	EV2 2) % ≤0,42	EV3 3) % ≤0,90	EV4 4) % ≤0,11	Nb+V % ≤0,09				
913146 *) 913146 *)	43831 43831	0,37 0,36	0,12 0,10	0,03 0,03	0,03 0,03				

*) The sample product is not part of the delivery

We hereby certify that the delivered material complies with the terms of the order. (Z01) QM–System: Certification as per ISO 9001 since 28 February 1990



llsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)



Langemann

ILSENBURGER

	ection cer EN 10204	ctificate 3.2				No . (A03) Page Date	1145686 4/5 13.11.2018
No. (A07) Purchaser (A06)	P1200120 Norsk St 4683 SOE NORWEGEN	al AS GNE	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN	1	
Product (B01) Steel grade (B02-B03)	Heavy pl	S420G2+M EN 10225 08/01 MDS-Y30 Rev.5 DIN EN 10029 A	Kl. A UG3 (sed objects is	Works order No. (A08) Dispatch note No. Inspection (A05) subject	0000096751 0087902191 12.11.2018 DNVGL N141JYW8 to special

Tensile test (C10-C29)

							-		
Specimen No. (COO)	Heat No. (B07)	Location (C01) 1) 2) 3)	Direct. (CO2) 4)	Cond. (B05) 5)	Type (C10) 6)	Yield point (C11) ReH N/mm2 420 - 540	Tensile strength (C12) Rm N/mm2 500 - 660	ReH/Rm ≤ 0,93	Elongation (C13) A.5 7) % ≥ 19
913318	43831	K4G	Q	TM	P	489	557	0,88	31
1) K: Top 2) 4: 1/4 Width 3) G: Thickness of product					•	5) TM : thermo 6) P : prisma 7) A5 : Lo=5,65			

4) Q: transversal

Specimen No. (C00)	Heat No. (B07)	Location (C01) 1) 2) 3)	Direct. (C02) 4)	Cond. (B05) 5)	Type of specimen (C40–C41)	Temperature (C03) ° C	Impact energy (C42–C43) ⊥ J ≥ 42	2	3	MW 6) J ≥60
913318	43831	к40	Q	TM	KV450	-040	118	111	114	114

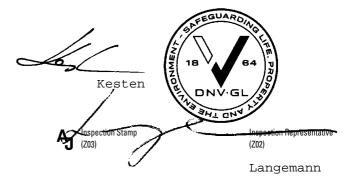
All plates have been ultrasonically tested according to EN 10160 07/99 in area and edges Class of areatesting: S1 Class of edgestesting: E2 Results: No faults were found.

We hereby certify that the delivered material complies with the terms of the order. $\ensuremath{\text{(Z01)}}$

QM–System: Certification as per ISO 9001 since 28 February 1990

(A04)

Ilsenburger Grobblech GmbH Veckenstedter Weg 10 D-38871 Ilsenburg (A01)



ILSENBURGER GROBBLECH

	ction certificate 3.2 N 10204				No . (A03) Page Date	1145686 5/5 13.11.2018
No. (A07) Purchaser (A06)	P12001207 Norsk Stal AS 4683 SOEGNE NORWEGEN	28.08.2018	No. (A07) Customer (A06)	Norsk Stal AS 4683 SOEGNE NORWEGEN	3	
Product (B01)	Heavy plate				Works order No. (A08)	0000096751
Steel grade (B02-B03)	and terms of delivery S420G2+M EN 10225 08/01 MDS-Y30 Rev.5 DIN EN 10029 A DIN EN 10163-2 312604 Application on consideration.	02/11 Kl. A UG3 board DNV -	,	sed objects is	Dispatch note No. Inspection (A05) subject	12.11.2018 DNVGL N141JYW8
do no which given	products are free of rad ot exceed the clearing l guarantees the complia in the Radiation Prote he unrestricted clearan	imit value ince with line ction Ordina	of 100 mit va ance(S) Bq/kg, alues StrlSchV)		

(StrlSchV Annex III, Section 5) for ferrous nuclides.

de l 18 Kesten DN ANT ON llsenburger Grobblech GmbH (Z03) inspection Representative Inspec Stamp Veckenstedter Weg 10 (Z02) D-38871 Ilsenburg Langemann

We hereby certify that the delivered material complies with the terms of the order. (Z01)

QM–System: Certification as per ISO 9001 since 28 February 1990

Trademark (A04)

(A01)

Certified Material	Test Report	Chiba Plant
According to EN 1	0204, type 3.1	Nippon Steel & Sumikin Welding Co., Ltd.
Test Report No. :	N06547	Purchaser's Spec. No. : NSTN-13 Rev.0
Date of issue :	June 29, 2017	Material Classification
Purchaser :	Norsk Sveiseteknikk AS	EN ISO 17632-A-T 46 6 1Ni M M 1 H5
Trade Designation :	NSSW SM-47A	AWS A5.36 E80T15-M21A8-Ni1-H4

1. 1.1.40011415				
Trade Designation	Size	Manuf. No.	Flux lot No.	Manuf. Date
NSSW SM-47A	1.2 mm	7U341AW996	7V311	June 12, 2017
Test type		Test	Unit	
EN ISO	14344 Lot c	elass : T1 (each flux lot), Tes	ting Schedule : 4	(exclude soundness test)

2. Welding Condition (Test type : ISO)

Base Met	al KL3 nm) 20tx250w	-		ding ition	Flat	5	Kind of Gas		80% Ar 20% CO ₂	CTWI) (mm)	20
	Current DC(+) (A)	Volt (V			Speed min)	Flo	w Rate of ((l/min)	Jas	Preheat 7 (°C)	ſemp.	Inter-p	oass Temp. (℃)
	270	30	0	3	80		25		140		135	~ 163
Spec.	270 ± 10	30 =	± 2	30	± 5		$20\sim 25$		$135 \sim$	163	135	$5 \sim 163$

3. Test results of All-Weld-Metal (Test type : ISO)

Tension Test	Conditions	Yield Strength (MPa)	Tensile Strength (MPa)	Elongation (%)
	AW	527	617	28
Spec.		460 Min.	$530 \sim 680$	20 Min. A5
Impact Test	Conditions	Temp. (°C)	Ind. Value (J)	Ave. (J)
I	AW	-60	93, 104, 115	104
Spec.		,	32 Min.	47 Min.
Spec.				

	Chemical composition (%)											
	С	Si	Mn	Р	S	Cu	Ni	Cr	Mo	V	Nb	
	0.07	0.59	1.26	0.009	0.008	0.25	1.02	0.02	0.01	0.01	0.01	
Spee	0.04/	0.40/	1.1/	0.020	0.020	0.3	0.80/	0.15	0.15	0.05	0.05	
Spec.	0.10	0.80	1.4/	Max.	Max.	Max.	1.10	Max.	Max.	Max.	Max.	

S, P and N of the strip : Below specified maximum

4. Hydrogen Content of Deposited Metal (Acc. to ISO 3690)

HDM (ml/100g)	Ave.	Spec.
1.0, 1.7, 1.0	1.2	5 Max.

Statement:

We hereby certify that the contents of this report are correct and accurate, and all operations performed by us or our subcontractors are in compliance with the requirements of the DATA SHEET No. : NSTN-13 Rev.0

No.	170639HFH	(MIT	9118)	Net Weight	
				711941AW006	17 00

Certified by

ISAO KANAUCHI Group Manager, Quality Control Dept.

Certified Mate	rial '	Test Report	Chiba Plant						
According to E	N 10	0204, type 3.1	Nippon Steel & Sumikin Welding Co., Ltd.						
Test Report No.	:	N06726	Purchaser's Spec. No. : NSTN-03 Rev.3						
Date of issue	:	September 6, 2017	Material Classification						
Purchaser	:	Norsk Sveiseteknikk AS	EN ISO 17632-A-T 46 4 Z P M 2 H5						
Trade Designati	on :	NSSW SF-3AM	EN ISO 17632-A-T 46 6 Z P M 2 H5 AWS A5.36 E81T9-M21A8-Ni1-H4						

Trade Designation	Size	Manuf. No.	Flux lot No.	Manuf. Date						
NSSW SF-3AM	1.2 mm	7S041MP960	7T051	August 02, 2017						
NSSW SF-3AM	1.2 mm	7S071MP960	7T051	August 03, 2017						
NSSW SF-3AM	1.2 mm	7S101MP960	7T051	August 04, 2017						
Test type		Test Unit								
EN ISO	14344 Lot c	14344 Lot class : T1 (each flux lot), Testing Schedule : 4 (exclude soundness test)								

2. Welding Condition (Test type : ISO)

Base Met	tal KL33 nm) 20tx250wx300L		Welding Position Flat		;	Kind of Gas		80% Ar 20% CO ₂	CTWI) (mm)	20	
	Current DC(+) (A)	Volt	age 7)		Speed min)	Flo	w Rate of ((l/min)	Jas	Preheat 7 (°C)	ſemp.	Inter-p	oass Temp. (°C)
	270	2	7	2	5		25		140		135	~ 163
Spec.	270 ± 10	27	± 2	25	± 5		$20\sim 25$		$135 \sim$	163	135	~ 163

3. Test results of All-Weld-Metal (Test type : ISO)

Tension Test	Conditions	Yield Strength (MPa)	Tensile Strength (MPa)	Elongation (%)
	AW	556	614	26
Spec.		460 Min.	$530 \sim 680$	20 Min. A5
Impact Test	Conditions	Temp. (°C)	Ind. Value (J)	Ave. (J)
	AW	-40	112, 126, 142	127
Spec.			32 Min.	47 Min.
	AW	-60	82, 80, 87	83
Spec.			32 Min.	47 Min.

	Chemical composition (%)											
	С	Si	Mn	Р	S	Cu	Ni	Cr	Mo	V	Nb	
	0.06	0.31	1.22	0.010	0.005	0.26	1.02	0.02	0.01	0.01	0.01	
Spee	0.03/	0.25/	1.00/	0.020	0.020	0.40	0.80/	0.15	0.35	0.05	Ref.	
Spec.	0.07	0.60	1.50	Max.	Max.	Max.	1.10	Max.	Max.	Max.		

S, P and N of the strip \vdots Below specified maximum

4. Hydrogen Content of Deposited Metal (Acc. to ISO 3690)

HDM (ml/100g)	Ave.	Spec.
1.4, 1.2, 1.3	1.3	5 Max.

Statement :

We hereby certify that the contents of this report are correct and accurate, and all operations performed by us or our subcontractors are in compliance with the requirements of the DATA SHEET No. : NSTN-03 Rev.3

No. 170830HFH	(MIT 9188)	Net Weight
		7S041MP9
		7S071MP0

Net Weight	_		
7S041MP960	:	2,575.0	kg
7S071MP960	:	4,275.0	kg
7S101MP960	:	1,375.0	kg

Certified Materia	l Test Report	Chiba Plant						
According to EN	10204, type 3.1	Nippon Steel & Sumikin Welding Co., Ltd.						
Test Report No. 3	N07653	Purchaser's Spec. No. : NSTN-03 Rev.4						
Date of issue :	April 23, 2018	Material Classification						
Purchaser :	Norsk Sveiseteknikk AS	EN ISO 17632-A-T 46 4 Z P M 2 H5						
Trade Designation	NSSW SF-3AM	EN ISO 17632-A-T 46 6 Z P M 2 H5 AWS A5.36 E81T9-M21A8-Ni1-H4						

Trade Designation	Size	Manuf. No.	Flux lot No.	Manuf. Date							
NSSW SF-3AM	1.2 mm	8X221MP996	8Y261	March 08, 2018							
Test type		Test Unit									
EN ISO	14344 Lot o	14344 Lot class : T1 (each flux lot), Testing Schedule : 4 (exclude soundness test)									

2. Welding Condition (Test type : ISO)

,	Base Metal KL33 (mm) 20tx250wx300			ding ition	Flat	5	Kind of Gas		80% Ar 20% CO ₂	CTWI) (mm)	20
	Current DC(+) (A)	Volt (V	0		Speed min)	Flo	w Rate of ((l/min)	Jas	Preheat 7 (°C)	lemp.	Inter-p	oass Temp. (°C)
	270	2'	7	2	5		25		140		135	~ 163
Spec.	270 ± 10	27 :	±2	25	± 5		$20\sim 25$		$135 \sim$	163	135	$5 \sim 163$

3. Test results of All-Weld-Metal (Test type : ISO)

Tension Test	Conditions	Yield Strength (MPa)	Tensile Strength (MPa)	Elongation (%)
	AW	535	606	27
Spec.		460 Min.	$530 \sim 680$	20 Min. A5
	0 1:1:		T 1 T7 1 (r)	
Impact Test	Conditions	Temp. (°C)	Ind. Value (J)	Ave. (J)
	AW	-40	137, 151, 162	150
Spec.			32 Min.	47 Min.
	AW	-60	115, 95, 74	95
Spec.			32 Min.	47 Min.

	Chemical composition (%)													
	C Si Mn P S Cu Ni Cr Mo V Nb B													
	0.05	0.27	1.15	0.010	0.004	0.30	1.03	0.03	0.01	0.01	0.02	0.003		
Shoo	0.03/	0.20/	1.00/	0.020	0.020	0.40	0.80/	0.15	0.35	0.05	Ref.	Ref.		
$\begin{array}{ c c c c c c c c c c c c c c c c c c c$														

S, P and N of the strip : Below specified maximum

4. Hydrogen Content of Deposited Metal (Acc. to ISO 3690)

HDM (ml/100g)	Ave.	Spec.
1.8, 1.2, 1.0	1.3	5 Max.

Statement :

We hereby certify that the contents of this report are correct and accurate, and all operations performed by us or our subcontractors are in compliance with the requirements of the DATA SHEET No. : NSTN-03 Rev.4

No.	180424HFH	(MIT	9588)	_	Net Wei

Net Weight			
8X221MP996	:	16,000.0	kg

Certified by <u>SHUSHIRO NAGASHIMA</u> Group Manager, Quality Control Dept.

This is electronically made and is not signed.

- Doc C302 All WPS's
 - WPS PL1-SW
 - WPS: PL3/4-DW50-Rep
 - WPS: PL5/6-DW15-Rep
 - WPS: PL7/8-DW5-Rep

J	S		ivers Stav		r		WELDIN					WPS Ref.: Date:	No.:	PL1-SW Magnus 2019.05	Larsson	Rev.2
Prod. b	y:	KIWA	TI.	11 (11 () () ()	- 9	C	lient:	Uni	versity of S	tavan	ger	Ref.	. star	nd: NS-E	N ISO 15	614-1:2017
Project	:	MSc -	Weld Pr	oximity I	NVES	6. Re	ef. spec.:	NO	RSOK M-1	01:20	11 Rev.5	Exa	m. bo	ody: Tekno	ologisk Ins	stitutt (KIWA)
Locatio	n:	Gener	al			R	ef. WPQR:	S42	20PL1-M							
Welding	g proc	ess		138	6				136							
Shieldir	ng gas	type		1 Arg	jon /1	8%CO2 (M21)	2	Argon /	18%C	O2 (M21)		3			
Weavin	g (yes	/no)		yes	5	max: 7	mm		yes	n	nax: 13	mm			max:	mm
Purging	gas t	ype		N/A				1/min							indix.	
Welding	a posit	ions		PF, vert	ical u	p			1		1		50 [°])		
Joint ty				BW							4	V		7		
Joint pr		tion		Waterje	cutti	ng, Grinding				-		t	<	\sim		
Cleanin	ig met	hod		Wire bri	lsh							1	<			
Backing	-			N/A					-	1		1	Fil	1	П	
Single/	Double	9		Single					1)	2	nose 05-1.0m		
Back g	ouging			N/A]	L			P	3-4mm		
Flux de	signat	ion		N/A]							
Flux ha	ndling			N/A								AREA				
Tungst	en ele	ctrode		N/A				mm					1			
Torch a	angle			70-90	0						A		/			
Stand of	off dist	ance		10-25				mm			120 22	/				
Nozzle	diame	ter(s)		10-20				mm	-							
Tack w	-			Genera			Rev: 0	_							_	
Identi	ficatio		_		1:											M max: 0,22
Part		Nam	ne/Grad	e		Standa	rd	Grou		eliver	ry cond.	Thic	kness (mm	s range		eter range [mm]
<u>t</u>	S4200	32+M,	MDS-Y3	0 Rev.5	EN	10225:2009		2.1	TM				15,0	00	5	00-<
11		-	MDS-Y3	_	EN	10225:2009		2.1	TM	_		_	15,0	00	5	00-<
->Iden	tifica	tion o	of filler	metal												
Index		E	Brand na	me			Classific	ation			Group			Filler	handling	
1	NSS	SW SM	- 47A			EN ISO 1763	2-A-T 46 6 1	Ni M I	M 1 H5		FM1	Su	ppli	ers recom	mendati	Lon
2	NSS	SW SF	- 3AM			EN ISO 1763 EN ISO 1763					FM1	Su	ppli	ers recom	mendati	ion
Weldi	ng Pa	rame	ters						E	quipn	nent:					
Pass		Dia.	Weldin	g Wir	e feed	d Curren	t Volta	ge			ling speed Ru	in Out Le	ength	Gas	F	leat input
no.		[mm]	metho		eed	[A]			Polarity	r	mm/min]	[mm]		[]/min]		[kJ/mm]
BW:		fund			*	- [0]					•	[]		[annu]	-	(a)
1(Root)	1	1,20	138		2.2	93 - 11:	3 13.2 -	16.2	DC+		42 - 88	N/A		20		1.0 - 2.0
2	2	1,20	136		7.1	192 - 23			DC+		14 - 226	N/A		20		1.3 - 2.5
Fill	2	1,20	136		7.1 7.1	192 - 23		_	DC+		15 - 264	N/A		20		1.1 - 2.5
CAP	2	1,20	136	_	/.1	184 - 22	4 15.0-	24.2	DC+		29 - 279	N/A		20		0.9 - 2.0
Heat t	reatm	lent			-							Method:		-		
		-	°C							_			_		6	9
Prehea PWHT		20	°C n			p. max: 150 °C Soaking		min/m	tment proc.:		min Heatin	i emp ig rate:	p. cor	ntrol: Digital co °C/h Co	ontact ther	
Remar											Addition	•	nclose	ed (¥es/No):	H H	
accord All we welds	ding elds shou ole.	to 15 on th ld be Follo	614-1 nis WPS perfo owing t	2017. S are ormed	to be	speed and e used for lose to th ld be perf	r master Ne origin	thes al w	is. Subs eld (WPG	seque (R) a	Date/Sig 2019.05	ndi C	K	LarssonUn S nandima Rat	tavang	,er
												_	1	m,		

Page 1

1 of

J	S		iver Stav	sity 'angei					PROC			Wf Re Da		PL3/4-D Magnus 2019.05.		Rev.2
Prod. b Project Locatio	:	KIWA MSc - Gener	Weld P	roximity IN	IVES.		t: spec.: WPQR:	NOF	versity of S RSOK M-1 0PL1-M				ef. stano xam. bo			614-1:201 7 stitutt (KIWA)
Weldin	g proce	ess		138					136							
Shieldi				1 Ara	n /18	3%CO2 (M21)		2	Argon /	18%C	O2 (M21)		3			
Weavir								-11					┥╹┝			
Purging	• •			yes		max: 7	mm		yes	m	ax: 13	mm			max:	mm
				N/A PF, vertic	alur		_	l/min					100			
Weldin	• ·	ions		BW	aiup							<u></u>			1	
Joint ty		ion			cuttir	ng, Grinding										
Joint pr		-		Wire brus		iy, Gimuny						1	(.4			
Cleanir		nod			in				-		Т	1	Fill		TT	2
Backing	-			N/A Single						and a second			1	_/	11	
Single/				Single							_		- D	nose 0.5 - 1	. 0 mm	
Back g				N/A										3-4mm		
Flux de	•	on		N/A								1998				
Flux ha				N/A	_											
Tungst		ctrode		N/A	0			mm				AFE		/	/	
Torch a	-	_	_	70-90	0		_	_						/		
Stand o				10-25				mm				-	-1-	The second second		
Nozzle		. ,		10-20		2		mm				50mm h	etween the	weld to as		
Tack w				General	_		Rev: 0									
Identi	ficatio	on of	parent	metal	I;	CE max: 0,42	C ma	ax: 0,	14 PCM	max:	0,22 ll	CE max	:0,42	C max: 0	,14 PCN	1 max: 0 , 22
Part		Nam	ne/Grad	e		Standard		Grou		eliver	y cond.	TI	nickness [mm]	-	Diam	eter range
1	S420G	62+M,	MDS-Y	30 Rev.5	EN 1	0225:2009		2.1	TM				15,0			500-<
H	S420G	62+M,	MDS-Y	30 Rev.5	EN 1	0225:2009		2.1	TM				15,0	0	ŧ	500-<
Identi	ficatio	on of	filler n	netal												
Index	1	E	Brand n	ame			Classific	cation			Gro	up		Fille	er handling]
1	NSS	W SM	- 47A		E	N ISO 17632-A-	-T 46 6 1		A 1 H5		FM1		Suppli	ers reco	mmendat	ion
2	NSS	W SF-	3AM			N ISO 17632-A					FM1		Suppli	ers reco	mmendat	ion
Weldi	ng Pa	rame	ters						F	quipm	ient:					
Pass no.	Index		Weldin	•		Current	Volta	age			ing speed	Run Out	Length	Gas	H	leat input
BW:		[mm]		spe	_	[A]	[V]	1		[r	nm/min]	{mr	n]	[l/min]		[kJ/mm]
1(Root)	1	1,20	138	_	2	93 - 113	13.2 -	16.2	DC+		12 - 88	N/.	A	20		1.0 - 2.0
2	2	1,20	136	7	1	192 - 235	21.2 -	26.0	DC+	1	14 - 226	N/	4	20		1.3 - 2.5
Fill	2	1,20	136	7		192 - 235	21.2 -		DC+		15 - 264	N/.		20		1.1 - 2.5
CAP	2	1,20	136	7	1	184 - 224	19.8 -	24.2	DC+	1:	29 - 279	N/	A	20		0.9 - 2.0
Heat t	reatm	ent	1									Metho	d:	÷.		
Prehea	t min:	70	°C	Interpass	temp	.max: 150	°c He	attreat	ment proc .:			Те	mp. con	trol: Digita	al contact th	ermometer
PWHT				nax:		C Soaking:		mi n /m			min He	ating rate			oling rate	1
Remar	ks:	is h				WPQ as the	origi			lwel		•		used (¥es/No		
this W Repair - 6.1	PS are r and l Prel	e to b preh heat	e used leat a and i	lformas s in No	ter rsok s te	thesis. M-101:2011 Emperature	-				25 011	Date/Sign 2019 05 0 Approved	9 / Magn	us Larsson	Univere Blavor	ity of
														Chandima R	atnavake	A Y

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J	S		ivers Stav	sity anger		SPE	CIFIC	PROC	I (WP	S)	Re	PS No.: ef.: ute:	PL5/6-DV Magnus 2019 05.	Larsson 09	Rev.2
Prod. b Project Locatio		KIWA ⁻ MSc Th Genera	nesis - V	Veld Proxim	,	nt: spec.: WPQR:	NOF	ersity of S RSOK M-10 DPL1-M				ef. stand xam. boo			i14-1:2017 titutt (KIWA)
Welding	g proce	ess		138				136							
Shieldir	ng gas	type		1 Argon	/18%CO2 (M21)		2	Argon /1	18%CO2	2 (M21)		3			
Weavin	a (ves	/no)		yes	max: 7	mm	-11	yes	ma	x: 13	mm			max:	mm
-	gas ty	_		N/A			1/min	,	ma	A. 19				max.	
Neldin		-		PF, vertical	up						1	0			
Joint ty			-	BW							1		7		
Joint pr		ion		Waterjet cu	Itting, Grinding							Cap	\sim		-
Cleanir	-	_	,	Wire brush							F	~			
Backing	3		1	N/A				~		. I		Till		IT	
Single/I	Double			Single						1		2	nose: 0.5 - 1.0		1
Back g	ouging			N/A					-			- LY	3-4mm		
Flux de	signati	on		N/A											
=lux ha	ndling			N/A							ASS.		1	1	
Fungst	en elec	trode	1	N/A			mm								
Forch a	Ingle			70-90	0							//	/		
Stand o	off dista	ance		10-25			mm			100	-				
lozzle	diame	ter(s)		10-20			mm								
Tack w	elding	proc.		General		Rev: 0					15mm b	etween the v	veld toes		
denti	ficatio	n of p	parent	metal	I: CE max: 0,42	2 C ma	ax: 0,:	14 PCM	max: 0	,22	: CE max	(0,42	C max: 0,	14 PCN	1 max: 0 , 22
Part		Nam	e/Grade	e	Standard		Group	D	elivery	cond.	т	hickness [mm]	U	Diam	eter range
1	S420G	32+M, I	MDS-Y3	0 Rev.5 E	N 10225:2009		2.1	TM				15,0	0	ŧ	500-<
1	S420G	2+M, I	MDS-Y3	0 Rev.5 E	N 10225:2009		2.1	TM				15,0	0	ŧ	500-<
denti	ficatio	on of f	filler m	etal											
ndex		В	rand na	me		Classific	cation			Gro	oup		Fille	r handling	I
1	NSS	W SM	- 47A		EN ISO 17632-A	-T 46 6 1		1 1 H5	F	Ml		Suppli	ers reco	mmendat	ion
2	NSS	W SF-	3AM		EN ISO 17632-A EN ISO 17632-A				EJ	M1		Suppli	ers reco	mmendat	ion
Weldi	ng Pa	rame	ters					E	quipme	nt:					
Pass no.	Index	Dia.	Weldin method	feed		Volta		Current / Polarity		g speed	Run Out	C C	Gas	ł	Heat input
BW:		fuuul		spee		[V]	1			*	(m]	[within i]		[KJ/mm]
(Root)	1	1,20	138	2.2	93 - 113	13.2 -	16.2	DC+	42	- 88	N/	A	20		1.0 - 2.0
2	2	1,20	136	7.1	192 - 235	21.2 -		DC+		- 226	N/		20		1.3 - 2.5
Fill	2	1,20	136	7.1	192 - 235	21.2 -		DC+		- 264	N/		20		1.1 - 2.5
CAP	2	1,20	136	7.1	184 - 224	19.8 -	24.2	DC+	129	- 279	N	A	20		0.9 - 2.0
leat t	reatm	ent				1					Metho	od:	-		
	t min [.]	70	°C	Interpass te	emp. max: 150	°c He	attreatr	nent proc.:			Τo	mp. cont	rol: Diaita	I contact th	ermometer
	s trutte	,0	°C m		°C Soaking:	5	min/m			in He	eating rate		0	oling rate:	°Ci
Prehea	min			ICLA.	- Suaking.					пе		info enclo		1	
Prehea PWHT Remar Repai:	ks: r WPS		ased o	on the sa	ame WPQ as the er thesis.	e origi	inal v	eld.All	l welds	on	Date/Sign	ature:	s Larsson	U)

NORWAY

J	S		iver: Stav	sity anger					PROC			WPS Ref.: Date:		PL7 /8-D W Magnus La 2019.05.09	rsson
Prod. b Project: Locatio		KIWA ⁻ MSc T Genera	hesis -	Weld Prox				NOF	versity of S RSOK M-1 0PL1-M				stand m. boo		ISO 15614-1:2017 gisk Institutt (KIWA)
Welding				138					136						
					- 140	000 (1404)				00/ 0	O2 (M21)				
Shieldir				-	n /18	3%CO2 (M21)		_ 2	Argoni		52 (10121)		3		
Weavin	•			yes		max: 7	mm		yes	m	nax: 13	mm			max: mm
Purging	gas ty	pe		N/A				l/min					60 ⁰		
Welding	g positi	ons		PF, vertic	al up)							(4)		
Joint ty	pe			BW											
Joint pr	eparati	on		Waterjet o	utti	ng, Grinding							Cap		
Cleanin	g meth	od		Wire brus	h						т	1	1 ill	7	**
Backing	9			N/A							- 1	/	111		
Single/I	Double			Single									2	nose: 0.5 - 1.0 mm	
Back g	ouging			N/A							y.		M	3-4mm	
Flux de	signati	on		N/A											
Flux ha	ndling			N/A										1	1
Tungste	en elec	trode		N/A				mm				AMERICA		/ /	
Torch a	ngle			70-90	0									/	
Stand of	off dista	ince		10-25				mm			42		/		
Nozzle	diamet	er(s)		10-20				mm							
Tack w	elding	oroc.		General			Rev: 0					5mm betwe	en the w	reld toes	
Identif	icatio	n of p	parent	metal	1:	CE max: 0,42	C ma	ax: 0,	14 PCM	max;	0,22 II	: CE max: 0	,42	C max: 0,14	PCM max: 0,22
Part		Nam	e/Grad	e		Standard		Grou	D	eliver	y cond.	Thic	kness [mm]	range	Diameter range [mm]
,f	S420G	2+M, I	MDS-YS	30 Rev.5	EN 1	0225:2009		2.1	TM				15,0	0	500-<
II	S420G	2+M, I	MDS-Y3	80 Rev.5	EN 1	0225:2009		2.1	TM				15,0	0	500-<
Identi	ficatio	n of f	filler m	netal											
Index		В	rand na	ame			Classific	cation			Gro	up		Filler h	andling
1	NSS	W SM	- 47A		E	N ISO 17632-A-	-T 46 6 1		A 1 H5		FM1	Su	ppli	ers recomm	endation
2	NSS	W SF-	3AM			EN ISO 17632-A- EN ISO 17632-A-					FM1	Su	ppli	ers recomm	endation
Weldi						_				quipm					1
Pass no.	Index	Dia.	Weldir metho	9	d	Current [A]	Volta		Current / Polarity		ling speed	Run Out Le	ength	Gas [l/min]	Heat input [kJ/mm]
BW:				-						·					
1(Root)	1	1,20	138	2.		93 - 113	13.2 -		DC+		42 - 88	N/A		20	1.0 - 2.0
2	2	1,20	136	7.		192 - 235	21.2 -		DC+		14 - 226	N/A		20	1.3 - 2.5
Fill	2	1,20	136	7.		192 - 235	21.2 -		DC+		15 - 264	N/A	-	20	1.1 - 2.5
CAP	2	1,20	136	7.		184 - 224	19.8 -	24.2	DC+	1	29 - 279	N/A		20	0.9 - 2.0
Heat t	reatm	ent								L		Method:		-	
			**	late as a c	10.0-	may: 150	1a 11-	ottra - 1	montaria						
Prehea		70					°c He		ment proc.:				. cont		ontact thermometer
-	KS: WPS		ased	on the s	same	C Soaking:	origi	min/m		welc		eating rate: Additional info Date/Signatu		°C/h Cooli sed (¥es/No):	ng rate: °C/r
Repair - 6.11	r and L Preb	preh neat	eat a and i	s in No:	rso) s te	M-101:2011 Amperature	Rev.5	sec	tion:			2019.05.09 /	Magn	us Larason (r. 1 Chandima Ratin	avanget v.



Manufacturing and welding process, characterization of the Steel S420G2+M

- Production Welding of Plates

- Doc D401 All Weld log's for Weld A
- Doc D402 All Weld log's for repair Weld B

• Doc D401 - All Weld log's for Weld A

All Weld log's for Weld A:

PL3-DW 50 (Weld A)

WPQR	nr.:					S420PL1	-M	Welding	method		138 / 13	36				
Date &	Place				2	5 April 201	9 KIWA	Gas/Purg	ge gas	Arg	on/18% CC	02 (M21)			- " _ "	
Plate II	D:					Plate 3 - D	W50	Welding	position		PF			1	mi	
Proced	ure sta	andard				PL1-S	N	Groove t	ype		BW		Treasure	541041.4M	4/3/	THE E Hatenat Stont + H
Welder						BI			fdistance		10-25 m	2.2.3		+	2 Jarm	
Materi	al / he	eat nr.:				43831-913	3182	Torch an	gle		70-90				Tim I	
Filler m	nat./t	oatch / Lot			NSSW	SM-47A, N	SSW SF-3AM	Nozzle di	iam		10-20 m	m				
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138	23.8	1.2	2.2	DC+	103	14.7	107	159	40.4	20	0	9	2.25
	2	Björn-Ivar	138		1.2	2.2	DC+	103	14.7	175	228	46.1	20	0	9	1.97
	3	Björn-Ivar	138		1.2	2.2	DC+	103	14.7	215	162	79.6	20	0	7	1.14
2	1	Björn-Ivar	136	45.0	1.2	8.2	DC+	211	23.8	235	77	183.1	20	0	12	1.65
	2	Björn-Ivar	136		1.2	8.2	DC+	211	23.8	250	95	157.9	20	0	12	1.91
3	1	Björn-Ivar	136	85.0	1.2	8.2	DC+	211	23.8	380	94	242.6	20	0	12	1.24
	2	Björn-Ivar	136		1.2	8.2	DC+	211	23.8	110	32	206.3	20	0	12	1.46
4	1	Björn-Ivar	136	100.0	1.2	8.2	DC+	211	23.8	485	146	199.3	20	0	13	1.51
5	1	Björn-Ivar	136	60.0	1.2	7.2	DC+	201	22.0	485	148	196.6	20	0	14	1.35
6	1	Björn-Ivar	136	117.0	1.2	7.2	DC+	201	22.0	485	151	192.7	20	0	17	1.38

PL4-DW 50 (Weld A)

WPQR	nr.:					S420PL1	-M	Welding	method		138 / 13	5			60'-	
Date &	Place				2	9 April 2019) KIWA	Gas/Purg	e gas	Argo	n/18% CO	2 (M21)		1	·	
Plate ID):					Plate 4 - D	W50	Welding	position		PF				asj'	1
Procedu	ure sta	andard				PL1-SV	V	Groove t	ype		BW		I Haterial	541042+M ISam	4 3	Emm I Haberial Stable 2+H
Welder	.: ID					BI		Stand off	distance		10-25 mm	n		4	The farm	+
Materia	al / he	at nr.:				43831-913	3182	Torch an	gle		70-90				1200	
Filler m	at./b	oatch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle di	am		10-20 mr	n				
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138	51.9	1.2	2.2	DC+	105	14.7	145	131	66.4	20	0	8	1.39
	2	Björn-Ivar	138		1.2	2.2	DC+	105	14.7	145	122	71.3	20	0	8	1.30
	3	Björn-Ivar	138		1.2	2.2	DC+	105	14.7	160	125	76.8	20	0	8	1.21
	4	Björn-Ivar	138		1.2	2.2	DC+	105	14.7	50	46	65.2	20	0	8	1.42
2	1	Björn-Ivar	136		1.2	8.3	DC+	209	23.0	265	74	214.9	20	0	12	1.34
	2	Björn-Ivar	136		1.2	8.3	DC+	209	23.0	230	71	194.4	20	0	12	1.48
3	1	Björn-Ivar	136	83.8	1.2	8.3	DC+	209	23.0	160	56	171.4	20	0	12	1.68
	2	Björn-Ivar	136		1.2	8.3	DC+	209	23.0	175	42	250.0	20	0	12	1.15
	3	Björn-Ivar	136		1.2	8.3	DC+	209	23.0	75	22	204.5	20	0	13	1.41
4	1	Björn-Ivar	136	77.6	1.2	8.3	DC+	209	23.0	390	121	193.4	20	0	16	1.49
	2	Björn-Ivar	136		1.2	8.3	DC+	209	23.0	105	33	190.9	20	0	16	1.51
5	1	Björn-Ivar	136	70.0	1.2	8.3	DC+	202	22.0	280	71	236.6	20	0	14	1.13
	2	Björn-Ivar	136		1.2	8.3	DC+	202	22.0	220	62	212.9	20	0	14	1.25
6	1	Björn-Ivar	136	40.3	1.2	8.3	DC+	202	22.0	305	94	194.7	20	0	17	1.37
	2	Björn-Ivar	136		1.2	8.3	DC+	202	22.0	195	63	185.7	20	0	17	1.44

PL5-DW 15 (Weld A)

WPQR	nr.:					S420PL1	M	Welding	method		138 / 136	5			60'-	
Date &	Place				2	9 April 2019	KIWA	Gas/Purg	e gas	Argo	n/18% CO2	2 (M21)			- ·	
Plate ID):					Plate 5 - D	W15	Welding	position		PF			*	esi	*
Proced	ure sta	andard				PL1-SW	/	Groove t	уре		BW		I Material	541042+M Ham	43/ "	Material Stille 62+4
Welder	:.: ID					BI		Stand off	distance		10-25 mn	n		+	The saran	-
Materia	al / he	at nr.:				43831-913	3182	Torch and	gle		70-90				3,200	
Filler m	at./b	oatch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle di	am		10-20 mm	n				
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138		1.2	2.1	DC+	98	14.7	95	79	72.2	20	0	7	1.20
	2	Björn-Ivar	138		1.2	2.1	DC+	98	14.7	130	108	72.2	20	0	7	1.20
	3	Björn-Ivar	138		1.2	2.1	DC+	98	14.7	105	78	80.8	20	0	7	1.07
	4	Björn-Ivar	138		1.2	2.2	DC+	107	14.7	85	59	86.4	20	0	7	1.09
	5	Björn-Ivar	138		1.2	2.2	DC+	107	14.7	105	65	96.9	20	0	7	0.97
2	1	Björn-Ivar	136		1.2	8.3	DC+	224	23.0	495	90	330.0	20	0	12	0.94
3	1	Björn-Ivar	136	101.8	1.2	8.3	DC+	230	23.0	310	122	152.5	20	0	16	2.08
	2	Björn-Ivar	136		1.2	8.3	DC+	229	23.0	185	53	209.4	20	0	16	1.51
4	1	Björn-Ivar	136	125.2	1.2	8.3	DC+	217	23.0	495	137	216.8	20	0		1.38
5	1	Björn-Ivar	136	62.0	1.2	8.3	DC+	204	22.0	450	124	217.7	20	0	15	1.24
	2	Björn-Ivar	136		1.2	8.3	DC+	204	22.0	45	18	154.3	20	0	15	1.75
6	1	Björn-Ivar	136	114.0	1.2	8.3	DC+	204	22.0	495	161	184.5	20	0	18	1.46

PL6-DW 15 (Weld A)

WPQR	nr.:					S420PL1	-M	Welding	method		138 / 13	6		2.3	- 60'	
Date &	Place				3	0 April 2019) KIWA	Gas/Purg	e gas	Argo	n/18% CO	2 (M21)]			
Plate II	D:					Plate 6 - D	W15	Welding	position		PF]	-	ters'	*
Proced	ure sta	andard				PL1-SV	V	Groove ty	/pe		BW		I Haterial		4/3/	ITTER I Haterial Stick2+K
Welde	r.: ID					BI		Stand off	distance		10-25 mr	n		+	1 datas	+
Materi	al / he	eat nr.:				43831-913	3182	Torch ang	gle		70-90]		1200	
Filler m	nat./t	batch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle di	am		10-20 mr	n				
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138	18.6	1.2	2.2	DC+	100	14.7	170	153	66.7	20	0	8	1.32
	2	Björn-Ivar	138		1.2	2.2	DC+	99	14.7	140	121	69.4	20	0	8	1.25
	3	Björn-Ivar	138		1.2	2.2	DC+	99	14.7	110	95	69.5	20	0	8	1.25
	4	Björn-Ivar	138		1.2	2.2	DC+	97	14.7	90	89	60.7	20	0	8	1.40
2	1	Björn-Ivar	136	46.0	1.2	8.3	DC+	238	23.0	495	172	172.7	20	0	12	1.90
3	1	Björn-Ivar	136	107.8	1.2	8.3	DC+	227	23.0	410	120	205.0	20	0	-	1.53
	2	Björn-Ivar	136		1.2	8.3	DC+	227	23.0	85	26	196.2	20	0	-	1.60
4	1	Björn-Ivar	136	136.0	1.2	8.3	DC+	214	23.0	495	156	190.4	20	0	15	1.55
5	1	Björn-Ivar	136	70.0	1.2	7.0	DC+	196	22.0	495	173	171.7	20	o	16	1.51
6	1	Björn-Ivar	136	89.0	1.2	7.0	DC+	202	22.0	495	163	182.2	20	0	17	1.46

PL7-DW 5 (Weld A)

WPQR	nr.:					S420PL1	-M	Welding	method		138 / 13	5				
Date &	Place)			3	0 April 2019) KIWA	Gas/Purg	e gas	Argo	n/18% CO	2 (M21)		1	- 60''	
Plate II	D:					Plate 7 - D	W5	Welding	position		PF			1	1	
Proced	ure st	andard				PL1-SV	V	Groove ty	/pe		BW		Trend	5410G1+HL Ha	4/3/	1
Welder	r.: ID					BI		Stand off	distance		10-25 mr	n	T. Patence	STADGLINL Har	12/1	IT Haterial Stariat + Ha
Materia	al / he	eat nr.:				43831-913	3182	Torch ang	gle		70-90			¥		
Filler m	nat. / I	batch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle di	am		10-20 mr	n			1200	
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138	59.0	1.2	2.2	DC+	97.0	14.7	250	242	62.0	20	0	8	1.38
	2	Björn-Ivar	138		1.2	2.2	DC+	92.0	14.7	180	168	64.3	20	0		1.26
	3	Björn-Ivar	138		1.2	2.2	DC+	98.8	14.7	65	66	59.1	20	0	-	1.47
2	1	Björn-Ivar	136	64.0	1.2	8.3	DC+	227.1	23.0	495	170	174.7	20	0		1.79
3	1	Björn-Ivar	136	97.0	1.2	8.3	DC+	227.2	23.0	495	145	204.8	20	0	15	1.53
4	1	Björn-Ivar	136	124.0	1.2	8.3	DC+	224.0	23.0	495	160	185.6	20	0	16	1.67
5	1	Björn-Ivar	136	67.0	1.2	7.0	DC+	202.0	22.0	495	173	171.7	20	0	17	1.55
6	1	Björn-Ivar	136	99.0	1.2	7.0	DC+	202.9	22.0	495	157	189.2	20	0	18	1.42

PL8-DW 5 (Weld A)

WPQR	nr.:					S420PL1	-M	Welding	method		138/13	6				
Date &	Place				3	0 April 2019	9 KIWA	Gas/Purg	e gas	Argo	n/18% CO	2 (M21)]	1	- 40°	
Plate II	D:	~ ~		Ì		Plate 8 - D	W5	Welding	osition		PF]	2	7	
Proced	ure sta	andard				PL1-SV	V	Groove ty	/pe		BW			1	4 31	1
Welder	r.: ID					BI		Stand off	distance		10-25 m	n	I Mutanul	543045+H Hm	- 12/.	IT Haberiah Stiolaz * M.
Materi	al / he	eat nr.:				43831-913	3182	Torch ang	gle		70-90]		Til farm	+
Filler m	nat./t	batch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle di	am		10-20 mr	n			1200	
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138		1.2	2.2	DC+	100	14.7	175	146	71.9	20	0	7	1.23
	2	Björn-Ivar	138		1.2	2.2	DC+	96	14.7	130	100	78.0	20	0	7	1.09
	3	Björn-Ivar	138		1.2	2.2	DC+	93	14.7	110	72	91.7	20	0	7	0.90
	4	Björn-Ivar	138		1.2	2.2	DC+	87	14.7	95	68	83.8	20	0	7	0.91
2	1	Björn-Ivar	136		1.2	8.3	DC+	223	23.0	495	174	170.7	20	0	13	1.80
3	1	Björn-Ivar	136	131.4	1.2	8.3	DC+	220	23.0	495	138	215.2	20	0	15	1.41
4	1	Björn-Ivar	136	134.2	1.2	8.3	DC+	225	23.0	290	90	193.3	20	0	15	1.61
	2	Björn-Ivar	136		1.2	8.3	DC+	225	23.0	205	64	192.2	20	0	15	1.62
5	1	Björn-Ivar	136	53.8	1.2	7.0	DC+	202	22.0	495	166	178.9	20	0	17	1.49
6	1	Björn-Ivar	136	116.5	1.2	7.0	DC+	189	22.0	495	167	177.8	20	0	18	1.40

• Doc D402 - All Weld log's for repair Weld B

All Weld log's for repair Weld B:

PL3-DW 50-Rep (Weld B)

WPQR						S420PL1	14	Welding	mothed		138 / 13	e				
Date &					<u> </u>	7 May 2019		Gas/Purg		Argo	n/18% CO2	-	1	1.00	- 40'	
Plate II						te 3 - DW50		Welding		74180	PF	- (11121)	1	7	7	
Proced		andard				PL3/4-DW50		Groove t		<u> </u>	BW			1	Tu st	1
Welder		anuaru			-	AF	о-кер	Stand off		<u> </u>	10-25 mr		I Robert	SHEGSING HE	12	man I Hutanut-Stilling + M.
Materi		at as .				43831-913	2102	Torch and			70-90	n			Til farm	*
								<u> </u>	,	<u> </u>			1		Sim	
Filler m	at. / b	oatch / Lot					SW SF-3AM	Nozzle di			10-20 mr					
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Arild	138	82.0	1.2	2.2	DC+	95	14.7	90	77	70.1	20	0	-	1.19
	2	Arild	138		1.2	2.2	DC+	95	14.7	70	58	72.4	20	0	-	1.16
	3	Arild	138		1.2	2.2	DC+	97	14.7	75	61	73.8	20	0	-	1.16
	4	Arild	138		1.2	2.2	DC+	98	14.7	85	66	77.3	20	0		1.12
	5	Arild	138		1.2	2.2	DC+	96	14.7	125	73	102.7	20	0	-	0.82
	6	Arild	138		1.2	2.2	DC+	96	14.7	160	99	97.0	20	0	6	0.87
2	1	Arild	136	74.0	1.2	8.3	DC+	229	23.0	185	43	258.1	20	0	11	1.22
	2	Arild	136		1.2	8.3	DC+	217	23.0	155	51	182.4	20	0	11	1.64
	3	Arild	136		1.2	8.3	DC+	213	23.0	170	62	164.5	20	0	11	1.79
3	1	Arild	136	75.0	1.2	8.3	DC+	218	23.0	205	40	307.5	20	0	13	0.98
	2	Arild	136		1.2	8.3	DC+	231	23.0	220	53	249.1	20	0	13	1.28
	3	Arild	136		1.2	8.3	DC+	231	23.0	75	21	214.3	20	0	13	1.49
4	1	Arild	136	101.0	1.2	8.3	DC+	220	23.0	270	68	238.2	20	0	-	1.27
	2	Arild	136		1.2	8.3	DC+	226	23.0	230	65	212.3	20	0	-	1.47
5	1	Arild	136	117.0	1.2	7.0	DC+	199	22.0	205	72	170.8	20	0	15	1.54
	2	Arild	136		1.2	7.0	DC+	204	22.0	200	69	173.9	20	0	15	1.55
	3	Arild	136		1.2	7.0	DC+	208	22.0	95	28	203.6	20	0	16	1.35
6	1	Arild	136	124.0	1.2	7.0	DC+	200	22.0	105	47	134.0	20	0	12	1.97
	2	Arild	136		1.2	7.0	DC+	203	22.0	195	59	198.3	20	0	12	1.35
	3	Arild	136		1.2	7.0	DC+	206	22.0	200	48	250.0	20	0	12	1.09

PL4-DW 50-Rep (Weld B)

WPQR	nr.:					S420PL1	-M	Welding r	nethod		138 / 13	6				
Date &	Place					9 May 2019	KIWA	Gas/Purg	e gas	Argo	n/18% CO2	2 (M21)		1	- 60' /	
Plate ID):				Pla	te 4 - DW50	- weld 2	Welding p	osition		PF			7	7	
Proced	ure sta	andard				PL3/4-DW5	0-Rep	Groove ty	/pe		BW		-		4 3	
Welder	.: ID					AF		Stand off	distance		10-25 mm	n	I Hatmid	STIDILTH Han	12/ "	man I Hutanut-Stilling+H.
Materi	al / he	at nr.:				43831-913	3182	Torch ang	le		70-90			*	The salar	
Filler m	at./b	oatch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle dia	am		10-20 mr	n			12mm	
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Arild	138	85.0	1.2	2.2	DC+	96	14.7	100	91	65.9	20	0	-	1.28
	2	Arild	138		1.2	2.2	DC+	98	14.7	110	89	74.2	20	0	-	1.17
	3	Arild	138		1.2	2.2	DC+	97	14.7	100	90	66.7	20	0		1.28
	4	Arild	138		1.2	2.2	DC+	98	14.7	105	75	84.0	20	0		1.03
	5	Arild	138		1.2	2.2	DC+	97	14.7	140	103	81.6	20	0	-	1.05
2	1	Arild	136	72.0	1.2	8.3	DC+	227	23.0	340	90	226.7	20	0	11	1.38
	2	Arild	136		1.2	8.3	DC+	227	23.0	160	50	192.0	20	0	11	1.63
3	1	Arild	136	74.0	1.2	8.3	DC+	225	23.0	250	66	227.3	20	0	-	1.37
	2	Arild	136		1.2	8.3	DC+	225	23.0	230	64	215.6	20	0	-	1.44
4	1	Arild	136	85.0	1.2	8.3	DC+	235	23.0	210	68	185.3	20	0	-	1.75
	2	Arild	136		1.2	8.3	DC+	235	23.0	270	77	210.4	20	0		1.54
5	1	Arild	136	76.0	1.2	7.0	DC+	202	22.0	285	98	174.5	20	0	18	1.53
	2	Arild	136		1.2	7.0	DC+	204	22.0	195	66	177.3	20	0	18	1.52
6	1	Arild	136	146.0	1.2	7.0	DC+	204	22.0	435	114	228.9	20	0	11	1.18

PL5-DW 15-Rep (Weld B)

WPQR	nr.:					S420PL1	-M	Welding	nethod		138 / 13	6				
Date &	Place					9 May 2019	KIWA	Gas/Purg	e gas	Argo	n/18% CO	2 (M21)	1	1.00	- 60'/	
Plate II):				Pla	te 5 - DW15	- weld 2	Welding	position		PF		1	2	7	
Proced	ure sta	andard				PL5/6-DW5	0-Rep	Groove ty	/pe		BW			1	451	
Welder	r.: ID					AF		Stand off	distance		10-25 mr	n	I Raterial	STREELING STOOM	12/1 "	E Hutanut Pfletil*H.
Materi	al / he	at nr.:				43831-913	3182	Torch ang	le		70-90			*		
Filler m	nat. / b	oatch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle di	am		10-20 mr	n			Elmen.	
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Arild	138	76.0	1.2	2.2	DC+	95	14.7	150	166	54.2	20	0	8	1.55
	2	Arild	138		1.2	2.2	DC+	99	14.7	125	125	60.0	20	0	8	1.46
	3	Arild	138		1.2	2.2	DC+	97	14.7	250	205	73.2	20	0	8	1.17
2	1	Arild	136	76.0	1.2	8.3	DC+	226	23.0	275	73	226.0	20	0	11	1.38
	2	Arild	136		1.2	8.3	DC+	226	23.0	200	63	190.5	20	0	11	1.64
3	1	Arild	136	80.0	1.2	8.3	DC+	231	23.0	215	55	234.5	20	0	12	1.36
	2	Arild	136		1.2	8.3	DC+	231	23.0	105	25	252.0	20	0	12	1.27
	3	Arild	136		1.2	8.3	DC+	222	23.0	135	37	218.9	20	0	12	1.40
4	1	Arild	136	74.0	1.2	8.3	DC+	229	23.0	130	44	177.3	20	0	13	1.78
	2	Arild	136		1.2	8.3	DC+	220	23.0	360	111	194.6	20	0	13	1.56
5	1	Arild	136	70.0	1.2	7.0	DC+	207	22.0	460	138	200.0	20	0	15	1.37
6	1	Arild	136	88.0	1.2	7.0	DC+	208	22.0	235	54	261.1	20	0	12	1.05
	2	Arild	136	110.0	1.2	7.0	DC+	201	22.0	255	58	263.8	20	0	-	1.01

PL6-DW 15-Rep (Weld B)

WPQR	nr.:					S420PL1	-M	Welding r	nethod		138 / 13	5				
Date &	Place				1	0 May 2019	KIWA	Gas/Purg	e gas	Argo	n/18% CO2	2 (M21)			- 10'	
Plate IC	D:				Pla	te 6 - DW15	- weld 2	Welding p	position		PF			1	1	
Proced		andard				PL5/6-DW5		Groove ty			BW		Transf	341643.+ML	4/3/	I Hatenat Stienz + M.
Welder						BI		Stand off			10-25 mr	n	T Comment	STEEGZING Bas	12 Jacon	International Protocol Pro-
Materi	al / he	at nr.:				43831-913	3182	Torch ang	;le		70-90				Tim.	
Filler m	nat. / t	oatch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle di	am		10-20 mr	n			1100	
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138	62.0	1.2	2.2	DC+	101	14.7	120	112	64.3	20	0		1.39
	2	Björn-Ivar	138		1.2	2.2	DC+	98	14.7	115	136	50.7	20	0		1.70
	3	Björn-Ivar	138		1.2	2.2	DC+	99	14.7	155	150	62.0	20	0	-	1.40
	4	Björn-Ivar	138		1.2	2.2	DC+	97	14.7	115	77	89.6	20	0	-	0.95
2	1	Björn-Ivar	136	100.0	1.2	8.3	DC+	239	23.0	495	154	192.9	20	0	-	1.71
3	1	Björn-Ivar	136	110.0	1.2	8.3	DC+	229	23.0	495	123	241.5	20	0	-	1.31
4	1	Björn-Ivar	136	148.0	1.2	8.3	DC+	233	23.0	495	140	212.1	20	0	13	1.52
5	1	Björn-Ivar	136	80.0	1.2	7.0	DC+	204	22.0	495	152	195.4	20	0	16	1.38
6	1	Björn-Ivar	136	101.0	1.2	7.0	DC+	198	22.0	495	161	184.5	20	0	14	1.42

PL7-DW 5-Rep (Weld B)

WPQR	nr.:					S420PL1	-M	Welding r	nethod		138 / 13	5				
Date &	Place				1	0 May 2019	KIWA	Gas/Purg	e gas	Argo	n/18% CO2	2 (M21)]	*	- 60'	
Plate ID):				Pla	te 7 - DW5	- weld 2	Welding p	position		PF				and	
Proced	ure st	andard				PL7/8-DW5	0-Rep	Groove ty	/pe		BW		I Ruberid	SHEDGENAL INC.	4/3/	I Hatenal Storath
Welder						BI		Stand off			10-25 mr	n		1	12 parm	+
Materia	al / he	at nr.:				43831-913	3182	Torch ang	ļle		70-90				Tann .	
Filler m	at. / t	oatch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle dia	am		10-20 mr	n				
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
<u> </u>					(000)	mymm		(A)	(*)	(1111)			(171111)	(L/min)	(1111)	near input to/initi
1	1	Björn-Ivar	138	64.0	1.2	2.2	DC+	100	14.7	150	130	69.2	20	0	-	1.27
	2	Björn-Ivar	138		1.2	2.2	DC+	99	14.7	115	120	57.5	20	0	-	1.52
	3	Björn-Ivar	138		1.2	2.2	DC+	100	14.7	235	168	83.9	20	0	-	1.05
2	1	Björn-Ivar	136	81.0	1.2	8.3	DC+	227	23.0	495	155	191.6	20	0	-	1.64
3	1	Björn-Ivar	136	70.0	1.2	8.3	DC+	224	23.0	495	134	221.6	20	0	14	1.39
4	1	Björn-Ivar	136	100.0	1.2	8.3	DC+	233	23.0	495	149	199.3	20	0	-	1.62
5	1	Björn-Ivar	136	87.0	1.2	7.0	DC+	206	22.0	495	148	200.7	20	0		1.36
6	1	Björn-Ivar	136	92.0	1.2	7.0	DC+	204	22.0	495	130	228.5	20	0	13	1.18

PL8-DW 5-Rep (Weld B)

WPQR	nr.:					S420PL1	·M	Welding r	nethod		138 / 130	5				
Date &	Place				1	0 May 2019	KIWA	Gas/Purg	e gas	Argo	n/18% CO2	2 (M21)		*	- 60'	
Plate ID):				Pla	ate 8 - DW5	- weld 2	Welding	oosition		PF			1	ton!	
Proced		andard				PL7/8-DW5	0-Rep	Groove ty			BW		Tant	5410411H Hart	4/3/	I Huteria PRess
Welder						BI		Stand off			10-25 mm	n	T LINGUA	L I	12 Jaron	Anna I Hateria Stielastic
Materia	al / he	at nr.:				43831-913	3182	Torch ang	le		70-90				111	
Filler m	nat. / k	batch / Lot			NSSW	SM-47A, NS	SW SF-3AM	Nozzle dia	am		10-20 mm	n			12mm	
String	Seq.	Welder	Method	Temp °C	Wire Ø (mm)	Wirefeed m/min	AC/DC+/DC-	Current (A)	Voltage (V)	Length (mm)	Time (s)	Speed mm/min	Gass (L/min)	Purge gas (L/min)	weld width (mm)	Heat input kJ/mm
1	1	Björn-Ivar	138	50.0	1.2	2.2	DC+	95	14.7	135	117	69.2	20	0	8	1.22
	2	Björn-Ivar	138		1.2	2.2	DC+	97	14.7	135	144	56.3	20	0	8	1.53
	3	Björn-Ivar	138		1.2	2.2	DC+	98	14.7	115	113	61.1	20	0	8	1.42
	4	Björn-Ivar	138		1.2	2.2	DC+	98	14.7	120	85	84.7	20	0	8	1.02
2	1	Björn-Ivar	136	80.0	1.2	8.3	DC+	224	23.0	495	150	198.0	20	0	12	1.56
3	1	Björn-Ivar	136	130.0	1.2	8.3	DC+	223	23.0	495	128	232.0	20	0	13	1.33
4	1	Björn-Ivar	136	74.0	1.2	8.3	DC+	223	23.0	495	144	206.3	20	0	14	1.49
5	1	Björn-Ivar	136	124.0	1.2	7.0	DC+	208	22.0	495	156	190.4	20	0	16	1.44
6	1	Björn-Ivar	136	70.0	1.2	7.0	DC+	208	22.0	495	141	210.6	20	0	-	1.30



Manufacturing and welding process, characterization of the Steel S420G2+M

- NDT, Mechanical Testing and Specimen Preparation

- Doc E501 NDT report Radiograpic Examination
- Doc E502 Qlabs Material Report 8612-2 (PL 3)
- Doc E503 Qlabs Material Report 8612-3 (PL 5)
- Doc E504 Qlabs Material Report 8612-4 (PL7)

• Doc E501 - NDT report - Radiograpic Examination

- Doc E5XX Report8076-19-DRT-1
- Doc E5XX Report8076-19-DRT-2-REV1
- Doc E5XX Report8076-19-DRT-3-REV1
- Doc E5XX Report8076-19-DRT-5
- Doc E5XX Report8076-19-DRT-6
- Doc E5XX Report8076-19-DRT-7



CLIENT / KUNDE UIS Stavanger			T O.NO / on wel	KUNDE O.NF	۲			E OF TEST 9-05-15	ING / KC	NTRO	OLLDATO			RAPPORT 9-DRT-1	NR.	PAGE/S 1 of/av	
DRAWING NO. / T N/A		I						K / KONTR 1, Bærhe			ERATOR / rin Erdei		ØR	AT O	ГАСНМІ	ENT / VE	DLEGG
OBJECT / KONTRO PL 3 - DW 50. Welding Positic Welder ID: AF / Material: S420	on: PF					1 1132		l, Dærne		201	<u>rin Eruei</u>			<u> </u>			
PROCEDURE / PROSEDYRE BPI-01.Radiog r	ranhic Exam	vination ISO 1	17636-	 >	R 0	EV		ENT OF TE FROLLOM		А	CCEPTAN KSEPTST	ANDARD					
MATERIAL TYPE S420	•			CE / OVERFL	ATE			VE / FUGI	EGEOME			W		PROCESS	/ SVEIS	EPROSES	35
EQUIPMENT / UT X-RAY TUBE / RØ Smart 225Kv								IATION SO HNIQUE / 1			ON / KILD	EPOSISJC	DN				
FOCAL SPOT SIZE	3 / BRENNFLI	3 KK								1 ★	2	3	4		5	6	7
		UBE VOLT kV	A 220	В	С			I.Q.I. SOUI SIDE KILDE		Λ		*		*		$\overline{\mathbb{A}}$	
EXP. DATA	DI EX m/	LMFOCUS ISTANCE mm XP.TIME Amin.	700 4,5			+		FILMSI TYPE OF I I.Q.I. TYPI	IDE 1.Q.I. / E				d			C	(\mathcal{V})
DETECTOR / DETEKTOR HD-IP Plus	SCREENS .	min. AND FILTERS / JG OG FILTRE m	<u> </u>	DETECTO SPATIAL 0,063			[SIZE / STØRRE	LSE	MIN. SN 168	R	TOTA UNSH 0.1	L IARPNESS		GNIFICA RSTØRRE	
WELD NO SVEIS NR	DATE AND DATO KL.		DIA. PIPE DIA RØR	FILM LOCATI FILM PLASSER		SENSITIVI FØLSOMH % WIRE NI	ΉET	WELDER SVEISER	MATERIA THICKNES TYKKELS mm	SS BEI	ALUATION DØMMELSE	DEFECT L FEILENS PI		REM BEMER	IARKS KNINGER	EXP. DATA A-D	TECH. TEGN. 1-7
Weld A	14-05-2			0 - 220		11		AF/BIK	15		epted					A	1
Weld A Weld B	14-05-2		Plate Plate	220 - 440 0 - 220		11		AF/BIK AF/BIK	15 15	_	epted			 		 	1
Weld B	14-05-2		late	220 - 440		11		AF/BIK	15		epted						1
										-						—	<u> </u>
	_										-					1	<u> </u>
																\downarrow	
100 = Sprekk 200 = 402 = Rotfeil 500 = COMMENTS / KO	Uregelmessig	form 501 = Sårka	lutting, S ant 600 =	lagg 400 = Bin Andre uregeln	defeil og nessighete	mangler er (spesif	nde gje fiser)	nnomsveisi	ng 401 =	Bindef	èil						
Result: Acc	epted																
REPAIRS MARKE			MERKET	ΓPÅ													
N3 NAME CERT. N		ETCH / SKISSE			UE CEDT	NO /N	12 NI A X	/N SERT. 1	ND		OPER	ATOR NA	MECER	.T. NO. / C		D NIA VA	LCEDT
	NO. / N3 NAVI	IN SERT. INK.		Sorin E			NZ INAN	IN SERT. I	NK.		NR.	n Erdei ((.1. NO. / C	FERAI	JK INA VI	V SEKI.
APPROVED / GOE)KJENT DA	ATO:		APPROV Approv	N			ATO:2019-	-05-15						D:2019-0	5-15	



CLIENT / KUNDE UIS Stavanger	•		T O.NO	KUNDE O.N	R			TE OF TEST	ING / KON	NTROI	LLDATO			RAPPORT NR. -DRT-2-REV	PAGE	
DRAWING NO. / T		0.						RK / KONTR on, Bærhe			RATOR / in Erdei	OPERATØI	R	ATTACH O	IMENT / V	EDLEGG
OBJECT / KONTRO PL 4 - DW 50. Welding Positic Welder ID: AF / Material: S420	on: PF				[<u> </u>		<u>.</u>								
PROCEDURE / PROSEDYRE BPI-01.Radiog r	anhic Exa	mination ISO	17636-	.7		REV O	ко	TENT OF TE NTROLLOM 0 %		Al	SEPTST	ICE STAND ANDARD 5-1:2016	DARD /			
MATERIAL TYPE S420	-			CE / OVERFI	LATE			OOVE / FUGE	EGEOMET			WE	LDING I 3/136	PROCESS / SVE	EISEPROSI	ESS
EQUIPMENT / UT X-RAY TUBE / RØ Smart 225Kv		R						DIATION SO CHNIQUE / 7			N / KILD	EPOSISJON	J			
FOCAL SPOT SIZE	E / BRENNF	LEKK						- 1		1 ✿	2	3	4	5	6	7
	7	TUBE VOLT kV	A 220	B	C		D	I.Q.I. SOUI	-	Λ		*	4	* *	Ā	æ
EXP. DATA		FILMFOCUS DISTANCE mm	700					- KILDES FILMSI							\bigcirc	(\downarrow)
	1	EXP.TIME mAmin. Cimin.	4,5					TYPE OF I I.Q.I. TYPE W 6 FE <				÷			Ŭ	v
DETECTOR / DETEKTOR HD-IP Plus		S AND FILTERS ING OG FILTRE mm	/	DETECT SPATIA 0,063		SIC / DLUTION		PIXEL PIXEL 25 m i	STØRREL	.SE	MIN. SN 168	R	TOTAL UNSH		MAGNIFIC FORSTØRR	
WELD NO SVEIS NR	DATE AN DATO KI		DIA. PIPE DIA RØR	FILM LOCAT FILM PLASSE		SENSITIV FØLSOMI % WIRE N	HET	WELDER SVEISER	MATERIAL THICKNESS TYKKELSE mm	S BED	LUATION ØMMELSE	DEFECT LO FEILENS PLA		REMARKS BEMERKNINGI	ER DATA A-D	A TEGN. 1-7
Weld A	_		Plate	0 - 220	_	W 11			15		epted				A	1
Weld A Weld B			Plate Plate	220 - 440 0 - 220		W11 W11			15 15	-	epted	270 		517	A A	1
Weld B			Plate	220 - 440	_	W11			15		epted				A	1
				<u> </u>							-					
																_
									<u></u>							-
100 = Sprekk 200 = 402 = Rotfeil 500 =	Hulrom, por Uregelmessi	rer 300 = Fast inneof form 501 = Sårk	slutting, S	Slagg 400 = Bi	ndefeil	og mangle	ende g	gjennomsveisi	ng 401 = E	Bindefe	-il					
COMMENTS / KO				i indie dregen	ine oorge	(open		,								
Result: Acc	epted															
	-															
REPAIRS MARKE		ARASJONER AV KETCH / SKISSE		ΓPÅ												
N3 NAME CERT. N ()	NO. / N3 NA	VN SERT. NR.		N2 NAN Sorin I			N2 N/	AVN SERT. N	NR.		NR.	ATOR NAM		T. NO. / OPERA	TØR NAV	'N SERT.
APPROVED / GOD	OKJENT I	DATO:			ved /0	GODKJEN Godkjen		DATO:2019-	05-15			ATOR / OP		R DATO:201	9-05-15	



CLIENT / KUNDE UIS Stavanger			T O.NO / on wel	KUNDE O.N.	R			E OF TEST	ING / KON	NTROL	LDATO				RAPPORT		PAGE/ 1 of/a	
DRAWING NO. / T		0.						K / KONTR n, Bærhei			RATOR / in Erdei		RATØR		ATT 0	TACHMI	ENT / VI	EDLEGG
OBJECT / KONTRO PL 5 - DW 15. Welding Positio Welder ID: AF / Material: S420	n: PF																	
PROCEDURE / PROSEDYRE BPI-01.Radiogr	anhic Exa	mination ISO	17636-	2		REV O		ENT OF TE		Ak	CCEPTAN SEPTST	AND	ARD	ARD /				
MATERIAL TYPE S420				CE / OVERFL	ATE			OVE / FUGE	EGEOMET			<u> </u>	WEL	DING I /136	PROCESS	/ SVEIS	EPROSE	SS
EQUIPMENT / UTS X-RAY TUBE / RØ Smart 225Kv		R						DIATION SO HNIQUE / T			N / KILD	EPOS	SISJON					
FOCAL SPOT SIZE 3	E / BRENNF	LEKK	1					11		1 ★	2		3	4		5	6 ▲	7
		TUBE VOLT kV	A 220	B	C		D	I.Q.I. SOUI SIDE KILDES		Λ			*	3	*		Λ	
EXP. DATA		FILMFOCUS DISTANCE mm EXP.TIME mAmin. Cimin.	700 4,5					FILMSI TYPE OF I I.Q.I. TYPE W 6 FE<	DE .Q.I. /		•)) (\bigcirc	đ		\mathbf{D}	Ó	(\mathbf{y})
DETECTOR / DETEKTOR HD-IP Plus	SCREEN	S AND FILTERS ING OG FILTRE	/	DETECT SPATIAI 0,063				PIXEL	STØRREL	.SE	MIN. SN 168	IR		TOTA UNSH 0.1	L ARPNESS		GNIFIC. RSTØRR	ATION / ELSE
WELD NO SVEIS NR	DATE AN DATO KI		DIA. PIPE DIA RØR	FILM LOCAT FILM PLASSEI	RING	SENSITIV FØLSOMF % WIRE N	HET R	WELDER SVEISER	MATERIAL THICKNESS TYKKELSE mm	BED	LUATION ƏMMELSE	FEILI	FECT LOC ENS PLAS		BEMERI	ARKS KNINGER	EXP. DATA A-D	TECH. TEGN. 1-7
Weld A			Plate	0 - 220		V 11	_		15		epted	60			2011		A	1
Weld A Weld B			Plate Plate	220 - 440 0 - 220		V11 V11			15 15	-	epted				 		A	1
Weld B			Plate	220 - 440		V11	_		15		pted						A	1
100 = Sprekk 200 = 402 = Rotfeil 500 = COMMENTS / KOI Result: Acco	Uregelmess MMENTAR	g form 501 = Sårk	slutting, S ant 600 =	Slagg 400 = Bir Andre uregelr	ndefeil o nessighe	g mangler ter (spesi	nde gj fiser)	ennomsveisi	ng 401 = E	Bindefe	il	<u> </u>						
REPAIRS MARKE	D ON / REP	ARASJONER AV	MERKET	ГРÅ														
	EKT S	KETCH / SKISSE			IE CERT		J2 N A	VN SERT. N	JR		OPED	ATO	RNAM	IE CEP	T. NO. / O	PERATO	AR NAV	NSFRT
0				Sorin E	Erdei (0	0816)					NR. Sorii	n Erd	lei (08	16)				IN SEK1.
APPROVED / GOD	OKJENT I	DATO:			ved /G	odkjen odkjen		DATO:2019-	05-15			M	r / OPI	ERATØ	R DATO	D:2019-0	5-15	



CLIENT / KUNDE UIS Stavanger		CLIENT X-ray o		KUNDE O.NI ds	R			OF TESTI -05-15	NG / KON	ITROL	LDATO		RT NO. / 8076-1			PAGE 1 of/a	
DRAWING NO. / T N/A	EGNING NO.							/KONTR , Bærhei		-	RATOR / n Erdei	OPERAT	ØR	A 0		MENT / V	EDLEGG
OBJECT / KONTRO PL 6 - DW 15. Welding Positio Welder ID: AF / Material: S420								<u>,</u>									
PROCEDURE / PROSEDYRE BPI-01.Radiog r	aphic Examina	ation ISO 1	7636-2	2		REV D		NT OF TE ROLLOM		AK	SEPTST	NCE STAN ANDARD					
MATERIAL TYPE S420	/ MATERIALTY		SURFA As We	CE / OVERFL Ided	ATE		GROOV BW	VE / FUGE	GEOMET	RI			ELDING 88/136		SS / SVEI	ISEPROSI	ESS
EQUIPMENT / UTS X-RAY TUBE / RØ Smart 225Kv								ATION SC INIQUE / T			N / KILD	DEPOSISJO	DN				
FOCAL SPOT SIZE 3	E / BRENNFLEKK	K.								1 ₩	2	3		4	5	6	7
	TUBE	E VOLT kV	A 220	В	С	D	— s	.Q.I. SOUF		Ā		~		¥	*	Ā	\oplus
EXP. DATA	DIST	ANCE mm	700					✔ KILDES FILMSI	DE 7			\mathbb{Q}				\bigcirc	(\mathcal{Y})
	EXP.1 mAmi Cimin	in.	4,5				I.	TYPE OF I. .Q.I. TYPE N 6 FE <	1 1 1 1 1								
DETECTOR / DETEKTOR HD-IP Plus	SCREENS AN SKJERMING (Cu 0,3 mm			DETECTO SPATIAL 0,063				PIXEL PIXEL 25 mi	STØRREL		MIN. SN 168	IR	TOTA UNSI 0.1	AL HARPNE		AGNIFIC DRSTØRF	
WELD NO SVEIS NR	DATE AND TIMI DATO KL.		DIA. PIPE DIA RØR	FILM LOCATI FILM PLASSEF		SENSITIVI FØLSOMH % WIRE NF	ET	WELDER SVEISER	MATERIAL THICKNESS TYKKELSE mm	EVA BED@	LUATION MMELSE		OCATION LASSERING		EMARKS ERKNINGEF	R EXP DATA A-D	A TEGN. 1-7
Weld A	14-05-201	-		0 - 220		/ 11		<u> </u>	15		pted					A	1
	14-05-201			220 - 440		/11		•	15		pted					A	1
Weld B Weld B	14-05-201		late late	0 - 220 220 - 440		/11 /11			15 15		pted	100 360		515 515		A	1
100 = Sprekk 200 = 402 = Rotfeil 500 = COMMENTS / KOI	Uregelmessig forr							nomsveisi	ng 401 = B	indefei	il						
Result: Acco	epted																
REPAIRS MARKE	D ON / REPARAS		IERKET	Γ PÅ													
N3 NAME CERT. N ()	NO. / N3 NAVN S	ERT. NR.		N2 NAM Sorin E			2 NAV	N SERT. N	JR.		NR.	RATOR NA n Erdei (RT. NO. /	OPERA	TØR NAV	'N SERT.
APPROVED / GOD	DKJENT DATC):		APPROV Approv	ved /G	odkjent		ATO:2019-	05-15			RATOR / C	/	ØR DA	ATO:2019	-05-15	



CLIENT / KUNDE UIS Stavanger			T O.NO / on wel	KUNDE O.NF	ł			E OF TESTI 9-05-15	ING / KON	NTROL	LLDATO			RAPPORT N 9-DRT-6		AGE/S	
DRAWING NO. / T N/A	EGNING NO.	<u> </u>						K / KONTRO n, Bærhei			RATOR / in Erdei	OPERATØ	R	ATTA O	CHMEN	IT / VE	DLEGG
OBJECT / KONTRO PL 7 - DW 5. Welding Positio Welder ID: AF / Material: S420	n: PF								<u></u>	13011							
PROCEDURE / PROSEDYRE BPI-01.Radiogr	aphic Exam	ination ISO :	17636-	2	R	EV		ENT OF TES TROLLOM		Ak	SEPTST	ICE STANE ANDARD 5-1:2016					
MATERIAL TYPE S420	•		1	CE / OVERFL	ATE			OVE / FUGE	GEOMET			WE		PROCESS / S	VEISEI	PROSES	3S
EQUIPMENT / UTS X-RAY TUBE / RØ Smart 225Kv								DIATION SC HNIQUE / T			N / KILD	PEPOSISJON	N				
FOCAL SPOT SIZE 3	E / BRENNFLE	КК								1	2	3	4	1 5		6	7
	Т	BE VOLT kV	A 220	В	С	Ι)	I.Q.I. SOUF SIDE		Ā		••		• •	1	Ĩ.	F
EXP. DATA	FII	LMFOCUS STANCE mm	700			+		✓ KILDES FILMSI		A					ble	5	$\langle T \rangle$
	EX mA	TP.TIME Amin. nin.	4,5					TYPE OF I. I.Q.I. TYPE W 6 FE <	.Q.I. /						21/		
DETECTOR / DETEKTOR HD-IP Plus		AND FILTERS / G OG FILTRE m	1	DETECTO SPATIAL 0,063				PIXEL PIXEL 25 mi	STØRREL	.SE	MIN. SN 168	IR	TOTA UNSH 0.1	L IARPNESS		NIFICA TØRRI	ATION / ELSE
WELD NO SVEIS NR	DATE AND 1 DATO KL.		DIA. PIPE DIA RØR	FILM LOCATI FILM PLASSEF	ING	SENSITIVI FØLSOMH % WIRE NI	IET R	WELDER SVEISER	MATERIAL THICKNESS TYKKELSE mm	BEDO	LUATION ØMMELSE	DEFECT LO FEILENS PLA		REMARI BEMERKNI	KS NGER	EXP. DATA A-D	TECH. TEGN. 1-7
Weld A Weld A	14-05-2		Plate Plate	0 - 220 220 - 440		11	_		15 15		epted epted	 395		 515		A A	1
Weld B	14-05-2			0 - 220		11			15		epted	130		517		A	1
Weld B	14-05-2	:019 F	Plate	220 - 440	w	11		AF/BIK	15	Acce	epted	250		517		A	1
													-				
100 = Sprekk 200 = 402 = Rotfeil 500 = COMMENTS / KON	Uregelmessig f	form 501 = Sårka						ennomsveisii	ng 401 = B	Bindefe	il						
Result: Acce	epted																
REPAIRS MARKE			MERKET	ΓPÅ													
N3 NAME CERT. N ()	NO. / N3 NAVN	I SERT. NR.		N2 NAM Sorin E			12 NA'	VN SERT. N	JR.		NR.	ATOR NAI		T. NO. / OPE	RATØR	R NAVN	J SERT.
APPROVED / GOD	KJENT DA	TO:		APPROV Approv	ved /Go			DATO:2019-	05-15			•	PERATØ	DR DATO:2	019-05-	15	



CLIENT / KUNDE UIS Stavanger			T O.NO / on wel	KUNDE O.NF	2			E OF TEST 9-05-15	ING / KO	NTROI	LLDATO			RAPPORT NR. 9-DRT-7	PAGE	
DRAWING NO. / T. N/A	EGNING NC).						K / KONTR 1, Bærhe			RATOR / in Erdei	OPERAT	ØR	ATTACI O	HMENT / VI	EDLEGG
OBJECT / KONTRO PL 8 - DW 5. Welding Positio Welder ID: AF / Material: S420	n: PF							<u>, balle</u>			<u>Ili ciue</u> i					
PROCEDURE / PROSEDYRE BPI-01.Radiogr	anhic Exar	mination ISO :	17636-	2	R	EV		ENT OF TE FROLLOM		Al	KSEPTST	NCE STAN ANDARD 5-1:201				
MATERIAL TYPE				CE / OVERFL	ATE			VE / FUGI	EGEOME			W		PROCESS / SV	EISEPROSE	ESS
EQUIPMENT / UTS X-RAY TUBE / RØ Smart 225Kv		1						IATION SO HNIQUE / ')N / KILD	EPOSISJO	DN			
FOCAL SPOT SIZE 3	E / BRENNFL	EKK								_1 ✿	2	3	4	5	6	7
		UBE VOLT kV	A 220	B	С			I.Q.I. SOU SIDE ✔KILDE		Λ		*	3	* *		
EXP. DATA	E n	ILMFOCUS DISTANCE mm XP.TIME DAmin. Vimin.	700 4,5			+	[FILMS TYPE OF I I.Q.I. TYPI	IDE I.Q.I. / E						\bigcirc	(\mathcal{Y})
DETECTOR / DETEKTOR HD-IP Plus	SCREENS	AND FILTERS / NG OG FILTRE	<u></u>	DETECTO SPATIAL 0,063					SIZE / STØRREI	LSE	MIN. SN 168	IR	TOTA UNSH 0.1	ARPNESS	MAGNIFIC FORSTØRR 1	
WELD NO SVEIS NR	DATE ANI DATO KL.		DIA. PIPE DIA RØR	FILM LOCATI FILM PLASSER		SENSITIVI FØLSOMH % WIRE N	IET	WELDER SVEISER	MATERIAI THICKNES TYKKELSI mm	S BED	UATION ØMMELSE	DEFECT I FEILENS PI		REMARKS BEMERKNING	GER DATA A-D	A TEGN. 1-7
Weld A	14-05			0 - 220		11		AF/BIK	15		epted	100		517/2011	A	1
Weld A Weld B	14-05- 14-05-		Plate Plate	220 - 440 0 - 220		11		AF/BIK AF/BIK	15 15	-	epted epted	340		517	A A	1
Weld B	14-05		Plate	220 - 440		11		AF/BIK	15	-	epted			 	A	1
				<u> </u>							-					
				<u> </u>								ļ				_
				<u> </u>								1				
				1												_
100 = Sprekk 200 = 402 = Rotfeil 500 =	Hulrom, pore	er 300 = Fast innes	lutting, S	$3 \log 400 = Bin$	Idefeil og	manglei	nde gjer	nnomsveisi	ng 401 = 1	Bindefe	eil	<u> </u>				
COMMENTS / KON			<u>ant 000 –</u>	Andre uregenn	lessignete	/ (spesh	11501)									
Result: Acce	ented															
REPAIRS MARKEI		ARASJONER AVI ETCH / SKISSE	MERKET	ſ PÅ												
N3 NAME CERT. N ()				N2 NAM Sorin E			J2 NAV	/N SERT. 1	NR.		NR.	RATOR NA		T. NO. / OPER	ATØR NAV	N SERT.
APPROVED / GOD	KJENT D	ATO:		APPROV Approv	r			ATO:2019	-05-15			ATOR / C		OR DATO:201	19-05-15	

• Doc E502 - Qlabs Material Report 8612-2 (PL 3)

		MATE	RIAL REPO	RT	
Certificate no: 8612-2	Rev A	Order no: 8612	0	Quality Lab AS	
^{Client:} Master Thesis	UIS	Test date/Report date 22.05.2019 - 27.06.2019	quality lab	Industrivegen 54 4110 Forsand Norge	
		Client contact Mattias Larsson	Equipment identification: Mechanical testing of W		TEST 248
Pages 4		Surveyor Mattias & Magnus Larsson	Standard ISO 15614-1:2017		
Additional information:		Quality Lab Confirm	s that the results are c	connected to the object tes	sted.
		Macro examinatio	n according to NS-E	N 17639:2013	
Etchant		Nital	Visual examinat	ion:	X5
			10 11 12 /18 45 8 9 10 11 12 314	314 <u>- 117 - 119 11</u> - 117 - 119 11	
			Weld A. A left side		
				12134 17 18 15	
		-j <mark>9-j2</mark> -j <u>3</u>		- <u> </u> 17 - <u> </u> 18 <u> </u> 19	
	Com	ment: The Macro are inside Imperfectio	Weld B. B left side the acceptance criter ons are inside quality le		14

MATE	ERIAL	REP	ORT

Certificate no:	Rev	Order no:
8612-2	А	8612
Client:		Test date/Report date
Master Thesis	UIS	22.05.2019 - 27.06.20



Quality Lab AS Industrivegen 54 4110 Forsand



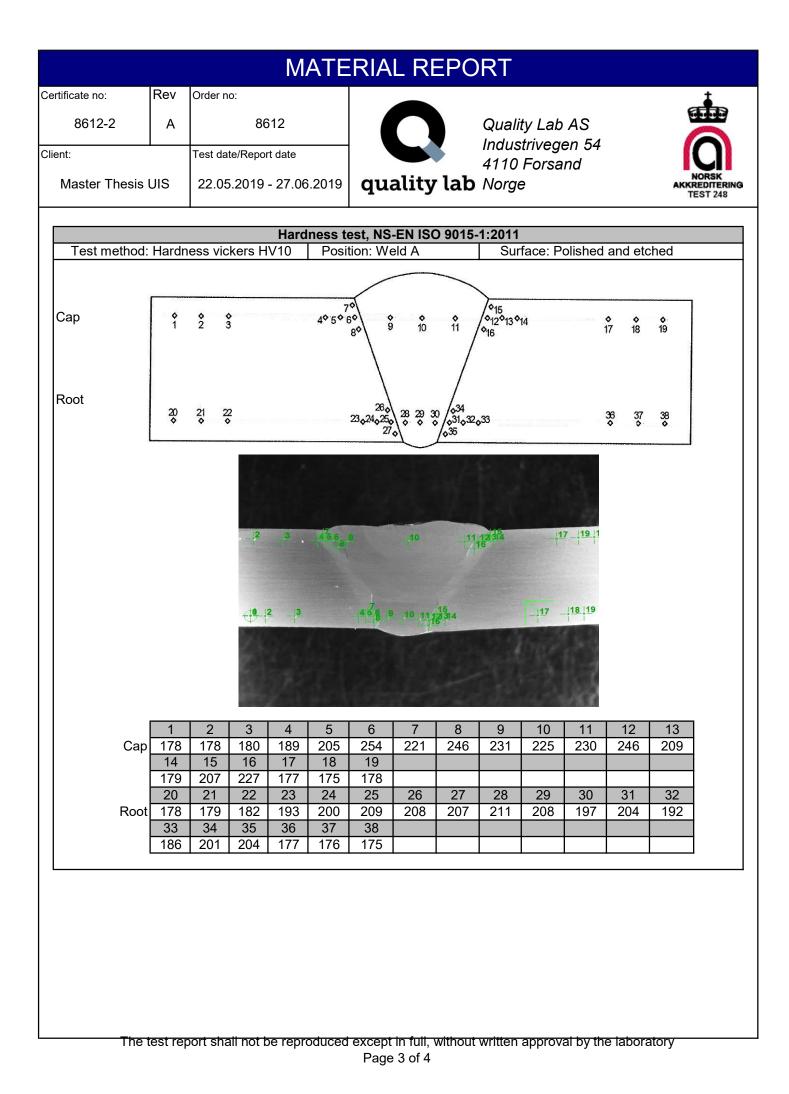
2019 quality lab Norge

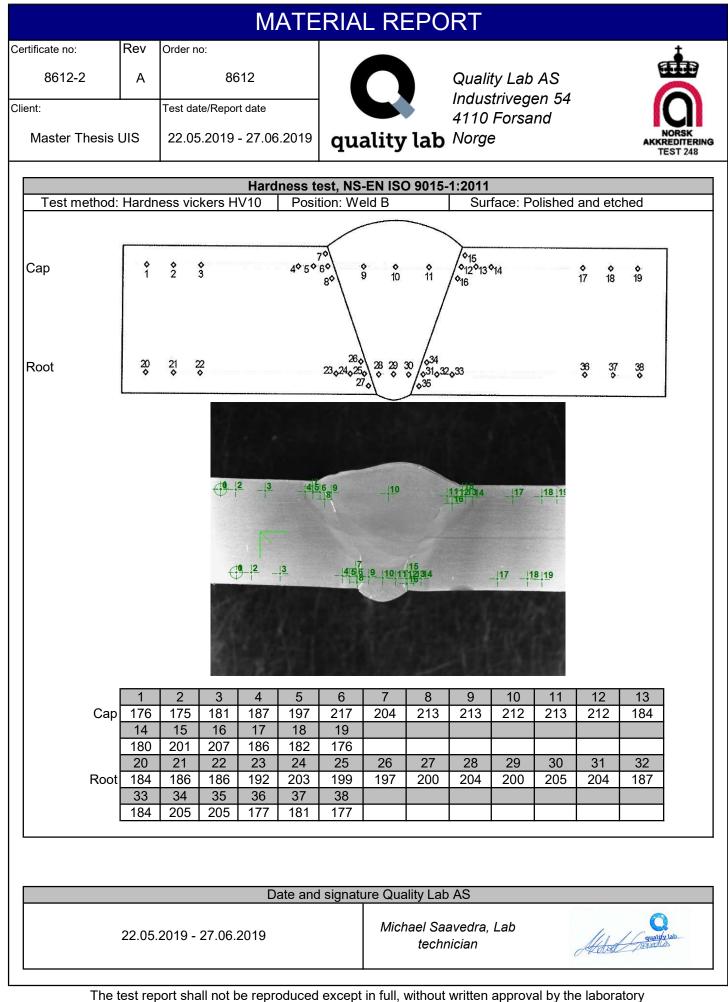
Tensile test,	NS EN ISO 4136-1:2	2012 / ISO 6892-1:201	16 Method A.1
Test ident:	Area [mm²]	R _m [Mpa]	Fracture
Cross weld sample 1	104,66	528	Base material
Cross weld sample 2	102,96	525	Base material

Charpy V ir	npact test, K\	/8, NS-EN ISO	148-1:2016; 1	SO 901	6:2012		
Test ident	Dimension	Notch	Test temp	Sing	le value	es [J]	Average
Test ident	[mm]	Orientation	[⁰ C]	1	2	3	[J]
Weld	10x10x55	Т	-40	121	120	112	118
FL	10x10x55	Т	-40	186	124	163	158
FL+2	10x10x55	Т	-40	395	341	404	380
FL+5	10x10x55	Т	-40	196	215	212	208

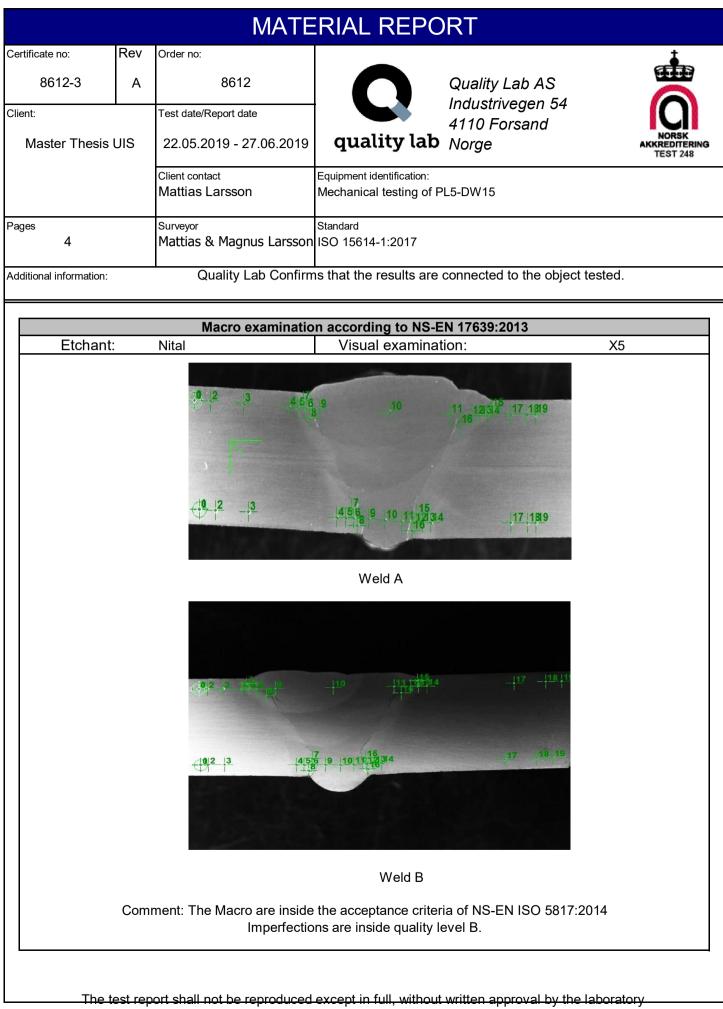
MPI examination according to ISO 23278:2015 level B (non accreditated test)

MPI performed, No findings Accepted





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			MATE	RIA	L REPO	RT		
Certificate no:	Rev	Order no:						t
8612-3	А	86	512				ty Lab AS trivegen 54	
Client:		Test date/Report	date				Forsand	
Master Thesis	SUIS	22.05.2019	- 27.06.2019	qu	ality lab	-		NORSK AKKREDITERING TEST 248
		Tensile test,	NS EN ISO 41	136-1:2	2012 / ISO 689	2-1:201	6 Method A.1	
Test ident: Area [m					R _m [Mpa	a]	Fracture	
Cross	102,58		537		Base mater	ial		

Charpy V impact test, KV8, NS-EN ISO 148-1:2016; ISO 9016:2012

Notch

Orientation

Т

Т

Т

Т

MPI examination according to ISO 23278:2015 Level B (non accreditated test)

MPI performed, No findings Accepted

530

Test temp

l₀Cl

-40

-40

-40

-40

Base material

3

91

143

372

327

Average

[J]

92

173

368

336

Single values [J]

2

84

206

366

319

1

100

169

365

361

104,11

Dimension

[mm]

10x10x55

10x10x55

10x10x55

10x10x55

Cross weld sample 2

Test ident

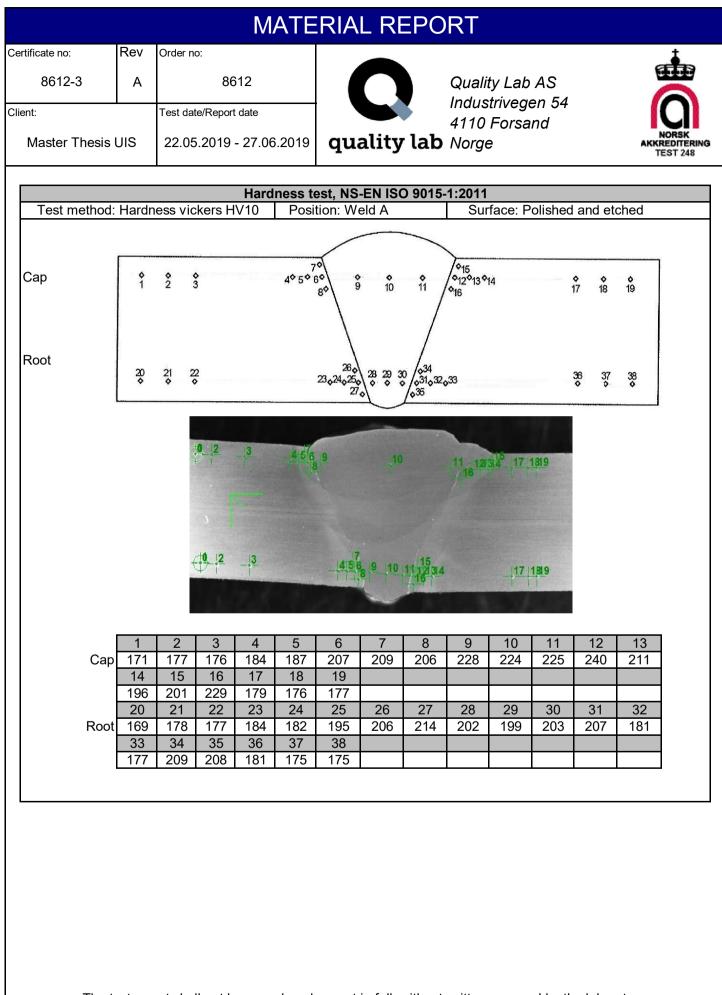
Weld

FL

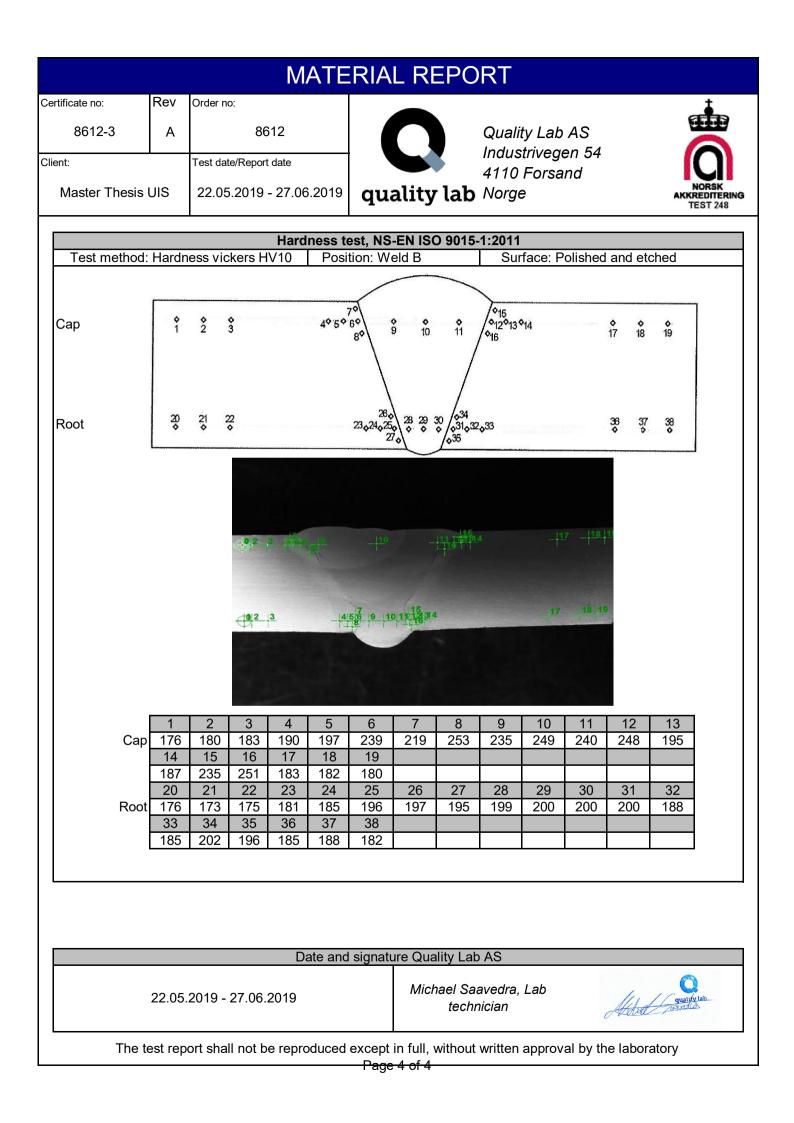
FL+2

FL+5

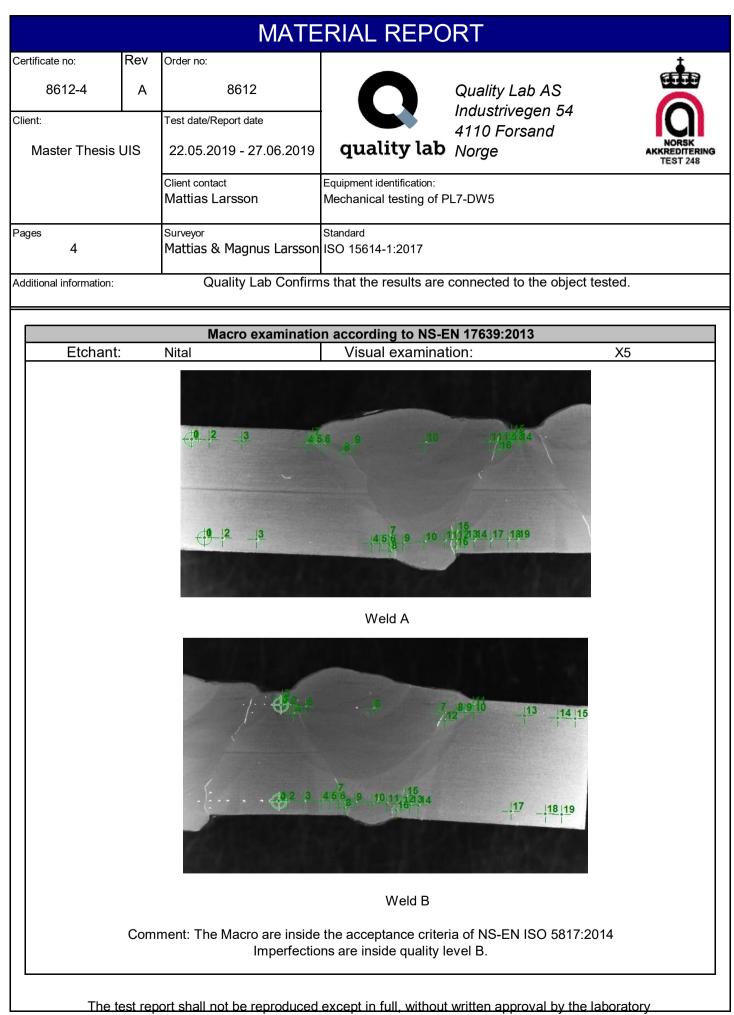
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• Doc E504 - Qlabs Material Report 8612-4 (PL7)



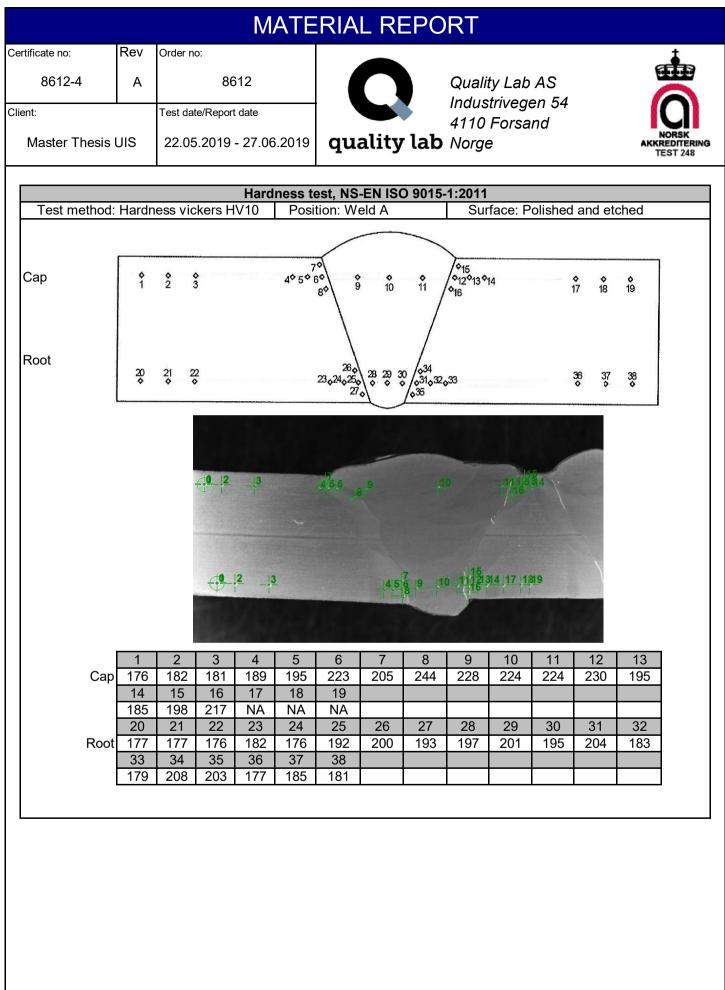
			MATE	RIA	L REPC	RT						
Certificate no:	Rev	Order no:						1				
8612-4	A	86	612			Quality Lab AS Industrivegen 54						
Client:		Test date/Report date										
Master Thesis UIS		22.05.2019 - 27.06.2019		4110 Forsand quality lab Norge				NORSK AKKREDITERING TEST 248				
Tensile test, NS EN ISO 4136-1:2012 / ISO 6892-1:2016 Method A.1												
Test ident:		t:	Area [mr	n²]	R _m [Mpa]		Fracture					
Cross weld sample 1			101,63		534		Base material					
Cross weld sample 2			105,61		533		Base material					
		Charpy V im	pact test, KV	/8, NS-I	EN ISO 148-1:	2016; I	SO 9016:2012					

Charpy V impact test, KV8, NS-EN ISO 148-1:2016; ISO 9016:2012												
Test ident	Dimension	Notch	Test temp	Single value		es [J]	Average					
Test ident	[mm]	Orientation	[⁰ C]	1	2	3	[J]					
Weld	10x10x55	Т	-40	107	84	81	91					
FL	10x10x55	Т	-40	203	281	141	208					
FL+2	10x10x55	Т	-40	245	354	396	332					
FL+5	10x10x55	Т	-40	343	306	301	317					

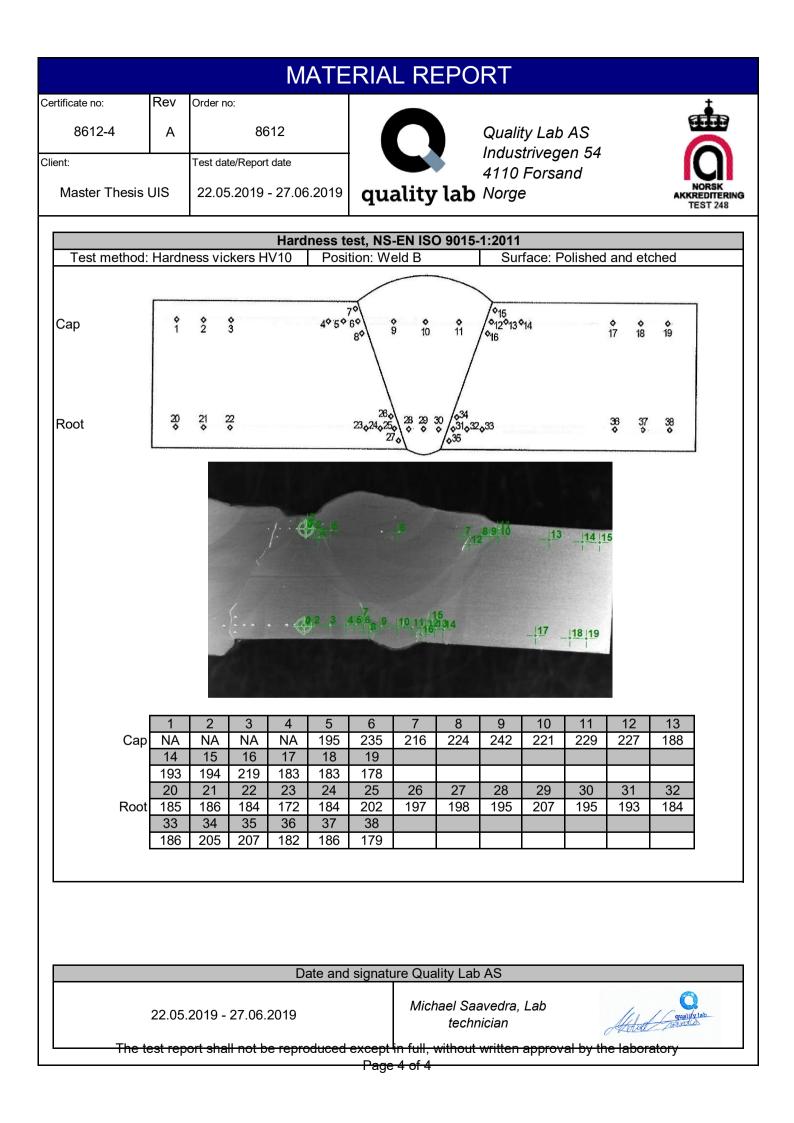
MPI examination according to ISO 23278:2015 level B (non accreditated test)

MPI performed, No findings Accepted

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Microstructural Examination

• The chapter "Microstructural Examination" has no attachments.



Fatigue Testing

- Doc G701 Fatigue test Prior to fabrication specimens
- Doc G702 Fatigue test Unwelded base metal specimens
- Doc G703 Fatigue test Production test plate specimens

• Doc G701 - Fatigue test - Prior to fabrication specimens

Area = olould be 205 m 2 ! J= F - 98.6 = 481.2 MPs

T	T	

Project name Specimen ID F1 Date II/6~1 G

R= 0,1 Larsson Master thesis pilot

Fatigue test log - rev 0

481.2 Stress range (ΔS) 98.6 Mpa Load range ∆F kN 250 000 Target life Cycles Max load 109.6 kN 328.8 11.0 Area mm² Min load kΝ 60,3 Cyclic rate Hz Mean load kΝ 60 3 49.3 Test rig Machine load MTS 809 ± kN ΔF=ΔS*Area/1000 Fmax=ΔF/(1-R) Fmin=Fmax-ΔF

Fmean=(Fmax+Fmin)/2

F=Fmean±∆F/2

Date	Time	Cycles	Observation, comments	Sign
11/6-19	9:30	0		
1 11	36269.45	- 3676	the Liten spreta ches	7.4
11/6 - 19		12195	Soude Soude Milling	
			ALLAND ALIAN	
				1
		Ø	Failure Criterion.	
			Complete fracture	
			Initiated at weld cap	
			N=12 195	
		5.	Complete fracture Initiated at weld cap N=12 195 AS = 481.2 MPa	
				0
				1
			-	
			×	
5-2-			-	
		- Andre and a state of the second		

Ø

Area stant be 205 mm² 5 - F = 98640 - 481.2 Mrn A = 205

 $\Delta F = \Delta S^* Area / 1000$

 $Fmax=\Delta F/(1-R)$

Fmin=Fmax-∆F

Fmean=(Fmax+Fmin)/2

59100

= 288.7 _________



Date

Project name Specimen ID Date

481.2

R= 0,1 Larsson Master thesis pilot

	300			98.64	
Stress range (∆S)		Мра	Load range ∆F	59,18	kN
	250			109.6	
Target life	100000	Cycles	Max load	65,75	kN
	8			11. 6	
Area	328,8	mm²	Min load	6,58	kN
			я	60,28	
Cyclic rate	10	Hz	Mean load	36,16	kN
			60.28	49.32	
Test rig	MTS 809	Machine load		29,59	± kN

 $F=Fmean\pm\Delta F/2$ Observation, comments Sign Cycles Time The Stat WHE ale. PVA comp 9.40 76/19 0 New Load: 300 mpa Pause 26895 13.10 2 Rostatel -new Lond 32303.27 7:50 VG Le 38570 Brall 8.70 in Failure Criterion: 0 complete fracture Initiated at weld root N= 38570 DS = 481.2 MPa

3 -> 6) continue offor

Fatigue test log - rev 0



3

 (\mathcal{L})

Project name Specimen ID Date R= 0,1 Larsson Master thesis pilot F 2 11/6-20 W

$\mathcal{L} = O_{\mathcal{L}}$ Stress range (ΔS)	200	Мра	Load range ∆F	40.3	kN
Target life		Cycles	Max load	44.7	kN
A: t _K w Area	8.2×24.7 = 201.3	mm²	Min load	4.4	kN
Cyclic rate	163015	Hz	Mean load	24.6	kN
Test rig	MTS 809	Machine load	24.6	20,2	± kN

 $\Delta F = \Delta S^* \text{Area}/1000 \approx \frac{200 \cdot 201.3}{1000}$ Fmax= $\Delta F/(1-R) \approx 40.3/(1-0.1)$ Fmin=Fmax- ΔF

Fmean=(Fmax+Fmin)/2

 $F=Fmean\pm\Delta F/2$

	Date	Time	Cycles	Observation, comments Sig	zn
)	11/6-2019	13.30	0		2.6
	e.	15.55	10000	2,38 ma/0.1 mg	The
	12/6-2019	0800	670 000	No crachs	2.6
	<i>u</i>	9:35	Flavo		7.6
	⁶ 1	10:05	245000	0,47 / 0,13	
	И	12:25	870000		Ze
	- V -	14.27	977 900	0,42 / 0,12	
	-11-	16:22	1082965	/ · / M	lattics
	13/6	7:55	192000		2 *
	~ lom	10.00	2037347	-11- 0.425/0.12 4	lattics
		10:14	2050 000	-11- 0,44/0,13	er
	- LL -	11.37	2718780	0, 44 (0,13	
	-14	11:42	2124441	Stopped test to continue with	specman ML
		15:30	+ 0		luttias
	16/6-2019	11:13 -	1066318	0.40/0.10	Ц
	18/6-2019	09:24	4 3 559 350	0.40/0.10	24
	18/6-2029	17:33	4000000	237/009 1	Keyn
	19/6-2019	09-47	4876000	0,40/0,09	lign
	-11-	17:13	5277795	0.39/0.09 1	lateias
	20/6-2014	09:33	6157710	0.40/0,10 ~	- 11-
	21/6-2019	11:02	7534716	STOP - u - Ma	attices
		TOT =	9659157		
┟					
-			8	Run out	
				N=9654157	
┟				$\Delta S = 200 M Pa$	
╞	-				
+					
$\left \right $					
$\left \right $					
L					



Project name Specimen ID Date R= 0,1 Larsson Master thesis pilot 두닉 1³/ &

Stress range (∆S)	300	Мра	Load range ∆F	63.75	kN	A ΔF=ΔS*Area/1000 = 300.217.5
Target life	(000 000	Cycles	Max load	70.87	kN	β = Fmax = ΔF/(1-R) = 5.25
Area	8.5×25 = 212.5	mm²	Min load	7.08	kN	\sim Fmin=Fmax- $\Delta F = 29.83 - 63.25$
Cyclic rate	10	Hz	Mean load	38.96	kN	? Fmean=(Fmax+Fmin)/2 = 70.83 - 7.05
Test rig	MTS 809	Machine load	38.96	31.88	± kN	F=Fmean± $\Delta F/2$ 63.75

Date	Time	Cycles	Observation, comments	Sign	
13/6-19	11:46	0		Mattia	
(:	12:26	24 851	- 2.32/- 2.78	44	4 d = 0,46 mm
-11-	12:58	43752 312278	-231/-277	ligns	
19/6	7:40	312278	Bindd	Jugars - 24	
					1
		0			
			Complete fracture at weld toe cap		1
			N = 312278]
			ΔS = 300 MPa		1
]
]
]
]
]
]
]
]
]
			*]
]
]
]
-			5		
]
			1]
]
					1
					1
					1
	х.				



Project name Specimen ID Date

R= 0,1 Larsson Master thesis pilot F^3

Stress range (∆S)	300	Мра	Load range ∆F	66	kN	ΔF=ΔS*Area/1000
Target life	320 000	Cycles	Max load	73.33	kN	Fmax=∆F/(1-R)
Area	25%8.8 = 220	mm²	Min load	7.33	kN	Fmin=Fmax-∆F
Cyclic rate	10	Hz	Mean load	40,33	kN	Fmean=(Fmax+Fmin)/2
Test rig	MTS 809	Machine load	40.33	33	± kN	F=Fmean±∆F/2

Date	Time	Cycles	Observation, comments	Sign	1
14.06.2019	1	0	Misaligned specimen 0.71 (0.23	Mattias	1
- L	71.79	26 300	076(0,28	Man	1
- 16-	(2:1)	57 967	0,77/0,29	Madia]
	13:38	110 095	0,78/0,20	- 4	
- in	13:50		0.77/ 0.29	Mayn	G H
	25:28	12,400	0.77-	Myny	
15/06-2014	12:16	924104	Fracture	Matias	
		6	Failure Criterion:	-	
			Complete fracture in base material		
			N = 924104		
			$\Delta S = 300 M Pa$		1
				2	
]
2 2					
					1
				1	

Test (1) (Fran test 6) Fatigue test log - rev 0 R= 0,1

R= 0,1 Master Hesis Lassim



Project name Specimen ID Date

Stress range (ΔS)	300	Мра	Load range ∆F	58.03	kN
Target life	500 000 .	Cycles	Max load	64.48	kN
24.8 × 7.8 Area	= 193.44	mm²	Min load	6.45	kN
Cyclic rate	13,15	Hz	Mean load 🛼	35.47	kN
Test rig	MTS 809	Machine load	35.47	29.02	± kN

F2

∆F=∆S*Area/1000	67.704
Fmax=∆F/(1-R)	75.23
Fmin=Fmax-∆F	7.52
Fmean=(Fmax+Fmin)/	2 41.38
F=Fmean±∆F/2	41.38 ± 32.85

AS=300	Date	Time	Cycles	Observation, comments	Sign	drspl.
	1/7-2019		0	Re-grinded, cont. form (6)	Matters	
	u.	1832	200 000	0,54/0,14	4	0,40.
	2/7-2019 62.1	Carolo	878000	0.55/0.15	Martins	
	h	Gall	987000	li	hadins	
	-1-2-2411-	13 31	1220000	254/014	Magni	
	11	1905	1520 464		MEHIE	
	И	2033	1600 000	И	11	
	317-2019	07.27	2189 000	0.55/ 10.15		
	in	09:27	2300 000	STUP	Madra	
5=350	h	0933	Ő	350 MPa 0.66/0.19	"	0.47
	1.	13:06	190 800	668/0.20	Magn	
	1	15:02	295 000	0.68/0.21	din	
	ч	17-39	432064	1.0	Madies	
	4/4-2019	07:32	747488			
				*		
	al an		nĝo	Etizyue Specimen F2. Tested at 3 differ	ant as	
	29		Test (3)& (6):	Run out?		-
			1.0	S= 200 MPA		-
				N = 9654157		4
			× ($\int_{\mathbb{R}^{n-1}} f_{n} ^{2} dx = \ f_{n}\ ^{2} \ f_{n}\ ^$		4
			Test (17) .	Run out.		
				We stoped at 2300 000 cycles at AS=3.	OD MPa. We	4
				continued to test at AS= 350		4
			· . ·	4 5 = 300 Mla		-
				N = 2300 000 (Stoped)		4
			. 's:	2		
			0	Failure criterion;		4
				Complete fracture at wold root		4
	4			AS=350 MPG		4
				N = 747 488		4
						4

• Doc G702 - Fatigue test - Unwelded base metal specimens

Test

Fatigue test log - rev 0

\cap			R=	0,1				
	Project nar	me	Larsson Master		lot			
	Specimen		BM2	thesis pr		,		
U)	Date							
			27/6-2019			15	0 234.8	
						57	21	
Stress range (∆S)	350	Mpa 👘	Load range ∆F	82.2	kN	<u>Θ</u> = ΔF=ΔS*A	rea/1000	
	1.1		-			1	A	
Target life	1.0	Cycles	Max load	91.3	kN .	🔞 🤋 Fmax=4	∆F/(1-R) °, /	
	23.2×F.6	1	1					
Area	= 234.8	mm²	Min Îoad	9.13	kN	🦒 🔩 Fmin=F	max-ΔF	
4. 	10						BC	
Cyclic rate	10	Hz	Mean load	50.2	kN	D ₁ Fmean=(Fm	nax+Fmin)/2	
<u>×</u>			F : 0	41.1				
Test rig	MTS 809	Machine load	50.2	-11. (± kN	F=Fmea	an±∆F/2	
			Verifierst au	Magmis.				_
Date	Time	Cycles	Obse	rvation,	comme	ents	Sign	
21/6-2019	11:23	0			0	83 10.20	Maura	0
~~	12:41	47200			0,	8310,2	high	
	17:12	62700			0	83/0.2	chi	
	15.21	942957			O.		cen	
- (-	17:10	20800			0.9	33/0.2	Myn	
-11-	18:33	258389			0,	82/0.19	Mattia	
	19.52	305000			0,9	3/0.79	Magan	
22/6-2019	08:35	763:00				310,20	un	
- 11 -	09:38	806000				y/ 0, 4 21	Majoris	
-1	10-26	83000			85		dayon	
-4-	77:0g	822020	-			510.20	dins	
!_	11:18	865 000		2		4/0.20	Mart.23	
	17:55	285 000				3/0.20	ingas	_
- (12:36	901 000				10.20	Myn,	
-1-	14:24	972 933	NO CRACH	د		3/0.20	Mattics	_
23/6-2019	10:15	1680000				10.20	Mag	4
		1730 020	29 e - 1		0.85	10.27	dign	•
- 1	13:13	1767000				10,22	Mija	-
	14:21	1828 00				1022	Mayn	4
- (-	15:24	1886000				10,20	Ming_	-
~ (_	16:32	1950000			0.54	10.20	Mign	-
	18:00	2025000				10.20	Mayn	-
-1	19:13	2084000				675	mon	
24/6-2019	08:35	2760 000				0.19	Mayrus	Pri
<u> </u>	29.41	2760746		ð	,03/	0.20 0.63	dryne	- Wil
· · ·			0					- ^`
			Run out:					-
			N = 27607 $\Delta S = 350 M$					-
			23 - 350 PC	15.		4		-
								-
,		-						1
		1. South	State -					-

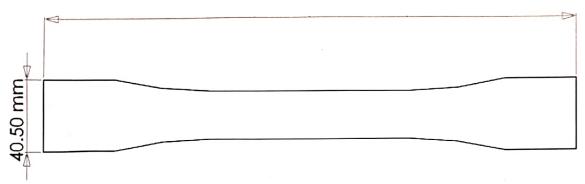
inse pertined

Visnel Test Log Funout AS = 350 MPZ N = 2760 776

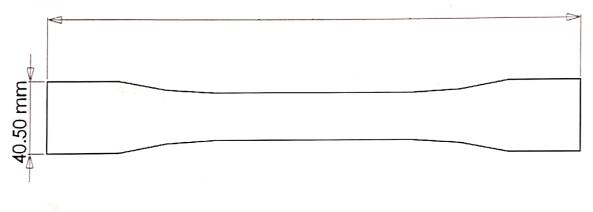
Test: 7

Name: BMZ Mogmas Lanson.

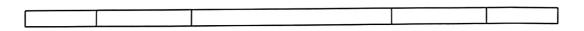




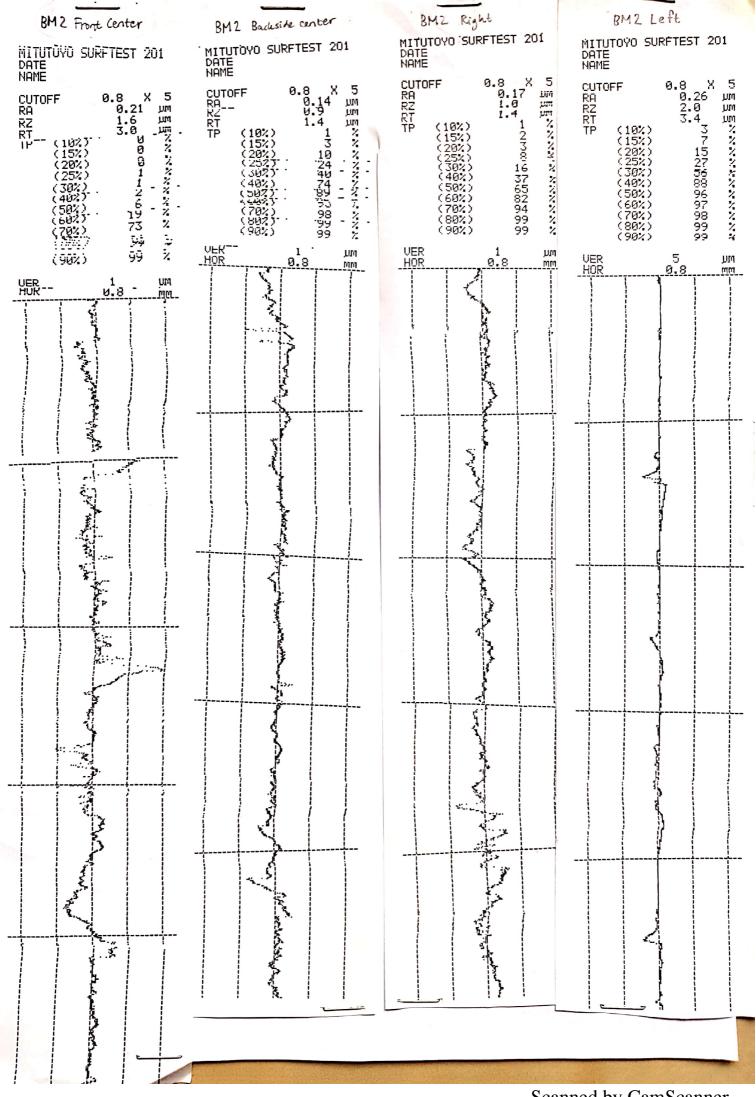
Back side



<u>Right side</u>



<u>Left side</u>



Scanned by CamScanner

Fatigue test log - rev 0

Signature : allen Approved : Math



Project name Specimen ID Date R= 0,1 Larsson Master thesis pilot BM1 24.06.2 019

Stress range (∆S)	430	Мра	Load range ∆F	102.13	kN	ΔF=
Target life	1000 000	Cycles	Max load	173.4g	LNI	5
	1000 000	Cycles		15.10	KIN	Fn
Area	237.51	mm²	Min load	17.35	kN	Fn
Cyclic rate	15	Hz	Mean load	62.42	kN	Fmear
Test rig	MTS 809	Machine load	62,42	51.07	± kN	F=

∆F=∆S*Area/1000

Fmax=∆F/(1-R)

⁻min=Fmax-∆F

Fmean=(Fmax+Fmin)/2

F=Fmean±∆F/2

Date	Time	Cycles	Observation, comments	Sign]
24.06.2019	10.34	0	2.19/1.30	denn	
- 1 -	11.46	9247	1.66 / 0 7 3 0.93	Jam-	Fielfal Tog av 1-st & instrict specimen izer.
	17.090	15643	10.49/70.48	Im	specinen izen.
			Failure criterion:		
			complete fincture		
			N = 15643		
			$\Delta S = 430 MPa$		
					1
					1
					1
					1
				,	
					1
					1
			-		1
					1
					1
		<i>x</i>			
					1
	•				•

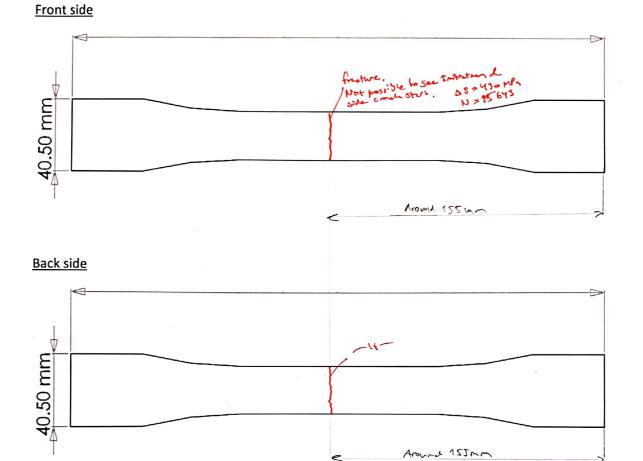
Visual Test (05

The voud test log was created after BMI fatiguetest. was performent.

Test: 8 · It can be confirmed that no transverse scretches were visible & roughness was below < 20pm Name: BM1

before testing started

Megnus Larsson

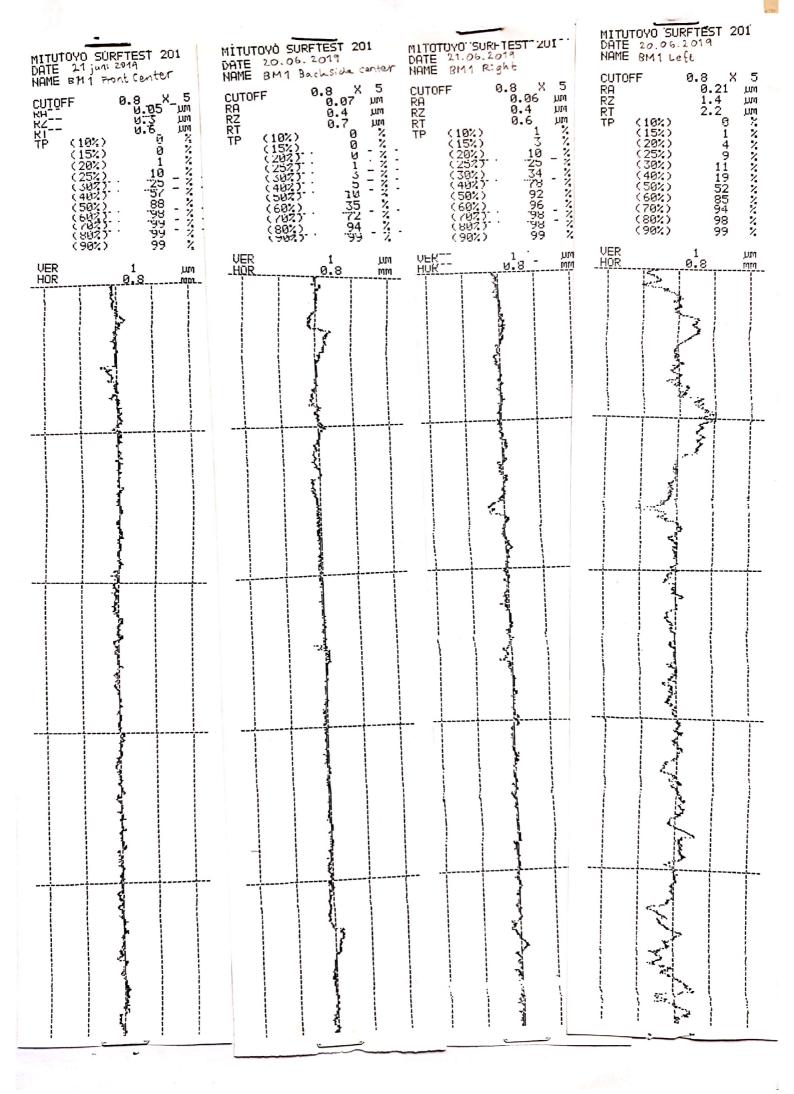


<u>Right side</u>



Left side





J	Project nar Specimen I Date		Or La	0,1 15 -, son ,		er Thesi	3	
Stress range (ΔS)	400	Мра	Load range ∆F	91.28	kN	ΔF=ΔS*A	rea/1000	
Target life		Cycles	Max load	101.44	kN	Fmax=/	\F/(1-R)	
Area	8.45.270		Min load	10.14	kN	Fmin=F	max-∆F	
Cyclic rate	13	Hz	Mean load	55.79	kN	Fmean=(Fm	nax+Fmin)/2	
Test rig	MTS 809	Machine load	55.79	45.64	± kN	F=Fmea	an±∆F/2	
Date	Time	Cycles	Obse	ervation, o	comme	ents	Sign Matias Maba	displ
24.06.2019	14:55	0			1,	01/0.25	Madia,	0.77 -
i.	17:16	110 000			Ι.	13/0.36	Mabre	0,77.
11	19:25	242563				18/041	Malti	tr
	20:09	1			1.	18 10.41	Myan-	
		426615	Frecture					
			E.V. C.					
			Failure Cri					1
		4	Complete		_			
			N= 42661					
			DS = 4001	1Pa			1	
								1
			1					
								`

Visual Test log

The voul test log was created after, BMS Retry we test was performed

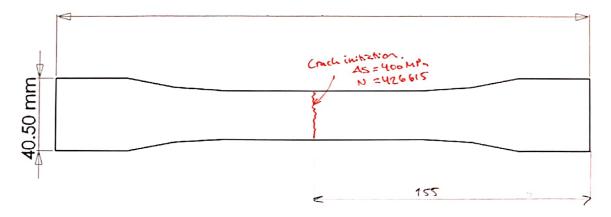
Test: 9

· It can be confirmed that no transverse scretches were visible & roughness was below I 20 pm . be fore Lesting was stirted

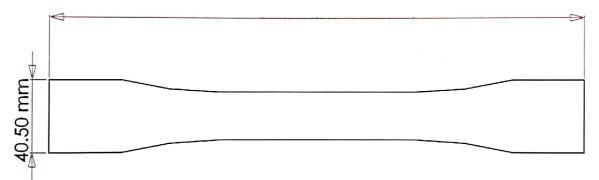
Name: BMS

Magnus Caroson

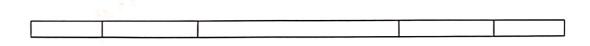
Front side - Not checked with visual fast



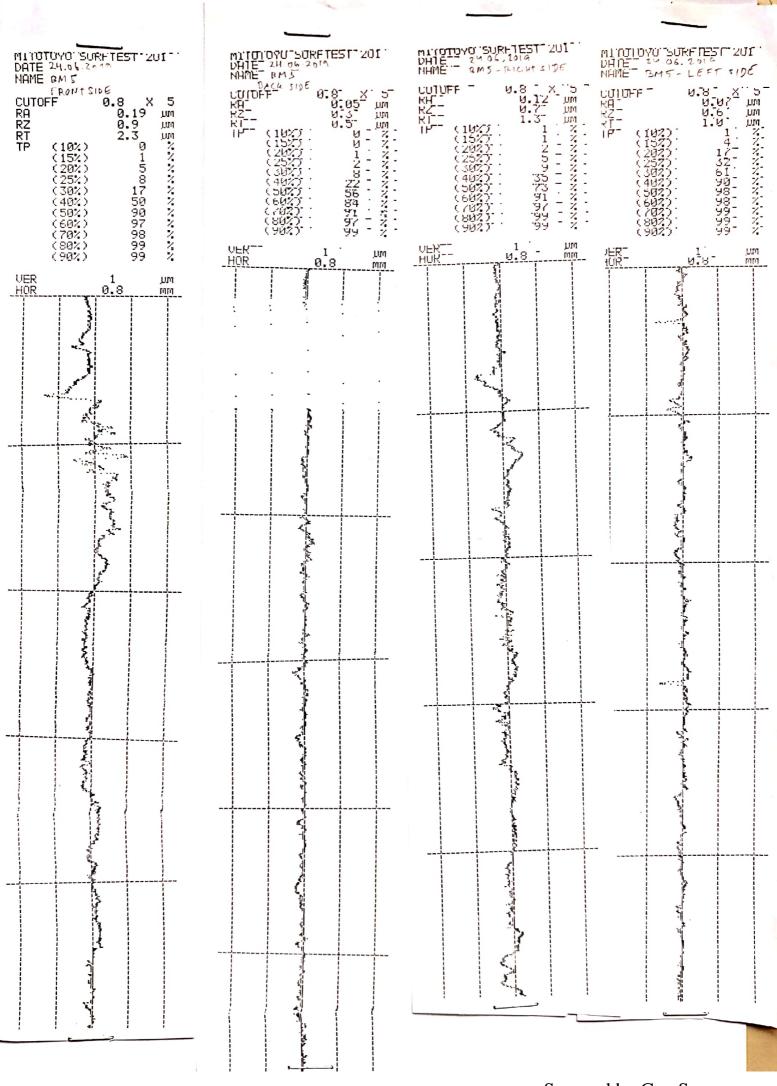
Back side



Right side



Left side

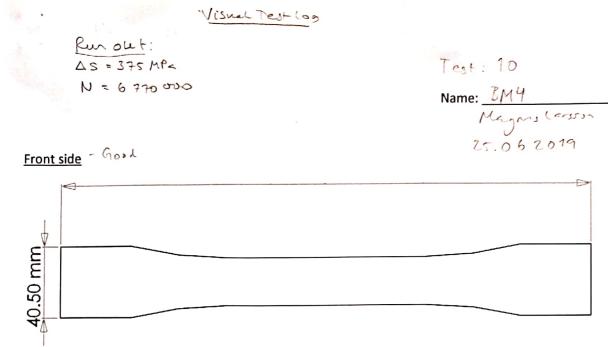


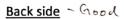
Scanned by CamScanner

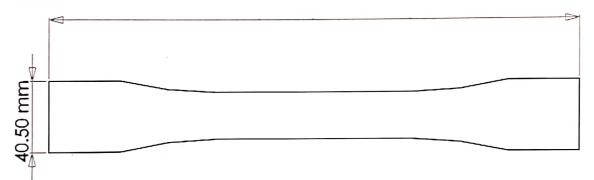


Fatigue test log - rev 0

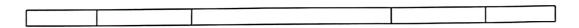
uS	Project na Specimen Date		Mas BM	0,1 T~ 7 W 14 /6 -20		Larsson		
Stress range (ΔS)	375	Мра	Load range ∆F	87.7	kN	∆F=∆S*A	rea/1000	
Target life		Cycles	Max load	97.47	kN	Fmax=/	∆F/(1-R)	
27.2×8.4 Area	237.92	mm²	Min load	9.75	kN	Fmin=F	⁻ max-ΔF	
Cyclic rate	13	Hz	Mean load	53.61	kN	Fmean=(Fm	1ax+Fmin)/2	
Test rig	MTS 809	Machine load	53.61	43.85	± kN	F=Fmea	an±∆F/2	
	I							last -
Date	Time	Cycles	Obse	rvation, o			Sign	,69 mm
25.06.2019	12:50	0				0.92/0.23	Mart	.69 mm
u	13:00	7700				94/0.26	ι	.68
ii	15:00	103300			C	, 44/ 0,30	10	.69
t.	16:14	160 000			1	00/0.31	m	u
٢٠	17:40	22687n			0	95/0.30	dezz	"
ΕL	19:18	304 020				10	Mart K)	u
~ U	20:04	775 000			٥,	29/0:20	Myn	x
26:06.201 m	10:00	191400				L.	halt ~	n
	12:00	1079000				ĥ	in	1
и	14:25	120000				И	u	n
ir	15:30	125000		÷.,		in	n	1
Ĺ	1950	1450 000				ñ	-1	1
27.06.2019	08:09	2027000				0,99/030	Man	1
L	13:00	2250 000				1	Matter	1
4	1730	2470000				4	Li	1
n	1870	2 500 000	5105	ал. Г		2	n	1
28.06.2019	03:06	3143 000				1.00/0.30	1	1
n	1917	3666 -00				0 9 4/0,27		1
29.06:2019	08:13	4271 000		2		1.00 0.30		1
		4432000				949 10.30	M	1
30.06.2019	11:14	55 40 000				0.29/0.30	"Li	1
		575000				1.00/0.30		1
1.07.2012		55000				100.00	elin	0,70
11		6770600	STOP			11	Madres	
	7 6 6 7	S TPU GLO				((WIELENC J	1
			Run out:					1
	1		N= 6770	ഹാറ				1
			$\Delta s = 375 M$					1
			07310					1
							-	1
								1
								1
								1
	I							_







Right side - Good



Left side - Good

N1101040'SURFIEST 201 DHIE 25.06.2010 NHME BM4 Fool Side COTOFF 0.8' X'S KA 0.18' um KZ 0.18' um KI 0.18' um KI 0.18' um (157) 0'X' (157) 0'X' (202) 0'X' (202) 0'X' (202) 0'X' (202) 0'X' (202) 0'X' (202) 0'X' (202) 0'X' (202) 0'X' (202) 0'X' (302) 0'X' (302) 0'X' (302) 0'X' (302) 0'X' (302) 0'X' (302) 0'X' (302) 1'X' (302) 1'X' (902) 7'X'	M1'0TDY0'SURFTEST-201- DHTE: 25.06.2010 NHME: BM4 BAASAA CUTOFF 0.8 X 5 KH 0.8 X 5 KH 0.8 LM KI 0.8 LM KI 0.8 LM KI 0.8 LM KI 1.8 LM (157) 1.8 (157) 1.8 (207) 3 X (307) 3 X (307) 5 X (307) 3 X	MINTOVU SURFIEST 201 DHIE 2005.2010 NHME 004 RightSide CUIVEF 0.8 X 5 RH 0.05 um RZ 0.4 um RI 0.8 um IF (10%) 1 X (15%) 3 X (15%) 3 X (25%) 20 X (30%) 40 X (30%) 40 X (50%) 99 X (60%) 99 X (80%) 99 X	UTUYU SURFIEST 201 E 26,06.2014 E 674 Left SAC UFF U.8 X (182) 2 (152) 4 (282) 9 (282) 42 (382) 42 (382) 42 (382) 97 (582) 97 (882) 99 (882) 99 (882) 99
VER 1 µm HUR U.8 mm	(257): 3 - 2 (307): 3 - 2 (407): 6 - 2 (507): 34 - 2 (607): 79 - 2 (607): 79 - 2 (607): 79 - 2 (707): 95 - 2 (807): 99 - 2 (907): 99 - 2	(25%) (30%) (40%) (40%) 84 - 2 (40%) 98 - 2 (50%) 97 - 2 (60%) 99 - 2 (60%) 99 - 2 (70%) 99 - 2 (90%) 99 - 2 (90%) 90 - 2 (90%) (9	(502) 94 (602) 97 (702) 99 (802) 99 (902) 99 (902) 99 (902) 99 (902) 99 K ⁻ 0.8 m
		and the second	المراجع
	No and the second secon		and the second second
			A second se
		A Star	



10 C

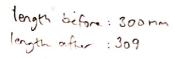
Project name Specimen ID Date

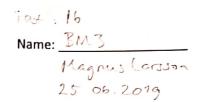
Sec.

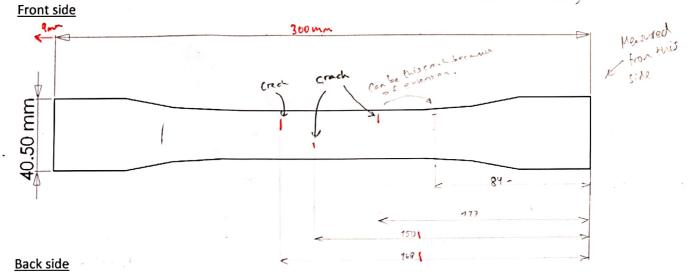
weld proximity investigation BM3 22/7-19

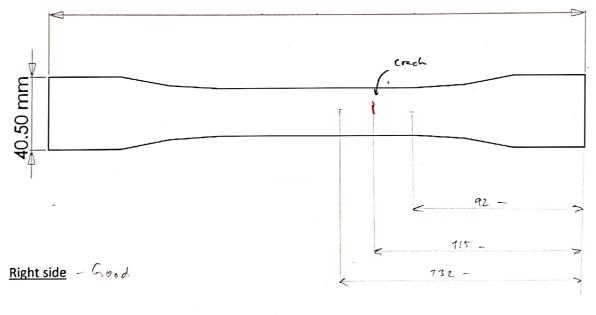
Stress range (ΔS)	400	Мра	Load range ΔF	96.06	kN	∆F=∆S*Area/1000
Target life	500 000	Cycles	Max load	106.73	kN	Fmax=∆F/(1-R)
27.32× F.7 Area	= 240.14	mm²	Min load	10.67	kN	Fmin=Fmax-∆F
Cyclic rate	13	Hz	Mean load	58.20	kŇ	Fmean=(Fmax+Fmin)/2
Test rig	MTS 809	Machine load	58.70	48.07	± kN	F=Fmean±∆F/2

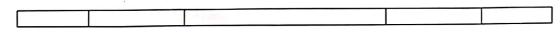
Date	Time	Cycles	Observation, comments Temp	Sign	
22/7-19	1019	1000	6.18/5.28 23	Matini	1.0.r
- ['] 4	10 3.2	5000	6.53/5.73 1	4	1.5
- 'u	1045	16 000	7.41/6.60 4	-	0.82
LL	1128	50 272	7.82/7.01 1.	Li	0.81
и	1247	111 102	7.88/ 7.07 "	~	
	1726		790/709 "		0.81
n	1521	221000	7.91/7.10 4	u	
~	1721	371000	292/2/10 11		82
L.	1853	394 201	Fracture in the middle	15	
					-
			Failure criteria.		
			AS= 400 MP2		
			N = 394701		
	- N				
			Fracture in the middle Not		7
			100% fracture 3 crack initiates		7
		- 60	on the back		7
					1
	~				1
	_				1
					1
					1
					1
					1
					1
					1
					1
					1
					1
					4
*******				· · ·	
					-





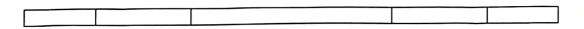


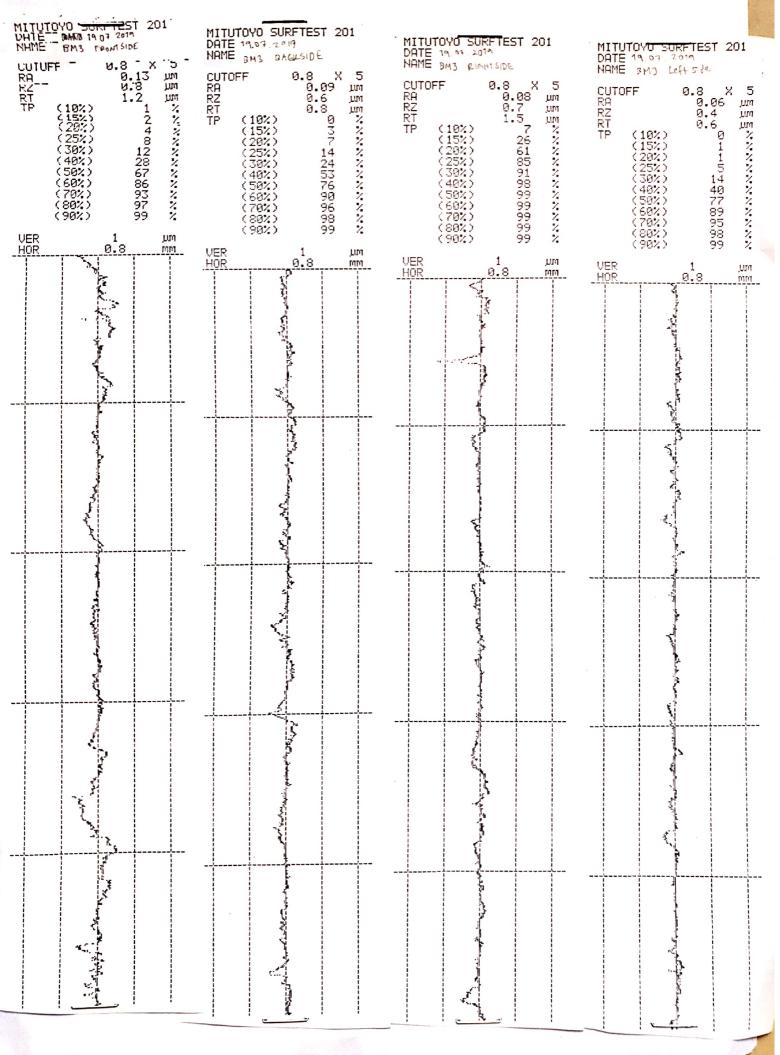




Left side - Good

-





• Doc G703 - Fatigue test - Production test plate specimens

Test(12) & (13) Yest leef

Fatigue test log - rev 0



Project name Specimen ID Date

R = 0,1Master Thesis 8612-2 2A (B) 04.07.2019

Stress range (∆S)	375 MP0	Мра	Load range ∆F	87.01	kN	
. A. C	T					
Target life		Cycles	Max load	96.68	kN	
8.60126.98						
Area	232.03	mm²	Min load	9.67	kN	
	10					1
Cyclic rate	13	Hz	Mean load	53.17	kN	Fr
						1
Test rig	MTS 809	Machine load	53.17	43.51	± kN	

 $\Delta F = \Delta S^* Area / 1000$

1/2

 $Fmax=\Delta F/(1-R)$

Fmin=Fmax-∆F

mean=(Fmax+Fmin)/2

 $F=Fmean\pm\Delta F/2$

Picture Date Time Cycles Adispl. Observation, comments Temp Sign 4/7-2019 10:37 0 - 11 ----10:42 3580 23.1 dlen -0.68 0,97/0,29 - 11 -11:13 26084 1.00/0.37 23.3 denn _11_ 17:36 45460 Mign 1.01/0,33 236 - 1. -12:27 85300 23.5 107/0.33 de - 11 -14:24 176 645 1,01/0.32 24.0 May--11-14:52 198210 24.1 Ma 7.01/032 -11-15:46 240 000 1.00/0.32 24.4 dein 17:11 - 11 -1,00/037 307 000 Menn-s 24. 2 _ 11_ 18:19 360 000 24.7 dina 0,9910.31 -11-20:40 469 000 101/032 23.8 Manus 5/7-2019 09:21 1063000 1.00 10.32 23.5 u 10:30 1175 000 - 1' -7.07 10.33 23.1 Man 17:10 1148 000 ninet : ora del - 11-101 23.1 M. 10.32 -11-1190 000 12:02 Bands are torming 1.01 / 0.33 22.8 Mannu Bands for both weigs Se nute (1) 133003 - 1'-13:30 1193 000 1.01/0.32 22 8 Mannis Check Error. 26:30 1258 000 Honne . 1- ---1.01/0.32 22.9 7325000 - 1. -17:40 1.01/0.32 23.0 deyn. ~ !! -1393000 19:07 107/033 22.6 Mannus - 10 21.23 1500 000 102/033 22.6 daynus 20190706-Bands is the same 617-2019 08:26 2019 000 22.6 101/0.32 Mayon, 082307 -1.-11:07 2139000 23.2 101/032 Mans ~ 11-12:57 dign 2227 000 100/0.32 23 1 Bonds 1: the Jame - (1-15.50 2360 000 1.00/0.32 23.8 dun - 11-18:56 2505 000 7.00/0,31 dunn 24.1 Bendy is the serie 7-2019 10:37 3234 000 den 1.00 1032 23:0 (_ 17:06 3543000 Naja 23.9 100/0.32 - 11-19:06 3647 000 Magnes 24.1 100/0.31 Bonds is the Same 8/7 - 2019 dugne. 07:41 4 225 083 23.3 1,4025 1.00/0.32 19:45 1. -1 800 000 24.4 Myn 1.00/0.32 21:06 - 1 -4875 000 100 10.3 2 24 Mannys 3 23:45 -1--000 000 1.00/0.31 24. 3 Mayn, 0.69

(I)

Run out: N= 5000 000 AS= 375 MP-

Scanned by CamScanner

+ 14m



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Fatigue test log - rev 0



2/2

Project name Specimen ID Date R= 0,1 Larsson Master thesis pilot 8612 - 2 - 2A(B)8/7 - 2019

Stress range (∆S)	HODMA	Мра	Load range ∆F	92.812	kN	
Target life		Cycles		103.124		
Area	232.05	mm²		10-		
Cyclic rate	12	Hz		56.778	kN	Fi
Test rig	MTS 809	Machine load		46,406		

 $\Delta F = \Delta S^* Area / 1000$

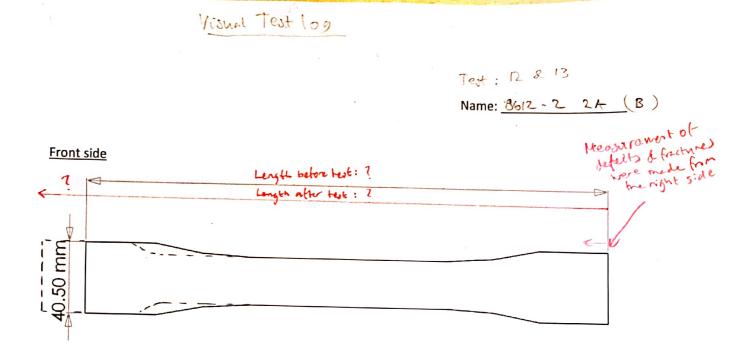
 $Fmax=\Delta F/(1-R)$

Fmin=Fmax-∆F

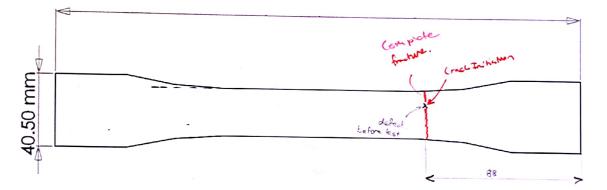
⁻mean=(Fmax+Fmin)/2

F=Fmean±∆F/2

Date	Time	Cycles	Ohannati	
8/7-2019	23:50	0	Observation, comments	Sign
8/7-2019 9/7-2019	10:15	439 404	Reads on the surface the same	dlagm.
	10.15	139 464	Bonds on the surface heatingness such when creak proves.	Maym. Hayn-
		•	Failure Criterion:	
			Complete fracture BM N = 439 404	
			N = 439 404	
			AS = 400 MPG	
				_



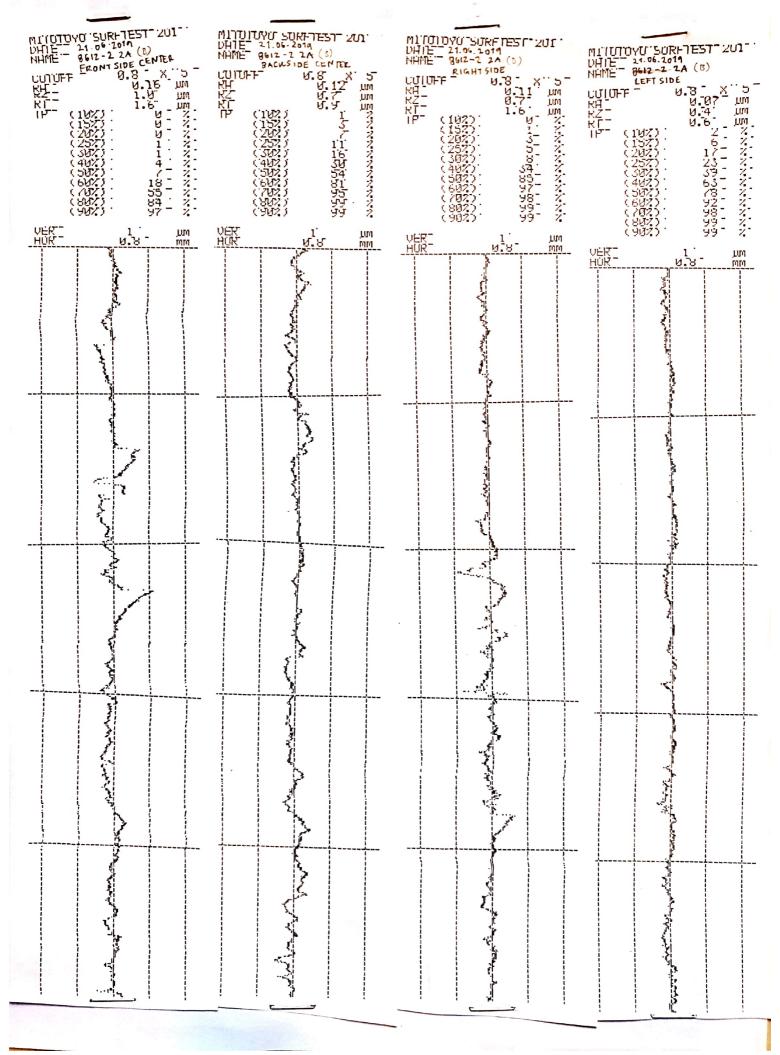
Back side



Right side



Left side

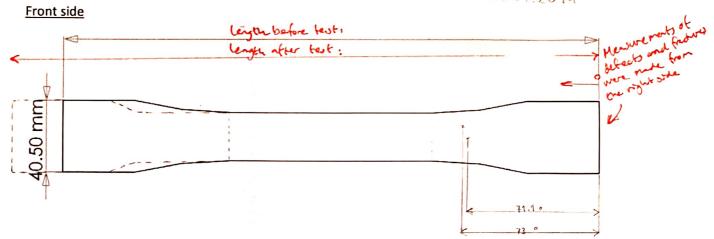


Visual test log

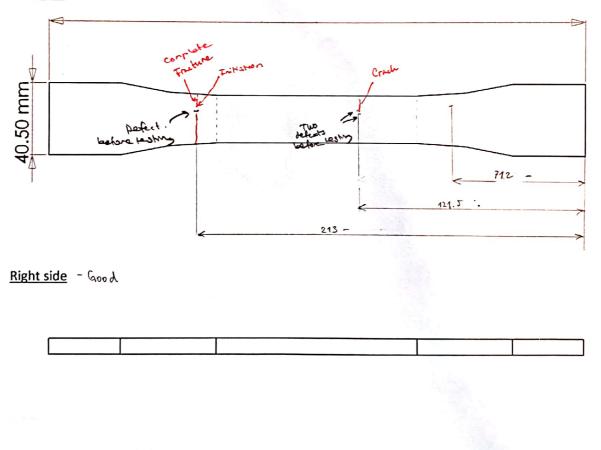
Test: 14

Name: 86(2 - 3 A3

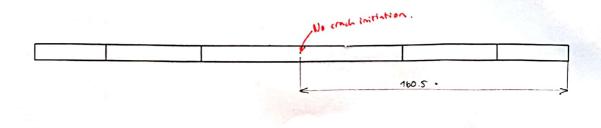
Magnus Larsson 06.07.2019



Back side



<u>Left side</u>



Test (14

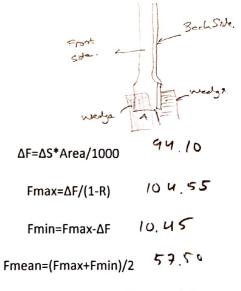
Fatigue test log - rev 0



Project name Specimen ID Date

R= 0,1 Larsson Master thesis pilot 8612 - 3 A3 9.07.2029

Stress range (ΔS)	375 MPa	Мра	Load range ∆F	88.22	kN
Target life		Cycles	Max load	98 oz	kN -
27.04 × 8.70 = Area	235.248	mm²	Min load	9,8	kN
Cyclic rate	13	Hz	Mean load	53.9	kN
Test rig	MTS 809	Machine load	539	44.11	± kN

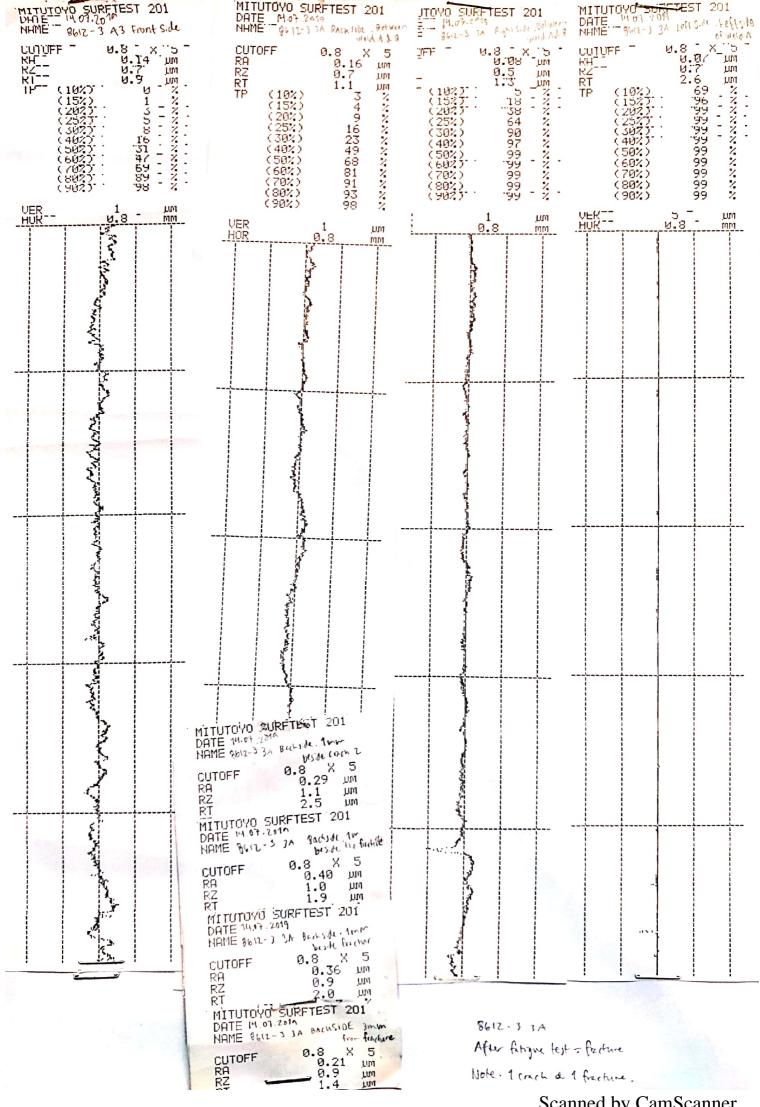


F=Fmean±ΔF/2 52.50 ± 4205

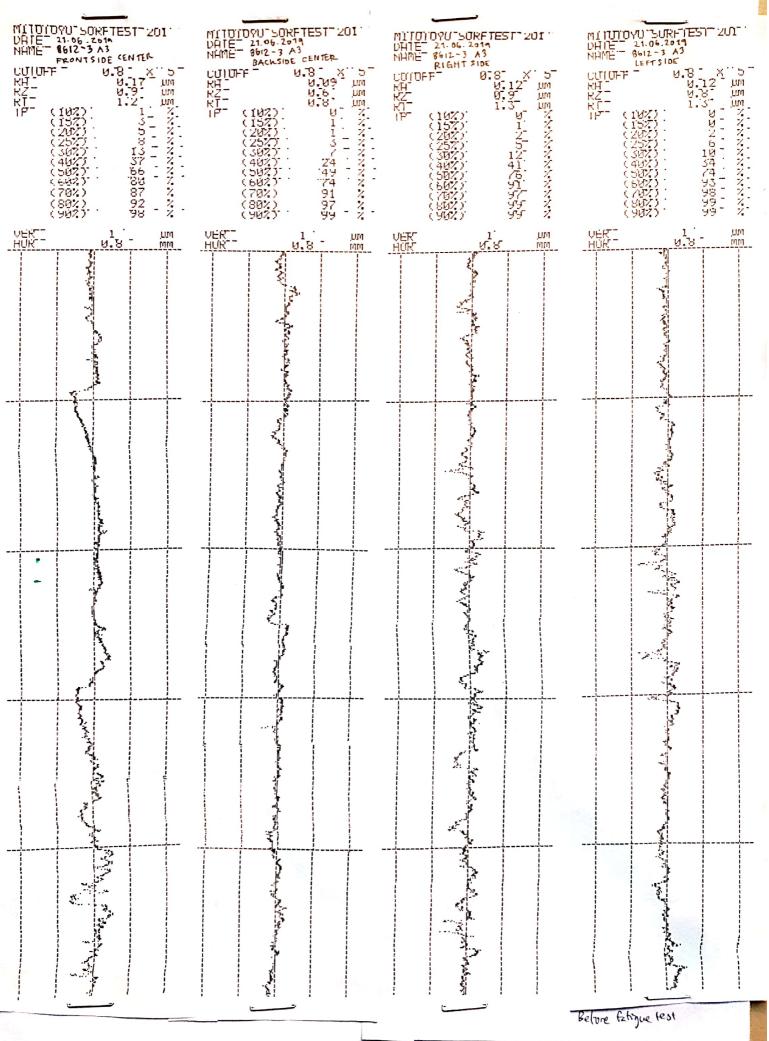
1

	Date	Time	Cycles	Observation, comments	STemp	Sign	
	9/7 -2019	10:10	0	0,98/029		degn.	8
	-11-	10:22	13 800	1.02/0.33	23.4	Magnes,	
	- (12:16	105 000	see son bendson myst 1.05/036	23.9	Manne,	
	-1	16:20	284 600	1.04 / 0.35	24.3	den -	
	- 1	18:11	379 600	1.04/0.35	24.8	very-	
	10/7 -2019	08:13	1 036 000	Some oxp bind. 1.05/0.35	23.5	Mayn	
	-1-	12:54	1255600	1.05/0.36	24.2	lega-	
	-1-	16:05	1386 000	1,04/0.35	24.9	din -	
	~!	18:11	1 502700	1.07/0.34	25:4	Mz_	
	- 1	19:24	1560 000	703/034	25.2	1 mm	
	11/7-2019	08:31	2 173 000	Same. 1.04/0.35	24.3	Man	
	-1-	10:43	2 277 000	204/035	24.7	lim	
	-1	14:20	2 449 000	104/035	24.4	diza	
	-11-	17:19	2680 000	1.034/0,34	25.1	Madias	
	12/7 -2014	08.08	3279 000	som new bins 105/035	27.1	clean m.	* *
	· · · -	1337	3532678	11	24.4	Maxin)	
	- 11 -	16:59	3693600	1.03/0.34	24.9	Myn	
	v	14112	3800 000			Magres	
	13/2-2010	01 34	4 375 300	1.04/0.35	24.8	plagno	
	11	18:28	4883979	11	-ti	Mag. 23	0,64
	1.	1958	4953344		- 11		0,69
MPA		2003	4-11000	1400 MPa 1,11/0,37	h		0,74
12	17	2010	7500	2.34/1.63			0,26
	14/7-19	06:24	483574	Fracture	4	11	
				Run out : We stoped at 495	3 344 cycle		
				at 375 MPn. We continued th			
				at 400MPa			
						S	
				Failure Conterion:		- Charles	
				Complete fracture BM			1.1
				N = 483574			
				AS = 400 MPm]

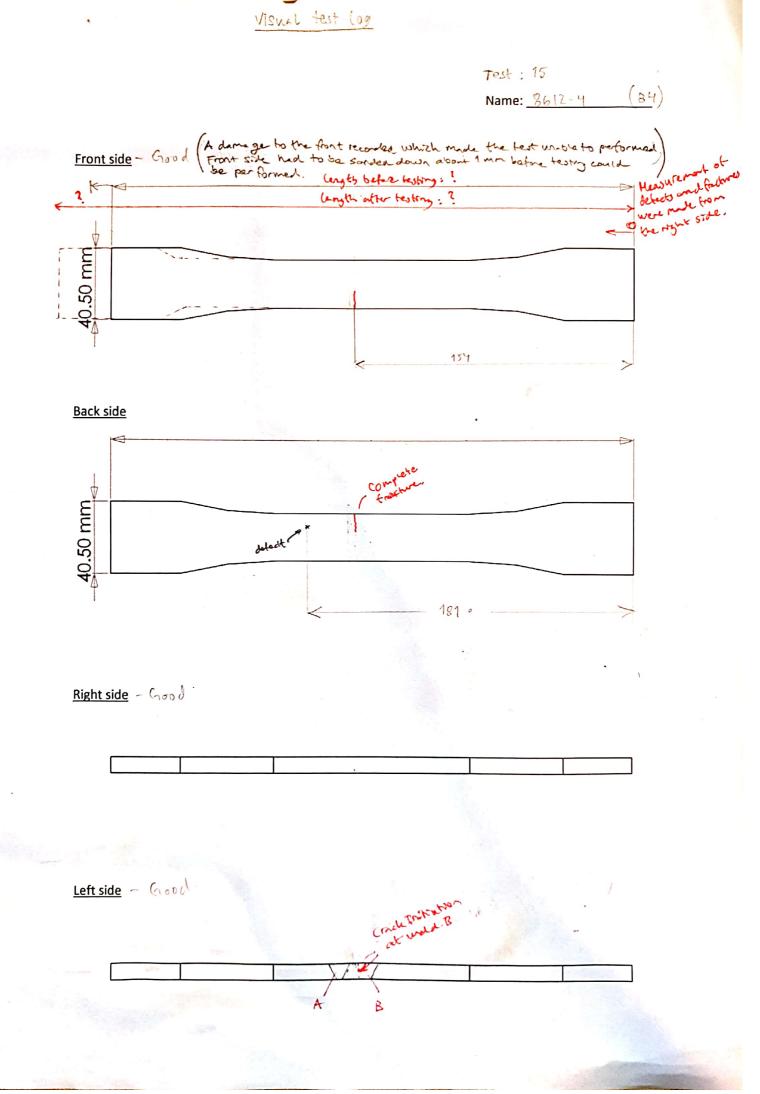
From visual inspection log & Fracture. Occure at the same place as reg. defect. (rech 2 -> Also here was a defeel registred.

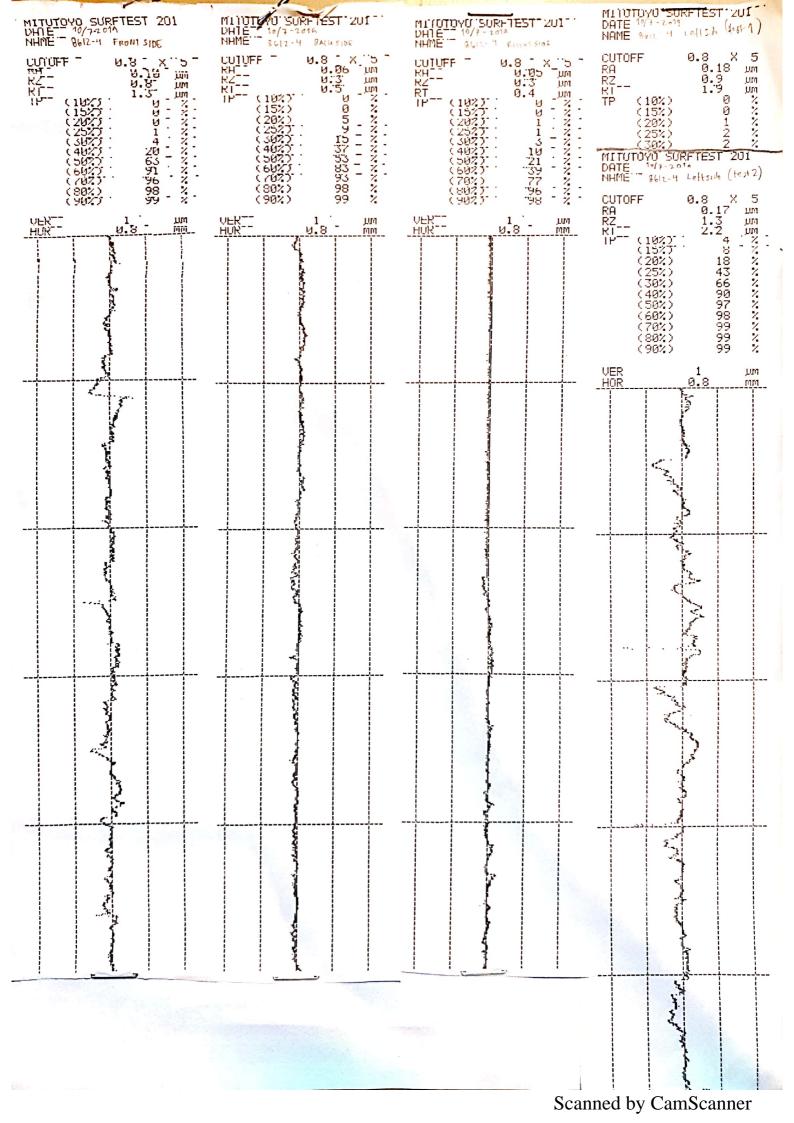


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2		ter Statistics		and the set of the set		in the second	Ta Fa	iure criterio	<u>^</u>
5)			F	atigue test log - r	ev 0		e Runau	H: We Stoped N=5000 0	oo at AS= 37
	uS	Project na Specimen Date		8612	0,1 10, Thesis 1- 4 7. 2019	Larsso		e Criteria 2:084 237 400 MPS	tb. Lookslike word B
	Stress range (ΔS)	375Mla	Мра	Load range ∆F	77, 794	kN	ΔF=ΔS*A	rea/1000	82.34
		5000000	Cycles	Max load	85.77	kN	Fmax=	∆F/(1-R) ⁶	11.49
	27.05 × 7.61 Area	205.851	mm²	Min load	8.577	kN	Fmin=F	-max-ΔF c	1.15
	Cyclic rate	13	Hz	Mean load	47,174	kN	Fmean=(Fn	nax+Fmin)/2	50.32
	Test rig	MTS 809	Machine load	47.174	38.6	± kN	+F=Fmea	an±∆F/2	41.17
				Ohsa	aution c	omme	nts TEMP	Sign	
	Date	Time	Cycles		vation, t	alla	26 23. +	Materas	
	14/7-19	1055	2200	Start		6103		den-	F
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	-1	20:13	436300					have	-
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	- <u>(</u>))	14:22	1287657			1	25.2	11	-
	1	1810	1467123			11			-
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	16/7-2019	08:01	2172000	100.00		510.2	1 -	Mayms.	
		09:59	2 204000			5/02		Mayney	-
	-1	12:50	2 336 000			10.29		Myn	4
		21:27	2.737 000			1/ 02		2-	4
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	~~~	13:22	3.485000	10	0.95	102		Ms-	_
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Project name Specimen ID Date R= 0,1 Mastecithesis 8612-2 2A(00) 27/2-19 1.45

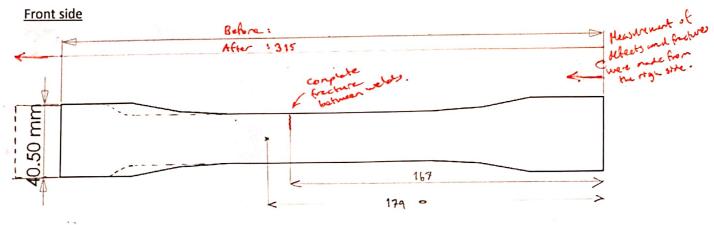
Stress range (ΔS)	400	Мра	Load range ΔF	97.33	kN	∆F=∆S*Area/1000
Target life	400000	Cycles	Max load	108.03	kN	Fmax=∆F/(1-R)
27.25 × 8.91 Area	= 243.0	mm²	Min load	10.80	kN	Fmin=Fmax-∆F
Cyclic rate	13	Hz	, Mean load	59.42	kN	Fmean=(Fmax+Fmin)/2
Test rig	MTS 809	Machine load	59.42	48.62	± kN	F=Fmean±∆F/2

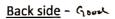
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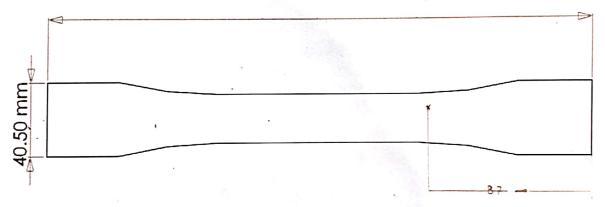
Visuel check before fatigue testing.

Test: 17 Name: 8612 - 2 2A (BB) Maynus Larsson 23.06.2019

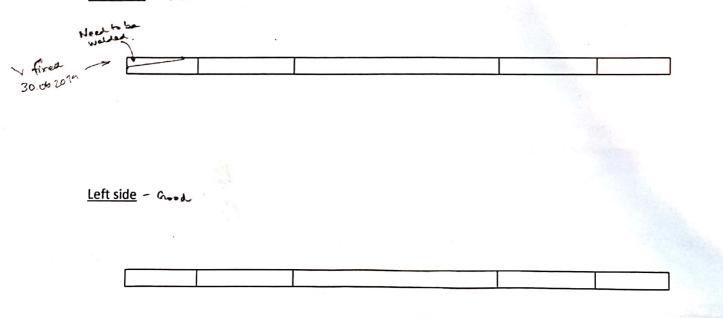
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Right side - Good



100

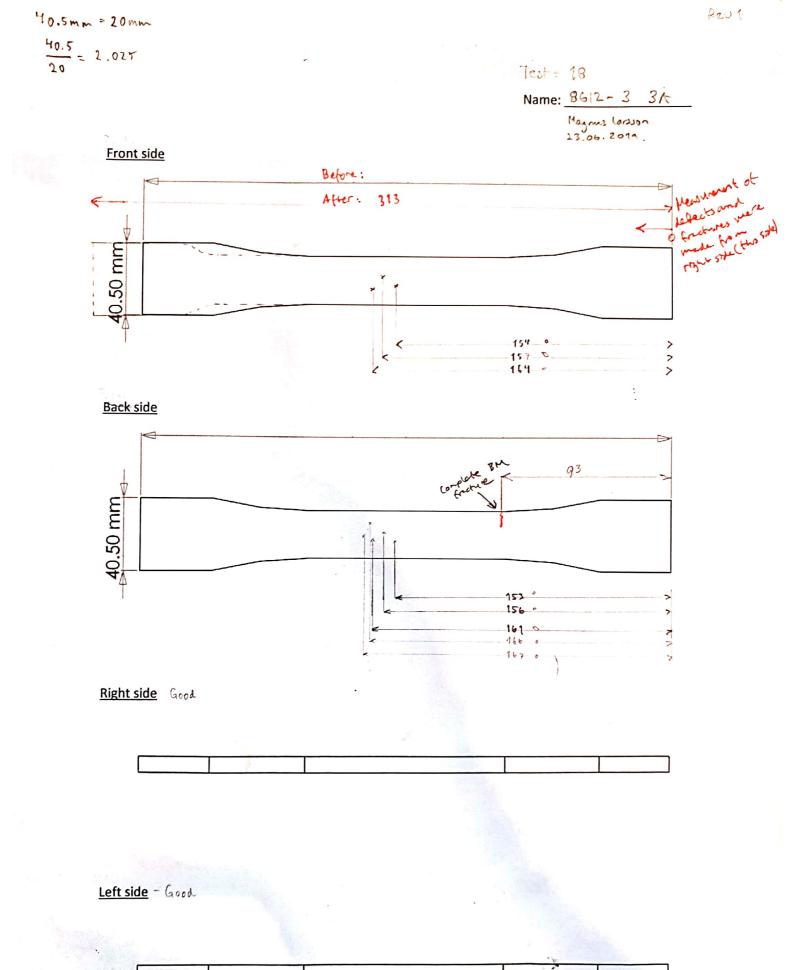


18

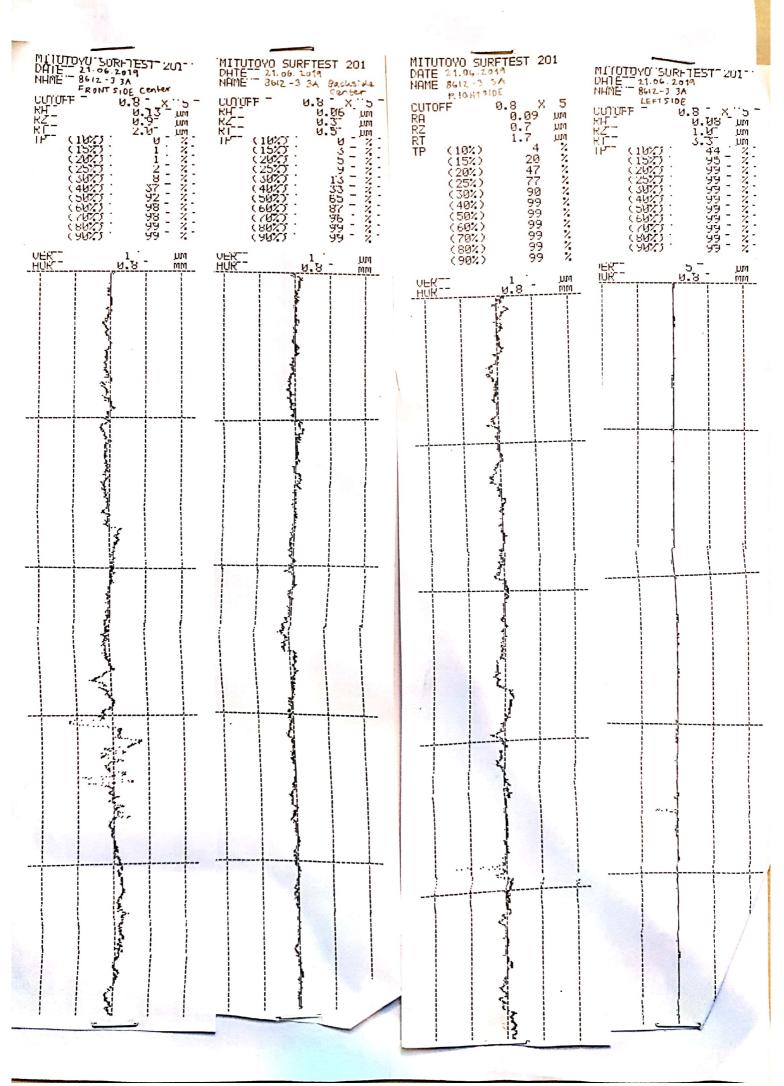
Project name Specimen ID Date R= 0,1 Master Thesis 8612-3 3A 23/2-14

Stress range (∆S)	400	Мра	Load range ∆F	97.63	kN	ΔF=ΔS*Area/1000
Target life	350	Cycles	Max load	138.47	kN	Fmax=∆F/(1-R)
27.27x 8.95 Area	244.07	mm²	Min load	10.84	kN	Fmin=Fmax-∆F
Cyclic rate	(3	Hz	Mean load	59.66	kN	Fmean=(Fmax+Fmin)/2
Test rig	MTS 809	Machine load	59.66	48.52	± kN	F=Fmean±∆F/2

Date	Time	Cycles	Observation, comments	Sign
23/7-19	1114	600	2. 22/1.94	Mattin)
( <b>I</b> N	1231	60 000	4. 71/2.60	
~	1432	154 400	4.47/3.67	
~	1646	250 0.0	4,45/3.61	
n	1823	300 000	Cruch ; nitig eirn	ч
	1841	305242	crache Proposetion 502/48	o u
	1848	306418	Fracture	~ ~
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# **Residual Stress Measurements**

• The chapter "Residual Stress Measurements" has no attachments.